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Novel photocatalytic activity of Cu@V co-doped TiO₂/PU for CO₂ reduction with H₂O vapor to produce solar fuels under visible light

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ABSTRACT

In this study, Cu and V co-doped TiO₂ deposited on polyurethane (Cu@V-TiO₂/PU) was synthesized as a catalyst for the reduction of CO₂ with H₂O vapor to preferentially produce CH₄ as a valuable solar fuel under visible light. The Cu and V dopants defected into the TiO₂ lattice, leading to the formation of Ti³⁺ and oxygen vacancies in the lattice. The Ti³⁺ formed in the doped TiO₂ lattice created an intermediate band between the valence band and the conduction band of TiO₂, leading to an increase in the electronhole pair separation efficiency of TiO₂. The oxygen vacancies existing on the surface of the photocatalyst could induce new adsorption sites to adsorb CO₂. The generated electrons and holes reacted with the adsorbed CO₂ and with H₂O vapor to produce CO and primarily CH₄. Therefore, the Cu@V-TiO₂/PU photocatalysts successfully utilized visible light as the energy source and H₂O vapor as a reductant to reduce CO₂ to CO and CH₄. The Cu@V-TiO₂/PU photocatalysts also supplied sufficient electrons and holes for the selective reduction of CO₂ to CH₄ rather than CO. The 2Cu@4V-TiO₂/PU photocatalyst, with Cu/TiO₂ and V/TiO₂ ratios of 2 and 4 wt.%, respectively, exhibited the highest photocatalytic activity for CO₂ conversion into solar fuels. The production rates of CH₄ and CO produced from the CO₂ reduction by the 2Cu@4V-TiO₂/PU photocatalyst under visible light were 933 and 588 µmol g⁻¹ cat. h⁻¹, respectively.

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1. Introduction

Carbon dioxide (CO_2) emission has become a major cause of the greenhouse effect, leading to global warming and climate change [1,2]. Atmospheric concentrations of CO_2 have greatly risen since the Industrial Revolution and the current rate of ambient concentration increase is alarming. Therefore, many studies have been conducted to reduce CO_2 concentrations [1–5]. The utilization of solar energy for the conversion of CO₂ into valuable fuels is one promising technology, which not only mitigates the increasing level of CO₂ in the environment but also provides alternative fuels simultaneously to overcome the energy crisis [6]. Since Inoue et al.'s pioneering work demonstrating the photocatalytic reduction of CO₂ to organic compounds, research on the photocatalytic conversion of CO_2 has received increasing attention [2,7–10]. The photocatalytic reduction utilizes ultraviolet (UV) and/or visible light as the excitation source for photocatalysis to generate electron-hole pairs, which can reduce CO₂ with H₂O adsorbed onto the photocatalyst surface to form valuable products such as carbon monoxide (CO), methane (CH₄), methanol (CH₃OH), formaldehyde (HCHO), and formic acid (HCOOH), depending upon the available electrons [4]. Among the various photocatalysts, titanium dioxide (TiO₂), which is a cheap, nontoxic, and abundant semiconductor and resistant to photocorrosion, has been most widely used. However, the major challenges of the TiO₂ photocatalyst are its wide band gap (\sim 3.2 eV for anatase and 3.0 eV for rutile) and fast recombination of photogenerated electrons and holes, resulting in low efficiency of photocatalysis and limited utilization of solar energy [8]. The photocatalytic conversion of CO₂ by the TiO₂ photocatalyst is also limited because of its poor adsorption capability for CO₂ [11].

For improving the photocatalytic activity of TiO_2 by enhancing the visible response and reducing the recombination rate of the photogenerated electrons and holes, many strategies have been proposed, including the loading of metals, metal oxides, and doping nonmetal ions [2,8,11–16]. Xiong et al. reported that modification of TiO_2 with metal elements reduced the recombination of the photogenerated electron-hole pairs and extended the spectral response from UV to the visible area, simultaneously [17]. In our previous studies, Cu and V were individually used as metal elements to enhance the photocatalytic activity of TiO_2 for air purification purposes [14,15,18,19]. The obtained results indicated that these metals acted as dopants (incorporated into the TiO_2 lattice) and supporters (existing in the form of metals or metal oxides







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distributed on the surface of the TiO₂ layer) to enhance the photocatalytic activity of the TiO₂ photocatalyst. Thus, the synthesized Cu and V single-doped TiO₂ exhibited very high photocatalytic activity, even under visible light irradiation. More recently, numerous studies have doped TiO₂ with two different dopants (co-doping) to synergistically improve the photocatalytic activity of the photocatalyst [12,13,20-22]. Therefore, in this study, we combine Cu and V as co-dopants to synergistically enhance the photocatalytic activity of TiO₂ for CO₂ conversion to produce valuable fuels under visible light. In addition, the co-doping is expected to increase defects or disorders in the TiO₂ lattice. Paulino et al. reported that the disorders in the TiO₂ lattice influenced CO₂ adsorption, activation, and dissociation processes [2]. Rodriguez et al. also reported that the presence of defects on the TiO₂ surface induced the formation of new adsorption sites to capture CO₂, after which the electrons stored in the oxygen vacancies were spontaneously transferred to CO₂, which in turn was finally reduced [23]. Therefore, the co-doping is also expected to increase disorder in the TiO₂ lattice, leading to an increase in CO₂ adsorption or photocatalytic reduction of CO₂. In our previous studies, we investigated the use of porous polyurethane (PU), a honeycomb structure material, as a substrate to immobilize the Cu-doped TiO₂ and V-doped TiO₂ photocatalysts [14,15,19]. This synthesized structure not only overcame the disadvantages faced by the powder photocatalysts but also increased the adsorption ability of the photocatalysts. Therefore, in this study, we again used PU as the substrate to immobilize Cu@V co-doped TiO2 (Cu@V-TiO2/PU) in order to enhance the surface area and CO₂ adsorption capacity and thereby attain high photocatalytic activity and CO₂ adsorption capacity to overcome all the drawbacks of CO₂ conversion by photocatalysis.

2. Experimental

2.1. Photocatalyst preparation

We used a mixture of toluene, toluene-2,4-diisocyanate, and anhydrous triethylamine to activate pristine PU to introduce isocyanate groups onto its surface [14,15,19,24]. Titanium tetraisopropoxide and γ -aminopropyltriethoxysilane were used as precursors for synthesis of amino titanosiloxane, which contains amine groups (NH₂). The isocyanated PU was immersed in the synthesized amino titanosiloxane solution to promote the reaction between the isocyanate groups of the isocyanated PU and the NH₂ groups of the amino titanosiloxane to form urea bonds, which fixed the titanosiloxane on the PU surface. To synthesize Cu@V codoped TiO₂/PU, a mixture of 0.1 M Cu(NO₃)₂ and 0.1 M NH₄VO₃ solution was added to the titanosiloxane fixed on PU. The obtained material was cleaned by a 1 M oxalic acid solution. Finally, the Cu@V-TiO₂/PU was irradiated and calcined by UV-C light (60 W) for 5 h under nitrogen at 200 °C. Seven different Cu@V-TiO₂/PU materials were synthesized by adjusting the added volumes of $Cu(NO_3)_2$ and NH_4VO_3 solutions. The TiO₂/PU ratio in the synthesized materials was approximately 20 wt.%. The total weight ratios of both dopants to TiO₂ (total Cu/TiO₂ and V/TiO₂ ratios) were fixed at 6 wt.%, and the weight ratio of each individual Cu and V dopant was varied between 0 and 6 wt.% at intervals of 1 wt.%. The synthesized materials were labeled xCu@yV-TiO₂/PU, where x and y indicate the weight ratios of Cu/TiO₂ and V/TiO₂, respectively.

2.2. Photocatalyst characterization

The synthesized Cu@V-TiO₂/PU photocatalysts were analyzed by a Thermo Fisher K-Alpha X-ray Photoelectron Spectrometer (XPS) system. The obtained XPS spectra were fitted by Gaussian multipeak shapes to characterize the elemental states of copper, vanadium, and titanium in the synthesized photocatalysts. A Hitachi S4700 scanning electron microscope (SEM) was used to observe the surface morphology of the Cu@V-TiO₂/PU, which was coated with Pt before SEM analysis to increase the conductivity of the photocatalyst surface. A JEOL TEM-2010F system was used to obtain transmission electron microscopy (TEM) and highresolution TEM (HR-TEM) images of the synthesized Cu@V-TiO₂/PU. The optical absorption ability of the shredded Cu@V-TiO₂/PU photocatalyst was characterized by an Evolution 300 spectrophotometer (UV-1700 Shimadzu). X-ray diffraction (XRD) spectra of the Cu@V-TiO₂/PU were obtained using a Bruker AXN model equipped with a Cu K α radiation (λ = 1.5418 Å) source and operated at a scan rate of 0.02° s⁻¹ over the 2 θ range 10–80°.

2.3. Conversion experiments

Photocatalytic reduction of CO₂ with H₂O vapor was conducted by a continuous system comprising a gas generator, reaction chamber, and analyzer. The gas generator consisted of a cylinder of CO₂ gas (99.99%), flow rate meters, and a humidifier. The reaction chamber was a dark cover cask ($50 \times 25 \times 50$ cm) containing two white light bulbs (EFTR 20EX-D, Kumho Co., Ltd.) and a reactor $(15 \times 4 \times 2 \text{ cm})$, the top and bottom parts of which were made of quartz to allow easy passage of visible light ($400 < \lambda < 700$ nm). The reactor was placed in the center of the reaction chamber. Two 20 W white bulbs were placed at the top and the bottom of the reaction chamber to generate visible light in the range 400–700 nm with a power density of 0.05 W/cm^2 for the photocatalysis. The dark and visible light conditions for photocatalysis were achieved by turning the bulbs off and on, respectively. Before the photocatalytic conversion experiments, the reactor containing 2 g (36 cm³) of the synthesized porous Cu@V-TiO₂/PU was purged three times with high-purity CO₂ gas. Then 50 mL/min CO₂ that had been passed through water (303 K) was admitted to the reactor. Thus, the residence time of CO₂ in the reactor was 144 s and the space velocity, calculated by dividing the volumetric flow by the reactor volume, was 25 h^{-1} . The reactor temperature was constant at 32 °C during the conversion experiments. To determine gaseous products, a 100 µL sample was automatically injected into the GC system, which consisted of a Varian CP-3800 gas chromatograph (GC) equipped with a methanizer and a flame ionization detector using a packed column (Porapak Q 80/100 2×2 mm) at intervals of 20 min. The reaction start time ($t_0 = 0$) was assumed to be the time when the CO₂ gas stream containing H₂O was admitted to the reactor. The production rate was calculated based on the total weight of synthesized catalyst (2 g).

3. Results and discussion

3.1. Material characteristics

3.1.1. Surface morphology and dopant states

Fig. 1 shows the SEM and elemental mapping of Ti, Cu, and V in the selected areas in the SEM images of the synthesized materials. In the TiO₂/PU, TiO₂ was smoothly immobilized on the PU surface as a thin layer (Fig. 1A). The surfaces of the Cu-TiO₂/PU, V-TiO₂/PU, and Cu@V-TiO₂/PU materials were rougher than that of TiO₂/PU due to the presence of both Cu and V on the metal-doped photocatalysts (Fig. 1B–D). High-resolution XPS spectra with Gaussian multipeak shapes of Cu2p_{3/2} and V2p_{3/2} peaks were obtained to indicate the elemental states of copper and vanadium in the synthesized Cu@V-TiO₂/PU (Fig. 2). The XPS spectra show that the Cu2p_{3/2} peaks consist of two different peaks at 932.18 and 933.48 eV, corresponding to the binding energies of Cu2p_{3/2} in the Cu⁺ and Cu²⁺ states, respectively [25,26]. The Cu⁺ and Cu²⁺ exist



Fig. 1. SEM and elemental mapping of selected areas in SEM images of TiO₂/PU (A), 6Cu-TiO₂/PU (B), 6V-TiO₂/PU (C), and 3Cu@V-TiO₂/PU (D).



Fig. 2. XPS spectra of $Cu2p_{3/2}$ (A) and $V2p_{3/2}$ (B) in $3Cu@3V-TiO_2/PU$.

in the form of either dopants (incorporated into the TiO₂ lattice) or Cu₂O and CuO oxides (distributed on the surface of the TiO₂ layer) [14,18,26,27]. The XPS spectra also show that the V2 $p_{3/2}$ peaks consisted of two different peaks, observed at 516.0 and 517.4 eV, which were attributed to the binding energies of V2 $p_{3/2}$ in the V⁴⁺ and V⁵⁺ states, respectively [28,29]. The V⁵⁺ state of vanadium corresponded to the state of V₂O₅ oxide distributed on the TiO₂

surface [15,28–30]. The V⁴⁺ ions exist as dopants incorporated into the TiO₂ lattice by substitutional replacement of the Ti⁴⁺ ions to form a Ti–O–V bond [15,31,32]. The various formed particles, such as CuO and Cu₂O oxides on the surface of the Cu-TiO₂/PU, V₂O₅ on the surface of the V-TiO₂/PU, and CuO, Cu₂O, and V₂O₅ on the surface of the Cu@V-TiO₂/PU, increased the roughness and hence the surface area of the metal-doped TiO₂ photocatalysts. Table 1 indi-

Table 1
Ti ³⁺ /Ti ⁴⁺ ratios, surface areas, band gap energies, and production rates of the synthesized materials.

	Ti ³⁺ /Ti ⁴⁺ ratios (%)	Surface areas (m ² /g)	Band gap energy (eV)	Production rates $(\mu mol g^{-1} cat. h^{-1})$	
				CO	CH4
TiO ₂ /PU	0	110.6	3.20	0	0
6Cu-TiO ₂ /PU	25.4	166.3	2.85	431	772
5Cu@1V-TiO ₂ /PU	30.9	190.9	2.80	464	803
4Cu@2V-TiO2/PU	33.1	226.7	2.77	501	836
3Cu@3V-TiO2/PU	34.4	232.2	2.70	577	902
2Cu@4V-TiO ₂ /PU	35.3	236.5	2.66	588	933
1Cu@5V-TiO ₂ /PU	33.9	221.3	2.73	531	865
6V-TiO ₂ /PU	30.1	192.6	2.83	453	794

cates that the BET surface areas of the metal-doped photocatalysts were much higher than that of the TiO₂/PU. Table 1 also shows that the surface areas of the co-doped photocatalysts were much higher than those of the single-doped photocatalysts. This was because the co-doping effect of Cu and V inhibited the aggregation of CuO, Cu₂O, and V_2O_5 particles formed on the surface of TiO₂ in the Cu@V-TiO₂/PU. Due to this inhibition, the formed particles were more uniformly distributed and their fractions were smaller than those in the single-doped photocatalysts [29]. Therefore, the surface areas of the Cu@V-TiO₂/PU photocatalysts were much higher than those of the Cu-TiO₂/PU and V-TiO₂/PU photocatalysts. The incorporation of Cu and V in the TiO₂ lattice also distorted the formation of the TiO₂ layer, which increased the surface areas of the synthesized co-doped photocatalysts. The optimal combination weight ratios of 2 wt.% Cu/TiO₂ and 4 wt.% V/TiO₂ led to the largest BET surface area of 236.5 (m^2/g) for 2Cu@4V-TiO₂/PU.

3.1.2. Microstructure and co-doping effects

Fig. 3 shows the XRD patterns of the synthesized TiO_2/PU , Cu-TiO_2/PU, V-TiO_2/PU, and Cu@V co-doped TiO_2/PU . The obtained results indicate that all the TiO_2 in the TiO_2/PU and V- TiO_2/PU existed in the anatase phase [29]. However, the XRD patterns of Cu-TiO_2/PU and Cu@V-TiO_2/PU shows additional peaks at 27.5° and 36.1°, corresponding to rutile peaks of TiO_2 (Fig. 3). Rutile peaks occurred due to the Cu dopants, which accelerated the anatase-rutile phase transformation or the formation of the rutile phase in the TiO_2 [33,34]. In addition, the (101) anatase peak of the TiO_2 in the XRD patterns of both single- and co-doped materials was more left-shifted and broader than that in the TiO_2/PU . This was due to the metal dopants incorporated into the crystal struc-



Fig. 3. XRD patterns of TiO₂/PU, Cu-TiO₂/PU, V-TiO₂/PU, and 3Cu@3V-TiO₂/PU.

ture of TiO₂, leading to distortions in the crystal lattice [35–38]. The Cu and V dopants defected into and substituted for the Ti⁴⁺ ions in the TiO₂ lattice by forming Cu–O–Ti and Ti–O–V bonds, respectively [39,40]. This substitution led to the formation of Ti³⁺ ions or oxygen vacancies in the metal-doped TiO₂/PU materials [41,42]. The obtained XPS results of the doped TiO₂ materials confirmed the existence of Ti^{3+} in the TiO_2 lattice (Fig. 4). The calculated Ti³⁺/Ti⁴⁺ ratios, which are proportional to the ratios (area under the Ti^{3+} peak)/(area under the Ti^{4+} peak) in the obtained XPS spectra of the synthesized metal-doped TiO₂, are also listed in Table 1. The obtained results show that the Ti^{3+}/Ti^{4+} ratio in the V single-doped photocatalyst (6V-TiO₂/PU) was higher than that in the Cu single-doped photocatalyst (6Cu-TiO₂/PU). This was because the ionic radius of V^{4+} (0.72 Å) is closer to that of Ti^{4+} (0.74 Å) than is that of Cu^{2+} (0.87 Å), and hence the V⁴⁺ was more effectively doped into the TiO₂ lattice than the Cu²⁺ was, which increased the oxygen vacancy (or formation of Ti³⁺) in the V single-doped photocatalyst more than that in the Cu single-doped photocatalyst [31,32]. The Ti³⁺/Ti⁴⁺ ratio in the Cu@V-TiO₂/PU co-doped photocatalyst was higher than that in either the Cu or V single-doped photocatalyst. This indicates that the Cu and V co-doping further enhanced the defects in the TiO₂ lattice, thus increasing the formation of oxygen vacancies or Ti^{3+} compared with the case of the single-doped with Cu and V. In the co-doped photocatalysts, when the V/TiO₂ ratio was increased to 4 wt.% (2Cu@4V-TiO_2/PU), the Ti^{3+}/Ti^{4+} ratio increased to a maximum of 35.3%, but then decreased slightly with further increase in the V/TiO₂ ratio. Therefore, the optimal combination weight ratio of 2 wt.% Cu/TiO2 and 4 wt.% V/TiO2 led to the most defects in the TiO₂ lattice.

3.1.3. Optical properties and band gap energy

The optical absorption spectra of the synthesized TiO₂/PU, Cu-TiO₂/PU, V-TiO₂/PU, and Cu@V-TiO₂/PU materials indicate that the synthesized TiO₂/PU inherited all the optical absorption properties of TiO₂, which only absorbs UV light with the absorption edge at \sim 370 nm (Fig. 5) [42,43]. However, the optical absorption of the twin-metal-doped TiO₂/PU materials was observed in both the UV and visible regions. The significant enhancement of the optical absorption in the visible region of the twin-metal-doped TiO₂/PU was due to the synergistic contribution of Cu and V dopants. The dopants defected into the TiO₂ lattice led to oxygen vacancies and the formation of Ti³⁺ in the TiO₂ lattice. The formed Ti³⁺ in the doped TiO₂ lattice created an intermediate band between the valence band and the conduction band of TiO₂, which increased the electron-hole pair separation efficiency of TiO₂ and the light absorption enhancement of the metal-doped TiO₂/PU [44]. The formed oxides (CuO, Cu_2O , and V_2O_5) distributed on the surface of the TiO₂ layer also combined with TiO₂ to create a suitable semiconductor-semiconductor system, which increased the electron transfer from the valence band to the conduction band



Fig. 4. XPS spectra of Ti2p_{3/2} in TiO₂/PU (A), Cu-TiO₂/PU (B), V-TiO₂/PU (C), and 3Cu@3V-TiO₂/PU (D).



Fig. 5. UV–Vis absorption spectra of the synthesized TiO_2/PU and $Cu@V-TiO_2/PU$.

of TiO_2 and thereby enhanced the light absorption of the metaldoped TiO_2/PU [45]. The Cu and V co-doped TiO_2/PU photocatalysts inherited all the advantages of these individual single dopants to enhance their light absorption efficiency. The obtained results indicate that $2Cu@4V-TiO_2/PU$, with Cu/TiO_2 and V/TiO_2 ratios of 2 and 4 wt.%, respectively, showed the greatest optical absorption ability in the visible light region. A Tauc plot combined with the Kubelka– Munk method was used to estimate the band gap energies of the synthesized photocatalysts [46,47]. In the Tauc plot, a linear fit was drawn from the transformed curves to the photon energy axis (*hv*). The energy value at the point of intersection of the line and the horizontal axis is the band gap energy of the material (Fig. 6). The obtained results indicate that the band gap energies of the synthesized $Cu@V-TiO_2/PU$ were positively related to the Ti^{3+}/Ti^{4+} ratios (Table 1). As mentioned, the formed Ti^{3+} acted as an intermediate band between the valence band and the conduction band of TiO₂, thereby decreasing the band gap energy.

3.2. CO₂ reduction

3.2.1. Photocatalytic reduction mechanism

The experiments with CO_2 reduction by the synthesized photocatalysts were conducted under two different conditions. Under condition A, 50 mL/min CO_2 containing water was passed through the synthesized photocatalysts under dark conditions for 180 min and then irradiated with visible light for an additional 360 min.



Fig. 6. Tauc plot of the synthesized TiO₂/PU and twin-metal-doped TiO₂/PU.

Under condition B, the visible light was immediately provided from the beginning of the reduction experiments (in the same CO₂ feed environment). The obtained results show that the synthesized TiO₂/PU and doped TiO₂/PU could not reduce CO₂ under dark conditions. Under visible light, the TiO₂/PU also did not exhibit any photocatalytic activity for CO₂ reduction: however, the doped TiO₂ photocatalysts reduced a certain amount of CO₂ into CO and CH₄. Fig. 7 shows the production rates for CO and CH₄ from the photocatalytic reduction of CO₂ by 3Cu@3V-TiO₂/PU. Under both conditions A and B, the production rates for CO and CH₄ gradually increased to a stable value after a certain irradiation time. However, in the early irradiation period, condition A produced CO and CH_4 at higher rates than condition B did (Fig. 7), because the photocatalysts adsorbed a certain amount of CO₂ and H₂O on their surfaces during the darkened period of condition A (preadsorption). As mentioned, the adsorption is the initial step for CO₂ conversion. Thus, the photocatalyst immediately reduced the adsorbed CO₂ to produce CO and CH₄ when it was irradiated with visible light after the darkened period. Under condition B, the photocatalysts were irradiated immediately with visible light when the CO₂ containing H₂O reached their surfaces. Therefore, the photocatalysts needed a certain time to adsorb CO₂ and H₂O, which slowed down the reduction. However, the adsorption of the CO₂ and H₂O on the

photocatalyst gradually reached a balanced state because the CO₂ conversion experiment was conducted in an online system, in which 50 mL/min CO₂ containing water was continuously admitted to the photocatalyst. The adsorbate (CO_2 and H_2O) on the surface of each photocatalyst in the balanced state under the two experimented conditions was similar because in the balanced state the adsorption depends on the interactions between the photocatalyst and the adsorbate. Therefore, the CO₂ reduction rate and efficiency of the photocatalysts under the two provided conditions were similar after a certain irradiation time (when the "preadsorption" effects of condition A no longer existed). The Cu and V dopants defected into the TiO₂ lattice, leading to oxygen vacancies and the formation of Ti³⁺ in the lattice and thereby enhancing the separation of electron-hole pairs in the Cu@V-TiO₂/PU photocatalysts. The formed CuO, Cu₂O, and V₂O₅ oxides distributed on the TiO₂ surface acted as an electron sink (or electron transfer) to prevent recombination of the generated electrons and holes, also enhancing the electron-hole separation efficiency. Therefore, the synthesized Cu@V-TiO₂/PU photocatalysts generated a significant number of electron-hole pairs, even under visible light. The generated holes reacted with H₂O vapor to produce protons (H⁺). The generated electrons reduced CO₂ to certain carbon radicals (reduced CO₂) such as $^{-}CO_2$ and C^* [48,49]. Finally, the reduced CO₂ reacted with the formed protons to produce CO and CH₄ (Fig. 8). The overall reactions to form CO and CH_4 are shown by the following equations:

$$2H_2O + 4h^+ \to 4H^+ + O_2, \tag{E2}$$

$$CO_2 + 2H^+ + 2e^- \rightarrow CO + H_2O \quad (E^0_{redox} = -0.48 \ V), \eqno(E3)$$

$$CO_2 + 8H^+ + 8e^- \rightarrow CH_4 + 2H_2O \quad (E_{redox}^0 = -0.24 \text{ V}),$$
 (E4)

$$2H^+ \rightarrow H_2.$$
 (E5)

Eqs. (E3) and (E4) show that 2 mol protons and 2 mol electrons are needed to generate 1 mol CO, whereas the formation of 1 mol CH_4 requires 8 mol protons and 8 mol electrons. In addition, a certain time was required to produce protons (via reaction (E2)). More time was also spent in diffusion of the produced protons to the carbon



Condition B: Light was provided after a darkened period Condition B: Light was provided from beginning of experiments

Fig. 7. Rates of CO and CH₄ production from the CO₂ photocatalytic reduction of 3Cu@3V-TiO₂/PU under different conditions.



Fig. 8. Mechanism of CO₂ reduction by doped TiO₂ photocatalysts.

Table 2	
Production rates of the photocatalytic conversion o	$f CO_2$

Materials	Production rates (μ mol g ⁻¹ cat. h ⁻¹)					CO ₂ flow rate (mL/min)	Light sources and reductants	Ref.
	CH ₄	CH₃OH	CO	H ₂	02			
Cu@V-TiO ₂ /PU	935	-	588	30	2160	50	vis + Water vapor	The study
Cu-TiO ₂	18	-	763	-	-	20	UV + H ₂	[51]
Co-Fe	10	-	218	13	-	15	UV + NaCl + water	[53]
Au@CdS/TiO2	41.6	-	0.6	18	109	-	vis + Water vapor	[54]
PO ₄ -C ₃ N ₄	73	-	38	-	-	10	vis + NaHCO ₃ solution	[55]
Au-TiO ₂	-	13	1237	-	-	_	vis + H ₂ gas	[52]
Pt ²⁺ -Pt ⁰ /TiO ₂	64	-	55	394	290	200	UV + Water vapor	[17]
CuCoO ₄	-	-	1150	245	-	_	vis + TEOA + Water	[56]
Ag/CFs	-	475	-	-	120	20	vis + Water	[57]
Pt/LaPO ₄	220	-	-	40	-	-	UV + Water	[58]
Au-In/TiO ₂	15	-	8982	-	-	-	$UV + H_2$	[59]
$ZnO/g-C_3N_4$	5	19	38.7	-	-	-	vis + Water	[60]

radicals. Therefore, in the early irradiation period, there was a small number of protons participating in reactions with CO₂. Thus, in this period, the CO was preferably produced rather than CH₄ due to lack of protons. However, after a certain irradiation time, the produced amount of CH₄ surpassed that of CO because the generation of CH_4 ($E_{redox}^0 = -0.24$ V) is thermodynamically more feasible than that of CO ($E_{redox}^0 = -0.48$ V) when there are sufficient protons and electrons. In the present study, the co-doping of Cu and V significantly enhanced the electron-hole separation of the TiO₂. Therefore, the synthesized materials generated a significant number of electrons and holes, even under visible light. The generated electrons were easily transferred to the photocatalyst surface by the CuO, Cu₂O, and V_2O_5 oxides (distributed on the TiO₂ surface) to react with protons, which were abundantly produced from the reduction of H_2O by the generated holes, leading to selective conversion of CO2 into CH4 rather than CO.

3.2.2. Optimizing doping contents

The rates of production of CO and CH₄ on synthesized Cu@V-TiO₂/PU with different doping ratios after approximately 5 h irradiation by visible light (when production rate was stable) are shown in Table 1. The obtained results indicate that the production rates of CO and CH₄ were positively related to the concentration of the formed Ti³⁺ (or oxygen vacancies) in the TiO₂ lattice. This is consistent with the possible roles of the Ti³⁺ and oxygen vacancies in the reduction of CO₂. The oxygen vacancies existing on the surface of the photocatalyst introduced new adsorption sites to adsorb CO₂. As mentioned in Section 3.2.1, the formation of Ti³⁺ enhanced the separation of electron–hole pairs of the photocatalysts and the generated electrons and holes participated in multiple reactions for the reduction of CO₂ by H₂O vapor [23,50]. Table 1 also shows that 2Cu@4V-TiO₂/PU exhibited the highest CO₂ reduction efficiency. The rates of CH₄ and CO production from the photocatalytic reduction of CO₂ with 2Cu@4V-TiO₂/PU under visible light were 933 and 588 (µmol g⁻¹ cat. h⁻¹), respectively.

Table 2 show the rates of production for products of the photocatalytic reduction of CO_2 by our synthesized material and other recently reported materials [17,51–60]. In a previous study, UV irradiation was mostly used as the energy source to initiate the photocatalytic reduction of CO_2 . The reductants, such as H_2 gas, and the electron donors, such as NaOH and KHCO₃, were also used to enhance the CO_2 reduction efficiency or selectivity. Therefore, the real application of the studies for industrial reduction of CO_2 is limited, because of high energy consumption and safety issues. In contrast, our synthesized Cu@V-TiO₂/PU materials showed a novel photocatalytic activity for the reduction of CO_2 to produce CO and CH₄, because only water vapor was used as the reductant and visible light as the energy source. Thus, our obtained results are expected to open a new era in the applications of photocatalysts for the reduction of CO_2 to produce valuable fuels.

4. Conclusion

In this study, Cu and V co-doped Cu@V-TiO₂/PU photocatalysts were synthesized for CO₂ conversion using visible light as the energy source and H₂O vapor as a reductant to produce valuable fuels. Both Cu and V defected into the TiO₂ lattice, leading to the formation of Ti³⁺ and oxygen vacancies in the lattice. However, as the ionic radius of V⁴⁺ is closer to that of Ti⁴⁺ than is that of Cu²⁺, V⁴⁺ was doped into the TiO₂ lattice more effectively than

Cu²⁺. The Ti³⁺ formed in the doped TiO₂ lattice created an intermediate band between the valence band and the conduction band of TiO₂, which increased the electron-hole pair separation efficiency of TiO₂ and enhanced the light absorption by the metal-doped TiO₂/PU. The oxygen vacancies existing on the surface of the photocatalyst induced new adsorption sites to adsorb CO₂, and the generated electrons and holes reduced the adsorbed CO₂ with H₂O vapor to produce CO and CH₄. The 2Cu@4V-TiO₂/PU photocatalyst, with Cu/TiO₂ and V/TiO₂ ratios of 2 and 4 wt.%, respectively, exhibited the highest rate of defection into the TiO₂ lattice, leading to the maximum formation of Ti³⁺ and oxygen vacancies with the highest reduction efficiency of the doped photocatalyst. The rates of CH₄ and CO production from the photocatalytic reduction of CO₂ with H₂O vapor by 2Cu@4V-TiO₂/PU under visible light were 933 and 588 µmol g⁻¹ cat. h⁻¹, respectively. The synthesized photocatalysts generated sufficient electrons and holes for the reduction of CO_2 . Therefore, CH_4 was selectively produced rather than CO.

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Appendix A. Supplementary material

Supplementary data associated with this article can be found, in the online version, at http://dx.doi.org/10.1016/j.jcat.2016.10.030.

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