# Syntheses, Structures, and Characterization of New Lead(II)-Tellurium(IV)-Oxide Halides: $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{X}_{2}$ and $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{X}_{2}(\mathrm{X}=$ Cl or Br ) 

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Received May 24, 2002


#### Abstract

The syntheses, structures, and characterization of four new lead(II)-tellurium(IV)-oxide halides, $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{X}_{2}$ and $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{X}_{2}(\mathrm{X}=\mathrm{Cl}$ or Br$)$ are reported. The materials are synthesized by solid-state techniques, using $\mathrm{Pb}_{3} \mathrm{O}_{2} \mathrm{Cl}_{2}$ or $\mathrm{Pb}_{3} \mathrm{O}_{2} \mathrm{Br}_{2}$ and $\mathrm{TeO}_{2}$ as reagents. The compounds have three-dimensional structural topologies consisting of lead-oxide halide polyhedra connected to tellurium oxide groups. In addition, the $\mathrm{Pb}^{2+}$ and $\mathrm{Te}^{4+}$ cations are in asymmetric coordination environments attributable to their stereoactive lone pair. We also demonstrate that $\mathrm{Pb}_{3}-$ $\mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{X}_{2}$ and $\mathrm{Pb}_{2} \mathrm{TeO}_{4} \mathrm{X}_{2}$ can be interconverted reversibly through the loss or addition of $\mathrm{TeO}_{2}$. X -ray data: $\mathrm{Pb}_{3}-$ $\mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$, monoclinic, space group $\mathrm{C} 2 / m$ (No. 12), $a=16.4417(11) \AA, b=5.6295(4) \AA, c=10.8894(7) \AA, \beta=$ 103.0130(10) ${ }^{\circ}, Z=4 ; \mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Br}_{2}$, monoclinic, space group $\mathrm{C} / \mathrm{m}$ ( No . 12), $a=16.8911(8) \AA$, $b=5.6804(2) \AA$, $c=11.0418(5) \AA, \beta=104.253(2)^{\circ}, Z=4 ; \mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$, orthorhombic, space group Bmmb (No. 63), $a=5.576(1)$ $\AA, b=5.559(1) \AA, c=12.4929(6) \AA, Z=4 ; \mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Br}_{2}$, orthorhombic, space group Bmmb (No. 63), $a=$ $5.6434(4) \AA, b=5.6434(5) \AA, c=12.9172(6) \AA, Z=4$.


## Introduction

The interest in synthesizing new mixed-metal oxide halide materials exists not only for their potential ion-exchange behavior but also for their optical properties as well as interesting crystal chemistry. We have previously reported that a low-temperature ( $160{ }^{\circ} \mathrm{C}$ ) aqueous method can be employed for the synthesis of new metal oxide halides. ${ }^{1}$ Specifically, we synthesized and characterized $\mathrm{Pb}_{3}\left(\mathrm{SeO}_{3}\right)\left(\mathrm{SeO}_{2}-\right.$ $\mathrm{OH}) \mathrm{Cl}_{3}$ and $\mathrm{Pb}_{3}\left(\mathrm{SeO}_{3}\right)_{2} \mathrm{Cl}_{2}$ utilizing this reflux technique and demonstrated that an irreversible transformation occurs between the two materials with the loss of HCl . In this paper, we report the syntheses, structures, and characterization of new $\mathrm{Pb}(\mathrm{II})-\mathrm{Te}(\mathrm{IV})$-oxide halides, $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{X}_{2}$ and $\mathrm{Pb}_{3}{ }^{-}$ $\mathrm{TeO}_{4} \mathrm{X}_{2}(\mathrm{X}=\mathrm{Cl}$ or Br$)$. In addition, we demonstrate that a reversible transformation occurs between the two materials through the loss or addition of $\mathrm{TeO}_{2}$.

## Experimental Section

Reagents. PbO (99.9+ \%, Aldrich), $\mathrm{PbCl}_{2}$ ( $99 \%$, Alfa Aesar), and $\mathrm{TeO}_{2}$ ( $99.9 \%$ Aldrich) were used as received. $\mathrm{Pb}_{3} \mathrm{O}_{2} \mathrm{Cl}_{2}$ and $\mathrm{Pb}_{3} \mathrm{O}_{2} \mathrm{Br}_{2}$ were synthesized by heating a stoichiometric mixture of

[^0]PbO and the corresponding lead halide in air at $500{ }^{\circ} \mathrm{C}$ for 1 day. The powder X-ray diffraction patterns for $\mathrm{Pb}_{3} \mathrm{O}_{2} \mathrm{Cl}_{2}$ and $\mathrm{Pb}_{3} \mathrm{O}_{2} \mathrm{Br}_{2}$ matched those previously reported. ${ }^{2-4}$

Synthesis. Single crystals of $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ were grown by combining $\mathrm{Pb}_{3} \mathrm{O}_{2} \mathrm{Cl}_{2}\left(0.2658 \mathrm{~g}, 3.67 \times 10^{-4} \mathrm{~mol}\right)$ and $\mathrm{TeO}_{2}(0.2342$ $\left.\mathrm{g}, 1.47 \times 10^{-3} \mathrm{~mol}\right)$ in a fused silica tube that was subsequently evacuated and sealed. The mixture was heated at $550{ }^{\circ} \mathrm{C}$ for 1 day and cooled at a rate of $6^{\circ} \mathrm{C} \mathrm{h}^{-1}$ to room temperature. A few clear colorless crystals were observed (roughly $2 \%$ of the bulk powder). Bulk, polycrystalline $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ was synthesized by combining $\mathrm{Pb}_{3} \mathrm{O}_{2} \mathrm{Cl}_{2}\left(0.3466 \mathrm{~g}, 4.79 \times 10^{-4} \mathrm{~mol}\right)$ and $\mathrm{TeO}_{2}(0.1529 \mathrm{~g}, 9.58$ $\times 10^{-4} \mathrm{~mol}$ ). The mixture was introduced into a fused silica tube that was subsequently evacuated and sealed. The tube was heated to $575^{\circ} \mathrm{C}$ for 1 day and then furnace cooled to room temperature. An off-white powder was recovered.

The other phases, $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Br}_{2}, \mathrm{~Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$, and $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Br}_{2}$, were synthesized as polycrystalline powders. All attempts to grow single crystals were unsuccessful. Each material was synthesized by combining stoichiometric amounts of $\mathrm{Pb}_{3} \mathrm{O}_{2} \mathrm{Br}_{2}\left(\mathrm{~Pb}_{3} \mathrm{O}_{2} \mathrm{Cl}_{2}\right)$ with $\mathrm{TeO}_{2}$. The mixtures were introduced into separate fused silica tubes

[^1]Table 1. Crystallographic Data for $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$

| fw | 1043.67 |
| :--- | :--- |
| space group | $C 2 / m($ No. 12) |
| $a(\AA \AA)$ | $16.4417(11)$ |
| $b(\AA \AA)$ | $5.6295(4)$ |
| $c(\AA)$ | $10.8894(7)$ |
| $\beta(\mathrm{deg})$ | $103.013(10)$ |
| $V\left(\AA^{3}\right)$ | $982.02(11)$ |
| $Z$ | 4 |
| temp $\left({ }^{\circ} \mathrm{C}\right)$ | $25.0(2)$ |
| $\lambda(\AA)$ | 0.71073 |
| $\rho\left(\mathrm{~g} \mathrm{~cm}^{-3}\right)$ | 7.059 |
| $\mu\left(\mathrm{~cm}^{-1)}\right.$ | $576.65 \mathrm{~cm}^{-1}$ |
| $R(F)^{a}$ | 0.0617 |
| $R_{\mathrm{w}}\left(F^{2}\right)^{b}$ | 0.158 |
| ${ }^{a} R=\Sigma\| \| F_{\mathrm{o}}\left\|-\left\|F_{\mathrm{c}}\right\|\right\| / \Sigma\left\|F_{\mathrm{o}}\right\| \cdot{ }^{b} R_{\mathrm{w}}=\left[\Sigma w\left(\left\|F_{\mathrm{o}}{ }^{2}\right\|-\left\|F_{\mathrm{c}}{ }^{2}\right\|\right)^{2} / \Sigma w\left(F_{\mathrm{o}}{ }^{2}\right)^{2}\right]^{1 / 2}$. |  |

that were subsequently evacuated and sealed. For $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Br}_{2}\left(\mathrm{~Pb}_{3-}\right.$ $\mathrm{TeO}_{4} \mathrm{Cl}_{2}$ and $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Br}_{2}$ ) the tubes were heated to $575{ }^{\circ} \mathrm{C}(650$ ${ }^{\circ} \mathrm{C}$ ) for 1 day and cooled at a rate of $6{ }^{\circ} \mathrm{C} \mathrm{h}^{-1}$ to room temperature. With $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Br}_{2}\left(\mathrm{~Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}\right.$ and $\left.\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Br}_{2}\right)$ off-white (yellow) polycrystalline powders were recovered.

Single-Crystal Structure Determination. The structure of $\mathrm{Pb}_{3}{ }^{-}$ $\mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ was determined by standard crystallographic methods. For $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ a colorless column ( $0.04 \mathrm{~mm} \times 0.08 \mathrm{~mm} \times 0.20$ mm ) was used for single-crystal measurements. Room-temperature intensity data were collected on a Siemens SMART diffractometer equipped with a $1 \mathrm{~K} C C D$ area detector using graphite- monochromated Mo $\mathrm{K} \alpha$ radiation. A hemisphere of data was collected using a narrow-frame method with scan widths of $0.30^{\circ}$ in $\omega$ and an exposure time of $25 \mathrm{~s} /$ frame. The first 50 frames were remeasured at the end of the data collection to monitor instrument and crystal stability. The maximum correction applied to the intensities was $<1 \%$. The data were integrated using the Siemens SAINT program, ${ }^{5}$ with the intensities corrected for Lorentz, polarization, air absorption, and absorption attributable to the variation in the path length through the detector faceplate. $\psi$-scans were used for the absorption correction on the hemisphere of data. The data were solved and refined using SHELXS-97 and SHELXL-97, respectively. ${ }^{6,7}$ All calculations were performed using the WinGX-98 crystallographic software package. ${ }^{8}$ Crystallographic data, atomic coordinates, and thermal parameters for $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ are given in Tables 1 and 2 .

Powder Diffraction and Crystal Structure Refinement. The X-ray powder diffraction data were collected on a Scintag XDS2000 diffractometer at room temperature ( $\mathrm{Cu} \mathrm{K} \alpha$ radiation, $\theta-\theta$ mode, flat plate geometry) in the $2 \theta$ range $3-110^{\circ}$ with a step size of $0.02^{\circ}$ and a step time of 10 s . The diffraction patterns were analyzed using the Rietveld ${ }^{9}$ method with the FULLPROF program. ${ }^{10}$ An asymmetry correction was applied to the low-angle reflections. The scale was refined initially, followed in subsequent iterations by the zero point error, cell constants, peak shape parameters, atomic parameters, and overall isotropic temperature factors. For $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6}{ }^{-}$ $\mathrm{Cl}_{2}$, a powder refinement was undertaken attributable to the somewhat large errors observed in the single-crystal refinement.

[^2]Table 2. Atomic Coordinates for $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$

| atom | $x$ | $y$ | $z$ | $U_{\text {eq }}{ }^{a}\left(\AA^{2}{ }^{2}\right)$ |
| :--- | :---: | :--- | :--- | :--- |
| $\mathrm{Pb}(1)$ | $0.26200(6)$ | 0.0000 | $0.20191(9)$ | $0.0113(4)$ |
| $\mathrm{Pb}(2)$ | $0.02126(6)$ | 0.0000 | $0.20025(9)$ | $0.0108(4)$ |
| $\mathrm{Pb}(3)$ | $0.16092(7)$ | 0.5000 | $0.39346(10)$ | $0.0145(4)$ |
| $\mathrm{Te}(1)$ | $0.10312(10)$ | 0.5000 | $0.05086(15)$ | $0.0088(4)$ |
| $\mathrm{Te}(2)$ | $0.37188(10)$ | 0.5000 | $0.40736(15)$ | $0.0088(4)$ |
| $\mathrm{Cl}(1)$ | $0.3179(4)$ | 0.5000 | $0.0993(6)$ | $0.0166(14)$ |
| $\mathrm{Cl}(2)$ | $-0.0410(5)$ | 0.5000 | $0.3016(8)$ | $0.0252(17)$ |
| $\mathrm{O}(1)$ | $0.1309(8)$ | $0.263(3)$ | $0.1830(12)$ | $0.016(3)$ |
| $\mathrm{O}(2)$ | 0.0000 | $0.288(4)$ | 0.0000 | $0.024(5)$ |
| $\mathrm{O}(3)$ | $0.3935(13)$ | 0.5000 | $0.5810(19)$ | $0.018(2)$ |
| $\mathrm{O}(4)$ | $0.2913(9)$ | $0.262(3)$ | $0.3859(13)$ | $0.018(2)$ |

${ }^{a} U_{\mathrm{eq}}$ is defined as one-third of the trace of the orthogonalized $\mathbf{U}_{i j}$ tensor.

Table 3. Summary of Crystallographic Powder X-ray Diffraction and Refinement Data for $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ and $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$

|  | $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ | $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$ |
| :---: | :---: | :---: |
| $a($ A $)$ | 16.4349(5) | 5.576(1) |
| $b(\AA)$ | 5.6255(1) | 5.559(1) |
| $c(\AA)$ | 10.8802(3) | 12.4929(6) |
| $\beta$ (deg) | 103.050(1) |  |
| $V\left(\AA^{3}\right)$ | 979.94(1) | 385.9(1) |
| space group | C2/m (No. 12) | Bmmb (No. 63) |
| observns | 1380 | 352 |
| $\chi^{2}$ | 2.55 | 3.25 |
| $R_{\mathrm{p}}{ }^{a}$ | 0.131 | 0.169 |
| $R_{\text {wp }}{ }^{\text {b }}$ | 0.167 | 0.215 |
| $R_{\text {exp }}{ }^{c}$ | 0.104 | 0.119 |
| $R_{\text {Bragg }}{ }^{\text {d }}$ | 0.069 | 0.087 |
| ${ }^{a} R_{\mathrm{p}}=\Sigma \mid I_{\mathrm{o}}-1$ <br> ${ }^{d} R_{\text {Bragg }}=\Sigma \mid I_{\text {k(obs }}$ calculated integrat derived from an squares refinement | ${ }_{\mathrm{wp}}=\left[\Sigma w \mid I_{\mathrm{o}}-I_{\mathrm{c}}\right.$ <br> $/ \Sigma I_{\text {k(obs) }}$, where ies, $I_{\mathrm{k}}$ is the Bra gation scheme | 2. ${ }^{c} R_{\text {exp }}=R_{\mathrm{wp}} /(\chi$ are the observed $y$, and $w$ is the we process of the le |

The peaks were indexed on a monoclinic cell, with the refinements of the unit cell constants performed using a least-squares method. The structural refinements were carried out in space group $C 2 / m$ (No. 12) with a starting model based on the single-crystal data. A total of 44 parameters, including 13 profile parameters, were used in the refinement. For $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$ the peaks were indexed on an orthorhombic cell, with refinement of the unit cell constants performed by a least-squares method. The structural refinements were carried out in space group Bmmb (No. 63) with a starting model based on the structure of orthorhombic $\mathrm{PbBiO}_{2} \mathrm{Cl}$. To model $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$ with the orthorhombic $\mathrm{PbBiO}_{2} \mathrm{Cl}$ structure, statistical disorder must occur between the $\mathrm{Pb}^{2+}$ and the $\mathrm{Te}^{4+}$ cations. A variety of disorder models are possible; however, the model that gave the best fit to the data and made the most chemical sense is to statistically disorder $\mathrm{Pb}(2)$ and $\mathrm{Te}(1)$ (vide infra). In addition, although the unit cell is metrically tetragonal, refinements using higher symmetry tetragonal space groups resulted in large errors. A total of 20 parameters, including 12 profile parameters, were used in the refinement. The results of the powder refinements, atomic coordinates, thermal parameters, and bond distances are given in Tables 3-5. $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Br}_{2}$ and $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Br}_{2}$ are isostructural with $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ and $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$, respectively. Tables 6 and 7 gives the refined unit cell, space group, $h k l, d_{\mathrm{obs}}, d_{\text {calc }}, I_{\mathrm{obs}}$, and $I_{\text {calc }}$ for $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Br}_{2}$ and $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Br}_{2}$, respectively.

Infrared Spectroscopy. Infrared spectra were recorded on a Matteson FTIR 5000 spectrometer in the $400-4000 \mathrm{~cm}^{-1}$ range, with the sample pressed between two KBr pellets.

Thermogravimetric Analysis. Thermogravimetric measurements were carried out on a TGA 2950 thermogravimetric analyzer (TA Instruments). The samples were contained within platinum

Table 4. Fractional Atomic Coordinates, Isotropic Temperature Factors ( $\AA^{2}$ ), and Occupancies for $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$

| atom | $x$ | $y$ | $z$ | $U_{\text {eq }}\left(\AA^{2}\right)$ | occ |
| :--- | :---: | :---: | :--- | :---: | :---: |
| $\mathrm{Pb}(1)$ | 0.00000 | 0.2500 | $0.39063(19)$ | $0.02504(9)$ | 1.0 |
| $\mathrm{~Pb}(2)$ | 0.00000 | 0.2500 | $0.09796(23)$ | $0.0219(1)^{a}$ | 0.5 |
| $\mathrm{Te}(1)$ | 0.00000 | 0.2500 | $0.09796(23)$ | $0.0219(1)^{a}$ | 0.5 |
| $\mathrm{Cl}(1)$ | 0.00000 | 0.2500 | $0.74746(114)$ | $0.0123(30)$ | 1.0 |
| $\mathrm{O}(1)$ | $0.767(3)$ | 0.0000 | 0.00000 | $0.0116(58)$ | 1.0 |

${ }^{a}$ These atoms were constrained to have the same atomic coordinates and thermal parameters.

Table 5. Selected Bond Distances ( $\AA$ ) for $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}{ }^{a}$ and $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}{ }^{b}$

|  | $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ |  |  |  |
| :--- | :--- | :--- | :--- | :---: |
| $\mathrm{~Pb}(1)-\mathrm{O}(1)$ | $2.586(14) \times 2$ | $\mathrm{~Pb}(2)-\mathrm{Cl}(2)$ | $3.270(12) \times 2$ |  |
| $\mathrm{~Pb}(1)-\mathrm{O}(4)$ | $2.447(15) \times 2$ | $\mathrm{~Pb}(3)-\mathrm{O}(1)$ | $2.600(14) \times 2$ |  |
| $\mathrm{~Pb}(1)-\mathrm{Cl}(1)$ | $3.237(12)$ | $\mathrm{Pb}(3)-\mathrm{O}(4)$ | $2.544(16) \times 2$ |  |
| $\mathrm{~Pb}(1)-\mathrm{Cl}(2)$ | $3.176(12)$ | $\mathrm{Pb}(3)-\mathrm{O}(4)$ | $2.778(15) \times 2$ |  |
| $\mathrm{~Pb}(2)-\mathrm{O}(1)$ | $2.374(14) \times 2$ | $\mathrm{~Pb}(3)-\mathrm{O}(3)$ | $2.985(15) \times 2$ |  |
| $\mathrm{~Pb}(2)-\mathrm{O}(2)$ | $2.677(13) \times 2$ | $\mathrm{~Pb}(3)-\mathrm{Cl}(2)$ | $3.244(12)$ |  |
| $\mathrm{Pb}(2)-\mathrm{O}(3)$ | $2.48(2)$ |  |  |  |
| $\mathrm{Te}(1)-\mathrm{O}(1)$ | $1.938(13) \times 2$ | $\mathrm{Te}(2)-\mathrm{O}(3)$ | $1.84(2)$ |  |
| $\mathrm{Te}(1)-\mathrm{O}(2)$ | $2.044(12) \times 2$ | $\mathrm{Te}(2)-\mathrm{O}(4)$ | $1.861(16) \times 2$ |  |
|  | $\mathrm{~Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl} l_{2}$ |  |  |  |
| $\mathrm{~Pb}(1)-\mathrm{O}(1)$ | $2.452(12) \times 4$ | $\mathrm{~Pb}(1)-\mathrm{Cl}(1)$ | $3.305(12) \times 2$ |  |
| $\mathrm{~Pb}(1)-\mathrm{Cl}(1)$ | $3.271(12) \times 2$ | $\mathrm{~Pb}(2) / \mathrm{Te}(2)-\mathrm{O}(1)$ | $2.257(12) \times 4$ |  |

${ }^{a}$ Single-crystal data. ${ }^{b}$ Powder diffraction data.
Table 6. Powder XRD Data for $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Br}_{2}{ }^{a}$

| $h$ | $k$ | $l$ | $d_{\text {calc }}$ | $d_{\text {obs }}$ | $I_{\text {calc }}$ | $I_{\text {obs }}$ |
| ---: | :--- | :--- | :--- | :--- | ---: | ---: |
| 2 | 0 | 0 | 8.186 | 8.231 | 2 | 3 |
| -2 | 0 | 1 | 7.446 | 7.486 | 13 | 14 |
| 0 | 0 | 2 | 5.351 | 5.369 | 5 | 5 |
| -4 | 0 | 1 | 4.182 | 4.190 | 9 | 8 |
| 3 | 1 | 0 | 3.935 | 3.941 | 10 | 10 |
| -4 | 0 | 2 | 3.723 | 3.723 | 1 | 1 |
| 1 | 1 | 2 | 3.645 | 3.651 | 27 | 26 |
| -2 | 0 | 3 | 3.612 | 3.625 | 2 | 1 |
| 4 | 0 | 1 | 3.543 | 3.458 | 1 | 1 |
| -3 | 1 | 2 | 3.479 | 3.484 | 8 | 9 |
| -1 | 1 | 3 | 3.088 | 3.092 | 100 | 100 |
| 4 | 0 | 2 | 2.922 | 2.925 | 43 | 42 |
| -5 | 1 | 1 | 2.899 | 2.903 | 97 | 92 |
| 1 | 1 | 3 | 2.866 | 2.870 | 4 | 4 |
| 0 | 2 | 0 | 2.840 | 2.843 | 50 | 47 |
| -6 | 0 | 1 | 2.815 | 2.826 | 2 | 3 |
| -2 | 0 | 4 | 2.751 | 2.758 | 1 | 1 |
| 0 | 0 | 4 | 2.675 | 2.678 | 1 | 1 |
| 5 | 1 | 1 | 2.608 | 2.612 | 4 | 4 |
| -4 | 0 | 4 | 2.545 | 2.543 | 2 | 2 |
| 0 | 2 | 2 | 2.509 | 2.510 | 2 | 1 |
| -1 | 1 | 4 | 2.476 | 2.480 | 1 | 1 |
| 4 | 0 | 3 | 2.411 | 2.414 | 2 | 2 |
| -4 | 2 | 1 | 2.350 | 2.352 | 3 | 3 |
| 1 | 1 | 4 | 2.321 | 2.322 | 2 | 2 |
| -4 | 2 | 2 | 2.258 | 2.259 | 1 | 1 |

${ }^{a}$ Refined unit cell $a=16.8911(8) \AA, b=5.6804(2) \AA, c=11.0418(5)$ $\AA$, and $\beta=104.253(2)^{\circ}$ and space group $C 2 / m$ (No. 12).
crucibles and heated at a rate of $2{ }^{\circ} \mathrm{C} \mathrm{min}{ }^{-1}$ from room temperature to $900{ }^{\circ} \mathrm{C}$ in flowing nitrogen.

## Results and Discussion

The isostructural materials $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ and $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Br}_{2}$ have three-dimensional structural topologies consisting of lead-oxide halide polyhedra linked to tellurium-oxide groups (see Figures 1and 2). In both materials the $\mathrm{Pb}^{2+}$ cations are linked to both oxygen and chloride (bromide)

Table 7. Powder XRD Data for $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Br}_{2}{ }^{a}$

| h | k | 1 | $d_{\text {calc }}$ | $d_{\text {obs }}$ | $I_{\text {calc }}$ | $I_{\text {obs }}$ |
| :---: | :---: | :---: | :---: | :---: | ---: | ---: |
| 0 | 0 | 2 | 6.459 | 6.490 | 4 | 5 |
| 1 | 1 | 1 | 3.812 | 3.822 | 19 | 18 |
| 1 | 1 | 3 | 2.927 | 2.932 | 100 | 100 |
| 0 | 2 | 0 | 2.822 | 2.826 | 21 | 18 |
| 1 | 1 | 5 | 2.169 | 2.171 | 2 | 2 |
| 0 | 0 | 6 | 2.153 | 2.156 | 6 | 7 |
| 0 | 2 | 4 | 2.125 | 2.128 | 6 | 7 |
| 0 | 1 | 6 | 1.995 | 1.997 | 18 | 22 |
| 1 | 3 | 1 | 1.768 | 1.768 | 2 | 3 |
| 0 | 2 | 6 | 1.712 | 1.722 | 8 | 6 |
| 2 | 0 | 6 | 1.711 | 1.713 | 7 | 6 |
| 2 | 2 | 4 | 1.697 | 1.699 | 4 | 5 |
| 1 | 1 | 7 | 1.675 | 1.677 | 7 | 6 |
| 1 | 3 | 3 | 1.649 | 1.650 | 16 | 18 |
| 0 | 0 | 8 | 1.615 | 1.617 | 1 | 4 |
| 2 | 2 | 6 | 1.463 | 1.469 | 7 | 15 |
| 4 | 0 | 0 | 1.411 | 1.412 | 3 | 3 |
| 0 | 2 | 8 | 1.401 | 1.402 | 1 | 1 |

${ }^{a}$ Refined unit cell $a=5.6434(4) \AA, b=5.6434(5) \AA$, and $c=12.9172$ (6) $\AA$ and space group Bmmb (No. 63).


Figure 1. Ball-and-stick diagram of $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ in the $a-c$ plane.
forming $\mathrm{PbO}_{4} \mathrm{Cl}_{4}\left(\mathrm{Br}_{4}\right), \mathrm{PbO}_{5} \mathrm{Cl}_{3}\left(\mathrm{Br}_{3}\right)$, and $\mathrm{PbO}_{8} \mathrm{Cl}(\mathrm{Br})$ groups for $\mathrm{Pb}(1), \mathrm{Pb}(2)$, and $\mathrm{Pb}(3)$, respectively, whereas the $\mathrm{Te}^{4+}$ cations are only bonded to oxygen forming $\mathrm{TeO}_{4}$ and $\mathrm{TeO}_{3}$ groups for $\mathrm{Te}(1)$ and $\mathrm{Te}(2)$, respectively. Both $\mathrm{Pb}^{2+}$ and $\mathrm{Te}^{4+}$ are in highly asymmetric coordination environments attributable to their stereoactive lone pair. The lone pairs point into the layer, between the halide anions. The $\mathrm{Pb}-\mathrm{O}$ and $\mathrm{Pb}-\mathrm{Cl}$ bond distances range 2.447(15)-2.994(15) and 3.236(12)$3.270(12) \AA$, respectively, whereas the $\mathrm{Te}-\mathrm{O}$ bond distances range $1.84(2)-2.044(12) \AA$. The closest $\mathrm{Te}-\mathrm{Cl}$ contacts are at a distance of greater than $3.2 \AA$. Bond valence sums ${ }^{11,12}$ for $\mathrm{Pb}^{2+}$ range from 1.83 to 2.19 and for $\mathrm{Te}^{4+}$ are 3.89 and 4.18.

[^3]

Figure 2. Ball-and-stick diagram of $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ in the $b-c$ plane.


Figure 3. Ball-and-stick diagram of $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$ in the $b-c$ plane. Note the large, clear spheres are statistically disordered between $\mathrm{Pb}^{2+}$ and $\mathrm{Te}^{4+}$.
$\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$ and $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Br}_{2}$ also have three-dimensional structural topologies with lead-oxide halide polyhedra linked to tellurium oxide groups (see Figure 3). The materials are isostructural with orthorhombic $\mathrm{PbBiO}_{2} \mathrm{Cl}^{13}$ and $\mathrm{PbSbO}_{2} \mathrm{Cl}$ (Nadorite). ${ }^{14-16} \mathrm{The} \mathrm{Pb}-\mathrm{O}$ bond distances are $2.452(12) \AA$ with $\mathrm{Pb}-\mathrm{Cl}$ bond distances of $3.274(12)$ and $3.305(12) \AA$, whereas the $\mathrm{Te}-\mathrm{O}$ bond distances are 2.252(12) $\AA$. Similar

[^4]Table 8. Infrared Spectroscopy Data $\left(\mathrm{cm}^{-1}\right)$ for $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{X}_{2}$ and $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{X}_{2}(\mathrm{X}=\mathrm{Cl}$ or Br$)$

|  | $v(\mathrm{~Pb}-\mathrm{O})$ |  | $v(\mathrm{Te}-\mathrm{O})$ |  |  | $v(\mathrm{Te}-\mathrm{O}-\mathrm{Pb})$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: |
|  | 701 | 534 | 750 | 669 | 636 | 422 |
| $\mathrm{~Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ | 522 | 746 | 659 | 632 | 416 |  |
| $\mathrm{~Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Br}_{2}$ | 696 | 509 |  | 661 | 628 | 439 |
| $\mathrm{~Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$ |  | 599 |  | 653 | 613 | 441 |
| $\mathrm{~Pb}_{3} \mathrm{TeO}_{4} \mathrm{Br}_{2}$ |  | 499 |  |  |  |  |

to $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$, the closest $\mathrm{Te}-\mathrm{Cl}$ contacts are greater than $3.2 \AA$. The relationship between the stoichiometries of $\mathrm{PbBiO}_{2} \mathrm{Cl}$ and $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$ may be understood as follows. If the formula of $\mathrm{PbBiO}_{2} \mathrm{Cl}$ is "doubled", " $\mathrm{Pb}_{2} \mathrm{Bi}_{2} \mathrm{O}_{4} \mathrm{Cl}_{2}$ " is obtained. If the two $\mathrm{Bi}^{3+}$ cations are replaced by one $\mathrm{Te}^{4+}$ and one $\mathrm{Pb}^{2+}$ cation, " $\mathrm{Pb}_{2}(\mathrm{PbTe}) \mathrm{O}_{4} \mathrm{Cl}_{2} " \equiv \mathrm{~Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$ may be formulated. As stated earlier the $\mathrm{Te}^{4+}$ can be disordered over both $\mathrm{Pb}^{2+}$ sites, without any detriment to the refinement. However, this would result in larger, chemically unreasonable, $\mathrm{Te}-\mathrm{O}$ bond lengths (ca. $2.452(12) \AA$ ). Thus, we decided to statistically disorder $\mathrm{Pb}(2)$ and $\mathrm{Te}(1)$ resulting in four equal $\mathrm{Te}-\mathrm{O}$ bonds distances of 2.257 (12) $\AA$ (see Tables 4 and 5).

Infrared Spectroscopy. The infrared spectra for $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6}$ $\mathrm{Cl}_{2}, \mathrm{~Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Br}_{2}, \mathrm{~Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$, and $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Br}_{2}$ revealed a host of $\mathrm{Pb}-\mathrm{O}, \mathrm{Te}-\mathrm{O}$, and $\mathrm{Te}-\mathrm{O}-\mathrm{Pb}$ vibrations. Table 8 summarizes the infrared data. The additional $\mathrm{Pb}-\mathrm{O}$ and $\mathrm{Te}-\mathrm{O}$ stretches observed in $\mathrm{Pb}_{2} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{X}_{2}$ compared with $\mathrm{Pb}_{2-}$ $\mathrm{TeO}_{4} \mathrm{X}_{2}$ are attributable to the different coordination environments of the $\mathrm{Pb}^{2+}$ and $\mathrm{Te}^{4+}$ cations in the materials. In $\mathrm{Pb}_{2} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{X}_{2}$ the $\mathrm{Pb}^{2+}$ are in 8- and 9-fold coordination environments, whereas in $\mathrm{Pb}_{2} \mathrm{TeO}_{4} \mathrm{X}_{2}$ the $\mathrm{Pb}^{2+}$ is only in an 8 -fold environment. Similarly for $\mathrm{Te}^{4+}$, in $\mathrm{Pb}_{2} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{X}_{2}$ the $\mathrm{Te}^{4+}$ are in 3- and 4-fold environments, whereas in $\mathrm{Pb}_{2}-$ $\mathrm{TeO}_{4} \mathrm{X}_{2}$ the $\mathrm{Te}^{4+}$ is only in a 3-fold environment. All of the assignments are consistent with those previously reported. ${ }^{17-19}$

Thermogravimetric Measurements. For $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ $\left(\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Br}_{2}\right)$, the TGA measurements revealed one transition between 560 and $750^{\circ} \mathrm{C}$ corresponding to a weight loss of $15.57 \%$ ( $14.21 \%$ ). The weight loss is consistent with the following reaction:

$$
\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{X}_{2} \rightarrow \mathrm{~Pb}_{3} \mathrm{TeO}_{4} \mathrm{X}_{2}+\mathrm{TeO}_{2} \uparrow
$$

Here $\mathrm{X}=\mathrm{Cl}$ or Br . The calculated weight loss for this reaction for $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}\left(\mathrm{~Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Br}_{2}\right)$ is $15.29 \%$ (14.09\%). For $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}\left(\mathrm{~Pb}_{3} \mathrm{TeO}_{4} \mathrm{Br}_{2}\right)$, the TGA measurements also revealed one transition between 650 and $850{ }^{\circ} \mathrm{C}$ corresponding to a weight loss of $31.46 \%$ ( $37.62 \%$ ). The weight loss is consistent with the following reaction:

$$
\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{X}_{2} \rightarrow \mathrm{~Pb}_{2} \mathrm{TeO}_{4}+\mathrm{PbCl}_{2} \uparrow
$$

Here $\mathrm{X}=\mathrm{Cl}$ or Br . The calculated weight loss for this reaction for $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}\left(\mathrm{~Pb}_{3} \mathrm{TeO}_{4} \mathrm{Br}_{2}\right)$ is $31.81 \%$ (37.72\%).

One of the most interesting aspects of the reported materials is their interconvertibility. As demonstrated by the TGA measurements, when $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ is heated above 560

[^5]

Figure 4. Schematic structural representation of the interconversion between $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ and $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$. The circled atoms indicate where $\mathrm{TeO}_{2}$ may be lost, or added, during the transformation.
${ }^{\circ} \mathrm{C}$, the material loses $\mathrm{TeO}_{2}$ and forms $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$. Interestingly, when $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$ is combined with $\mathrm{TeO}_{2}$ and heated in an evacuated and sealed quartz tube at $600^{\circ} \mathrm{C}$ for 12 h , $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ is re-formed. The interconversion, which also occurs with the bromine analogues, indicates a great deal of stability in the lead(II)-tellurium(IV)-oxide "sheets". The aforementioned reaction is outlined below, and Figure 4 gives a schematic structural representation of the loss, or addition,
of $\mathrm{TeO}_{2}$ between $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ and $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$.

$$
\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2} \underset{+\mathrm{TeO}_{2}, 600^{\circ} \mathrm{C}}{\stackrel{-\mathrm{TeO}_{2}, 560^{\circ} \mathrm{C}}{\leftrightarrows}} \mathrm{~Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}
$$

Conclusion. We have reported the synthesis and characterization of four new mixed-metal oxide halide materials, $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{X}_{2}$ and $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{X}_{2}(\mathrm{X}=\mathrm{Cl}$ or Br$)$. All of the materials have three-dimensional structural topologies, with lead-oxide halide polyhedra linked to tellurium-oxide groups. The stereoactive lone pairs on the $\mathrm{Pb}^{2+}$ and $\mathrm{Te}^{4+}$ point into the layer, between the halide anions. Interestingly, the materials may be interconverted through the loss or addition of $\mathrm{TeO}_{2}$. Synthetically, we prepared all the reported materials by a combination of an oxide halide, $\mathrm{Pb}_{3} \mathrm{O}_{2} \mathrm{X}_{2}(\mathrm{X}$ $=\mathrm{Cl}$ or Br ), and an oxide, $\mathrm{TeO}_{2}$. This suggests that other three-dimensional materials are possible. We are pursuing this avenue of synthetic research and will be reporting on the new materials shortly.

Acknowledgment. We wish to thank the reviewers for perceptive criticism of an earlier version of the manuscript. We thank the Robert A. Welch Foundation for support. We wish to acknowledge the Center for Materials Chemistry at the University of Houston (CMC-UH) for support. This work was also supported by the NSF-Career Program through Grant DMR-0092054, and an acknowledgment is made to the donors of the Petroleum Research Fund, administered by the American Chemical Society, for partial support of this research. P.S.H. is a Beckman Young Investigator.

Supporting Information Available: Refined powder X-ray diffraction patterns for $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$ and $\mathrm{Pb}_{3} \mathrm{TeO}_{4} \mathrm{Cl}_{2}$ and a CIF for $\mathrm{Pb}_{3} \mathrm{Te}_{2} \mathrm{O}_{6} \mathrm{Cl}_{2}$. This material is available free of charge via the Internet at http://pubs.acs.org.

IC025750J


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    (1) Porter, Y.; Halasyamani, P. S. Inorg. Chem. 2001, 40, 2640.

[^1]:    (2) Vincent, H.; Perrault, G. Bull. Soc. Fr. Mineral. Cristallogr. 1971, 94, 323.
    (3) Berdonosov, P. S.; Dolgikh, V. A.; Popovkin, B. A. Mater. Res. Bull. 1996, 31, 717.
    (4) Pasero, M.; Vacchiano, D. Neues Jahrb. Mineral., Monatsh. 2000, 563-569.

[^2]:    (5) SAINT, version 4.05 ed.; Siemens Analytical X-ray Systems, Inc.: Madison, WI, 1995.
    (6) Sheldrick, G. M. SHELXS-97-A program for automatic solution of crystal structures; University of Goettingen: Goettingen, Germany, 1997.
    (7) Sheldrick, G. M. SHELXL-97-A program for crystal structure refinement; University of: Goettingen: Goettingen, Germany, 1997.
    (8) Farrugia, L. J. WinGX: An integrated system of publically available windows programs for the solution, refinement, and analysis of singlecrystal X-ray diffraction data. J. Appl. Crvstallogr. 1998, 32, 837.
    (9) Rietveld, H. M. J. Appl. Crystallogr. 1969, 2, 65.
    (10) Rodriguez, J. C. FULLPROF Program: Rietveld Pattern Matching Analysis of Powder Patterns; ILL: Grenoble, France, 1990.

[^3]:    (11) Brown, I. D.; Altermatt, D. Acta Crystallogr. 1985, B41, 244.
    (12) Brese, N. E.; O'Keeffe, M. Acta Crystallogr. 1991, B47, 192.

[^4]:    (13) Gillberg, M. Arkiv Mineral. Geol. 1961, 2, 565.
    (14) Sillen, L. G.; Melander, L. Z. Kristallogr. 1941, 103, 420.
    (15) Giuseppetti, G.; Tadini, C. Period. Mineral. 1973, 42, 335.
    (16) Porter, Y.; Halasyamani, P. S. Z. Naturforsch. 2002, 57b, 360.

[^5]:    (17) Brooker, M. H.; Irish, D. E. J. Chem. Phys. 1970, 53, 1083.
    (18) Bart, J. C. J.; Petrini, G. Z. Z. Anorg. Allg. Chem. 1984, 509, 183.
    (19) Nakamoto, K. Infrared and Raman Spectra of Inorganic and Coordination Compounds, 5th ed.; John Wiley \& Sons: 1997.

