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ARTICLE

Integrating CuInSe₂ Nanocrystals with Polymeric Carbon Nitride Nanorods for Photocatalytic Water Splitting

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Developing photocatalysts with improved photoactivity and efficiency has remained an enduring theme both fundamentally and technologically in the field of photocatalysis. Polymeric carbon nitride (CN) has been widely exploited as an earth-abundant photocatalyst for water redox reactions. Nevertheless, the limited visible-light utilization rate and the high recombination rate of photoinduced charge carriers give rise to the moderate photocatalytic reactivity of CN in water splitting. Herein, p-type CuInSe₂ nanocrystals are prepared by a solvothermal approach and then immobilized with n-type CN nanorods through self-assembly and thermal treatment process, forming a CuInSe₂/CN hybrid photocatalyst. Benefiting from the p-n heterojunction, 3% CuInSe₂/CN nanocomposite photocatalyst exhibits a three-fold increase in the hydrogen evolution rate (HER) than that of bare CN nanorods owing to the strengthened visible-light capturing capability and improved separation rate of photoexcited charge carriers. This work paves new avenues for the construction of p-n heterojunction photocatalysts for solar fuel production.

1. Introduction

Heterogeneous photocatalysis, which is recognized as a feasibility approach for resolving energy and environmental problems, has stimulated great research interest in the past 40 years.¹ Photocatalytic water splitting has drawn great attention attributed to its potential to produce hydrogen fuels by exploiting renewable solar energy. The creation of photocatalysts with superior photocatalytic performance remains to be the major task of photocatalytic water splitting.²

In the past decades, polymeric carbon nitride (CN) has been widely developed and utilized as a promising photocatalyst for water splitting and other photoredox reactions owing to the advantages of visible-light absorption, stability, sustainability and low preparation cost.³ Nevertheless, the photocatalytic reactivity of pristine CN is relatively moderate owing to the limited utilization rate of the solar spectrum and serious recombination of photoexcited charge carriers.⁴ Nanostructure engineering is one of the efficient pathways to overcoming such impediments of CN.⁵

Our group demonstrated that CN nanorods could serve as an effective photocatalyst for water splitting because the nanorod structure could facilitate photogenerated electron separation and offer more active sites.⁶ Additionally, extensive effort has been made to further promoting the photocatalytic performance of CN by constructing heterojunction photocatalysts by coupling CN with other semiconductors.⁷ Particularly, when p-type semiconductor is coupled with n-type semiconductor, the separation of photoinduced charges is efficiently improved owing to the formation of a built-in electric field across the p-n junction region.⁸ Thus, the p-n junctions formed in the nanocomposite photocatalysts could efficiently promote the photocatalytic performance for water splitting and other photoredox reactions.⁹

Recently, I-III-VI₂ compounds, especially for ternary copper indium diselenide (CuInSe₂) nanocrystals, are recognized as promising materials for photovoltaic and photocatalytic applications owing to the high absorption coefficients, optimal band gap energy, superior radiation stability and low toxicity.¹⁰ CuInSe₂ nanocrystals could be colloiddally prepared at low cost to generate products of highly uniform nanoparticles with controllable size.¹¹ Meanwhile, the optical absorption of CuInSe₂ with the low bandgap of ca. 1.0 eV is located in the region of visible and near-infrared light and greatly matches the solar spectrum.¹² Furthermore, the hybridization of CuInSe₂ with semiconductors has been demonstrated to result in enhanced photocatalytic activity owing to the better visible-light capturing capability and higher

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charge separation efficiency owing to the formation of heterojunction.¹³

Herein, we report on the synthesis, properties, and photocatalytic reactivities of CuInSe₂/CN hybrid photocatalysts. The CuInSe₂ nanocrystals are prepared by a solvothermal approach and then coupled with CN nanorods by self-assembly and thermal treatment method to generate CuInSe₂/CN hybrid photocatalysts. The physicochemical properties of CuInSe₂/CN nanocomposites are investigated by different characterization methods, and the activities of CuInSe₂/CN hybrids in photocatalytic water splitting are explored.

2. Experimental

2.1. Synthesis of polymeric carbon nitride (CN) nanorods

CN nanorods were synthesized according to ref. ⁶. The chiral mesoporous silica template (1 g) was dispersed and stirred in 100 mL 1 mol/L HCl solution at 80 °C for one day, followed by centrifugation and drying at 80 °C overnight. Then the silica template after acidification (0.5 g) and cyanamide (3.0 g) were mixed for 4 h in a flask connected to a vacuum line, and then sonicated at 60 °C for 4 h. Then the mixture was stirred at 60 °C for one day, and washed with H₂O, followed by centrifugation and drying at 80 °C for 12 h. Then the solids were heated at 550 °C for 4 h with a ramping rate of 2.3 °C·min⁻¹ under flowing N₂. Afterward, the obtained yellow hybrids were treated with 4 mol/L NH₄HF₂ solution (**Caution!**) for 12 h to remove the silica template. After being washed with water and ethanol for several times, the solids were dried at 80 °C for 12 h under vacuum. The yield of CN nanorods is ca. 0.24 g.

2.2. Synthesis of CuInSe₂ nanocrystals

CuInSe₂ nanocrystals were synthesized according to the literature.^{9a, 12a} 4 mmol CuCl, 4 mmol InCl₃·4H₂O, 8 mmol Se powder, and 20 mL oleylamine were loaded in a 50 mL three-neck flask. The mixture was heated to 80 °C under an argon atmosphere with stirring for 0.5 h, vigorously stirred and quickly heated to 220 °C and incubated for 4 h, and then naturally cooled down to room temperature. Black CuInSe₂ nanocrystals were isolated by adding 20 mL chloroform, precipitating with 10 mL ethanol, and centrifuging at the speed of 5000 rpm for 10 min. The precipitation/centrifugation/dispersion cycle was repeated for three times. The CuInSe₂ nanocrystals solution was achieved by dispersing the as-obtained sediment in chloroform, while the CuInSe₂ nanocrystals solid was obtained by drying the suspension in an oven at 80 °C overnight.

2.3. Synthesis of CuInSe₂/CN hybrid photocatalyst

The appropriate amount of CuInSe₂ nanocrystals was mixed with 0.2 g CN nanorods powders and 10 mL chloroform. The mixture was stirred for 0.5 h, ultrasonic dispersion for 10 min and stirred with the oil bath heating at 70 °C to get the dry powders. Finally, the powders were heated at 300 °C for 30 min in the nitrogen atmosphere to obtain the CuInSe₂/CN hybrid. The as-prepared composite of CuInSe₂ nanocrystals

modified CN nanorods were marked as x% CuInSe₂/CN, x (x = 1, 3, 5, or 10) is the mass ratio of CuInSe₂ to CN.

2.4. Photocatalytic hydrogen evolution

Photocatalytic hydrogen evolution arrays were performed in a Pyrex top-irradiation reaction vessel linked to a glass closed gas system. Photocatalyst powder (50 mg) was dispersed in an aqueous solution (100 mL) containing triethanolamine (10 vol.%) via sonication for 5 min. 3 wt.% Pt was loaded on the surface of photocatalyst by the in-situ photodeposition approach using H₂PtCl₆·6H₂O. The suspension was evacuated for several times to remove air thoroughly before illuminated under a 300 W Xe-lamp. The wavelength of the incident light was controlled by using an appropriate long pass cut-off filter. The temperature of the reaction solution was kept at 12 ± 1 °C by a flow of cooling water. The generated gases were analyzed by gas chromatography equipped with a thermal conductive detector (TCD) with Argon as the carrier gas. Other experimental sections of synthetic procedure, characterization, are described in ESIT[†].

3. Results and discussion

3.1. Morphology

The surface morphologies of CN after decoration with CuInSe₂ were inspected by scanning emission microscope (SEM) and transmission electron microscope (TEM) techniques. SEM image of CuInSe₂/CN shows the uniform nanorod-like morphology with the length of ca. 1.5 μm and the outer diameter of ca. 0.2 μm (Fig. S1, ESIT[†]). The coupling process with CuInSe₂ does not damage the nanorod morphology of CN (Fig. S2, ESIT[†]). The TEM analysis indicates that small CuInSe₂ nanocrystals with a size of approximately 20 nm are distributed onto the surface of CN nanorods (Fig. 1, Fig. S3-S4, ESIT[†]). The high-resolution transmission electron microscopy (HRTEM) image of CuInSe₂/CN showed that the CuInSe₂ nanoflakes are sporadic and adhered to the surface of CN nanorods compactly. The lattice fringe of *d* = 2.0 Å nm for (220)/(204) plane and the lattice fringe of *d* = 3.4 Å for (112) plane belonged to the tetragonal CuInSe₂ nanocrystals. C, N, Cu, In and Se elements were found in the energy-dispersive X-ray spectroscopy (EDX) spectra, verifying the successful generation of CuInSe₂/CN hybrids (Fig. S5, ESIT[†]).

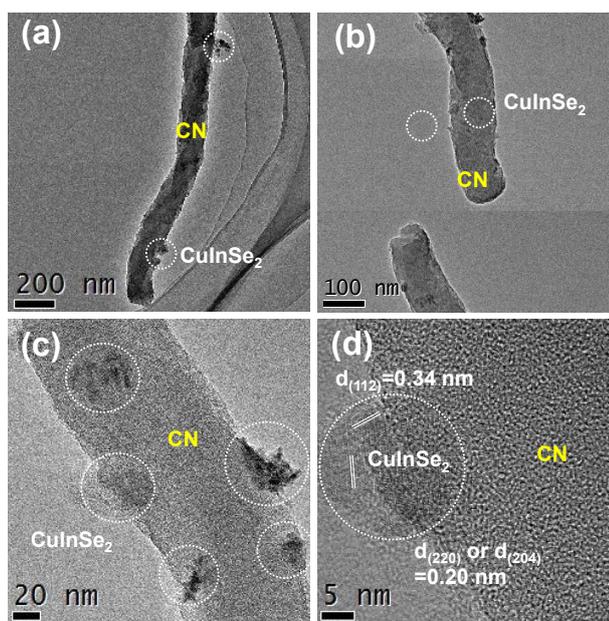


Fig. 1 (a-c) TEM and (d) HRTEM images of $\text{CuInSe}_2/\text{CN}$ hybrids. The locations of CuInSe_2 nanocrystals are marked by white circles.

3.2. Structural characterization

X-ray diffraction (XRD) patterns were tested to reveal the chemical structure of $\text{CuInSe}_2/\text{CN}$ composites. Two diffraction peaks are observed in the XRD pattern of CN nanorods, which are in good agreement with the melon-based polymeric carbon nitride reported in the previous literature.¹⁴ The diffraction peaks at 13.0° and 27.4° correspond to (100) plane with the interplanar separation of 6.72 \AA and the (002) plane with the interlayer d-spacing of 3.36 \AA , respectively (Fig. 2a).¹⁵ The diffraction peaks of the CuInSe_2 sample were indexed to two different phases of tetragonal and cubic CuInSe_2 (Fig. S6a, ESI†). The CuInSe_2 phase marked with '*' have been evolved into the tetragonal crystal structure (JCPDS 040-1487) with its lattice parameters $a = b = 5.786 \text{ \AA}$ and $c = 11.627 \text{ \AA}$, while the phase marked with '#' have been evolved into the cubic crystal system (JCPDS 023-0208) with its lattice parameters $a = b = c = 5.58 \text{ \AA}$.¹⁶ As shown in Table S1, EDS from fields of CuInSe_2 nanocrystals gave an average Cu/In/Se composition of 0.38:0.17:0.45, which is near the target 0.25:0.25:0.5 ratio, considering the error of the EDS detector (approximately 2 atom %). The XRD patterns of $\text{CuInSe}_2/\text{CN}$ composites present the characteristic diffraction peaks that belonged to both CN and CuInSe_2 , reflecting the successful hybridization of CuInSe_2 and CN.

The Fourier transform infrared (FT-IR) spectra of $\text{CuInSe}_2/\text{CN}$ composite were presented in Fig. 2b. The peaks at 810 cm^{-1} and $1200\text{-}1600 \text{ cm}^{-1}$ in the FT-IR spectra of CN and $\text{CuInSe}_2/\text{CN}$ are indexed as the breathing mode of s-triazine units and the stretching mode of carbon and nitrogen heterocycles, respectively.¹⁷ Besides, all samples possessed the broad bands in the regions of 2360 cm^{-1} and $3000\text{-}3800 \text{ cm}^{-1}$,

originating from the physically adsorbed carbon dioxide and water in the atmosphere, respectively. DOI: 10.1039/D0DT00865F

The chemical state of the $\text{CuInSe}_2/\text{CN}$ hybrid was assessed by the X-ray photoelectron spectroscopy (XPS). Six major elements (C, N, O, Cu, In and Se) are examined in the XPS survey spectra of $\text{CuInSe}_2/\text{CN}$ hybrid (Fig. 3). In the C 1s core level spectrum, the two peaks located at 284.8 eV and 288.0 eV correspond to sp^2 C-C bonds and sp^2 -hybridized carbon in the N-containing aromatic ring (N-C=N) of CN, respectively.¹⁸ In the N 1s core level spectrum, the two peaks centering at 398.5 eV and 400.0 eV belong to the sp^2 -hybridized nitrogen in triazine rings (C=N=C) and tertiary nitrogen N-(C)₃ groups of heptazine heterocyclic ring units for CN, respectively. The peak at 401.2 eV corresponds to amino groups (C-N-H) owing to the incomplete polymerization of CN. Additionally, the binding energies of Cu 2p 3/2 and Cu 2p 1/2 at 954.3 eV and 935.0 eV are assigned to the oxidation state of Cu^+ . The binding energies of In 3d 3/2 and In 3d 5/2 at 453.2 eV and 445.6 eV are also in good accordance with the oxidation state of In^{3+} . The binding energies of at 62.5 eV and 57.7 eV correspond to Se 3d 3/2 and Se 3d 5/2, respectively.¹⁹ The O 1s core level at 532.9 eV (Fig. 3g) is ascribed to surface adsorbed water. No additional signal corresponding to the C-O or N-O bond is observed for the $\text{CuInSe}_2/\text{CN}$ hybrid.²⁰ The determination of C, N, O, Cu, In and Se elements confirms the formation of $\text{CuInSe}_2/\text{CN}$ hybrid.

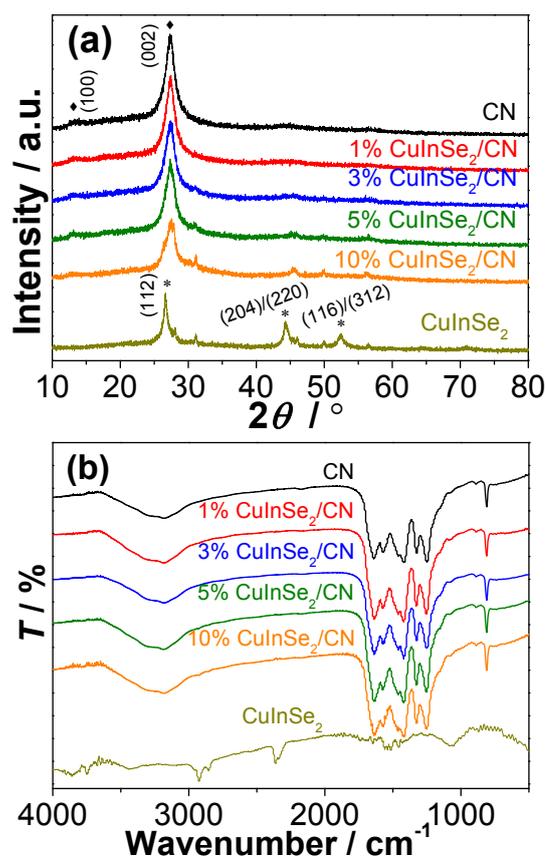


Fig. 2 (a) XRD patterns, and (b) FT-IR spectra of $\text{CuInSe}_2/\text{CN}$ hybrids.

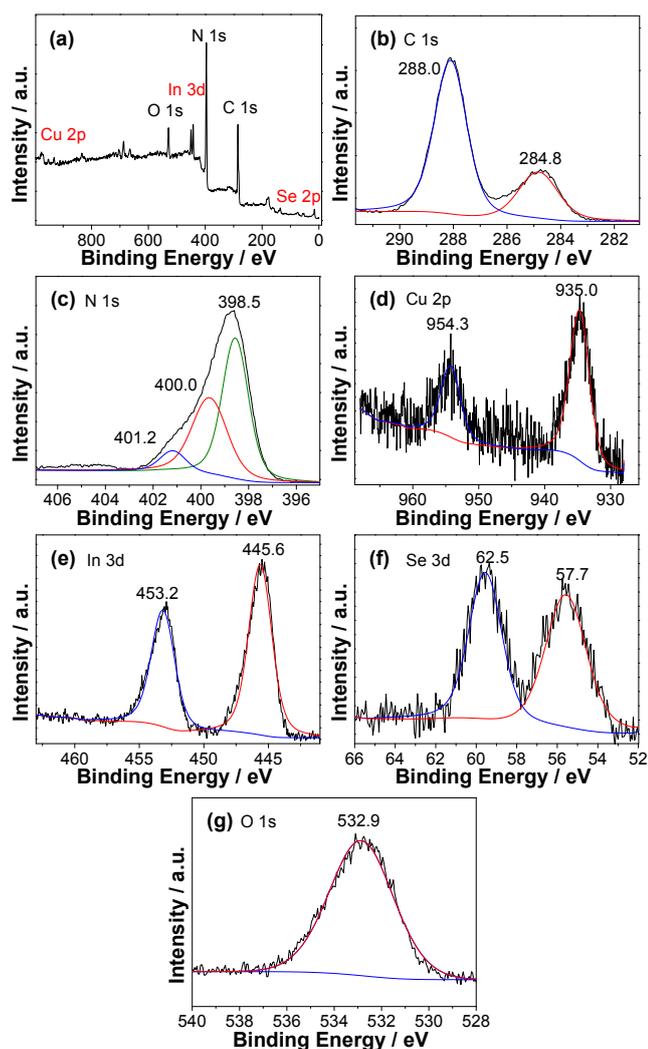


Fig. 3 (a) XPS survey, (b) C 1s, (c) N 1s, (d) Cu 2p, (e) In 3d, (f) Se 3d and (g) O 1s spectra of 3% CuInSe₂/CN.

3.3. Optical properties

The optical properties of CuInSe₂/CN were studied by ultraviolet and visible diffuse reflectance spectroscopy (UV-Vis DRS). Fig. 4a presents the UV-Vis DRS spectra of bare CN and CuInSe₂/CN hybrids. Pristine CN exhibits an optical absorption with an edge at 460 nm. Compared to pure CN, all the CuInSe₂/CN hybrids present stronger absorption capability in visible-light section ascribed to the absorption of CuInSe₂ (Fig. S6b-c, ESI[†]). As the CuInSe₂ content increases, the absorbance of the CuInSe₂/CN samples enhances gradually. These results demonstrate the greatly improved light-harvesting capability after incorporating with CuInSe₂, which is favorable for the optimized photocatalytic performance. The changes of light absorption ability can also be demonstrated by the color transition of the samples. The color of CuInSe₂/CN hybrid shifts from earthy yellow to grayish yellow with increasing CuInSe₂ content (Fig. S7a, ESI[†]).

The separation efficiency of photoinduced charge-carriers was explored by photoluminescence spectra (PL). The maximal PL intensity of pristine CN among all these samples indicates

the high exciton energy of CN (Fig. 4b). The PL intensities of CuInSe₂/CN hybrids are much lower than that of the pristine CN due to the inhibited energy-wasteful process by introducing CuInSe₂ onto CN surface to construct an ideal composite system. With the increasing content of CuInSe₂, the PL intensities of the CuInSe₂/CN composites reduced gradually. According to the time-resolved photoluminescence spectra (Fig. S7b, ESI[†]), the average lifetime (τ_{av}) of CN and 3% CuInSe₂/CN composite are determined to be 1.56 ns and 1.14 ns, respectively (Table S2, ESI[†]). The shortened emission lifetime confirms the facilitated separation of photogenerated charge carriers of CuInSe₂/CN nanocomposites. Coating with CuInSe₂ indeed inhibits the recombination of photogenerated charge carriers.

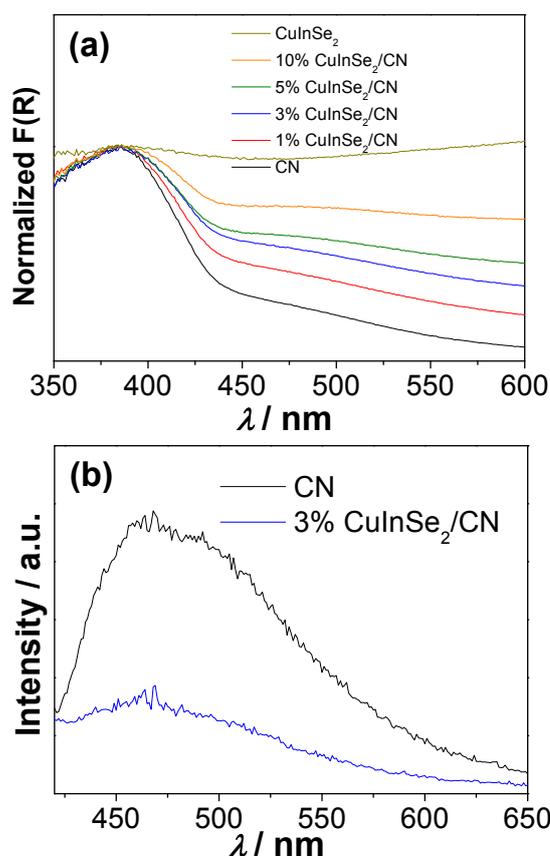


Fig. 4 (a) UV-Vis DRS spectra and (b) PL spectra of CuInSe₂/CN samples.

3.4. Photoelectrochemical measurements

The charge separation ability of CuInSe₂/CN was studied by the photo-electrochemical tests (Fig. 5). 3% CuInSe₂/CN sample exhibited an enhanced transient photocurrent response than pristine CN. Electrochemical impedance spectroscopy in the dark revealed that 3% CuInSe₂/CN shows a significantly decreased diameter in comparison with CN. These results illustrate the promoted separation of photoexcited charge carriers of CuInSe₂/CN hybrid.

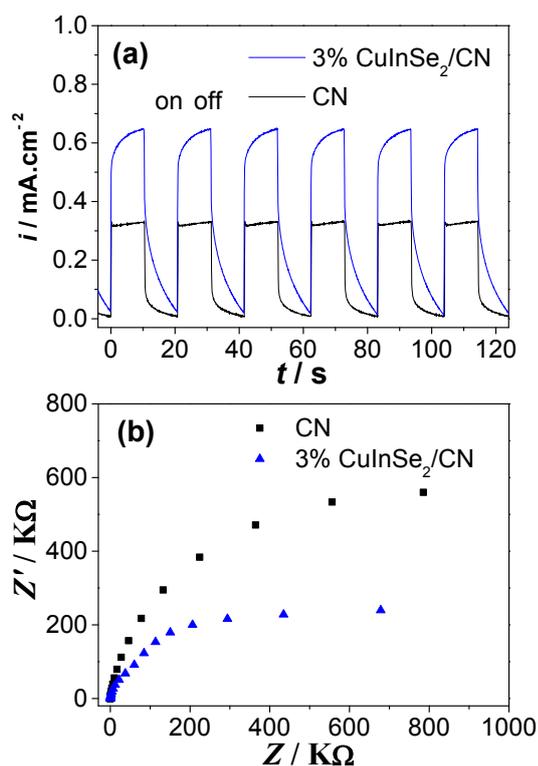


Fig. 5 (a) Periodic on/off photocurrent response, and (b) Nyquist plots of electrochemical impedance spectroscopy of $\text{CuInSe}_2/\text{CN}$ samples.

3.5. Photocatalytic H_2 evolution activities

The photocatalytic reactivities of $\text{CuInSe}_2/\text{CN}$ hybrids in water splitting were tested. As shown in Fig. 6a, the hydrogen evolution rate (HER) of bare CN nanorods is only $74 \mu\text{mol h}^{-1}$. The HER of CuInSe_2 is negligible ($0.3 \mu\text{mol h}^{-1}$) owing to the poor dispersion of bare CuInSe_2 in water solution and the severe recombination of photogenerated charge carriers. The influence of CuInSe_2 content on the HER of $\text{CuInSe}_2/\text{CN}$ hybrid photocatalysts was evaluated. The HER increases initially with the increasing CuInSe_2 amount and then reaches a maximum when the amount of CuInSe_2 loaded is about 3%. Notably, the 3% $\text{CuInSe}_2/\text{CN}$ sample presents the maximum HER of $240 \text{ mol}\cdot\text{h}^{-1}$, which is ca. 3 times higher than that of bare CN.

Moreover, the catalyst lifetime is also of great significance besides the photocatalytic activity. A recycling test of the photocatalytic activity of 3% $\text{CuInSe}_2/\text{CN}$ hybrid was carried out (Fig. 6b). No significant variation was found in the photocatalytic activity from the first run to the fifth run. Furthermore, almost no changes were observed in the chemical structure and morphology of 3% $\text{CuInSe}_2/\text{CN}$ before and after photocatalytic water splitting reactions, as demonstrated by the result of XRD, FT-IR, and SEM analysis (Fig. S8-S9, ESI[†]). These results confirm the good stability of $\text{CuInSe}_2/\text{CN}$ hybrid in photocatalytic water splitting. Fig. 6c shows the wavelength-dependent HERs of 3% $\text{CuInSe}_2/\text{CN}$. The photocatalytic activities of 3% $\text{CuInSe}_2/\text{CN}$ are in good accordance with its UV-Vis DRS spectrum, demonstrating that

the water splitting reaction is driven by the captured photon for $\text{CuInSe}_2/\text{CN}$ hybrid.

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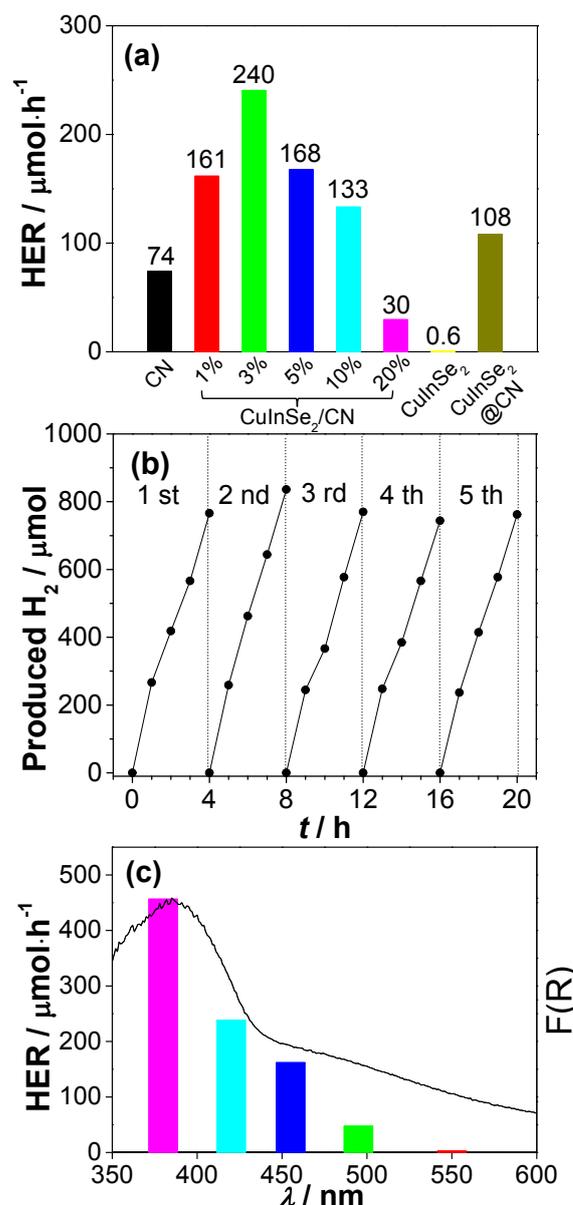


Fig. 6 (a) The HERs of $\text{CuInSe}_2/\text{CN}$ samples with different amount of CuInSe_2 , (b) Long-term water splitting test of 3% $\text{CuInSe}_2/\text{CN}$ hybrid with visible light ($\lambda > 420 \text{ nm}$) illumination, and (c) HERs of 3% $\text{CuInSe}_2/\text{CN}$ hybrid under the light irradiations with different wavelength.

3.6. Band structure and charge separation process

The energy band structures of CN and CuInSe_2 are investigated. As shown in the Mott-Schottky plots (Fig. S10, ESI[†]), the positive slope of CN corresponds to an n-type semiconductor, while a negative slope of CuInSe_2 corresponds to a p-type semiconductor. The flat-band potentials of CN and CuInSe_2 were estimated to be -0.98 and 1.11 V vs. RHE , which were close to the conduction band positions (E_{CB}) of CN and valence band positions (E_{VB}) of CuInSe_2 , respectively. As shown

in Fig. S11c, a positive slope was observed for 3% CuInSe₂/CN, and the flat band potential was determined to be -0.93 V vs. RHE, which was more positive than that of CN. The optical band gaps (E_g) of 3% CuInSe₂/CN hybrids, CN nanorods, and CuInSe₂ nanocrystals were estimated from Tauc plots of $(\alpha h\nu)^2$ vs. photon energy ($h\nu$) (Fig. S11, ESI[†]). The E_g of CN nanorods, 3% CuInSe₂/CN hybrids, and CuInSe₂ nanocrystals were respectively determined to be 2.72 eV, 2.65 eV, and 1.75 eV.^{16c} According to the equation: $E_{VB} = E_{CB} + E_g$, the E_{VB} of CN nanorods and 3% CuInSe₂/CN hybrids were calculated to be 1.74 and 1.72 V vs. RHE (close to the determined E_{VB} of XPS valence band spectra in Fig. S12, ESI[†]), and the E_{VB} of CuInSe₂ nanocrystals was calculated to be -0.64 V vs. RHE.

The energy band structures and photoinduced charge transfer pathways in the CuInSe₂/CN composites are put forward in Fig. 7. Both the photoinduced electron-hole pairs are generated for CN and CuInSe₂ with visible light illumination. When CN is in contact with CuInSe₂ to form a heterojunction, their Fermi energy level (E_F) reach equilibration. Meanwhile, the energy bands of CN shift downward along with the E_F , whereas the energy bands of CuInSe₂ shift upward along with the E_F . As a result of the built-in electric field, the electrons in the CB of CuInSe₂ transfer to the CB of CN, whereas the holes in the VB of CN migrate to the VB of CuInSe₂. The electrons accumulated on the CB of CN will transfer to the Pt nanocrystals and then initiate the H₂ evolution reaction. Since the valence band edge potential of 3% CuInSe₂/CN hybrid (1.72 V vs. RHE) and CuInSe₂ (1.11 V vs. RHE) are more positive than the redox potential of TEOA/TEOA⁺ (0.82 V vs. RHE), the holes accumulated on the VB of CuInSe₂ can oxidize triethanolamine (TEOA).²¹ Thus, the formation of p-n junctions in CuInSe₂/CN hybrid photocatalyst indeed inhibits the recombination of charge carriers, thus greatly benefiting the photocatalytic reactivity for water splitting.

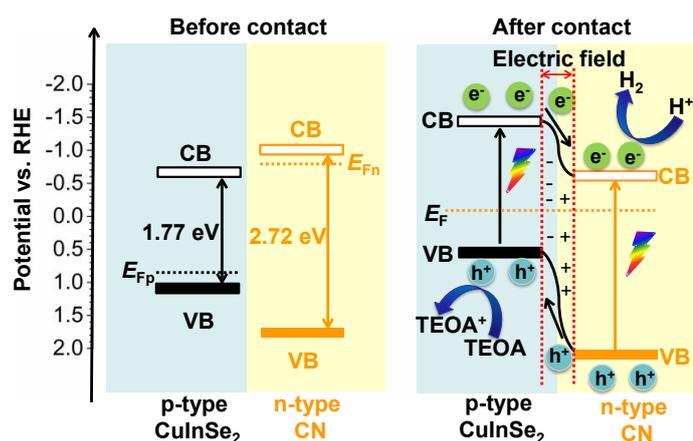


Fig. 7 Energy band structures of CuInSe₂ and CN before formation of the heterojunction and the band structure for CuInSe₂/CN heterojunction and charge separation process under illumination

4. Conclusions

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In summary, n-type CN nanorods are coupled with p-type CuInSe₂ nanocrystals through the self-assembly and thermal treatment process to produce p-n heterojunction photocatalyst for water splitting. The CuInSe₂/CN nanocomposite photocatalyst exhibits greatly increased HER than the individual counterparts. The optimum hydrogen evolution rate of 3% CuInSe₂/CN reaches 240 $\mu\text{mol}\cdot\text{h}^{-1}$, which is 3-fold higher than that of CN nanorods. The better photocatalytic performance of the CuInSe₂/CN hybrid originates from the improved visible-light utilization capability and the accelerated separation of photoexcited electron-holes. This investigation paves new avenues for the construction of p-n heterojunction photocatalysts for sustainable photoredox reactions.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

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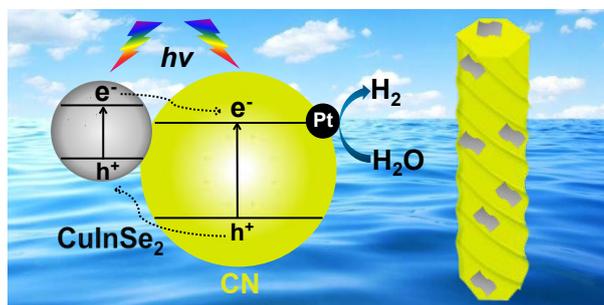
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