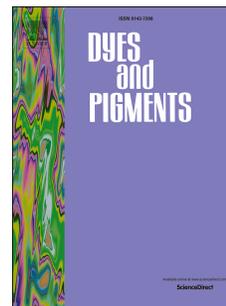


# Accepted Manuscript

Facile synthesis of a narrow-bandgap strong-donor-*alt*-strong-acceptor copolymer of poly(5,6-difluorobenzo-[c][1,2,5]-thiadiazole-*alt*-5*H*-dithieno[3,2-*b*:2',3'-*d*]pyran) via direct C-H arylation polymerization

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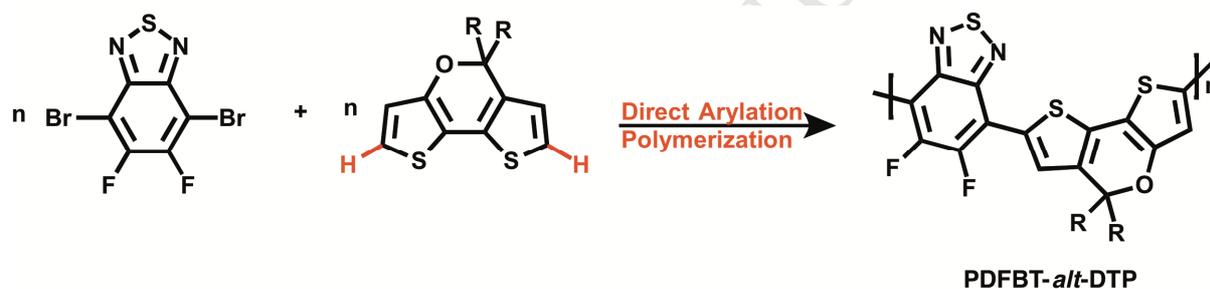
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## Graphical abstract

**Facile Synthesis of a Narrow-Bandgap Strong-Donor-*alt*-Strong-Acceptor Copolymer of Poly(5,6-Difluorobenzo-[c][1,2,5]-thiadiazole-*alt*-5*H*-Dithieno[3,2-*b*:2',3'-*d*]pyran) via Direct C-H Arylation Polymerization**Amsalu Efrem,<sup>a</sup> Kai Wang,<sup>a</sup> Mingfeng Wang,<sup>\*,a</sup><sup>a</sup> School of Chemical and Biomedical Engineering, Nanyang Technological University, 62 Nanyang Drive, Singapore 637459

1 **Facile Synthesis of a Narrow-Bandgap Strong-Donor-*alt*-Strong-**  
2 **Acceptor Copolymer of Poly(5,6-Difluorobenzo-[c][1,2,5]-**  
3 **thiadiazole-*alt*-5*H*-Dithieno[3,2-*b*:2',3'-*d*]pyran) *via* Direct C-H**  
4 **Arylation Polymerization**

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8  
9 **ABSTRACT**

10 A narrow bandgap D-A conjugated polymer based on 5*H*-dithieno[3,2-*b*:2',3'-*d*]pyran  
11 alternating with 5,6-difluoro-2,1,3-benzothiadiazole (denoted as **PDFBT-*alt*-DTP**) was  
12 synthesized *via* direct C-H arylation polymerization (DAP) as an atomically efficient protocol.  
13 The optimal reaction condition for the DAP gave the target polymers with moderate  
14 number-average molecular weights ( $M_n$ ) using optimized catalytic condition of Pd<sub>2</sub>(dba)<sub>3</sub>/(*o*-  
15 MeOPh)<sub>3</sub>P/PivOH/K<sub>2</sub>CO<sub>3</sub> in *o*-xylene. UV-vis-NIR absorption spectra of the obtained polymers  
16 show the presence of aggregation and interchain interaction in thin films. <sup>1</sup>H-NMR  
17 spectroscopic analysis indicates good C-H selectivity corresponding to an alternating strong-  
18 donor-*alt*-strong-acceptor conjugated backbone. The hole mobility of the resulting polymers  
19 in a magnitude of 10<sup>-4</sup> cm<sup>2</sup>V<sup>-1</sup>s<sup>-1</sup> was reached in bottom-gate top-contact field-effect  
20 transistors fabricated and tested under ambient conditions.

21

22 **Keywords:** benzothiadiazole, conjugated polymer, direct arylation polymerization, organic  
23 electronics.

## 24 1. Introduction

25 Conjugated organic semiconductors have interesting properties like solution processability  
26 which could enable cheap and flexible organic electronics devices such as organic field-effect  
27 transistors (OFETs)[1-6], organic photovoltaics (OPVs)[7-12], and organic light emitting  
28 diodes (OLEDs) to be readily obtained.[13, 14] Conventionally,  $\pi$ -conjugated polymers are  
29 synthesized by transition metal-catalyzed polycondensation reactions such as Suzuki–  
30 Miyaura and Migita–Kosugi–Stille couplings.[15-18] However, the great versatility and  
31 choice, these protocols require the preparation of high-purity organometallic reagents such  
32 as arylene diboronic acid and distannyl arylenes using flammable (*n*-BuLi) and toxic  
33 organotin precursors. Meanwhile, the polycondensations also produce a stoichiometric  
34 amount of toxic byproducts such as trialkyltin compound, which would become key issues in  
35 the future commercialization of conjugated polymers. Recently, direct C–H arylation  
36 polymerization (DAP) has attracted tremendous attention to synthesize  $\pi$ -conjugated  
37 polymers. DAP involves the coupling of an aryl halide monomer directly with another arene  
38 monomer without preactivation of the C–H bonds.<sup>[19-21]</sup> Therefore, it is an economically  
39 efficient and environmentally benign approach compared to the traditional coupling  
40 methods such as Stille and Suzuki couplings.[22-26]

41 Copolymerization of alternating electron donor and acceptor moieties has been proved to  
42 be a successful strategy to tailor the optoelectronic properties of conjugated  
43 copolymers.[27-31] Nowadays, a large library of donor and acceptor moieties are available  
44 for the synthesis of D–A conjugated polymers for organic electronics applications. For  
45 instance, some of the largely studied donor moieties include carbazole (Cz),  
46 benzodithiophene (BDT), dithienosilole (DTS), dithienogermole (DTG), and

47 cyclopentadithiophene (CPDT) units (Scheme 1). A structurally similar donor moiety, 5*H*-  
48 dithieno[3,2-*b*:2',3'-*d*]pyran (DTP) has an electron-donating oxygen atom in the pyran ring,  
49 which makes DTP a stronger electron donor moiety than the aforementioned moieties in  
50 Scheme 1. In addition to this, the inclusion of fused-ring structures in the conjugated  
51 backbone could enhance planarity and thus extended the conjugation length.[32-37]  
52 Therefore, copolymerizing DTP with a strong acceptor moiety like DFBT could give a low  
53 bandgap D-A conjugated polymer which possesses UV-vis absorption spectra extended to  
54 NIR region.

55 Recently, Dou et al. have reported the synthesis of a new low bandgap (1.38 eV) D-A  
56 conjugated polymer poly(5,6-difluoro-2,1,3-benzothiadiazole-*alt*-dithienopyran (denoted as  
57 **PDFBT-*alt*-DTP**) via Stille polymerization of 5*H*-dithieno[3,2-*b*:2',3'-*d*]pyran (**DTP**) as an  
58 electron donor and 5,6-difluorobenzo[*c*][1,2,5]thiadiazole (**DFBT**) as an electron acceptor  
59 moiety. **PDFBT-*alt*-DTP** as an electron donor mixed with PCBM as an electron acceptor in the  
60 bulk heterojunction (BHJ) configuration exhibited a high power conversion efficiency (PCE) of  
61 8%.[38] Nevertheless, the traditional C-C coupling technique of Migita–Kosugi–Stille  
62 couplings requires a tedious preactivation of C-H bonds to obtain reactive precursors  
63 (boronic acid or trialkyltin) using toxic (stannyl agents) and flammable (*n*-BuLi) agents,  
64 associated with the stoichiometric amount of toxic stannyl byproducts.[15-18] As a  
65 consequence, an environment-friendly, scalable reaction which retains high yield for mass  
66 production of these type of high performance conjugated polymers is required for the future  
67 large-scale production and commercialization of organic electronic devices (OFETs, OPVs).

68 Recently, we have employed direct arylation polymerization (DAP) to synthesize a series of  
69 linear D–A conjugated polymers as well as 2D/3D conjugated polymer networks and

70 investigated the factors including catalysts, solvents, ligands, bases, additives, the  
71 concentration of reactants, and phase transfer agents on the polymerization. Our synthetic  
72 optimization of the DAP reactions has led to a variety of high-quality polymers with minimal  
73 structural defects.[24, 25, 39-52] In contrast to Suzuki or Stille coupling, DAP avoids the  
74 tedious preactivation of C-H bonds in arenes as monomers using highly flammable (e.g. *n*-  
75 butyl lithium) and/or highly toxic stannyl agents that are often involved in Stille coupling,  
76 thus enabling facile synthesis of  $\pi$ -conjugated polymers in fewer synthetic steps, i.e. in a  
77 more atomically efficient way.[23-25, 39-42, 53-59]

78 In this article, we present the DAP synthesis of a narrow-bandgap D-A conjugated  
79 polymers, poly(5,6-difluoro-2,1,3-benzothiadiazole-*alt*-dithienopyran) (denoted as **PDFBT-**  
80 ***alt*-DTP**) with solubilizing alkyl chain attached on the donor moiety. Its synthetic  
81 optimization towards high molecular weight has been carried out via changing the reaction  
82 time, temperature, and concentration of monomers. The structure-property relationship of  
83 the polymers was characterized using nuclear magnetic resonance (NMR) spectroscopy, size-  
84 exclusion chromatography (SEC), thermogravimetric analysis (TGA), UV-vis-NIR  
85 spectroscopy, and cyclic voltammetry. The charge carrier properties of the synthesized  
86 polymers were characterized in OFET devices, which are fabricated in the air without any  
87 encapsulation and protective atmosphere.

## 88 **2. Experimental Section**

### 89 *2.1. Materials and Instrumentation.*

90 All chemicals and solvents were obtained from commercial sources and used without  
91 further purification unless otherwise stated. 4*H*-cyclopenta[1,2-*b*:5,4-*b'*]dithiophen-4-one  
92 were commercially obtained from SunaTech Inc. and 4,7-dibromo-5,6-

93 difluorobenzo[c][1,2,5]thiadiazole (**DBrDFBT**) were obtained from Derthon optoelectronics  
94 materials. Compound **2**, **3**, & **4** were synthesized according to reported procedures.[38]

95 <sup>1</sup>H NMR spectra were recorded on a Bruker AV 300 spectrometer with TMS as the internal  
96 reference. Molecular weights of the polymers were obtained from a Gel permeation  
97 chromatography (GPC) using CHCl<sub>3</sub> as eluent at 80 °C after calibration with standard  
98 polystyrene. UV–vis absorption spectra were recorded on a Cary 5000 spectrophotometer.  
99 Thermogravimetric analysis (TGA) was carried out on a Perkin-Elmer Diamond TGA/DTA, at a  
100 heating rate of 10 °C min<sup>-1</sup> from 25 to 800 °C under a nitrogen atmosphere. Cyclic  
101 voltammetry (CV) measurements were conducted on a three-electrode electrochemistry  
102 workstation (Model CHI006D) in 0.1 M anhydrous acetonitrile solution of  
103 tetrabutylammonium hexafluoro phosphate (Bu<sub>4</sub>NPF<sub>6</sub>) at 25 °C under a nitrogen atmosphere  
104 with ferrocene as an internal standard. A glassy carbon disk coated with a thin layer of the  
105 polymer film was used as working electrode, a silver wire as a pseudo-reference electrode  
106 and a platinum wire as counter electrode. Tapping-mode AFM images of pristine polymer  
107 films on OTS-modified Si/SiO<sub>2</sub> substrates at room temperature and annealed at 120 °C were  
108 taken by using asylum research MFP-3D AFM.

109 Bottom-gate/top-contact OFETs were fabricated using heavily doped Si wafer as the  
110 bottom gate electrode with 300 nm of the SiO<sub>2</sub> layer as the gate dielectric. Gold source and  
111 drain electrodes (60 nm) were patterned by the vacuum thermal evaporation. The substrate  
112 was cleaned in acetone, isopropanol, and dried on the hot plate at 110 °C then it was further  
113 cleaned with piranha solution (H<sub>2</sub>SO<sub>4</sub>: H<sub>2</sub>O<sub>2</sub>, 3:1 v/v), washed with deionized water and dried  
114 at 100 °C. After that, the substrate was immersed in the octadecyltrichlorosilane (ODTS)  
115 solution in toluene solvent overnight, rinsed with toluene to remove excess ODTS, and dried

116 on the hot plate at 110 °C. The polymers dissolved in dichlorobenzene at a concentration of  
117 5 mg mL<sup>-1</sup> by stirring for 1 day at 60 °C. The polymer films were spin-cast at 2000 rpm and  
118 annealed at 100 °C for 15 min. Finally, the gold electrode (60 nm) was thermally evaporated  
119 in vacuum through a shadow mask with two different dimension length (L) of 200 μm and  
120 channel width (W) of 4.0 mm. The field effect mobilities were extracted from the saturation  
121 regime using the equation:  $\mu_{\text{sat}} = (2 I_{\text{ds}} L) / (W C_i (V_g - V_{\text{th}})^2)$ , where  $I_{\text{ds}}$  denotes the saturation  
122 drain current,  $C_i$  is the capacitance ( $C_i = 10.8 \times 10^{-9}$  F cm<sup>-2</sup>) of SiO<sub>2</sub> dielectric,  $V_d$  is the gate  
123 bias, and  $V_{\text{th}}$  is the threshold voltage.

## 124 2.2. General synthesis procedure for PDFBT-**alt**-DTP polymer via direct arylation 125 polymerization (DAP)

126 A mixture of 5-(3,7-dimethyloctyl)-5-(3-methyloctyl)-5H-dithieno[3,2-b:2',3'-d]pyran  
127 (Compound **4**) (1 equiv.) and 4,7-dibromo-5,6-difluorobenzo[c][1,2,5]thiadiazole (**DBrDFBT**)  
128 (1 equiv.), palladium source, ligand, base, PivOH, and o-xylene was added in a reaction vial  
129 containing a magnetic stirring bar under nitrogen atmosphere. The vial sealed with an  
130 aluminum cap with PTFE/silicone septum and then heated in a preheated oil bath at 100 °C  
131 except for Entry 1, **E1-DTP** heated at 140 °C. After being cooled to room temperature, the  
132 reaction mixture was diluted with CHCl<sub>3</sub> and added dropwise to methanol (150 mL),  
133 precipitates were collected by filtration and then subjected to Soxhlet extraction with  
134 methanol, acetone, hexane, and chloroform sequentially each for 24 h. The chloroform  
135 and/or chlorobenzene fraction was concentrated and precipitated into methanol. The  
136 precipitates were collected by filtration and dried under vacuum before further  
137 characterization.

138 **Entry 1:** Pd<sub>2</sub>(dba)<sub>3</sub> (5 mol%), (*o*-MeOPh)<sub>3</sub>P (10 mol%), K<sub>2</sub>CO<sub>3</sub> (4 equiv.), PivOH (50 mol%),  
139 and the monomers (0.1 M in *o*-xylene) at 140 °C. Yield: 48% from CHCl<sub>3</sub> fraction.

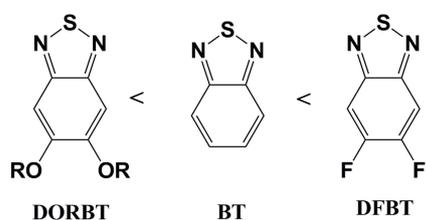
140 **Entry 2:** Pd<sub>2</sub>(dba)<sub>3</sub> (5 mol%), (*o*-MeOPh)<sub>3</sub>P (10 mol%), K<sub>2</sub>CO<sub>3</sub> (4 equiv.), PivOH (50 mol%),  
141 and the monomers (0.2 M in *o*-xylene) at 100 °C. Yield: 88% from CHCl<sub>3</sub> fraction.

142 **Entry 3:** Pd(OAc)<sub>2</sub> (5 mol%), (*o*-MeOPh)<sub>3</sub>P (10 mol%), K<sub>2</sub>CO<sub>3</sub> (4 equiv.), PivOH (50 mol%),  
143 and the monomers (0.2 M in *o*-xylene) at 100 °C. Yield: 80% from CHCl<sub>3</sub> fraction.

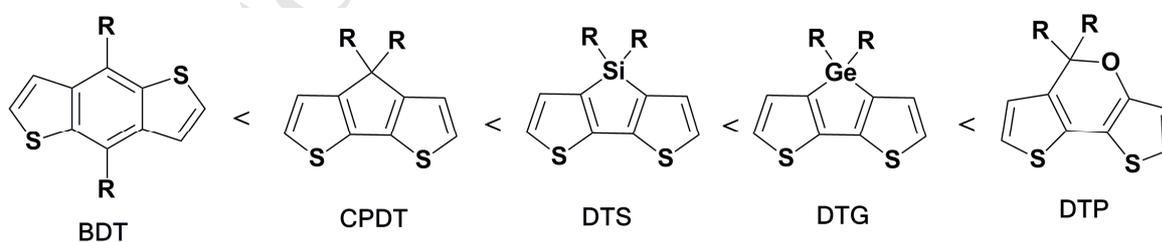
144 **Entry 4:** Herrmann's catalyst (5 mol%), (*o*-MeOPh)<sub>3</sub>P (10 mol %), K<sub>2</sub>CO<sub>3</sub> (4 equiv.), PivOH  
145 (50 mol%), and the monomers (0.2 M in *o*-xylene) at 100 °C. no polymerization.

146 **Entry 5:** Pd<sub>2</sub>(dba)<sub>3</sub> (5 mol%), (*o*-MeOPh)<sub>3</sub>P (10 mol%), K<sub>2</sub>CO<sub>3</sub> (4 equiv.), PivOH (50 mol%),  
147 and the monomers (0.3 M in *o*-xylene) at 100 °C. Yield: 83% from CHCl<sub>3</sub> fraction.

148 **Entry 6:** Pd<sub>2</sub>(dba)<sub>3</sub> (5 mol%), (*o*-MeOPh)<sub>3</sub>P (10 mol%), K<sub>2</sub>CO<sub>3</sub> (4 equiv.), PivOH (50 mol%),  
149 and the monomers (0.5 M in *o*-xylene) at 100 °C. Yield: 15% from CHCl<sub>3</sub> fraction.



150 **Increasing order of electron accepting strength** →

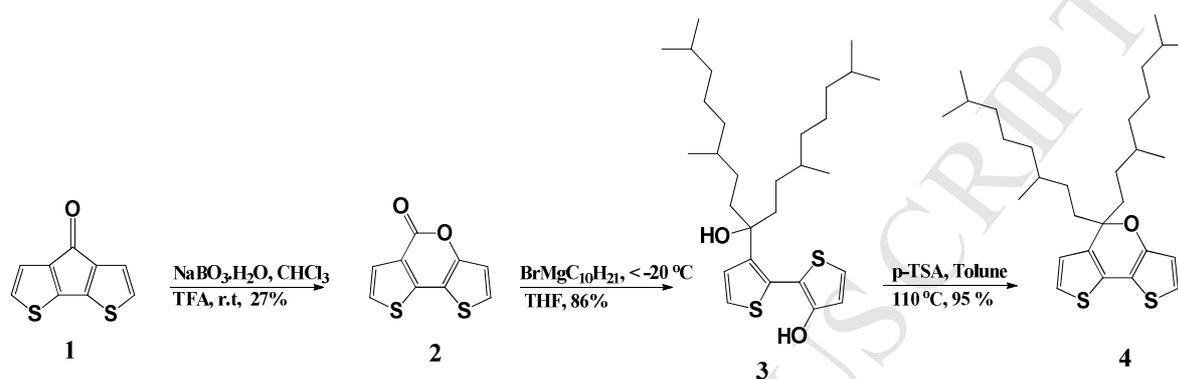


152 **Increasing order of electron donating strength** →

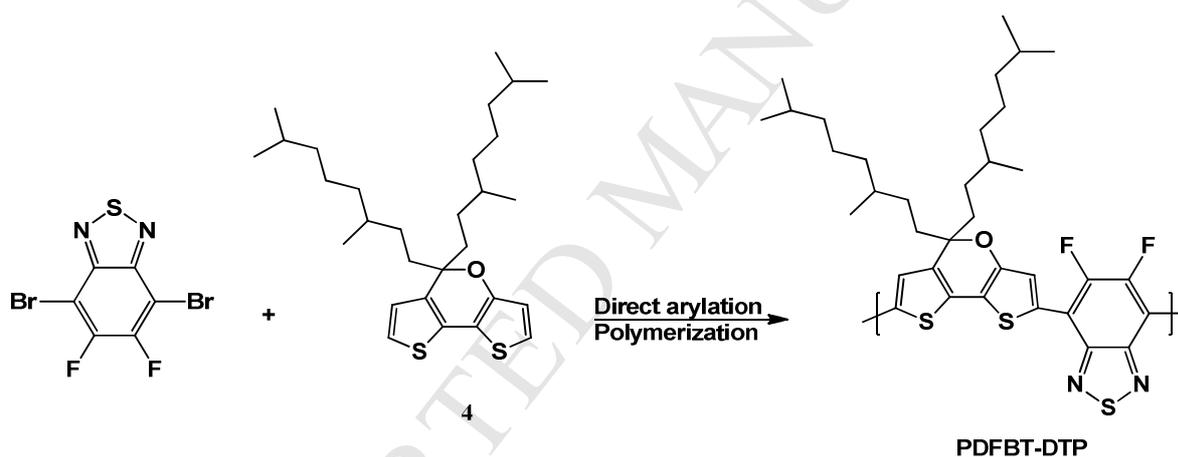
153

154 **Scheme 1.** Chemical structures of electron donor units (BDT, CPDT, DTS, DTG, and DTP) and  
 155 electron acceptor units (BT, DORBT, DFBT). (R = alkyl chains); arranged in increasing order of  
 156 electron donating and electron accepting strength.[60-62]

157



158



159

160 **Scheme 2.** Synthetic route to PDFBT- *alt*-DTP polymer *via* DAP.

161

### 162 3. Results and discussion

#### 163 3.1. Synthesis and structural characterization

164 The synthetic route to the donor moiety 5-(3,7-dimethyloctyl)-5-(3-methyloctyl)-5*H*-  
 165 dithieno[3,2-*b*:2',3'-*d*]pyran (Compound **4**, **DTP**) and its copolymerization with 4,7-dibromo-  
 166 5,6-difluorobenzo[*c*][1,2,5]thiadiazole (**DBrDFBT**) are shown in Scheme 2. The synthesis of  
 167 Compound **2** to **4** were conducted according to reported procedure.[38]

168 The optimization of the DAP reaction was carried out by changing conditions, such as  
169 palladium-catalyst, concentration of monomers, and reaction temperature (see Table 1) and  
170 the progress of reaction was monitored by the color change from dark red to dark cyan  
171 (which is the characteristics color of the **PDFBT-*alt*-DTP** polymer as reported in the  
172 literature).[38] First, Compound **4** and DBrDFBT (0.1 M) were reacted in the presence of  
173 Pd<sub>2</sub>(dba)<sub>3</sub> as the catalyst, (*o*-MeOPh)<sub>3</sub>P as the ligand, PivOH as the additive, K<sub>2</sub>CO<sub>3</sub> as the  
174 base in *o*-xylene at 140 °C for 24 h (**E1-DTP**), which gives **PDFBT-*alt*-DTP** polymer in a  
175 relatively low yield of 48% (*M<sub>n</sub>* of 6.8 kDa) from the chloroform fraction. The effort to further  
176 increase the yield and molecular weight was carried out by increasing the monomer  
177 concentration to 0.2 M and decreasing the reaction temperature to 100 °C (**E2-DTP**) to give  
178 **PDFBT-*alt*-DTP** polymer with an increased yield of 88% from chloroform fraction with  
179 comparable molecular weight (*M<sub>n</sub>* of 6.9 kDa) with **E1-DTP**. Further increase in concentration  
180 to 0.3 M (**E5-DTP**) provides a yield of 83% and *M<sub>n</sub>* of 6.6 kDa comparable to that of entry **E1-**  
181 **& E2-DTP**. Further increase of the monomer (**E6-DTP**) concentration to 0.5 M did not result  
182 in significant improvement of *M<sub>n</sub>* (7.1 kDa from chloroform fraction, Entry **E6-DTP**, Table 1).  
183 However, in Entry **E6-DTP** the yield in chloroform fraction has decreased dramatically to 15%  
184 with the presence of an insoluble residue which was later completely dissolved in  
185 chlorobenzene (*M<sub>n</sub>* of 11.4 kDa) with rather broad PDI (11.2), possibly attributed to side  
186 reactions, such as crosslinking or β-branching occurred in Compound **4**.

187

188

189

190

191 **Table 1.** DAP optimization entries for the synthesis of the **PDFBT-*alt*-DTP** polymers.

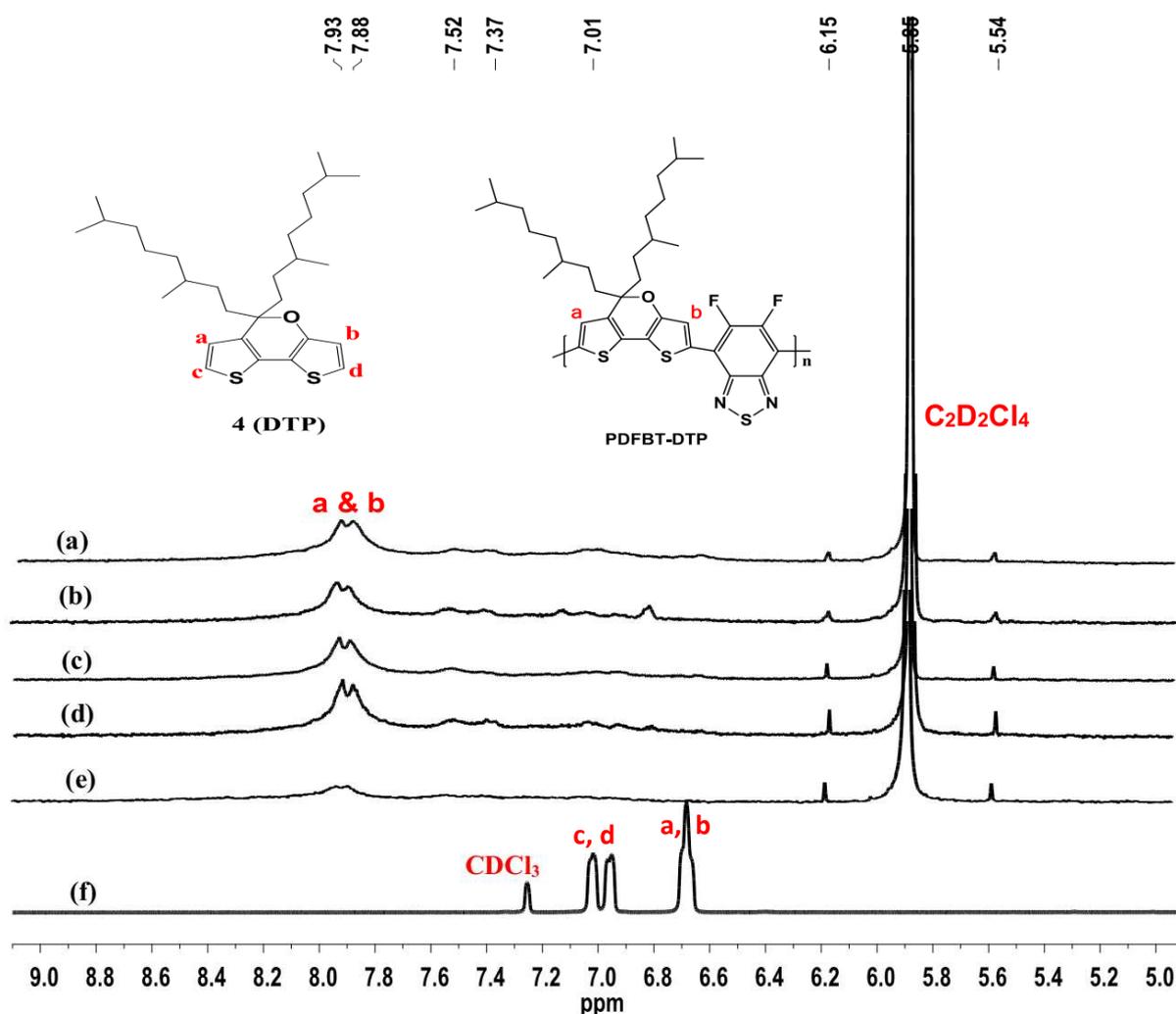
Entry no.	Catalyst	C <sub>monomer</sub> (M)	Temperature (°C)	Yield (%)	<sup>d</sup> M <sub>n</sub> (kDa)	M <sub>w</sub> /M <sub>n</sub>
<b>E1-DTP</b>	Pd <sub>2</sub> (dba) <sub>3</sub>	0.1	140	48	6.8	3.2
<b>E2-DTP</b>	Pd <sub>2</sub> (dba) <sub>3</sub>	0.2	100	88	6.9	8.5
<b>E3-DTP</b>	Pd(OAc) <sub>2</sub>	0.2	100	80	5.7	3.2
<b>E4-DTP</b>	Herrmann	-	100	<sup>c</sup> -	-	-
<b>E5-DTP</b>	Pd <sub>2</sub> (dba) <sub>3</sub>	0.3	100	83	6.6	4.5
<b>E6-DTP</b>	Pd <sub>2</sub> (dba) <sub>3</sub>	0.5	100	<sup>a</sup> 15	7.1	6.4
			100	<sup>b</sup> 72	11.4	11.2

192 <sup>a</sup>Yield of the polymer from chloroform fraction. <sup>b</sup>Yield of the polymer from chlorobenzene  
 193 fraction. <sup>c</sup>No precipitate in methanol. <sup>d</sup>Measured by GPC with polystyrene as standard and  
 194 CHCl<sub>3</sub> as eluent at 80 °C. For all entries; PivOH additive, P(*o*-MeOPh)<sub>3</sub> ligand, and K<sub>2</sub>CO<sub>3</sub> base  
 195 were used for a reaction time of 24 h.

196  
 197 Pd(OAc)<sub>2</sub> and Herrmann's catalyst were employed to gain more information about the  
 198 effect of a catalyst on the DAP reaction between Compound **4** and **DBrDFBT**. Thus, in **E3-**  
 199 **DTP**, Pd(OAc)<sub>2</sub> gave a polymer with a lower molecular weight (*M<sub>n</sub>* of 5.7 kDa), whereas  
 200 Herrmann's catalyst in **E4-DTP** gave no polycondensation product, where the color of the  
 201 reaction mixture after 24 h was dark purple and no precipitate was obtained from methanol.  
 202 Based on the aforementioned optimization results, the combination of Pd<sub>2</sub>(dba)<sub>3</sub>/*o*-  
 203 MeOPh)<sub>3</sub>P/PivOH/K<sub>2</sub>CO<sub>3</sub>/*o*-xylene is found to be the best DAP reaction condition for the  
 204 synthesis of high molecular weight **PDFBT-*alt*-DTP** polymer. These results indicate that the  
 205 *M<sub>n</sub>* of **PDFBT-*alt*-DTP** polymers obtained under the present DAP conditions remains lower  
 206 compared to that (*M<sub>n</sub>* = 28.5 kDa) of the same type of polymer synthesized through Stille  
 207 coupling polymerization.[38] One possible factor that contributes to such low-molecular-

208 weight polymers is the relatively low reactivity of the dibromo-DFBT monomer involved in  
 209 the DAP, which was also observed by our group in the DAP of dibromo-DFBT with  
 210 thienoisindigo-based monomers.[48]

211



212

213 **Figure 1.**  $^1\text{H-NMR}$  (300 MHz) spectra of the **PDFBT-*alt*-DTP** polymers synthesized *via* DAP, **E1-**  
 214 **DTP** (A), **E2-DTP** (B), **E3-DTP** (C), **E5-DTP** (D), and **E6-DTP-CB** (E) at 100 °C in  $\text{C}_2\text{D}_2\text{Cl}_4$ . **DTP**  
 215 monomer (**4**) (F) in  $\text{CDCl}_3$  at r.t.

216 To characterize the chemical structure of **PDFBT-*alt*-DTP** polymers synthesized *via* DAP,  
 217 high temperature (100 °C)  $^1\text{H-NMR}$  spectra in  $\text{C}_2\text{D}_2\text{Cl}_4$  as the solvent were collected (see

218 Figure 1). The extra little peaks found in the  $^1\text{H-NMR}$  spectra can be related to different end  
219 groups from the monomers. For instance, as shown in Figure 1, there are tiny peaks around  
220 7.52, 7.38 and 6.99 ppm, which could correspond to end groups from Compound **4** in the  
221 polymer. The selectivity of ( $\alpha$ )C-H over ( $\beta$ )C-H protons in Compound **4** is good as indicated  
222 by the presence of ( $\beta$ )C-H protons around 7.93 and 7.88 ppm in the  $^1\text{H-NMR}$  spectra of all  
223 entries of the DAP polymers. As described by Dou et al.,[38] it was difficult to get highly pure  
224 stannylated monomer of Compound **4** in the synthesis of **PDFBT-*alt*-DTP** by Stille  
225 polymerization and thus affected the quality of **PDFBT-*alt*-DTP** polymer and its device  
226 performance unless it was end-capped with bromobenzene. Therefore, employing direct  
227 arylation polymerization for the synthesis **PDFBT-*alt*-DTP** polymer might avoid stannylation-  
228 related problems.

### 229 3.2. Thermal Properties

230 The thermal stability of **PDFBT-*alt*-DTP** polymer was investigated using thermogravimetric  
231 analysis (TGA) with a heating rate of  $10\text{ }^\circ\text{C min}^{-1}$  under  $\text{N}_2$  atmosphere. The decomposition  
232 temperature ( $T_d$ , corresponding to 5% weight loss) is located at  $335\text{ }^\circ\text{C}$  for the **PDFBT-*alt*-DTP**  
233 polymer with the highest  $M_n$  obtained by DAP (**E2-DTP**) (see Figure S1), demonstrating its  
234 sufficient high thermal stability for the application in organic electronic devices.

### 235 3.3. Optical properties

236 The UV-VIS-NIR absorption spectra for **PDFBT-*alt*-DTP** polymers were acquired from  
237 solutions in chloroform and in chlorobenzene (for **E6-DTP** of chlorobenzene fraction) and  
238 spin-cast thin films prepared from chlorobenzene solution, respectively. The spectra are  
239 shown in Figure 2 and the relevant data are summarized in Table 2. Broad absorption bands  
240 across the visible region were observed for both solution and thin films. The absorption

241 maxima for **PDFBT-*alt*-DTP** polymers in solution (see Figure 2a) shows a red shift and an  
242 intense shoulder peak as the molecular weight increased. **E2-DTP** has  $\lambda_{\text{max}}$  ( $\lambda_{\text{shoulder}}$ ) of 711  
243 (776) nm which is red-shifted ca.10-15 (14) nm as compared to the polymers from the other  
244 entries. **E6-DTP** from CB fraction, despite its higher average molecular weight than other  
245 batches (e.g. **E1-5**), shows a blue shift of ca. 60 nm in  $\lambda_{\text{max,abs}}$ , which might be due to the  
246 presence of crosslinking or  $\beta$ -branching in the polymer backbone.

247 The thin film of **PDFBT-*alt*-DTP** polymers prepared by spin casting at 1500 rpm on a quartz  
248 substrate from their respective chlorobenzene solutions showed red-shifted absorbance  
249 (ca.10 nm for **E2-DTP**) as compared with its absorption spectra in chloroform. A shoulder  
250 peak at 780 nm with much increased intensity was observed in the thin-film spectra of **E2-**  
251 **DTP** as compared with its solution state, which indicates strong intermolecular interaction in  
252 solid thin films. However, the intensity of the peak at 780 nm relative to that at 719 nm has  
253 weaker intensity in **E2-DTP** polymer compared to the similar polymer synthesized by Stille  
254 polymerization,[38] which is probably due to the lower  $M_n$  of the polymer synthesized via  
255 DAP. Estimation from the onset absorption wavelength of their thin film spectra gave an  
256 optical bandgap ( $E_g^{\text{opt}}$ ) of 1.41 eV for the **E2-DTP** polymer, which is almost equal to that of  
257 the same type of polymer synthesized by Stille polymerization as previously reported (see  
258 Table 2).[38] Such a small bandgap is attributed to the intramolecular charge transfer  
259 between the strongly electron-donating DTP unit and strongly electron-withdrawing DFBT  
260 unit.[28-31, 63]

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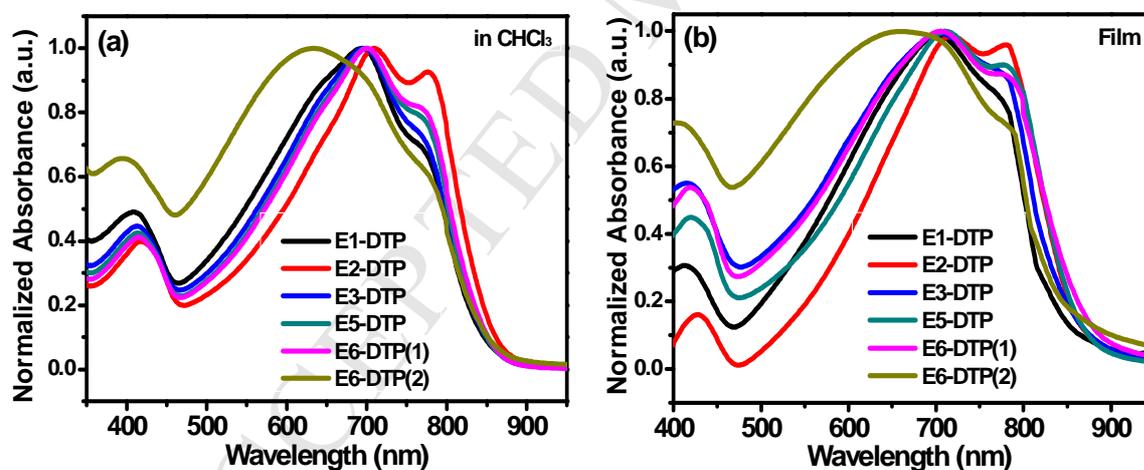
263

264 **Table 2.** Optical data of PDFBT-*alt*-DTP polymers.

Entry no.	In CHCl <sub>3</sub>		Film		
	$\lambda_{\text{max.abs}}(\lambda_{\text{shoulder.abs}})$ (nm)	$\lambda_{\text{onset.abs}}$ (nm)	$\lambda_{\text{max.abs}}(\lambda_{\text{shoulder.abs}})$ (nm)	$\lambda_{\text{onset.abs}}$ (nm)	$E_g^{\text{opt}}$ (eV)
E1-DTP	694 (767)	860	704 (770)	868	1.43
E2-DTP	711 (776)	874	719 (780)	878	1.41
E3-DTP	694 (767)	860	712 (776)	868	1.43
E5-DTP	698 (767)	860	712 (780)	876	1.42
E6-DTP (1) <sup>c</sup>	698 (767)	860	712 (780)	885	1.40
E6-DTP (2) <sup>d</sup>	638 (767) <sup>e</sup>	860 <sup>e</sup>	664 (786)	873	1.42

265 <sup>a</sup>Film at room temperature. <sup>b</sup> $E_g^{\text{opt}}(\text{eV})=1240/\lambda_{\text{onset}}[\text{nm}]$ . <sup>c</sup>E6-DTP (1) is CHCl<sub>3</sub> fraction and <sup>d</sup>E6-  
 266 DTP (2) is chlorobenzene fraction. <sup>e</sup>From chlorobenzene solution.

267



268

269 **Figure 2.** UV-vis-NIR absorption spectra of PDFBT-*alt*-DTP polymers (a) in CHCl<sub>3</sub>, except E6-  
 270 DTP (2) in chlorobenzene; and (b) in the thin film prepared by spin casting from their  
 271 chlorobenzene solutions. E6-DTP (1) is CHCl<sub>3</sub> fraction and E6-DTP (2) is chlorobenzene  
 272 fraction.

273

274

## 275 3.4. Electrochemical properties

276 The electrochemical properties of **PDFBT-*alt*-DTP** polymers were investigated by cyclic  
277 voltammetry (CV) using ferrocene (Fc/Fc<sup>+</sup>) as the standard (see Figure 3 and S2). It shows  
278 that the electrochemical oxidation and reduction process of **PDFBT-*alt*-DTP** polymers is  
279 reversible. Table 3 summarizes the HOMO/ LUMO energy levels and the bandgap of **PDFBT-**  
280 ***alt*-DTP** polymers. The oxidation was observed with an onset potential ( $E_{ox}$ ) at 0.83 to 1.05 V  
281 vs. Ag/Ag<sup>+</sup> and the onset reduction potentials ( $E_{red}$ ) were observed at -0.77 to -0.96 V vs.  
282 Ag/Ag<sup>+</sup>. Thus, the HOMO<sup>ele</sup>/LUMO<sup>ele</sup> energy levels were calculated by the following  
283 equation: HOMO<sup>ele</sup> =  $-(4.8 - E_{1/2, Fc/Fc^+} + E_{ox})$  eV and LUMO<sup>ele</sup> =  $-(4.8 - E_{1/2, Fc/Fc^+} + E_{red})$  eV,  
284 where  $E_{1/2, Fc/Fc^+}$  = 0.5 V. Therefore, HOMO<sup>ele</sup> of ca.-5.2 eV and LUMO<sup>ele</sup> of ca.-3.4 to -3.5 eV  
285 were calculated for all polymers from their corresponding  $E_{ox}$  and  $E_{red}$ , respectively. The  
286 LUMO<sup>opt</sup> calculated using  $LUMO^{opt} = HOMO^{ele} - E_g^{opt}$  is -3.77 eV for **E2-DTP**. The  
287 electrochemical bandgap ( $E_g^{ele}$ ) of **PDFBT-*alt*-DTP** polymers is in the range of 1.71 to 1.97 eV  
288 which is slightly larger than that of the optical bandgap ( $E_g^{opt}$ ) (1.40 to 1.43 eV). The  
289 discrepancy of  $E_g$  values lies within the range of experimental errors.[64] The HOMO level  
290 (5.18 eV) of **E2-DTP** polymer with a higher  $M_n$  is comparable with the HOMO level (5.2 eV) of  
291 Stille polymer as reported by Dou et al.[38] In addition to this, the electrochemical bandgap  
292 (1.76 eV) of the **E2-DTP** polymer is slightly higher than that of Stille polymer (1.62 eV). This  
293 difference could be explained based on  $M_n$ , where the Stille polymer has higher  $M_n$  of 28  
294 kDa, thus presumably a higher average conjugation length than that of **E2-DTP** polymer ( $M_n$  =  
295 6.9 kDa).

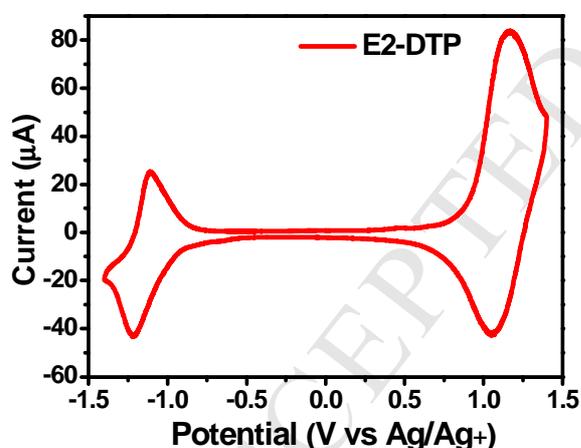
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297

298 **Table 3.** Electrochemical properties of PDFBT-*alt*-DTP polymers.

Entry no.	$E_{ox}$ (V)	$E_{red}$ (V)	$^a HOMO^{ele}$ (eV)	$^a LUMO^{ele}$ (eV)	$^b LUMO^{opt}$ (eV)	$^c E_g^{ele}$ <sup>299</sup> (eV) 300
E1-DTP	0.94	-0.77	-5.2	-3.5	-3.8	1.7
E2-DTP	0.88	-0.88	-5.2	-3.4	-3.8	1.8
E3-DTP	0.85	-0.91	-5.2	-3.4	-3.7	1.8
E5-DTP	0.83	-0.93	-5.1	-3.4	-3.7	1.7
E6-DTP (1) <sup>d</sup>	1.01	-0.96	-5.3	-3.3	-3.9	2.0
E6-DTP (2) <sup>e</sup>	1.05	-0.87	-5.4	-3.4	-3.9	2.0

305  $^a HOMO^{ele} = -(4.8 - E_{1/2, Fc, Fc+} + E_{ox, onset})$ ;  $LUMO^{ele} = -(4.8 - E_{1/2, Fc, Fc+} + E_{red, onset})$ .  $^b LUMO^{opt} =$   
 306  $E_g^{opt} + HOMO^{ele}$ .  $^c E_g^{ele} = LUMO^{ele} - HOMO^{ele}$ . <sup>d</sup>E6-DTP (1) is CHCl<sub>3</sub> fraction and <sup>e</sup>E6-DTP (2) is  
 307 chlorobenzene fraction.



308  
 309 **Figure 3.** Cyclic voltammetry curve of E2-DTP polymer from CHCl<sub>3</sub> fraction.

### 310 3.5. Field-effect transistor performance

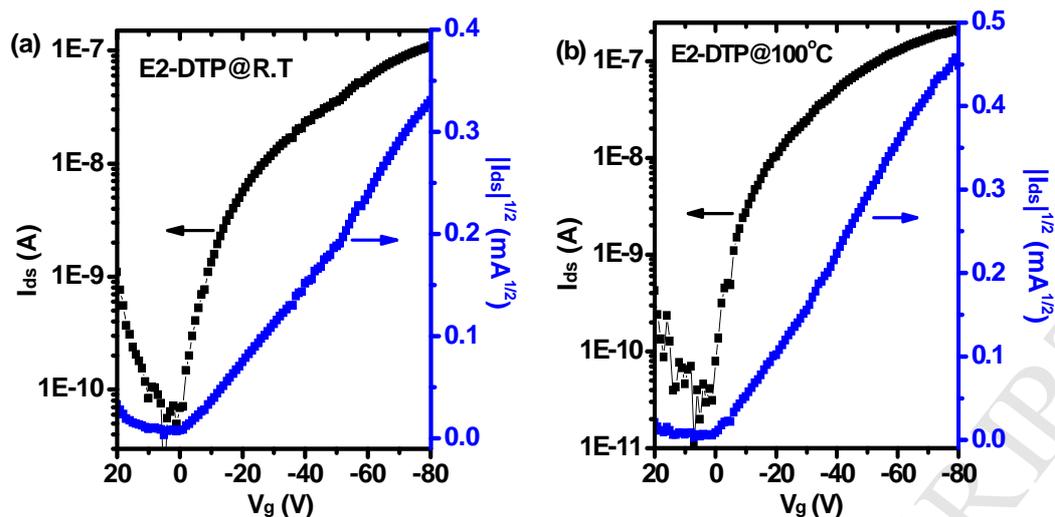
311 The charge-transport properties of PDFBT-*alt*-DTP polymers were studied by fabricating  
 312 bottom-gate, top-contact OTFT devices on OTS-18 modified Si/SiO<sub>2</sub> substrate. The  
 313 fabrication and measurement were done in the air without any protective atmosphere. The

314 device performances for selected polymer (**E2-DTP**) are summarized in Table 4 and their  
 315 transfer and output curves are shown in Figure 4 and S3, respectively. The hole mobility ( $\mu_h$ )  
 316 of the as-prepared OFET devices of **E2-DTP** polymer, which is derived from the slope of  
 317 transfer curve is  $1.0 \times 10^{-4} \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$ . After thermal annealing at 100 °C, a slightly increased  
 318  $\mu_h$  to  $1.7 \times 10^{-4} \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$  was obtained. The morphological studies of **E2-DTP** polymer thin  
 319 films investigated by AFM (see Figure S4) from its spin cast thin film on OTS-18 treated  
 320 Si/SiO<sub>2</sub> substrate at room temperature displayed less number of aggregates. However, after  
 321 annealing at 100 °C, the number of fine spherical aggregates has increased, which may affect  
 322 the  $\pi$ - $\pi$  stacking among the polymer chains. Consequently, a slight improvement in the hole  
 323 mobility of the **E2-DTP** polymer has observed.

324 **Table 4.** Performance of OFET devices for **PDFBT-*alt*-DTP** polymer (**E2-DTP**) semiconductor  
 325 on an OTS-18-treated substrate fabricated and measured under ambient conditions.

Entry no.	$M_n$ (kDa)	Annealing Temp. (°C)	$\mu_h$ ( $\text{cm}^2\text{V}^{-1}\text{s}^{-1}$ )	$V_{th}$ (V)	On/off Ratio
<b>E2-DTP</b>	6.9	R.T.	$1.0 \times 10^{-4}$	7.6	$10^3$ - $10^4$
		100	$1.7 \times 10^{-4}$	6.3	$10^3$ - $10^4$

326



327

328 **Figure 4.** Transfer curves of **E2-DTP** polymer (a) at r.t and (b) after annealing at 100 °C.

#### 329 4. Conclusion

330 In summary, direct arylation polymerization (DAP) has been used in the synthesis of low  
 331 bandgap 5*H*-dithieno[3,2-*b*:2',3'-*d*]pyran (**DTP**) based polymer, **PDFBT-*alt*-DTP**. A series of  
 332 DAP optimizations have led to an optimal catalytic condition, i.e. Pd<sub>2</sub>(dba)<sub>3</sub>/(*o*-  
 333 MeOPh)<sub>3</sub>P/PivOH/K<sub>2</sub>CO<sub>3</sub>/*o*-xylene (0.2 M), which resulted in the target polymers with *M<sub>n</sub>* up  
 334 to 6.9 kDa. The obtained **PDFBT-*alt*-DTP** polymers show a strong shoulder peak in the red  
 335 edge of their optical absorption spectra, which corresponds to the presence of strong π-π  
 336 interaction in solid films. Charge transport behavior of **PDFBT-*alt*-DTP** polymers shows a hole  
 337 mobility of  $1.7 \times 10^{-4} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ , evaluated in bottom-gate top-contact OFET devices  
 338 fabricated and measured under ambient conditions. These results indicate that further  
 339 optimization of DAP reaction conditions such as catalyst type, reaction time and  
 340 temperature, ligand type could lead to the higher molecular weight of the **PDFBT-*alt*-DTP**  
 341 polymer and consequently higher charge carrier mobility in its OFET device. Therefore, this  
 342 work further demonstrates the potential of DAP for the facile and efficient synthesis of a  
 343 variety of high-performance semiconducting polymers for optoelectronic applications.

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**Highlights**

- Direct arylation polymerization enables facile synthesis of a narrow-bandgap polymer.
- Polymers with alternating strong-donor/strong-acceptors are obtained.
- The charge transport properties are first examined in field effect transistors.