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1. Introduction

Solid rocket propellant is the power source for rockets and missile engines, and its performance directly affects the operational effectiveness and survivability of missile systems. Solid propellants are usually composed of oxidants, adhesives, fuels, burning rate catalysts and other components. Oxidants are the main energy components of the propellants, and their thermal decomposition characteristics significantly affect the combustion properties of solid propellants.^{1–3} Ammonium perchlorate (AP) is a commonly used oxidizer of solid rocket propellant owing to its high oxygen content, favorable physical and chemical stability.^{4–7}

Nanometer-scale transition metal oxides used as burning rate catalysts could effectively promote the thermal decomposition of the oxidant and improve the combustion properties of solid propellants.^{8–10} Numerous transition metal oxides, such as CuO, NiO, MnO₂, TiO₂ and Fe₂O₃ have been used as burning rate catalysts of solid propellants.^{11–15} Among them, Fe₂O₃ is widely used for the promotion of thermal decomposition owing to its abundance, low cost and non-toxic properties.^{16–18} Ma *et al.*¹⁸ studied the thermal decomposition properties of AP with and without Fe₂O₃ nanoparticles. The obtained Fe₂O₃–AP

Effect of rGO–Fe₂O₃ nanocomposites fabricated in different solvents on the thermal decomposition properties of ammonium perchlorate

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rGO-Fe₂O₃ composites were prepared using different solvents (distilled water, ethanol, *N*-methylpyrrolidone, ethyl acetate, *n*-butyl alcohol and *N*,*N*-dimethylformamide) *via* a simple solvothermal method and characterized by SEM, TEM, XRD, FT-IR, Raman and XPS techniques. Then, the catalytic action of rGO-Fe₂O₃ composites (prepared in different solvents), GO, rGO and pristine Fe₂O₃ on the thermal decomposition of ammonium perchlorate (AP) were studied using a TG-DSC instrument. DSC results showed that GO, rGO, pristine Fe₂O₃ and rGO-Fe₂O₃ composites could effectively promote the thermal decomposition of AP. Additionally, rGO-Fe₂O₃ fabricated in *N*,*N*-dimethylformamide (DMF) solvent possesses the best catalytic performance among the investigated materials. This result can be attributed to the better dispersibility of Fe₂O₃ nanoparticles on graphene oxide prepared in DMF solvent, which was confirmed by SEM and TEM imaging. The high temperature decomposition exothermic peak (HTP) and the apparent activation energy of AP are reduced by 119.6 °C and 173.3 kJ mol⁻¹, respectively, in the presence of Fe₂O₃-rGO (DMF) composite.

composite presented better thermal decomposition and combustion properties compared with those of pristine AP.

However, the large surface area of this nanoscale catalyst makes it easy to agglomerate, and the catalytic activity after agglomeration significantly reduced. Graphene and its derivatives have been used in the fields of catalyst loading and energetic materials owing to their large specific surface area, excellent mechanical properties, good electrical and thermal conductivity performance.¹⁹⁻²⁴ Dey et al.²³ fabricated rGO-Fe₂O₃ composite via microwave irradiation and ultrasonic treatment methods, and used it to catalyze the thermal decomposition of AP. The results indicated that the introduction of graphene could effectively promote the thermal decomposition of AP by improving the dispersion of Fe₂O₃ nanoparticles. As a commonly used method for nanomaterials fabrication, solvothermal method (water as the reaction medium) was also used for rGO-Fe₂O₃ composite fabrication. HTP of AP significantly reduced from 432 to 367 °C after mixing with the rGO-Fe₂O₃ composite.²⁴ It is well-known that the solvent in the solvothermal method has a great influence on the structure and properties of the as-synthesized materials. Additionally, the structure and properties of the catalytic materials directly affect the thermal decomposition properties of the catalyzed energetic materials.^{25,26} However, the effect of reaction solvents on properties of rGO-Fe2O3 composites and the effect of fabricated rGO-Fe₂O₃ composites on the thermal decomposition properties of AP are seldom reported.

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Based on this discussion, the effects of different solvents on the properties of graphene loaded ferric oxide (rGO– Fe_2O_3) nanocomposites, prepared *via* a simple one-pot solvothermal method, were studied using SEM, TEM, XRD, FT-IR, Raman and XPS techniques. In addition, the effects of the as-synthesized rGO, pristine Fe_2O_3 and rGO– Fe_2O_3 composites on AP decomposition were investigated. The apparent activation energy of AP was calculated using the Kissinger and Ozawa's methods with the obtained DSC results. Based on the abovementioned research strategy, the best rGO– Fe_2O_3 sample for thermal decomposition promotion of AP was determined, and the possible catalytic mechanisms on the thermal decomposition of AP was also analyzed (see Fig. 1).

2. Experiment

2.1 Materials

All the chemicals were of analytical grade and used without further purification. Graphite oxide (GO, C/O = 1) was purchased from Ashine Advanced Carbon Materials Co., Ltd., Changzhou. Distilled water, ethanol (EA, Xi'an Chemical Reagent Factory), *N*,*N*-dimethylformamide (DMF, Chron chemical Co., Ltd. of ChengDu), *N*-methylpyrrolidone (NMP, Xilong Scientific Co., Ltd. of SiChuan), ethylene glycol (EG) (EG, Sinopharm Chemical Reagent Co., Ltd.) and *n*-butyl alcohol (NBA, Chron chemical Co., Ltd. of ChengDu) were used as solvents, separately. Ammonia (Xilong Scientific Co., Ltd. of SiChuan, 25–28%) was used to adjust the pH value of solvents. Ferric nitrate hydrate (Fe(NO₃)₃·9H₂O) was purchased from Aladdin Inc. Ammonium perchlorate was obtained from Xi'an Modern Chemistry Research Institute.

2.2 Preparation of rGO, Fe₂O₃ and Fe₂O₃-rGO composites

Graphite oxide (60 mg) was ultrasonically dispersed in 50 mL of different solvents (six types) for 30 minutes, separately. Then, 0.2 g Fe(NO₃)₃·9H₂O (dissolved in 10 mL different solvents) was added into the above six dispersions under mechanical agitation. After mixing, the pH value of the solution was adjusted to 9–10 by adding ammonia. The reactants were transformed into 100 mL Teflon-sealed autoclave and heated at 180 °C for 24 hours. After cooling, the products were washed several times with distilled water and absolute alcohol. Then, the products were cured at 60 °C in a vacuum oven overnight and ground for further characterization. Fe₂O₃ and rGO were also fabricated using the same procedure except without the addition of Fe(NO₃)₃·9H₂O or GO.

2.3 Characterization

The morphology and size of GO, rGO, pristine Fe_2O_3 and rGO–Fe₂O₃ composites (prepared in different solvents) were characterized by scanning electron microscopy (SEM, Quanta600) and transmission electron microscopy (TEM, Tecnai G2 F20). Structure and composition were analyzed using X-ray diffraction (XRD, EMPYREAN), Fourier transform infrared spectroscopy (FT-IR, Bruker Tensor 27), Raman spectroscopy (RM2000, 532 nm) and X-ray photoelectron spectroscopy (XPS, Thermo scientific). XRD patterns were collected using a Cu K α source in the measurement angle range of $2\theta = 5-90^{\circ}$ with a scan rate of 8° min⁻¹. The catalytic action of the rGO–Fe₂O₃ composites (prepared in different solvents), GO, rGO and pristine Fe₂O₃ on the thermal decomposition of AP (the mass ratio of AP to additive is 5) were studied by TG-DSC (NETZSCH STA449F3) with a heating rate



Fig. 1 Illustration of rGO-Fe₂O₃ fabrication and its catalytic mechanism for the thermal decomposition of AP.

of 5, 10, 15 and 20 °C min⁻¹. Kissinger and Ozawa's methods were employed to obtain the kinetic parameters of AP before and after mixing with different additives.

3. Results and discussion

3.1 XRD analysis

XRD patterns of GO, as-prepared rGO, pristine Fe₂O₃ and rGO-Fe₂O₃ composites (prepared in different solvents) are presented in Fig. 2. The diffraction peaks of pristine Fe₂O₃ appear at 24.2°, 33.2°, 35.6°, 40.9°, 49.5°, 54.1°, 62.4° and 64.0°, which correspond well with the crystal planes of (012), (104), (110), (113), (024), (116), (214) and (300) of hematite, respectively (JCPDS No. 33-0664).³⁰ In addition, the characteristic diffraction peaks of Fe₂O₃ appear in the XRD patterns of rGO-Fe₂O₃ composites (fabricated in NBA, H₂O, NMP, EA and DMF solvents), demonstrating the successful formation of Fe₂O₃. However, the characteristic diffraction peaks of Fe₂O₃ are absent in the sample prepared in EG solvent, indicating that rGO-Fe₂O₃ could not be fabricated using EG as the solvent, probably because of the reducibility of EG. In addition, the characteristic (002) diffraction peak of graphite $(2\theta = 11^{\circ})$ could be seen in the pattern of the GO sample. Nevertheless, the peak at 11° disappears in the patterns of samples of rGO and rGO-Fe₂O₃ composites, confirming the reduction of GO after solvothermal treatment.

3.2 Morphology and size characterization

The morphologies and sizes of GO, rGO, Fe_2O_3 , and rGO- Fe_2O_3 samples (fabricated in different solvents) were characterized using SEM and TEM, and the results are shown in Fig. 3 and 4. SEM results (Fig. 3c) indicate that the Fe_2O_3 sample is composed of agglomerated spherical Fe_2O_3 nanoparticles with an average particle size of 130 nm. Evident

aggregation of the as-received graphite oxide (without ultrasonic treatment) could be seen in the SEM imagines. This is distinct from the TEM results which show that the oligomeric structure of graphene oxide is obtained after ultrasonic treatment of 30 min. GO can be considered as an oligomeric structure because ultrasonic treatment was performed in all preparation processes. Both SEM and TEM images (Fig. 3b) show the crimped structure of the as-prepared rGO, fabricated *via* the solvothermal method, which can be attributed to its extremely thin feature.

SEM results indicate that Fe₂O₃ nanoparticles were successfully anchored on the surface of graphene (Fig. 4(a)-(e)). However, the morphologies of the composites and the dispersion of the Fe₂O₃ nanoparticles were found to be significantly distinct for different solvents. When compared with the rGO-Fe₂O₃ composites fabricated in NBA, H₂O, NMP, EA and EG solvents, the composite obtained in DMF solvent presents a thinner structure (see Fig. 4). It is also significant that the distribution of Fe₂O₃ nanoparticles anchored on graphene is high even when using DMF as the reaction solvent, confirming that graphene could effectively prevent the aggregation of Fe₂O₃ nanoparticles. This phenomenon can be attributed to the better dispersion performance of DMF for both GO and rGO.^{27,28} Nevertheless, the poor dispersion of rGO in H₂O, EA and EG solvents results in an inferior interaction between rGO and Fe₂O₃. Moreover, the composite obtained in the EG solvent shows a different morphology from other samples. This can be attributed to the distinct composition (confirmed by XRD results).

3.3 FT-IR analysis

In order to confirm the reduction of GO and the interaction between rGO and Fe_2O_3 , FT-IR measurements were recorded to investigate the composition of GO, rGO, Fe_2O_3 and rGO–



Fig. 2 XRD results of GO, rGO, Fe_2O_3 and rGO- Fe_2O_3 composites. (a) XRD results of GO, rGO, Fe_2O_3 and rGO- Fe_2O_3 (DMF) samples. (b) XRD results of rGO- Fe_2O_3 composites prepared in different solvents.



Fig. 3 SEM and TEM images of GO, rGO, Fe₂O₃ and rGO-Fe₂O₃ (DMF) samples: (a) GO, (b) rGO, (c) Fe₂O₃, and (d) rGO-Fe₂O₃ (DMF).



Fig. 4 SEM and TEM images of rGO-Fe₂O₃ composites fabricated in different solvents: (a) NBA, (b) H₂O, (c) EA, (d) NMP, (e) DMF, and (f) EG.

Fe₂O₃ composites. The corresponding spectra are shown in Fig. 5. Absorption peaks at around 3420 cm⁻¹ (strong and broad) and 1625 cm⁻¹ in GO sample are derived from the O-H stretching vibration and bending vibration of adsorbed water molecules, respectively. Moreover, the stretching vibration of C=O (COOH groups) situated at the surface and edge of GO sheets can be observed at 1730 cm⁻¹. For the pristine

Fe₂O₃, the strong peaks appear at 563 and 478 cm⁻¹ (derive from the stretching vibration of Fe–O), which is consistent with previous studies that show the peaks of Fe–O appearing at around 470 cm⁻¹ and 580 cm⁻¹.²⁹ The spectra of rGO and rGO–Fe₂O₃ composites differ from that of the GO sample, as evidenced by the significantly weaker peaks at 1625, 1730 and 3420 cm⁻¹, confirming the reduction of GO.



Fig. 5 FT-IR results of GO, rGO, Fe₂O₃ and rGO-Fe₂O₃ composites: (a) FT-IR results of GO, rGO, Fe₂O₃ and rGO-Fe₂O₃ (DMF) samples; (b) FT-IR results of rGO-Fe₂O₃ composites prepared in different solvents.

Moreover, the distinction of the Fe–O peak wavenumbers between the Fe₂O₃ (peaks at 563 and 478 cm⁻¹) and rGO–Fe₂O₃ (peaks at 555 and 474 cm⁻¹) samples also confirm the chemical interaction between the metal oxide and graphene.

Based on this finding, we could suspect that van der Waals' forces (physical adsorption, weak interaction) and chemical interactions (chemical bond, strong interaction) coexist between Fe_2O_3 nanoparticles and rGO.²⁴



Fig. 6 XPS results of GO, rGO, Fe₂O₃ and rGO–Fe₂O₃ composites: (a) C 1s spectra of GO sample; (b) C 1s spectra of rGO and rGO–Fe₂O₃ (DMF) samples; (c) Fe 2p spectra of Fe₃O₄ and rGO–Fe₂O₃ (DMF) samples.

3.4 XPS analysis

XPS was performed to study the surface chemical composition and to identify the interaction between GO and Fe₂O₃. The results are shown in Fig. 6. The spectrum of C 1s (Fig. 6a) of GO can be divided into four main peaks at 284.6, 285.4, 288.5 and 288.8 eV, associated with C-C (sp²-C), C-OH (oxygen in hydroxyl), C=O and HO-O=C (carboxyl) groups, respectively. However, only three peaks, associated with C-C (sp²-C), C-OH (oxygen in hydroxyl) and C=O groups, appear in the C1s spectra of rGO sample. Moreover, the change in peak intensity (intensity reduction of C-OH and C=O groups, intensity increase of the C-C (sp²-C) group) confirm the reduction of GO to rGO. Particularly, the C 1s peak of the rGO-Fe₂O₃ (DMF) sample located at 284.4 eV (refer to carbon in C-O-H bond) shifts toward 0.3 eV lower binding energy in comparison with that of rGO owing to the stronger interaction between O and Fe, indicating the formation of Fe-O-C bond.31

Additionally, it was found that peaks at around 711, 720 and 725 eV appearing in the spectra of Fe_2O_3 and $rGO-Fe_2O_3$ (DMF) samples are related to Fe 2p, providing clear evidence for the existence of Fe^{3+} on the surface of graphene (Fig. 6c). XPS results show the chemical interaction between GO and Fe_2O_3 and also indicate the reduction of GO to rGO after solvothermal treatment.

3.5 Raman analysis

The Raman spectra of GO, rGO, pristine Fe_2O_3 and rGO– Fe_2O_3 (DMF) samples are presented in Fig. 7. The Raman peak at around 1340 cm⁻¹ (D band) is related to the defects and disorders in the hexagonal graphitic layers. In addition, the peak at 1580 cm⁻¹ (G band), corresponding to an E_{2g} mode of graphite, is related to the vibration of sp²-bonded carbon atoms in the 2-dimensional hexagonal lattice.²⁴ In the Raman spectrum of Fe_2O_3 sample, strong peaks at 214 cm⁻¹-

and 271 cm^{-1} are assigned to the A_{1g} mode and E_g mode, respectively. Typical peaks of hematite and graphene (D and G bands) appear in the Raman spectrum of the rGO-Fe₂O₃(DMF) composite, which reveals the formation of the rGO-Fe₂O₃ composite. The calculated I_D/I_G ratios of GO, rGO and rGO-Fe₂O₃ (DMF) samples are 2.75, 2.42 and 2.45, respectively. The distinction of the I_D/I_G ratios suggests an increase in sp² domains in the rGO and rGO-Fe₂O₃ (DMF) samples, confirming the reduction of GO to rGO, which is consistent with the XRD, FT-IR and XPS results. This also indicates the enhancement of the order degree after solvothermal treatment which concurs with previous studies.²³ Moreover, the enhanced I_D/I_G value of the rGO-Fe₂O₃ composite in comparison with that of rGO shows the decrease in the order degree, which is attributed to the interaction between Fe₂O₃ and rGO.

3.6 Thermal analysis

3.6.1 DSC results. The catalytic performance of GO, rGO, Fe_2O_3 and $rGO-Fe_2O_3$ composites on the thermal decomposition of AP was investigated by DSC. The results are shown in Fig. 8. The effects of different $rGO-Fe_2O_3$ samples (fabricated in different solvents) on the thermal decomposition of AP were also explored according to the DSC results. DSC results revealed that the thermal decomposition curve of pure AP contains one endothermic and two exothermic peaks. The endothermic peak appearing at 244 °C is attributed to the phase transition of AP from orthorhombic to cubic form. Exothermic peaks appear at 324 and 440 °C, corresponding to the LTP (low-temperature decomposition exothermic peak) and HTP of AP, indicating its full decomposition.

As can be seen from Fig. 8(a) and (b), in the DSC curve of AP, there is no distinct shift in the endothermic peak before and after the addition of GO, rGO, Fe_2O_3 and $rGO-Fe_2O_3$. However, the LTP and HTP of AP significantly change after



Fig. 7 RAMAN spectra of GO, rGO, Fe₂O₃ and rGO–Fe₂O₃ composites: (a) Raman peaks of rGO, Fe₂O₃ and rGO–Fe₂O₃ (DMF); (b) I_D/I_G values of GO, rGO and rGO–Fe₂O₃ (DMF).



Fig. 8 DSC and TGA results of pure AP, AP mixed with rGO, Fe_2O_3 and rGO– Fe_2O_3 at 10 °C min⁻¹: (a) DSC results of pure AP and AP mixed with GO, rGO, Fe_2O_3 and rGO– Fe_2O_3 (DMF); (b) DSC results of AP mixed with different rGO– Fe_2O_3 samples; (c) TGA results of AP, AP mixed with rGO, Fe_2O_3 , rGO– Fe_2O_3 (H₂O) and rGO– Fe_2O_3 (DMF).

mixing with GO, rGO, pristine Fe_2O_3 and $rGO-Fe_2O_3$ composites (see Table 1). The HTP of AP + GO and AP + rGO samples decrease by 68.3 and 79.2 °C, respectively, in comparison with the HTP of pure AP (440.8 °C). The better catalytic activity of GO and rGO on the thermal decomposition of AP resulted from the excellent thermal conductivity of the graphene materials, which was also confirmed by previous studies.^{32,33}

Table 1 Thermal decomposition temperatures of AP mixed with different catalysts at 10 $^{\circ}\text{C}$ min $^{-1}$

Sample no.	Energetic material	Catalysts				
		Name	Solvent	LTP/°C	HTP/°C	
1	AP	_	_	324.8	440.8	
2		GO	_	313.5	372.5	
3		rGO	H_2O	330.5	360.8	
4		Fe ₂ O ₃	H_2O	_	369.4	
5		rGO-Fe ₂ O ₃	DMF	_	321.2	
6			NMP	326.5	386.5	
7			H_2O	_	349.8	
8			EA	_	325.2	
9			NBA	_	322.8	
10			EG	324.7	344.7	

Additionally, the rGO-Fe₂O₃ samples (fabricated in different solvents) present different effects on the thermal decomposition temperature of AP (Fig. 8b). The HTP of AP mixed with rGO-Fe₂O₃ composites fabricated in H₂O, DMF, EA, EG, NMP and NBA solvents reduced by 91.0, 119.6, 115.6, 96.1, 54.3 and 118.0 °C, respectively. The HTP of AP + rGO-Fe₂O₃ (fabricated in DMF solvent) advanced the maximum, demonstrating that the rGO-Fe₂O₃ (DMF) sample exhibits the best catalytic performance on the thermal decomposition of AP. It is clear that compared with the rGO-Fe₂O₃ (H₂O) sample, the rGO-Fe₂O₃ composite fabricated in DMF solvent presents higher catalytic activity on the thermal decomposition of AP, which indicates that DMF is more suitable as a solvent for rGO-Fe₂O₃ composite fabrication. Combining DSC and SEM results, we can conclude that the enhanced catalytic property of rGO-Fe₂O₃ (DMF) can be attributed to the uniform distribution of Fe₂O₃ nanoparticles anchored on graphene, which could provide more active sites for the decomposition of AP.

3.6.2 TG results. Fig. 8(c) illustrates the thermogravimetric results of AP (heated from 50 to 500 °C) mixed with GO, rGO, pristine Fe_2O_3 , rGO– Fe_2O_3 (H₂O) and rGO– Fe_2O_3 (DMF). It can be seen that the pyrolysis behavior of AP contains two

stages corresponding to the high temperature and low temperature decomposition exothermic processes (see DSC results). The residue quality of AP + rGO–Fe₂O₃ (DMF) sample is clearly lower than that of AP + rGO and AP + Fe₂O₃ samples, which also confirmed the excellent catalytic performance of rGO–Fe₂O₃ (DMF). Moreover, the high residual quality of the rGO–Fe₂O₃ (H₂O) sample shows its unsuitability for AP decomposition, which can be attributed to the poor dispersion of Fe₂O₃ nanoparticles.

3.6.3 Dynamics calculation. The apparent activation energy (Ea) of the thermal decomposition process of AP mixed with different additives was evaluated by Kissinger and Ozawa's methods using DSC data (see Table 2), and the results are listed in the Table 3. The apparent activation energies of the high temperature decomposition process for AP + GO, AP + rGO, AP + Fe₂O₃ and AP + rGO-Fe₂O₃ (DMF) are reduced by 76.9, 164.3, 122.8 and 173.3 kJ mol⁻¹, respectively, in comparison with 293.1 kJ mol⁻¹ for pure AP. These results indicate that rGO-Fe₂O₃ (DMF) can effectively decrease the thermal decomposition temperature and apparent activation energy of AP, and presents an excellent catalytic effect on the thermal decomposition of AP.

3.7 Catalytic mechanism of the thermal decomposition of AP catalyzed by rGO-Fe₂O₃ composites

AP decomposition occurred *via* three steps: phase transformation, low temperature decomposition (LTD) and high temperature decomposition (HTD). In the first step, the phase of AP transforms from the orthorhombic phase to the cubic phase. LTD is a solid–gas multiphase reaction procedure, including decomposition and sublimation processes. HTD involves reactions between HNO and NO for the generation of N₂O, and the major products of HTD are N₂O, O₂, Cl₂, H₂O and some NO (shown in Fig. 1).²⁴ The electron transfer process from ClO₄⁻ to NH₄⁺ and from O₂ to the superoxide ion (O₂⁻) are the controlling steps of LTD and HTD, respectively.²³ Then, the generated O₂⁻ contributes to the decomposition of NH₃ in combination with the other products of HClO₄; eventually, the AP is completely decomposed.²⁴

The possible mechanism by which the introduction of GO, rGO, Fe₂O₃ and rGO–Fe₂O₃ composites promote the thermal

 Table 2
 Thermal decomposition temperatures of AP mixed with different catalysts at different heating rates

Sample	Energetic			β /°C min ⁻¹			
no.	component	Catalysts	$T/^{\circ}\mathrm{C}$	5	10	15	20
1	AP		LTP	307.9	324.8	329.0	339.3
			HTP	432.9	440.8	447.0	452.3
2		GO	LTP	300.2	313.5	329.0	339.3
			HTP	360.2	372.5	378.7	379.6
3		rGO	LTP	315.1	330.5	342.7	347.1
			HTP	346.8	361.6	375.6	377.7
4		Fe ₂ O ₃	LTP	_	_	_	_
			HTP	358.1	369.4	379.3	383.8
5		rGO-Fe ₂ O ₃	LTP	_	_	_	_
		(DMF)	HTP	310.2	321.2	332.2	341.0

Table 3 Kinetic parameters of AP obtained by Kissinger method and Ozawa's method

Sample		Kissinger method			Ozawa's method	
no.	$T_{\rm p}/^{\rm o}{\rm C}$	$E_{\rm k}/{\rm kJ}~{\rm mol}^{-1}$	$L_{\rm g} \left(A_{\rm k} / {\rm s}^{-1} \right)$	r _k	$E_{\rm o}/{\rm kJ}~{\rm mol}^{-1}$	r _k
1	LTP	124.2	8.73	0.986	127.5	0.988
	HTP	293.1	19.47	0.995	290.0	0.995
2	LTP	91.0	5.76	0.985	95.9	0.988
	HTP	216.2	15.55	0.981	215.8	0.983
3	LTP	116.7	7.89	0.996	120.5	0.997
	HTP	128.8	8.38	0.987	132.5	0.989
4	LTP	_	_	_	_	_
	HTP	170.3	11.73	0.996	172.1	0.997
5	LTP	_	_	_	_	_
	HTP	119.8	8.31	0.982	123.4	0.984

decomposition of AP are shown in Fig. 1. The excellent thermal conductivity and unique electronic structure of graphene materials (GO and rGO) benefit the heat transfer process and the rate controlling step of thermal decomposition of AP.^{32,33} Therefore, the addition of graphene materials (GO and rGO) significantly reduced the HTP of AP. According to the traditional electron-transfer theory, the partially filled 3d orbit of Fe³⁺ benefits the transfer of electrons. When AP is mixed with Fe₂O₃, the positive holes of Fe₂O₃ can accept electrons from AP, thus promoting the thermal decomposition of AP.²⁴ Additionally, graphene can effectively inhibit the agglomeration of Fe₂O₃ nanoparticles, providing more active sites for the thermal decomposition of AP. Moreover, its excellent conductivity contributes to the transport of electrons. Based on the above discussion, the introduction of graphene could not only significantly reduce the agglomeration of Fe₂O₃, but also accelerate the transport of electrons to speed up the controlling steps of LTD and HTD. Hence, rGO-Fe2O3 (DMF) displays higher catalytic activity than pristine Fe₂O₃.²⁴ In addition, the enhanced dispersibility of Fe₂O₃ nanoparticles anchored on graphene is the primary reason for rGO-Fe₂O₃ (DMF) presenting better catalytic activity compared with that of other rGO-Fe₂O₃ samples.

4. Conclusions

The effect of different solvents on the structure and properties of $rGO-Fe_2O_3$ composites and the effect of the assynthesized $rGO-Fe_2O_3$ composites on the thermal decomposition performance of AP were studied. $rGO-Fe_2O_3$ composites were successfully fabricated using different solvents *via* a solvothermal method, except when using EG as the solvent. The chemical interaction between rGO and Fe_2O_3 was confirmed by FTIR, XPS and Raman results. SEM results indicate that the distribution of Fe_2O_3 nanoparticles on graphene is more uniform while using DMF as the reaction medium, confirming that graphene effectively prevents the aggregation of Fe_2O_3 nanoparticles. In addition, DSC results also indicate that it is better to use DMF as the solvent for rGO-Fe_2O_3 preparation, which could effectively promote the thermal decomposition of AP with a significantly reduced decomposition temperature and apparent activation energy. Combining the results of SEM and DSC, we can infer that the higher catalytic activity of rGO–Fe₂O₃ (DMF) compared with that of other additives can be attributed to the excellent electroconductivity of graphene and better dispersibility of Fe₂O₃ nanoparticles, which could provide more catalytic sites for the thermal decomposition of AP.

Conflicts of interest

There are no conflicts to declare.

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