

Article

# Taking Advantage of Disorder: Small-Molecule Organic Glasses for Radiation Detection and Particle Discrimination

Joseph S. Carlson, Peter Marleau, Ryan A Zarkesh, and Patrick L Feng

J. Am. Chem. Soc., Just Accepted Manuscript • Publication Date (Web): 20 Jun 2017

Downloaded from http://pubs.acs.org on June 20, 2017

# Just Accepted

"Just Accepted" manuscripts have been peer-reviewed and accepted for publication. They are posted online prior to technical editing, formatting for publication and author proofing. The American Chemical Society provides "Just Accepted" as a free service to the research community to expedite the dissemination of scientific material as soon as possible after acceptance. "Just Accepted" manuscripts appear in full in PDF format accompanied by an HTML abstract. "Just Accepted" manuscripts have been fully peer reviewed, but should not be considered the official version of record. They are accessible to all readers and citable by the Digital Object Identifier (DOI®). "Just Accepted" is an optional service offered to authors. Therefore, the "Just Accepted" Web site may not include all articles that will be published in the journal. After a manuscript is technically edited and formatted, it will be removed from the "Just Accepted" Web site and published as an ASAP article. Note that technical editing may introduce minor changes to the manuscript text and/or graphics which could affect content, and all legal disclaimers and ethical guidelines that apply to the journal pertain. ACS cannot be held responsible for errors or consequences arising from the use of information contained in these "Just Accepted" manuscripts.



Journal of the American Chemical Society is published by the American Chemical Society. 1155 Sixteenth Street N.W., Washington, DC 20036 Published by American Chemical Society. Copyright © American Chemical Society. However, no copyright claim is made to original U.S. Government works, or works produced by employees of any Commonwealth realm Crown government in the course of their duties.

2 3

4 5

6 7

8 9 10

11

12

13

14

15

16

17

18

19

20

21 22 23

24

25

26

27

28

29

30

31

32

33

34

35

36

37

38

39

40

41

42

43

44

45

46

47

48

49

50

51

52

53

54

55

56

57

58

59

60

# Taking Advantage of Disorder: Small-Molecule Organic Glasses for **Radiation Detection and Particle Discrimination**

Joseph S. Carlson, Peter Marleau, Ryan A. Zarkesh, Patrick L. Feng\*

Sandia National Laboratories, 7011 East Ave., Livermore, CA 94550

ABSTRACT: A series of fluorescent silyl-fluorene molecules were synthesized and studied with respect to their photophysical properties and response toward ionizing neutron and gamma-ray radiation. Optically transparent and stable organic glasses were prepared from these materials using a bulk melt-casting procedure. The prepared organic glass monoliths provided fluorescence quantum yields and radiation detection properties exceeding the highest-performing benchmark materials such as solution-grown trans-stilbene crystals. Co-melts based on blends of two different glass-forming compounds were prepared with the goal of enhancing the stability of the amorphous state. Accelerated aging experiments on co-melt mixtures ranging from 0-100% of each component indicated improved resistance to recrystallization in the glass blends, able to remain fully amorphous for >1 month at 60°C. Secondary dopants comprising singlet fluorophores or iridium organometallic compounds provided further improved detection efficiency, as evaluated by light yield and neutron/gamma particle discrimination measurements. Optimized singlet and triplet doping levels were determined to be 0.05 wt. % 1,4-bis(2-methylstyryl)benzene (bis-MSB) singlet fluorophore and 0.28 wt. % Ir<sup>3+</sup>, respectively.

### INTRODUCTION

Luminescent organic materials have attracted significant recent attention due to their role in a variety of functional devices. One particular area of interest comprises organic light emitting diodes (OLEDs), which operate based on charge injection and radiative recombination in multi-layer devices.1 A related but more specialized application that is based on similar principles of charge transport and exciton mobility involves organic materials for the detection of ionizing radiation. This application serves as the basis for nuclear non-proliferation detection of illicit nuclear materials such as highly enriched uranium or plutonium. By analogy to OLEDs, the performance of an organic scintillator is controlled by the efficiency and kinetics of radiative decay processes that are associated with ion recombination and exciton transport. The ideal scintillator material can be characterized as having:

- 1) minimal optical self-absorption
- 2) fast emission timing (fast counting)
- 3) emission wavelength matching to photodetector (typically 300 to 500 nm)
- 4) high conversion of radiation energy to visible light (light yield)
- 5) discrimination between different types of radiation
- 6) energy resolution/spectroscopy

7) amenable to large sizes and various form factors, low cost, strength, stability

For these reasons, we describe here our efforts to improve the radiation detection performance and long-term stability of organic-based scintillators via control over the molecular and bulk properties of transparent organic glasses.

Organic scintillators possess the unique ability to discriminate ionization caused by fast neutron recoils on nuclei from that caused by Compton scattering of gamma-rays on electrons, owing to differences in the emission kinetics of the produced light pulses. These differences are evident in the relative fraction of light produced via prompt singlet fluorescence versus delayed triplet-triplet annihilation (TTA). In practice, nuclear recoils from fast neutron interactions produce a greater proportion of delayed luminescence than gamma-rays and can be identified by their characteristic pulse shape. This is due to a phenomenon known as ionization quenching, which leads to a reduction in the relative proportion of prompt fluorescence for neutron versus gamma-ray events.<sup>2</sup> This technique for identifying the type of incident particle is known as pulse-shape discrimination (PSD).<sup>3</sup> In mixed fluorophore systems, such as plastics and liquids, Förster resonant energy transfer also plays a role.

While this pulse-shape discrimination technique is effective in some materials such as trans-stilbene single crystals or liquid scintillation mixtures, there are several limitations that preclude their use in critical applications such as radiation portal monitors used at border crossings and ports-of-entry. First, PSD is easily disrupted by the presence of disorder or impurities. This is primarily due to a reliance upon TTA to provide the delayed emission component. TTA is a bimolecular recombination process that requires Dexter electronic interaction between two triplet excited states, the probability of which decreases exponentially as a function of distance. The presence of disorder or impurities decreases the effective triplet exciton lifetime due to a higher density of trapping sites that compete with TTA. Indeed, Arulchakkaravarthi et al. used X-ray rocking curve analysis to correlate degradation in the light yield and PSD with decreased structural order in melt-grown single crystals.4 Related work by Carman et al. has shown a sensitive dependence of the PSD upon the purity and degree of crystallographic perfection in solution-grown trans-stilbene.5 Knowledge gained from these studies have led to commercialization of high-quality trans-stilbene single crystals, although several limitations ACS Paragon Plus Environment

cost, limited size, and highly anisotropic scintillation response.<sup>6</sup>

The combination of these limitations has led to significant interest in non-crystalline organic scintillators based on polymers and organic liquids.<sup>2</sup> Various strategies have been employed to achieve neutron/gamma PSD in these materials, although the obtained scintillation light yields and discrimination performance have been found to be distinctly inferior to single crystals.<sup>7,8</sup> In this work, we report on a new class of amorphous organic scintillator that exceeds the scintillation performance of *trans*stilbene while enabling rapid fabrication via a bulk meltcasting procedure.

#### RESULTS AND DISCUSSION

# Organic Glass Scintillators: Stability and Performance on Practical Sizes

Considering that the photophysics of organic lightemitting diode (OLED) emitters<sup>1</sup> are strikingly similar to the principles of radioluminescence in scintillators, we evaluated fluorene as a potential fluorophore group. Despite possessing high quantum yields with a blue (ca. 420 nm) emission profile, fluorene and its derivatives have not been widely explored in the context of scintillators. Fluorene possesses an advantage in that it can be readily derivatized at the benzylic position to provide oxidative stability and enable facile modification of properties.<sup>9</sup>



Figure #. Glass forming compounds 1 - 3.

We found that compounds **1** – **3** based on fluorene chromophores were not only capable of providing greater scintillation light yields and PSD performance than *trans*-stilbene, but also were capable of being melted and cast into a transparent glasses in sizes relevant to radiation detection.<sup>10,11</sup> These finding provided the opportunity to explore several key objectives: (1) ability to circumvent single-crystal growth procedures as a route towards larger radiation detector elements, (2) ability to improve the photophysical properties via control over secondary dopant concentrations, (3) ability to achieve isotropic optical, mechanical, and transport properties.<sup>12</sup>

#### **Co-melt Formulation for Indefinite Stability**

Comparing scintillators of varying sizes can be challenging due to the fact that some performance variables are size dependent. For example, with overlap of the excitation and emission bands of the glass matrix, reabsorption of the scintillation light can decrease performance as a sample becomes larger, making small samples appear artificially higher performing. Alternatively, small samples have a greater chance of the secondary particle escaping the sample, resulting in incomplete energy deposition and therefore an artificially lower light yield measurement. Due to its higher mass, the secondary proton produced from neutron nuclear recoils will deposit ca. two orders of magnitude more of its energy over a given distance and as a result travel 1/100<sup>th</sup> the distance as an electron of similar energy and is therefore more likely to completely deposit its energy, even in small samples.<sup>2</sup>

Casting amorphous thin films with long term resistance to crystallization is a challenge frequently encountered in the OLED field that directly impacts the lifetime and conditions under which the device can be operated. To the best of our knowledge, small-moleculebased organic glasses have never been implemented in large scale applications (ca. > 1 cm<sup>3</sup>).<sup>13</sup> To make an accurate comparison between our formulation and the state of the art technology, samples had to be prepared on similar sizes to a reference material. To this end, a 2-gram glass sample of 1 was prepared to coincide with the size of benchmark materials on hand. After several weeks at room temperature the sample slowly began to turn opaque on the surface and eventually the entire sample, resulting from crystallization that was confirmed by powder X-ray diffraction (XRD) (See Supporting Information). Delayed crystallization would make organic glasses ineligible for practical use in the field. The observation of a glass transition temperature (T<sub>g</sub>) and lack of solid to liquid transition (T<sub>m</sub>) after multiple heating and cooling cycles in a DSC experiment is frequently used to qualitatively assess the stability of a glassy material. It is apparent that this is not true over long durations due to potential kinetic instability of the glassy state.

The glass to crystal transition can proceed by two modes: a glass-to-crystal transition originating from the bulk or originating from the surface.<sup>14</sup> There are known inhibition methods for both types of growth. For example, surface growth can be inhibited by applying a coating,<sup>15</sup> while bulk growth can be inhibited by polymer additives<sup>16</sup> or by blending compounds of similar structure.<sup>17</sup> Increased stability can also be gained when a glass is formed via vapor deposition.<sup>18</sup>

Table 1. Co-melt formulation effect on Tg and stability

1 : 2 (w/w) <sup>a</sup>	Tg (°C) <sup>b</sup>	Transparency at 80 °C <sup>c</sup>	
100 : 0	72.6	< 24 hours	
90 : 10	75.2	> 4 weeks	
70:30	80.2	> 4 weeks	
50 : 50	81.0	> 4 weeks	
30:70	89.4	> 4 weeks	
10:90	95.1	< 72 hours	
0 : 100	98.9	< 24 hours	

<sup>*a*</sup>200 mg samples. <sup>*b*</sup>T<sub>g</sub> value at the onset of the transition. <sup>*c*</sup>Evaluated by observation using a jeweler's loop.

At this juncture we instead sought to improve the stability of the glass scintillators via a formulation approach.

1

2

Since observation of T<sub>g</sub> in DSC is not a measure of long term glass stability, each sample was subjected to accelerated aging conditions; whereby, samples were aged at temperatures close to their T<sub>g</sub>, a technique known to accelerate crystallization by several orders of magnitude.<sup>14</sup> We found that mixtures of compounds **4** and **5** were indefinitely resistant to crystallization, even under these conditions (Table 1).



**Figure 2.** A 2 gram glass sample of 1 and 2 (90:10 w/w) after aging at 60 °C for 72 hours.

In a larger scale experiment, a formulation of 1 and 2 demonstrated the same resistance towards crystallization (Figure 2).

#### Singlet wavelength shifters

The nature of a glassy medium enables facile tuning of the fluorescence properties via incorporation of secondary dopants. Indeed, by casting a glass of 3 with 0.05% of 9.10diphenylanthracene (DPA) the Stokes' shift was increased to possess less self-absorption than undoped microcrystalline 3, indicative of efficient Förster Resonance Energy Transfer (FRET) (Figure 3). While it was observed that light yield increases in conjunction with DPA concentration, after a threshold concentration the particle discrimination ability diminished (Figure 4). On the 200 mg samples tested, it was determined that the maximum benefit for light yield and PSD was obtained with a DPA concentration of 0.05% (w/w), outperforming stilbene in both respects. A survey of other wavelength shifters emitting in the 400 - 500 nm range demonstrated a similar relationship between performance and concentration, with ca. 0.05% being the optimum concentration (vide infra).



**Figure 3.** Photoluminescence excitation and emission spectra of **3** in the microcrystalline form (red) and glass undergoing FRET with DPA (blue).



**Figure 4.** DPA doping concentration vs light yield and PSD in glass samples of **1**.

#### **Triplet Harvesting Dopants**

Taking further inspiration from the OLED literature and similar work in radioluminescent plastics,<sup>8,19,20</sup> we experimented with the radioluminescence response of glasses containing iridium triplet harvesting dopants. This strategy has been shown to be effective in maximizing the light yield when incorporated in polymer-based host materials at Ir<sup>3+</sup> concentrations of 0.8-5.5 wt%. Incorporating Bis[2-(4,6-difluorophenyl)pyridinato-C<sup>2</sup>,*N*](picolinato) iridium(III) (FIrpic) in an organic glass provided high scintillation light yields at Ir3+ concentrations as low as 0.03 wt% (Figure 5). The highest light yields were achieved at 0.13-0.28 wt% Ir3+, yielding 1.8-2 times the scintillation light output relative to stilbene. This result is notable since it corresponds to a light yield that is four times greater and an Ir<sup>3+</sup> doping level six times smaller than for a corresponding scintillator based on the OLED host material poly(9-vinylcarbazole).20 The presence of Ir<sup>3+</sup> confers a long fluorescence lifetime on the order of 1 µs, with pulse shapes possessing a very diminished prompt emission peak. The dominance of delayed emission precludes the ability to do particle discrimination;



**Figure 5**. Plot of the relative <sup>137</sup>Cs scintillation light yield as a function of Ir<sup>3+</sup> dopant concentration. The glass matrix was a 9:1 mixture of **1:2** and the Ir<sup>3+</sup> complex was FIrpic.

however, the increase in light yield at such a low  $Ir^{3+}$  concentration was an interesting discovery in its own right as one of the most sensitive radioluminescent organic materials known.<sup>2</sup>

# Radioluminescence of Organic Glasses: Light Yield, PSD, Radiation Timing, Neutron Response

### **Comparison to Benchmark Materials**

Using the best formulation from the co-melt and wavelength shifter concentration studies, we were able to cast 2 gram scintillators containing various dopants (Figure 6) for comparison to benchmark materials of similar dimension (Table 2). Apart from concentration, the identity of the wavelength shifter also had a significant impact of fluorescence and PSD. To our satisfaction, the fluorescence lifetime of singlet emitting doped glasses were faster than stilbene, with the exception of **G2**.



**Figure 6.** Singlet wavelength shifters and Iridium triplet harvesting dopant.

Sample <sup>a</sup>	Dopant (w/w)	Lifetime <sup>c</sup> (ns)	QY <sup>d</sup> (Φ)	Light Yield <sup>e</sup>	PSD <sup>f</sup>
Stilbene <sup>b</sup>	-	2.41	1.00	1.00	3.15
EJ-301	-	2.00 (89%) 16.21 (11%)	0.81	0.63	2.57
EJ-200	-	1.50	0.95	0.63	0.66
Gı	-	1.68	1.05	0.62	2.48
G2	0.05% DPA	8.09	0.74	1.11	2.53
G3	0.05% POPOP	1.52	0.92	1.11	3.56
G4	0.05% DPVBi	1.50	1.79	0.88	3.06
G5	0.05% MSB	1.45	1.50	1.10	3.73
G6	0.07% MSB	1.53	1.61	1.14	3.56
G7 <sup>g</sup>	1% FIrpic	1.01 µs	0.40	2.00	N/A

Table 2. Comparison of radioluminescence of 2 gramglass formulation to known standards

<sup>a</sup>Glass samples composed of 9:1 mixture of 1:2 <sup>b</sup>Single Crystal of similar size <sup>c</sup>Fluorescence lifetime measured at emission maximum <sup>d</sup>QY relative to stilbene, measured at the sample absorption maximum <sup>e</sup>Cs-137 relative light yield (478 keVee) <sup>f</sup>PSD-FOM at 478 keVee) <sup>g</sup>200 mg sample.

From this data,  $G_5$  was identified as possessing the best performance with 10% greater light yield and 18%

greater PSD than stilbene. Further analysis of a histogram plot of the data after pulse-processing revealed some interesting features (Figure 7a, b). There is excellent separation between the neutron and gamma bands down to low energy events, a desirable feature not present in organic liquids and plastics. When the binned histogram data is analyzed using the FOM definition and plotted as a function of energy, it is clear that the PSD capability of **G5** outperforms stilbene across the energy range (Figure 7c).

Another often overlooked aspect of scintillator performance is the neutron light yield. Since it is difficult to subject samples to a calibrated neutron energy source, a gamma calibrated PSD plot is typically reported using gamma-ray sources of known energies (i.e. keVee scale units). Often times a scintillator can be formulated to possess a higher PSD-FOM, but at the expense of the neutron sensitivity and light yield. To evaluate neutron sensitivity without utilizing a calibrated neutron source, we simply integrated the counts from the histogrammed data across the relevant energy region with respect to the PSD parameter for G5 and compared it to stilbene under identical conditions (Figure 7d). The larger neutron:gamma integral for G5 reveals that the high PSD-FOM values for G5 does not come at the expense of neutron sensitivity/light yield; in fact, the neutron sensitivity/light yield is higher than stilbene. Since fast neutrons have the highest interaction cross-section with hydrogen atoms, the neutron response is dependent on the hydrogen content of a scintillator. The identical H/C ratios (0.857) for both G5 and stilbene indicate that the improvement in neutron scintillation efficiency is attributed to the scintillation properties and not due to a difference in molecular formula.

Fast scintillation timing is also important for a number of radiation detection applications including correlated particle counting and imaging.<sup>21</sup> To demonstrate that the faster fluorescence timing of **G5** was duplicated in the scintillation response, a time correlated single-photon counting experiment was used to construct pulse shapes under <sup>137</sup>Cs gamma irradiation (See Supporting Information). As compared to stilbene and EJ-200 reference materials, the glass samples possess faster scintillation rise-time and decay characteristics. This is promising for applications such as active interrogation that demand fast counting as well as neutron discrimination.<sup>22</sup>

## Scale up and Practical Application

To demonstrate scalability, a casting of **G1** was made in a size and shape that is compatible with many photomultiplier tubes. The resulting sample possessed very good optical transparency and incurred no damage while handling (Figure 8). This observation was encouraging in that practical radiation detection with these materials are possible as a replacement for existing crystalline, plastic, and liquid scintillators. There is also the potential of casting into form factors for applications that are not yet realized.



**Figure 7**. PSD histograms for a) *trans*-stilbene crystal and b) **G5** glass obtained using an AmBe mixed n/γ-ray source. c) Plot of the neutron/gamma pulse-shape discrimination figure-of-merit as a function of gamma-ray energy for **G5** glass (red) in comparison to three reference scintillators. d) Total integral of neutron and gamma-ray counts in the energy range of 75-2100 keVee for *trans*-stilbene crystal (black) and **G5** glass (red).



Figure 8. A 2" diameter, ½" thick casting of G1 (19 grams).

#### CONCLUSION

The current work is the first example of organic glasses as a new class of materials for radiation detection. The development of a high quantum yield matrix in an easily processable form factor is the next evolutionary step in the field of radiation detection and discrimination. The ease by which additives can be incorporated into the matrix imparts versatility in designing new types of scintillators. As compared to benchmark materials, improvements were made across all metrics we measured, including light yield, pulse-shape discrimination, neutron response and timing. The combination of these improvements can lead to radiation detectors with better detection efficiency, which enhance national security and non-proliferation mission capabilities.

To the best of our knowledge, this is also the first example of a large (> 1 gram), indefinitely stable smallmolecule organic glass, which was achieved through formulation of two different scintillating compounds. This stable formulation provides a platform for many possibilities, including further development of high Z-doped glasses for gamma-ray spectroscopy.

# ASSOCIATED CONTENT

The Supporting Information is available free of charge on the ACS Publication website at DOI: **###**, which includes: experimental procedures for the synthesis and casting of glasses, photoluminescence spectra, radioluminescence data, DSC spectra and X-ray crystallographic data.

# AUTHOR INFORMATION

Corresponding Author

\*plfeng@sandia.gov

# Author Contributions

The manuscript was written through contributions of all authors.

# **Funding Sources**

This work was supported by the office of NA-22, NNSA, U.S. Department of Energy. Sandia National Laboratories is a multi-mission laboratory managed and operated by National Technology and Engineering Solutions of Sandia, LLC., a wholly owned subsidiary of Honeywell International, Inc., for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-NA0003525.

# Notes

The authors declare no competing financial interest.

# ACKNOWLEDGMENT

We would like to thank F. Patrick Doty and Mateusz Monterial for lively discussions.

# REFERENCES

(1) For a review of light-emitting diodes structure and function see a) Shirota, Y.; Kageyama, H. *Chem. Rev.* **2007**, *107*, 953–1010. b) Walzer, K.; Maennig, B.; Pfeiffer, M.; Leo, K. *Chem. Rev.* **2007**, *107*, 1233–1271. c) Jou, J.-H.; Kumar, S.; Agrawal, A.; Li, T.-H.; Sahoo, S. J. Mater. Chem. C. **2015**, 3, 2974–3002.

(2) Knoll, G. F. *Radiation Detection and Measurement, 4*<sup>th</sup> ed.; John Wiley & Sons: Hoboken, NJ, 2010.

(3) a) Berlman, I.B.; Steingraber, O.J. *Nucl. Instr. Meth. A* **1973**, 108, 587-591. b) Kaschuck, Y.; Esposito, B. *Nucl. Instr. Meth. A* **2005**, 551, 420-428.

(4) Arulchakkaravarthi, A.; Balamurugan, N.; Kumar, R.; Santhanaraghavan, P.; Muralithar, S.; Nagarajan, T.; Ramasamy, P. *Mat. Lett.* **2004**, 58, 1209–1211.

(5) Carman, L.; Zaitseva, N.; Martinez, H. P.; Rupert, B.; Pawelczak, I.; Glenn, A.; Mulcahy, H.; Leif, R.; Lewis, K.; Payne, S. *J. Cryst. Growth* **2013**, 368, 56–61.

(6) Schuster, P. F. Investigating the Anisotropic Scintillation Response in Organic Crystal Scintillator Detectors. Ph.D. Dissertation, UC Berkeley, Berkeley, CA, 2016. (7) Bourne, M. M.; Clarke, S. D.; Adamowicz, N.; Pozzi, S. A.; Zaitseva, N.; Carman, L. *Nucl. Instrum. Methods Phys. Res., Sect. A* **2016**, 806, 348–355.

(8) Feng. P. L.; Villone, J.; Hattar, K.; Mrowka, S.; Wong, B. M.; Allendorf, M. D.; Doty, F. P. *IEEE Trans. Nucl. Sci.* **2012**, *59*, 3312–3319.

(9) Grisorio, R.; Suranna, G. P.; Mastrorilli, P.; Nobile, C. F. *Adv. Funct. Mater.* **2007**, *17*, 538–548.

(10) Carlson, J. S.; Feng, P. L. *Nucl. Instrum. Methods Phys. Res., Sect. A* **2016**, *832*, 152–157.

(11) Wei, W.; Djurovich, P. I.; Thompson, M. E. Chem. Mater. 2010, 22, 1724–1731.

(12) With limited exceptions: a) Dawson, K. J.; Zhu, L.; Yu, L.; Ediger, M. D. *J. Phys. Chem.* **2011**, *115*, 455–463. b) Eich, M.; Looser, H.; Yoon, D. Y.; Twieg, R.; Bjorklund, G.; Baumert, J. C. *J. Opt. Soc. Am. B.* **1989**, *6*, 1590–1597.

(13) a) Shirota, Y. J. Mater. Chem. 2000, 10, 1–25. b) Molaire, M. F.; Johnson, R. W. J. Polym. Sci., Part A: Polym. Chem. 1989, 27, 2569–2592. c) Koop, T.; Bookhold, J.; Shiraiwa, M.; Pöschl, U. Phys. Chem. Chem. Phys. 2011, 13, 19238–19255.

(14) a) Xi, H.; Sun, Y.; Yu, L. J. Chem. Phys. 2009, 130, 094508-1-9. b) Cai, T.; Zhu, L.; Yu, L. Pharm. Res. 2011, 28, 2458-2466.

(15) Wu, T.; Sun, Y.; Li, N.; de Villiers, M. M.; Yu, L. *Langmuir*, **2007**, *2*3, 5148–5153.

(16) Powell, C. T.; Chen, Y.; Yu, L. J. Non-Cryst. Solids, 2015, 429, 122–128.

(17) Wu, Y.-C. M.; Molaire, M. F.; Weiss, D. S; Angel, F. A.; DeBlase, C. R.; Fors, B. P. *J. Org. Chem.* **2015**, 80, 12740–12745.

(18) Swallen, S. F.; Kearns, K. L.; Mapes, M. K.; Kim, Y. S.; McMahon R. J.; Ediger, M. D.; Wu, T.; Yu, L.; Satija, S. *Science*, **2007**, *315*, 353–356.

(19) Campbell, I. H.; Crone, B. K. *Appl. Phys. Lett.* **2007**, 90, 012117-1–3.

(20) Rupert, B.L.; Cherepy, N.J.; Sturm, B.W.; Sanner, R.D.; Payne, S.A. *Europhys. Lett.* **2012**, 97, 22002-1–4.

(21) a) Goldsmith, J. E. M.; Gerling, M. D.; Brennan, J. S. *Rev. Sci. Instrum.* **2016**, *87*, 083307. b) Monterial, M.; Marleau, P.; Pozzi, S. A. *IEEE Trans. Nucl. Sci.* **2017**, *in press*, DOI: 10.1109/TNS.2017.2647952.

(22) Runkle, R. C.; Chichester, D. L.; Thompson, S. J. Nucl. Instrum. Methods Phys. Res., Sect. A 2012, 663, 75–95.



For Table of Contents Only