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## Triple-shelled $\text{Mn}_2\text{O}_3$ hollow nanocubes: force-induced synthesis and excellent performance as the anode in lithium-ion batteries†

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In this paper, we report a novel structure of  $\text{Mn}_2\text{O}_3$ , the triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocube, as the anode material for high-energy lithium-ion batteries, synthesized through a programmed annealing treatment with cubic  $\text{MnCO}_3$  as precursor. This hierarchical structure is developed through the interaction between the contraction force from the decomposition of  $\text{MnCO}_3$  and the adhesion force from the formation of  $\text{Mn}_2\text{O}_3$ . The structure has been confirmed by characterization with XRD, FESEM, TEM, and HRTEM. The charge–discharge tests demonstrate that the resulting  $\text{Mn}_2\text{O}_3$  exhibits excellent cycling stability and rate capability when evaluated as an anode material for lithium-ion batteries. It delivers a reversible capacity of 606 mA h  $\text{g}^{-1}$  at a current rate of 500 mA  $\text{g}^{-1}$  with a capacity retention of 88% and a remaining capacity of 350 mA h  $\text{g}^{-1}$  at 2000 mA  $\text{g}^{-1}$ .

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### Introduction

With the increasing demand for large-scale electrochemical energy storage, rechargeable lithium-ion batteries with a high-energy density, high rate capability and long-term stability are urgently required.<sup>1,2</sup> Transition metal oxides have been widely investigated as anodes in lithium-ion batteries because of their higher theoretical specific capacities compared with the commercial graphite anode currently available.<sup>3–7</sup> Among the various transition metal oxide anodes,  $\text{Mn}_2\text{O}_3$  with a theoretical capacity of 1018 mA h  $\text{g}^{-1}$  is the most attractive owing to its abundance and environmentally friendly characteristics.<sup>8–10</sup> However, its major volume expansion during the charge–discharge process leads to a significant capacity loss and poor cycling stability, which greatly hinders its practical application in lithium-ion batteries.<sup>11,12</sup>

The synthesis of a hollow micro-/nanoscale structure with controlled size, shape and internal structure is a promising approach to improve the performance of transition metal oxide anodes.<sup>13–16</sup> The nano-size units reduce the path of lithium-ion diffusion and increase the number of reaction sites for lithium intercalation/deintercalation, and the interior hollow spaces buffer the volume changes during charge–discharge and thus enhance the rate capability and cyclic stability.<sup>17–20</sup> Much effort

has been devoted to developing new strategies for rational synthesis of complex hollow structures, such as multi-shelled hollow structures and yolk–shelled structures, as these complex structures are expected to improve the properties of lithium-ion batteries.<sup>21–24</sup> A general approach for the fabrication of multi-shelled hollow structures involves the employment of sacrificial templates, such as monodispersed polymer,<sup>25</sup> silica,<sup>13</sup> or carbon.<sup>26</sup> However, methods employing sacrificial templates usually require multiple steps and post-treatment, which might not be suitable for cost-effective and environmentally-friendly large-scale manufacture. It is still a great challenge to develop a template-free method that enables facile synthesis of multi-shelled hollow structures.

In this paper, we report a novel structure of  $\text{Mn}_2\text{O}_3$ , the triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocube, synthesized through a programmed annealing treatment with cubic  $\text{MnCO}_3$  as precursor. The crystal structure and morphology of the resulting product were characterized by XRD, FESEM, TEM, and HRTEM, and its performance as anode material in lithium-ion batteries, in terms of cyclic stability and capability, was evaluated with charge–discharge tests.

### Experimental section

#### Material synthesis

The cubic  $\text{MnCO}_3$  was synthesized by a precipitation reaction in a microemulsion.<sup>27–29</sup> Typically, 4 g of cetyltrimethyl ammonium bromide (CTAB) was dissolved in a mixture of 100 mL cyclohexane, 5 mL *n*-pentanol and 5 mL 0.8 M  $(\text{NH}_4)\text{HCO}_3$  aqueous solution. The mixture was stirred for 20 min until it became transparent, then 5 mL 0.4 M  $\text{MnSO}_4$  was dripped into the continuously stirred solution to produce a milky white

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solution. Finally, the white  $\text{MnCO}_3$  was filtered, washed several times with ethanol and distilled water, and dried in a vacuum at  $90^\circ\text{C}$ . The triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocubes were synthesized by a two-step annealing process, using the cubic  $\text{MnCO}_3$  as template. It was first sintered at  $300^\circ\text{C}$  for 1 h with a ramping rate of  $1^\circ\text{C min}^{-1}$  and then at  $600^\circ\text{C}$  for 1 h with a ramping rate of  $2^\circ\text{C min}^{-1}$  in air.

### Materials characterization

The morphology of the materials was observed by field emission scanning electron microscopy (FESEM, ZEISS Ultra 55), transmission electron microscopy (TEM, JEOL JEM-2100HR) and high-resolution TEM (HRTEM). The crystal structures of the samples were analyzed by X-ray diffraction (XRD, BRUKER D8 ADVANCE, Germany) with  $\text{Cu K}\alpha$  radiation. The surface area was determined by the Brunauer–Emmett–Teller method (BET, Micromeritics ASAP 2020 M) at liquid nitrogen temperature (77 K).

### Electrochemical measurements

The electrode was prepared by mixing 70 wt% active material with 20 wt% acetylene black and 10 wt% polyvinylidene difluoride (PVDF) binder, coating the mixture on an aluminum sheet, and then cutting the sheet into  $1\text{ cm} \times 1\text{ cm}$  samples. The typical mass of loaded active material was 5–7 mg. The CR2025 coin cell was assembled in an Ar-filled MBraun glove box using the prepared electrode as cathode, lithium film as anode, Celgard 2400 as separator, and 1 M  $\text{LiPF}_6$  in ethylene carbonate (EC) and dimethyl carbonate (DMC) (EC/DMC 1 : 2 by volume) as electrolyte. The charge–discharge tests were performed on a Land cell test system (Land CT2001A, China). Cells were cycled between 3 V and 0.01 V (vs.  $\text{Li/Li}^+$ ) at  $25^\circ\text{C}$ . Cyclic voltammetry was performed on a Solartron-1470E CellTest at  $25^\circ\text{C}$  between 3 V and 0.01 V at a sweep rate of  $0.1\text{ mV s}^{-1}$ .

## Results and discussion

Triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocubes are synthesized through a thermally driven contraction process during the decomposition of  $\text{MnCO}_3$ , a process which was developed in the fabrication of double-shelled  $\text{CoMn}_2\text{O}_4$ .<sup>15</sup> Uniform  $\text{MnCO}_3$  nanocubes prepared by the microemulsion method are applied as the precursor.<sup>27,28</sup> The formation of  $\text{Mn}_2\text{O}_3$  can be demonstrated by the thermogravimetric analysis curve for  $\text{MnCO}_3$  nanocubes (Fig. S1†). Thermal decomposition in air leads to the conversion of  $\text{MnCO}_3$  into manganese oxide,  $\text{MnO}_2$ ,  $\text{Mn}_2\text{O}_3$  or  $\text{Mn}_3\text{O}_4$ , depending upon the sintering temperature. The first steep weight loss ( $>300^\circ\text{C}$ ) is associated with the decomposition of  $\text{MnCO}_3$  to form  $\text{MnO}_2$  phase, while the second steep weight loss ( $>490^\circ\text{C}$ ) is the result of transformation of  $\text{MnO}_2$  to  $\text{Mn}_2\text{O}_3$ . Usually, to obtain the pure phase of  $\text{Mn}_2\text{O}_3$ , as guided by the thermogravimetric analysis curve, thermal decomposition should be conducted at a temperature above  $490^\circ\text{C}$  for a certain time. However, if the thermal decomposition process is simply stopped halfway,  $\text{MnCO}_3$  can only be partially decomposed, which could lead to the development of a manganese oxide layer on the surface of the particles, whereas the inner cores should remain as  $\text{MnCO}_3$ .<sup>30,31</sup>

As illustrated in Fig. 1, the fabrication process mainly consists of two annealing treatment steps. The  $\text{MnCO}_3$  nanocubes are first sintered at  $300^\circ\text{C}$  for 1 h with a ramping rate of  $1^\circ\text{C min}^{-1}$ , and partially converted into manganese oxide, leading to the formation of the first shell, which is mainly based on the heterogeneous contraction caused by non-equilibrium heat treatment.<sup>15,16,32</sup> During the calcination, a large temperature gradient ( $\Delta T$ ) is present along the radial direction, which leads to the formation of a manganese oxide shell on the surface of the  $\text{MnCO}_3$  core.<sup>32,33</sup> There are two forces in opposite directions acting on the interface between the manganese oxide shell and the  $\text{MnCO}_3$  core. One is the contraction force ( $F_c$ ) as a result of the large weight loss during the calcinations, which promotes the inward shrinkage of the  $\text{MnCO}_3$  core. The other is the adhesion force ( $F_a$ ) from the relatively rigid shell, which prevents its inward contraction. With  $\Delta T_1$  at the first heat treatment,  $F_c$  exceeds  $F_a$  so that the inner core contracts inward and detaches from the outer shell. It can be observed that, after the first heat treatment, the particle maintains its cubic shape but its surface becomes coarser due to the release of  $\text{CO}_2$  (Fig. S2A and B†). The TEM image of the product confirms the core–shell structure (Fig. S2C†). The XRD pattern provides evidence of the  $\text{MnCO}_3$  core from the partially decomposed  $\text{MnCO}_3$  nanocube (Fig. S2D†). Compared with the highly crystalline  $\text{MnCO}_3$  (Fig. 2), the intensity of the XRD peaks decreased considerably upon partial decomposition of  $\text{MnCO}_3$ . This process was accompanied by a color change of the material from milky-white to brown (Fig. S3†). Similar to the first heat treatment, the product obtained was then annealed at  $600^\circ\text{C}$  for 1 h with a ramping rate of  $2^\circ\text{C min}^{-1}$ . The same process as the first heat treatment occurs in the precursor cube, leading to the formation of the second shell. When the heating is prolonged at high temperature,  $F_c$  decreases rapidly and  $F_a$  surpasses  $F_c$ , leading to the reversed direction for movement of materials. As a result, the inner core shrinks outward, leaving a hollow cavity in the center.

The as-synthesized samples were characterized by XRD to determine their crystallographic structure. As shown in Fig. 2, the resulting  $\text{MnCO}_3$  exhibits a hexagonal phase (PDF#44-1472). All of the diffraction peaks of the  $\text{Mn}_2\text{O}_3$  can be indexed to a cubic phase (PDF#73-1826) by comparing them with the standard pattern. No residues or contaminants have been detected in the two XRD patterns, indicating the high purity of the product.

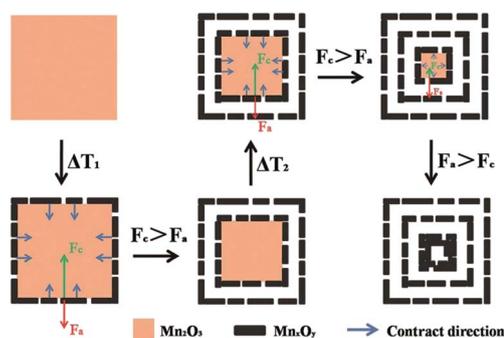


Fig. 1 Schematic illustration of the process of formation for triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocubes.

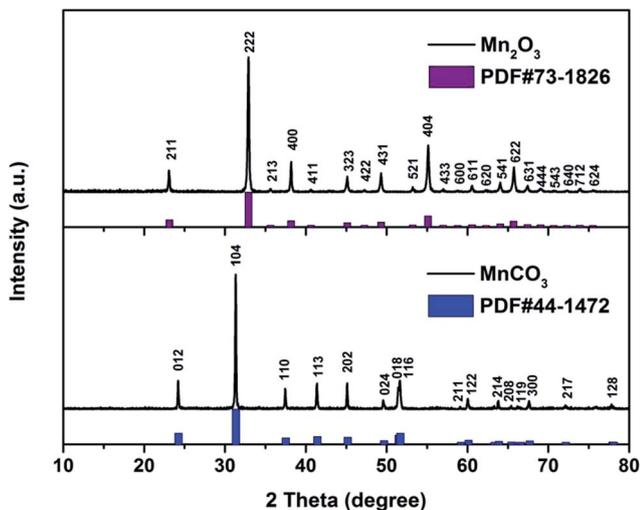


Fig. 2 XRD patterns for the  $\text{MnCO}_3$  nanocube and the triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocube.

The morphology and detailed structure of the samples were observed by field-emission scan electron microscopy (FESEM) and transmission electron microscopy (TEM). As can be seen from Fig. 3A and B, the  $\text{MnCO}_3$  obtained adopted a homogeneous cubic shape with edge length of around 200 nm. When  $\text{MnSO}_4$  and  $(\text{NH}_4)_2\text{CO}_3$  solutions were mixed,  $\text{MnCO}_3$  nucleated gradually in the microemulsion droplets. The selective interaction between the crystallographic planes of  $\text{MnCO}_3$  and CTAB, which acts as a morphology controller,<sup>27–29</sup> resulted in the cubic shape and microemulsion defines the nanoparticle size. After the programmed heat treatment, the  $\text{Mn}_2\text{O}_3$  product maintained the cubic shape, as shown in Fig. 3C. From the inset in Fig. 3C, it can be clearly observed that the  $\text{Mn}_2\text{O}_3$  cube adopts a porous architecture composed of nanosized subunits. The porous structure is attributed to gas evolution from the  $\text{MnCO}_3$  bulk during the thermal decomposition procedure to form manganese oxide. From the broken nanocubes with exfoliation of the shell as shown in Fig. 3D and E, the triple-shelled structure with the shell marked by the number and the hollow interior can be identified. Further FESEM images (Fig. S4†) indicate that the inner shells also retain the porous cubic shape. TEM images further confirmed the triple-shelled structure of the synthesized  $\text{Mn}_2\text{O}_3$  nanocube. As shown in Fig. 3F and G, clear gaps can be distinguished between the shells, and the inner core can be confirmed to adopt the hollow cubic structure. Fig. 3H presents the high-resolution TEM (HRTEM) image of the  $\text{Mn}_2\text{O}_3$  nanocube. The lattice fringes with interplanar spacing of about 0.54 nm can be indexed to the family of the (111) plane. The inset is the corresponding fast Fourier-transformed (FFT) pattern, indicating that the  $\text{Mn}_2\text{O}_3$  nanocube adopts a highly crystalline structure. The BET result illustrated in Fig. 4 indicates that the pore sizes of triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocubes are mainly less than 30 nm, contributing to a surface area of  $21.18 \text{ m}^2 \text{ g}^{-1}$ .

The electrochemical properties of the triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocube were investigated in coin cells by both cyclic voltammetry and galvanostatic charge–discharge test using a

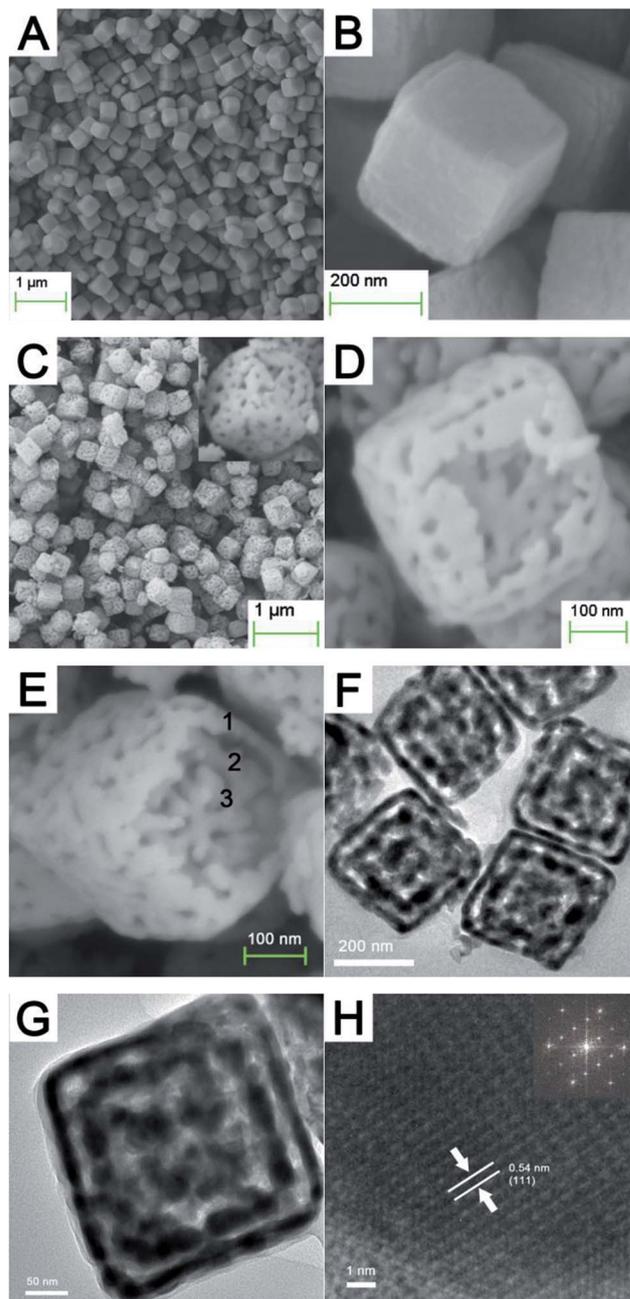


Fig. 3 FESEM images for the  $\text{MnCO}_3$  nanocube (A and B), and FESEM (C–E), TEM (F and G) and HRTEM (H) images for the triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocube.

lithium plate as reference and counter electrodes. The product obtained was first evaluated by cyclic voltammetry in the voltage range 0.01–3.0 V at a scan rate of  $0.1 \text{ mV s}^{-1}$ , as shown in Fig. 5A. During the first cathodic process, three main peaks were present centered at 1.25 V, 0.75 V and 0.2 V. The peaks located at 1.25 V and 0.2 V are attributed to the reduction of  $\text{Mn}^{3+}$  to  $\text{Mn}^{2+}$  and  $\text{Mn}^{2+}$  to  $\text{Mn}^0$ , respectively,<sup>9,10</sup> whereas the peak at 0.75 V is ascribed to the irreversible decomposition of the solvent in the electrolyte to form the solid–electrolyte interface (SEI).<sup>16</sup> The minor peak located at around 0.9 V may be attributed to the lithium-ion insertion into the conductive agent acetylene

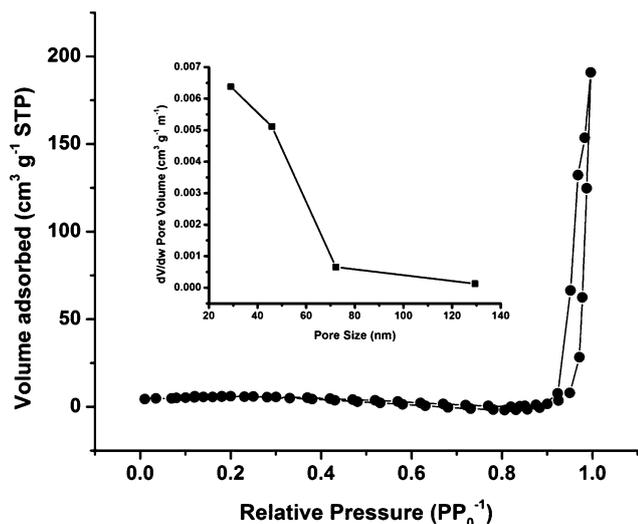


Fig. 4  $N_2$  adsorption/desorption isotherm for the triple-shelled  $Mn_2O_3$  hollow nanocube and the corresponding pore size distribution.

black.<sup>8,34</sup> There are two oxidation peaks at 1.25 V and 2.35 V in the following anodic process, which are associated with the oxidation of  $Mn^0$  to  $Mn^{2+}$  and  $Mn^{2+}$  to  $Mn^{3+}$ , respectively. Compared with the first cycle, two major peaks are observed in the following cycles, which correspond to the reduction/oxidation of  $Mn^{2+}$ . Note that the third curve and the fifth one overlap, indicating the good reversibility of the electrochemical reaction.

Fig. 5B and C illustrate the cycling performance of the triple-shelled  $Mn_2O_3$  hollow nanocube at  $100\text{ mA g}^{-1}$  current density between 3.0 V and 0.01 V and the corresponding discharge-charge profiles. The initial discharge and charge capacities are  $1200\text{ mA h g}^{-1}$  and  $845\text{ mA h g}^{-1}$ , respectively. The irreversible capacity loss of the first cycle can be attributed to the formation of the SEI film and the irreversible change between  $Mn^{3+}$  and  $Mn^{2+}$ , which correspond to the capacity in the voltage range between 1.5 V and 0.5 V. In the first discharge curve, two plateaus at around 1.3 V and 0.25 V could be attributed to the reduction of  $Mn^{3+}$  to  $Mn^{2+}$  and  $Mn^{2+}$  to  $Mn^0$  and the plateaus at about 1.3 V and 2.4 V could be associated with the oxidation of  $Mn^0$  to  $Mn^{2+}$  and  $Mn^{2+}$  to  $Mn^{3+}$ , respectively. After the first cycle, there are two main plateaus located at around 0.5 V and 1.5 V for the discharge and charge profiles, respectively, which are related to the  $Mn^{2+}/Mn^0$  redox reaction. This result is consistent with observations from cyclic voltammetry.

It is well known that the major volume expansion of transition metal oxide anodes during the charge-discharge process leads to serious capacity loss and poor cycling stability. The cyclic stability of the resulting triple-shelled  $Mn_2O_3$  hollow nanocube was evaluated with 100 charge-discharge cycles at current densities of  $200\text{ mA g}^{-1}$  and  $500\text{ mA g}^{-1}$  in the voltage range of 0.01–3.0 V. Fig. 6 presents the discharge capacity as a function of cycle number. All of the cells were first charged and discharged at a current density of  $100\text{ mA g}^{-1}$  for two cycles. The first discharge capacity is around  $718\text{ mA h g}^{-1}$  for the current density of  $200\text{ mA g}^{-1}$  and  $606\text{ mA h g}^{-1}$  for  $500\text{ mA g}^{-1}$ . Interestingly, the discharge capacities gradually decrease and

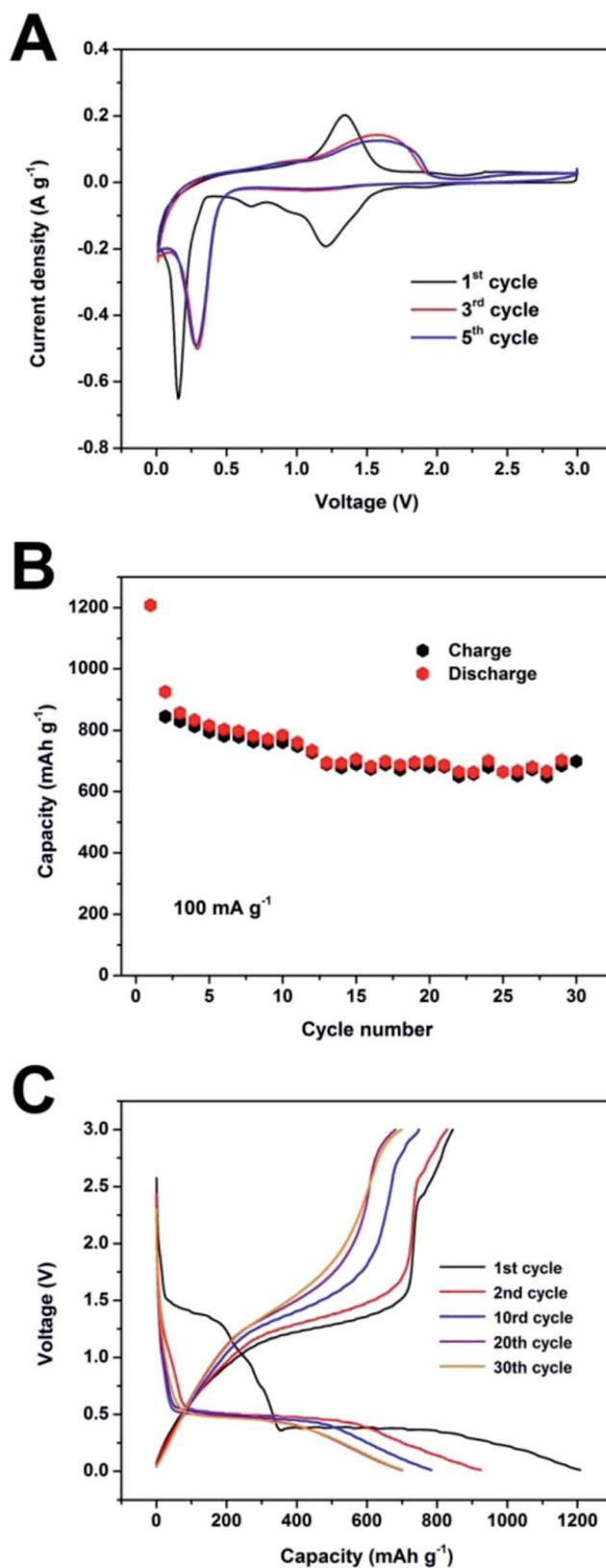


Fig. 5 (A) Cyclic voltammograms for the triple-shelled  $Mn_2O_3$  nanocube electrode at a scan rate of  $0.1\text{ mV s}^{-1}$  in the voltage range of 0.01–3.0 V. (B) Cycling performance in the voltage range of 0.01–3.0 V at a current density of  $100\text{ mA g}^{-1}$  and (C) the corresponding discharge-charge profiles.

then start to increase after certain cycles. This phenomenon can be attributed to the presence of a possible activation process in the electrode, which is commonly observed for transition metal oxides.<sup>4</sup> Note that the discharge capacities stay above the specific capacity of graphite ( $372 \text{ mA h g}^{-1}$ ), although they change slightly. The discharge capacity at  $500 \text{ mA g}^{-1}$  remains  $533 \text{ mA h g}^{-1}$  after 100 cycles with a capacity retention of 88%, indicative of the excellent cyclic stability of the triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocube. This excellent cyclic stability should be related to the structural stability of the triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocube. Fig. 7 presents the FESEM images of the electrode after cycling. Compared with the image for the pristine nanoparticles (Fig. 3D), the porous cubic structure is maintained after cycling although the cube size increases slightly due to the SEI layer formed, confirming the structural stability of the triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocube. Apparently, the hierarchical structure of  $\text{Mn}_2\text{O}_3$  provides space to buffer the mechanical strain induced by the volume change resulting from lithium insertion/extraction, maintaining the structural stability of  $\text{Mn}_2\text{O}_3$ .

To evaluate the rate capability, the resulting triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocube was cycled at various current densities

ranging from  $100 \text{ mA g}^{-1}$  to  $2000 \text{ mA g}^{-1}$ . Fig. 8A and B show the rate capability of the triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocube and the corresponding discharge–charge profiles. The triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocube delivers a capacity of  $350 \text{ mA h g}^{-1}$  at  $2000 \text{ mA g}^{-1}$ . To the best of our knowledge, this is one of the best rate performances for  $\text{Mn}_2\text{O}_3$  anodes that have been reported in the literature.<sup>8,9,35</sup> This excellent rate capability should be related to the nano-size units of  $\text{Mn}_2\text{O}_3$ . It should be noted that the discharge capacity recovers to  $700 \text{ mA h g}^{-1}$  when the current rate is returned back to  $200 \text{ mA g}^{-1}$ . These results indicate that the resulting triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocube simultaneously exhibits excellent cyclic stability and rate capability.

The excellent performance of the resulting triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocube can be attributed to its special structure. The hollow space buffers the mechanical strain induced by the volume change during cycling, while the nano-size units reduce the path of lithium-ion diffusion and increase the number of reaction sites for lithium insertion/extraction,

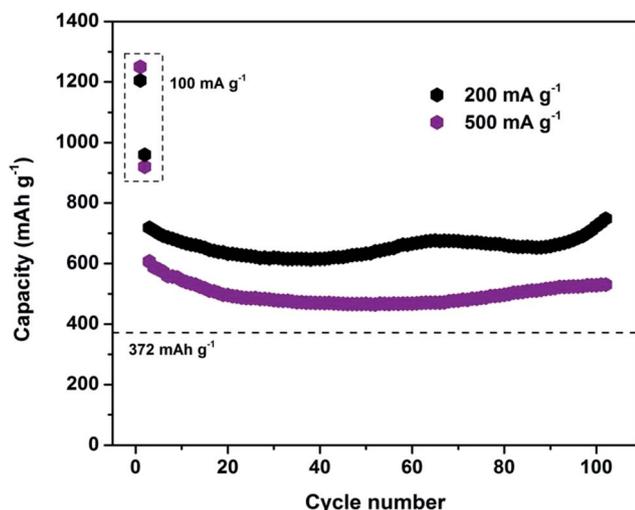


Fig. 6 Cyclic stability of the triple-shelled  $\text{Mn}_2\text{O}_3$  nanocube electrode at current densities of 200 and  $500 \text{ mA g}^{-1}$ .

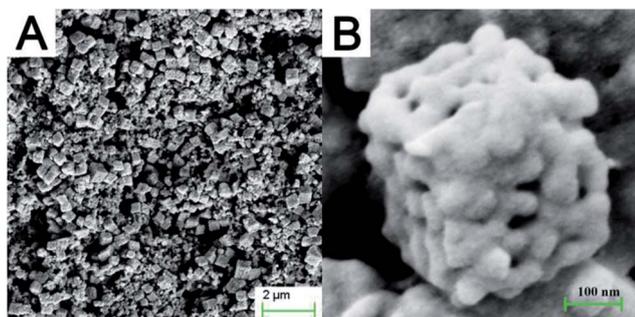


Fig. 7 FESEM images of the triple-shelled  $\text{Mn}_2\text{O}_3$  nanocube electrode after cycling. (B) is a higher magnification FESEM image.

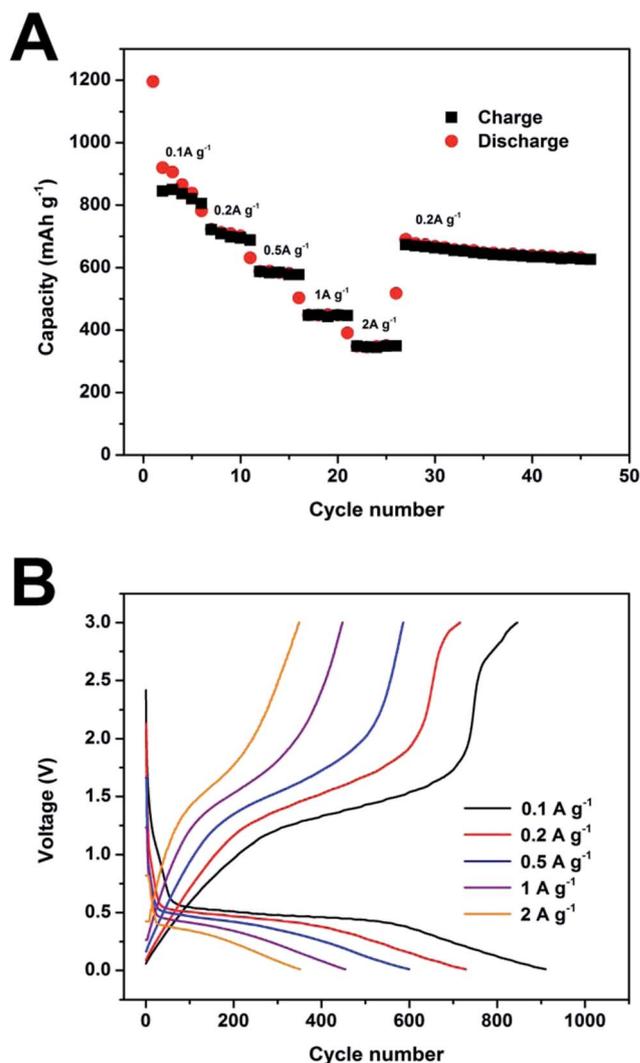


Fig. 8 (A) Specific capacity as a function of cycle number at different current densities and (B) the corresponding discharge–charge profiles.

leading to the excellent cyclic stability and rate capability of the resulting triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocube.

## Conclusions

In summary, we have developed a novel structure of  $\text{Mn}_2\text{O}_3$ , the triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocube with nanosize subunits, as the anode material for high-energy lithium-ion batteries. The nanocube was synthesized through a programmed annealing treatment with cubic  $\text{MnCO}_3$  as precursor. The resulting hierarchical structure is formed *via* the interaction between the contraction force from the decomposition of  $\text{MnCO}_3$  and the adhesion force from the formation of  $\text{Mn}_2\text{O}_3$ . When evaluated as an anode material for lithium-ion batteries, the triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocube exhibits excellent cyclic stability and rate capability. The hollow space buffers the mechanical strain induced by the volume change during cycling, while the nanosize units reduce the path of lithium-ion diffusion and increase the number of reaction sites for lithium insertion/extraction, leading to the excellent cyclic stability and rate capability of the resulting triple-shelled  $\text{Mn}_2\text{O}_3$  hollow nanocube.

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