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Jing Xu, Juan Chen, Yanhui Ao, Peifang Wang



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Communication

# 0D/1D AgI/MoO<sub>3</sub> Z-scheme heterojunction photocatalyst: Highly efficient visible-light-driven photocatalyst for sulfamethoxazole degradation Jing Xu<sup>1</sup>, Juan Chen<sup>1</sup>, Yanhui Ao<sup>\*</sup>andyao@hhu.edu.cn, Peifang Wang

Key Laboratory of Integrated Regulation and Resource Development on Shallow Lakes of Ministry of Education, College of Environment, Hohai University, Nanjing 210098, China

\* Corresponding author.

E-mail address: (Y. Ao).

<sup>1</sup> These two authors contribute equally to the work.

## Graphical Abstract



A novel AgI/MoO<sub>3</sub> Z-scheme photocatalyst was elaborately synthesized with a unique 0D/1D nanostructures. The results show that the as-prepared 0D/1D AgI/MoO<sub>3</sub> Z-scheme photocatalyst exhibited excellent activity for sulfamethoxazole (SMZ) degradation.

### ABSTRACT

Low dimension nano photocatalysts show great potential in the field of treating contaminated water for their large surface area and size effect. In this study, a 0D/1D Agl/MoO<sub>3</sub> Z-scheme photocatalyst with striking photocatalytic performance was constructed successfully. The one-dimensional MoO<sub>3</sub> nanobelts were prepared by a simple hydrothermal method, and then it was modified by AgI nanoparticles in a handy liquid phase deposition approach. When choosing sulfamethoxazole (SMZ) as the target contaminant, the rate constant value of the optimal 0D/1D Agl/MoO<sub>3</sub> composite could hit up to  $0.13 \text{ min}^{-1}$ , which is nearly 22.7 times and 31.7 times as that of pure MoO<sub>3</sub> ( $0.0058 \text{ min}^{-1}$ ) and AgI ( $0.0040 \text{ min}^{-1}$ ), respectively. A series of detailed characterizations give evidences that the charge transfer in the composite followed Z scheme mechanism. Therefore, efficient separation/transfer and the remained high redox activity of photogenerated carriers played a vital role in the sharply enhanced photocatalytic properties. The possible degradation pathways of SMZ were proposed based on the intermediates detected by high-performance liquid chromatography-mass spectrometry (HPLC-MS). Meanwhile, the magnificent cyclic stability makes the material a promising material in the practical application.

keywords: MoO3-based-photocatalyst, 0D/1D, Z-scheme, Sulfamethoxazole, Visible-light-responsive

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Pharmaceuticals and personal care products (PPCPs) recently have been frequently detected in various water environmental matrices, for lots of them are under heavy use and difficult to be naturally degraded [1,2]. What's worse, some of them, especially antibiotics, have already brought threat to human health and environmental safety even at a very low level [3-5]. However, traditional water treatment technologies have limitations in treating these emerging contaminants. Thus, finding proper methods to decompose and mineralize them is of great urgency. In all kinds of strategies, advanced oxidation processes have been proved versatile methods in treating contaminated water [6,7]. Among them, photocatalysis has been proven as a rather promising approach for its attractive solar energy conversion ability, outstanding redox ability, low cost, low toxicity, and talent to function under diversified ambient conditions [8-11]. To develop photocatalysts of high degradation and mineralization efficiency, various semiconductor catalytic materials have been studied. MoO<sub>3</sub>, as an intrinsic n-type semiconductor with high thermal and chemical stability, has attracted particular interest recently due to its optical properties, nontoxicity, and extensive sources [12-15]. As a transition-metal oxide, MoO<sub>3</sub> has a high positive potential of the valence band, which makes MoO<sub>3</sub> an attractive photocatalyst for the oxidation of pollutants. Moreover, based on the

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previous study, the shape and size of  $MoO_3$  prepared according to different synthetic strategies has significantly influence on its catalytic performance, among which, a one-dimensional nanostructured  $MoO_3$  has drawn attention due to its large surface area and abundant active sites for catalytic degradation [16,17]. However,  $MoO_3$  has some shortcomings, such as limited electrical conductivity, high recombination rate of charge carriers and low reduction capacity, which bring restriction to its practical application [18].

Recent studies have shown great success in improving catalytic performance of single materials by constructing heterojunctions which not only increase light harvesting capacity but speed up charge separation and transfer [19-23]. Among them, the Z-scheme system has become hotspot in the field of photocatalysis for improving redox capacity to a great extent [24, 25]. As it can keep the high reducing capacity from the reduction semiconductor with high conduction band (CB) position and the high oxidation capacity from the oxidation semiconductor with low valence band (VB) position [26-28]. Hence, choosing semiconductors with proper CB and VB edge positions to construct a Z-scheme system heterojunction is an excellent way to optimize the photocatalytic performance of semiconducting materials. Among all these semiconductors, AgI is an anticipated candidate for its good light sensitivity, benign photocatalytic performance, and suitable band position [29-31]. Recent studies have shown some successes in maximizing photocatalysts activity by building AgI-based direct Z-scheme heterojunction [32-34]. Thus, we believe that it is a worthwhile attempt to construct a Z-scheme system by combining AgI nanoparticles with MoO<sub>3</sub> nanobelts.

Against this background, a 0D/1D AgI/MoO<sub>3</sub> Z-scheme heterojunction photocatalyst was obtained, aiming to achieve materials with high activity for antibiotics degradation under visible light irradiation. SMZ was chosen as the main target contaminant to reveal the degradation performance and operational principle of 0D/1D AgI/MoO<sub>3</sub> composite materials in detail. In addition, a series of tests were conducted to systematically investigate the morphology, structure, optical properties and photoelectrochemical properties of the as-prepared materials to provide a reference for the investigation of mechanism and practical application of 0D/1D AgI/MoO<sub>3</sub> Z-scheme photocatalysts.



Fig. 1. Synthesis process of 0D/1D AgI/MoO<sub>3</sub> nanocomposites (a). SEM images of MoO<sub>3</sub> (b), AgI (c), AM-50 (d). TEM images of MoO<sub>3</sub> (e), AM-50 (f). HRTEM images of AM-50 (g)

The Fig. 1a depicts the synthesis process of 0D/1D AgI/MoO<sub>3</sub> composites, the details can be found in the Supporting information. The morphology and microstructure of the as-prepared samples were investigated by Scanning electron microscopy (SEM) and Transmission Electron Microscope (TEM) characterizations. It could be observed that MoO<sub>3</sub> shows obvious nanobelts morphology with an average width of about 180 nm and length of more than 3 µm (Figs. 1b and e). For pure AgI, uneven large particles clustering together were observed with an average size of more than 500 nm (Fig. 1c). However, in Fig. 1d, much smaller AgI particles closely anchor and adhere to the surface of MoO<sub>3</sub> nanobelts. This phenomenon validates that the growth and agglomeration of these AgI particles are suppressed in the formation process of the AgI/MoO<sub>3</sub> nanocomposites because of the interaction between the AgI and the MoO<sub>3</sub> surface, which is beneficial to charge transfer. The intimate interfaces between the MoO<sub>3</sub> and AgI can also imply that AgI/MoO<sub>3</sub> nanocomposites are heterojunction in structure rather than a physical mixture of two separate phases (Fig. 1f). The high-resolution transmission electron microscope (HRTEM) gives information about the lattice fringe of the materials. In the HRTEM image of AM-50 (Fig. 1g), the measured d-spacing of 0.33 nm and 0.20 nm corresponded to the (040) and (112) lattice planes of MoO<sub>3</sub> and AgI, respectively, which confirms the AgI particles modified MoO<sub>3</sub> nanobelts heterojunction materials are successfully constructed.



Fig. 2. XRD patterns of as-prepared samples (a)-(b). High-resolution XPS spectra of MoO<sub>3</sub> and AM-50: Mo 3d (c), O 1s (d). High-resolution XPS spectra of AgI and AM-50: Ag 3d (e), I 3d (f).

The crystal structures and phase purities of the as-prepared composites were determined by X-ray diffraction (XRD) analysis. As shown in Fig. 2a, the characteristic diffraction peaks of MoO<sub>3</sub> are at 12.8°, 23.4°, 25.7°, 27.3°, 29.7°, 39.0° and 58.8°, relating to the (020), (110), (040), (021), (130), (060) and (081) planes, respectively. It is in good agreement with the standard peaks for the orthorhombic phase of MoO<sub>3</sub> crystallites (JCPDS No. 05-0508). The pure AgI shows the diffraction peaks at 22.3°, 23.7°, 25.4°, 39.2°, 42.6° and 46.3°, which could be ascribed to (100), (002), (101), (110), (103) and (112) crystal planes of the hexagonal phase AgI (JCPDS No. 09-0374). Also, the AM-50 heterostructure obviously exhibits two groups of diffraction peaks that correspond to AgI and MoO<sub>3</sub>, respectively. Moreover, as the mass ratio of AgI increases, the peak intensity corresponding to (100), (002) and (112) lattice plane of AgI becomes stronger, indicating the 0D/1D AgI/MoO<sub>3</sub> composite materials are well synthesized (Fig. 2b). Moreover, it is obvious that no characteristic peaks of other impurities appear and the diffraction peak positions in the XRD patterns hardly change, indicating that the synthesized samples are pure with little structure change.

The surface compositions and chemical states of MoO<sub>3</sub>, AgI and AM-50 were further investigated by X-ray photoelectron spectroscopy (XPS). For Mo 3d, four bands can be observed, which correspond to two different oxidative states. The two peaks at 236.0 eV (Mo 3d<sub>3/2</sub>) and 232.8 eV (Mo 3d<sub>5/2</sub>) indicate the presence of Mo<sup>6+</sup> (Fig. 2c) [35]. While, the weaker spectra peaks at 234.6 eV (Mo  $3d_{3/2}$ ) and 231.7 eV (Mo  $3d_{5/2}$ ) are attributed to the existence of Mo<sup>5+</sup> [36], demonstrating the minor existence of oxygen vacancies introduced during the synthesis of the nanomaterial. Typically, the presence of these vacancies may in favor of the production of oxidized radical species, the improvement of photocatalytic performance, and the enhancement of light absorption [37,38]. The O 1s XPS spectrum of MoO<sub>3</sub> can be divided into two peaks at 530.6 eV and 531.7 eV, which can be assigned to lattice oxygen in molybdenum and surface hydroxyl groups, respectively [35] (Fig. 2d). We note that when the AgI are modified by MoO<sub>3</sub> the peaks of the Mo<sup>6+</sup> 4f and  $O^{2-}$  1s from MoO<sub>3</sub> in AM-50 display a slight shift toward a high binding energy, which can be attributed to the heterojunction interaction between MoO3 and AgI. Meanwhile, the Ag 3d core region of AgI shows two peaks at 374.5 and 368.5 eV (Fig. 2e), corresponding to Ag 3d<sub>3/2</sub> and Ag 3d<sub>5/2</sub>. And the I 3d XPS spectrum (Fig. 2f) shows two major peaks with binding energies of 631.2 eV and 619.7 eV, which can be ascribed to  $3d_{3/2}$  and  $3d_{5/2}$  of I<sup>-</sup> in AgI [34]. It is obviously that the binding energy of Ag<sup>+</sup> 3d and □ 3d in AM-50 all display slightly shift to the lower binding energy in comparison with pure AgI, indicating that the chemical coordination environment of Ag<sup>+</sup> and I<sup>-</sup> have changed to some extent because of strong interaction between AgI and MoO<sub>3</sub>. It is rational to infer that heterojunction have formed between the two materials, which can promote the migration of electron-hole pairs and enhance the photocatalytic performance and stability.



**Fig. 3.** Photocatalytic degradation of SMZ with AgI/MoO<sub>3</sub> composites (Blank: SMZ degraded under visible light without the addition of photocatalyst) (a); capture experiment of AM-50 for SMZ degradation (b); cycle experiments of AM-50 for SMZ degradation (c); photocatalytic degradation of ACE with AM-50 (d); the 3D EEM spectra of the SMZ solution in AgI/MoO<sub>3</sub> visible system: 0 min (e), 5 min (f), 15 min (g) and 20 min (h).

The photocatalytic performance of the as-prepared samples was evaluated by the degradation of SMZ under visible light irradiation. As exhibited in Fig. 3a, SMZ is rarely degraded without the addition of photocatalysts, indicating SMZ is quite stable under visible light irradiation. Meanwhile, pure MoO<sub>3</sub> and AgI exhibit poor photocatalytic activity, whose degradation rate in 20 min are just 10% and 8%, respectively. The photocatalytic degradation efficiency of SMZ is 13.6%, 47.4%, 97.6%, 59.6% and 46.1% for the AM-10, AM-30, AM-50, AM-70, and pm-AM-50 after irradiation for 20 min. Among them, the AM-50 exhibits the highest photocatalytic activity with the rate constant of 0.13 min<sup>-1</sup> (Fig. S2 in Supporting information). Noted that all the 0D/1D AgI/MoO<sub>3</sub> composites display markedly higher photocatalytic degradation of SMZ than bare MoO<sub>3</sub> and AgI. And the AM-50 nanocomposites exhibit much better photocatalytic activity than that of pm-AM-50, which imply the form of heterostructure interaction between AgI nanoparticles and MoO<sub>3</sub> nanobelts rather than simple physical mixture. The heterostructure brings outstanding photocatalytic performance and promotes the transfer and separation of photogenerated charge carriers, which has already been confirmed by photocurrent curves (PC) and electrochemical impedance spectroscopy (EIS) tests (Fig. S4 in Supporting information). With the increase of AgI load, the photocatalytic performance of 0D/1D AgI/MoO<sub>3</sub> heterostructure composites first increases and then decreases. This probably because that excessive AgI particles have an intense propensity of agglomeration, which restrains their valid contact with MoO<sub>3</sub> nanobelts.

To better reflect the degradation process of SMZ by AgI/MoO<sub>3</sub> composites, the transformation behavior of SMZ during the reaction was investigated by 3D excitation-emission matrix (EEM) spectra (Fig. 3). The general characteristics of the EEM plots for the sample AM-50 show that SMZ has one main peak located at  $E_x/E_m = 260-300/315-380$  nm [39]. Before irradiation, strong extra signals presented in SMZ solution (Fig. 3e). With irradiation time extending, this peak intensity begins to decrease, which attributed to the degradation of SMZ. When the irradiation time increases to 20 min (Fig. 3h), the peak strength of SMZ becomes weak, showing that the SMZ molecules were almost completely degraded by AM-50 composite. The possible degradation pathways of SMZ achieved according to the results of HPLC-MS are shown in Fig. S5 (Supporting information).

To investigate the photocatalytic mechanism, trapping experiments were carried out to quench the main reactive species involved in the degradation of SMZ. The results are shown in Fig. 3b, when using ethylenediaminetetraacetic acid disodium salt (EDTA-2Na) as the direct hole scavenger, a remarkable inhibitory effect on the degradation efficiency of SMZ is observed. In this experiment, and the rate constant *k* value (Fig. S2) is 0.02 min<sup>-1</sup>, which is far below that of control experiments of AM-50 without scavengers (0.13 min<sup>-1</sup>). When isopropanol alcohol (IPA) was introduced or Argon was passed through, 64% and 46% of SMZ were degraded. Therefore, hole (h<sup>+</sup>), superoxide ( $^{\circ}O_2^{-}$ ) and hydroxyl ( $^{\circ}OH$ ) are all working species in the degradation of SMZ. Among them, h<sup>+</sup> is the leading radical. Moreover, the AM-50 sample maintains effective photocatalytic degradation stability during repeated tests (Fig. 3c). Therefore, it is obvious that AM-50 could work as a high-efficiency photocatalyst for the removal of SMZ in water.

In order to verify the photocatalytic performance of the AM-50, Acetaminophen (ACE) was chosen as another contaminant for degradation under visible-light irradiation. As shown in Fig. 3d, 91.6% of ACE can be removed by AM-50 heterojunction under visible light for 25 min. While the degradation rate for pure  $MoO_3$  and AgI are 1% and 32%, respectively. The results indicate that the AM-50 heterostructures do have outstanding photocatalytic degradation capability.



Fig. 4. ESR spectra of the DMPO-'O<sub>2</sub><sup>-</sup> (a), and DMPO-'OH (b) adducts recorded with pristine AM-50 in the dark and under visible light irradiation. Scheme diagram of possible photocatalytic mechanism of AM-50 (c).

The electron spin resonance (ESR) technology was used to detected short-lived active species in the degradation process. As shown in the diagram Fig. 4, typical peaks of  $O_2^-$  were clearly recorded in DMPO- $O_2^-$  spectra after 10 min visible light irradiation, while there is no obvious signal appeared in the dark. As for OH, compared with the dark condition, four evident peaks of 1:2:2:1 proportion appeared under irradiation, which is powerful evidence of the generation of OH. The above results demonstrate convincingly that  $O_2^$ and OH are produced and participated in the photocatalytic degradation process of SMZ.

In the light of the active species verification test and the band position of each photocatalytic material obtained from the diffuse reflectance spectra (DRS) (Fig. S3 in Supporting information) and empirical equation, the possible carriers transfer path and photocatalytic mechanism of 0D/1D AgI/MoO<sub>3</sub> heterostructures is proposed. Under the irradiation of visible light, the photo-generated electrons would be inspired and then transfer form VB to CB both in MoO<sub>3</sub> and AgI. If the carriers transfer following a traditional type-II heterojunction (Fig. S6 in Supporting information), the electrons on the CB of AgI would shift to the CB of MoO<sub>3</sub> and the holes on the VB of MoO<sub>3</sub> would move to the VB of AgI, leaving electrons on the CB (0.35 V) of MoO<sub>3</sub> and holes on the VB (2.39 V) of AgI which cannot satisfy the reduction potential of  $O_2/O_2^-$  (-0.33 V vs. NHE) [40] and the oxidation potential of  $H_2O/OH$  (2.68V vs. NHE) [32]. Hence, a Z-system heterojunction may be a reasonable way to explain the photocatalytic reaction mechanism. As is illustrated in Fig. 4c, after being inspired by visible light, the electrons at the CB of AgI (-0.43 V) and the holes at the VB (3.44 V) of MoO<sub>3</sub> could be retained, which ensures a strong redox capability for the generation of reactive species, such as 'O<sub>2</sub><sup>-</sup> and 'OH, that are highly efficient for the degradation of SMZ. The assumption is consistent with the results of the above tests.

Overall, a novel 0D/1D AgI/MoO<sub>3</sub> Z-system heterostructure was successfully synthesized. The optimum composite sample AM-50 presented outstanding photocatalytic performance for both SMZ and ACE degradation. Compared with single MoO<sub>3</sub> and AgI, the degradation efficiency of SMZ by AM-50 increased by 87% and 89%, respectively. Meanwhile, the increasement of ACE degradation efficiency depended on AM-50 was up to 90% and 59%, when compared with MoO<sub>3</sub> and AgI, respectively. The 0D/1D nano heterostructure plays the key role in the increased exposure of active sites and the accelerated separation of photogenerated charge carriers, which brings excellent photocatalytic properties and stability. It is the first time that the MoO<sub>3</sub>-based photocatalytic samples have been used in SMZ degradation, which is a worthwhile attempt that widen the road of MoO<sub>3</sub>-based photocatalysts in PPCPs degradation. Based on this study, the construction of other low dimension MoO<sub>3</sub>-based heterostructure system materials could be considered.

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