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# Crystal Structure and Nontypical Deep-Red Luminescence of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$

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**ABSTRACT:** Rare-earth doped nitridosilicates exhibit outstanding luminescence properties and have been intensively studied for solid-state lighting. Here, we describe the new nitridolithosilicate $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ with extraordinary luminescence characteristics. The compound was synthesized by solid-state metathesis reaction in sealed Ta ampules. The crystal structure was solved and refined on the basis of single-crystal X-ray diffraction data. $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ crystallizes in the monoclinic space group $C2/m$ (no. 12) [$Z = 4$, $a = 5.966(1)$, $b = 9.806(2)$, $c = 11.721(2)$ Å, $\beta = 99.67(3)°$, $V = 675.9(2)$ Å$^3$] and exhibits a layered anionic network made up of edge- and corner-sharing $LiN_4$ tetrahedra and $[Si_2N_6]^{10-}$ bow-tie units. The network charge is compensated by $Ca^{2+}$ and $Mg^{2+}$ ions. Upon irradiation with UV to blue light, red emission at exceptionally long wavelengths ($\lambda_{em} = 734$ nm, fwhm ≈ 2293 cm$^{-1}$) is observed. According to emission in the near infrared, application in LEDs for horticultural lighting appears promising.

## INTRODUCTION

Nitridosilicates are characterized by its manifold structural chemistry and intriguing materials properties. Due to the high sensitivity to air and moisture of both starting materials and products, synthesis of less condensed nitridosilicates is quite challenging. Over the last years, different synthetic approaches leading to nitridosilicates have been elaborated. These include high-temperature reactions, precursor routes, ammonothermal syntheses and flux methods by using, for example, liquid sodium.[1-9] Solid-state metathesis reactions have gained significant importance as well, since a variety of ternary or higher nitrides have recently been synthesized by this approach.[10-15] Thereby, key feature is the employment of reactive starting materials by coproducing a metathesis salt, which on one hand drives the reaction thermodynamically and on the other hand concurrently acts as reactive flux.

Common structural motifs of nitridosilicates are $SiN_4$ tetrahedra, which may be linked through vertices and/or edges, forming more or less condensed anionic substructures ranging up to three-periodic anionic frameworks. The resulting structural diversity of nitridosilicates can be ascribed to the N atoms, which are able to connect up to four adjacent tetrahedral centers. Cations like alkaline earths ($AE^{2+}$), $Li^+$ or $Mg^{2+}$ are distributed among the voids of the anionic networks and balance their charges. As some examples in literature have already shown, the number of new nitrides can be further increased by partial substitution of $Si^{4+}$ by $Li^+$, $Mg^{2+}$ or $Al^{3+}$.[12-13,16-19] Complete exchange of $Si^{4+}$ by $Ga^{3+}$ or $Ge^{4+}$ is also possible and results in the strongly related compound classes of nitridogallates and nitridogermanates, respectively (see homeotypical compounds $Ba[Mg_3SiN_4]$, $Sr[Mg_3GeN_4]$, and $M[Mg_2Ga_2N_4]$ with $M$ = Sr,Ba).[17,20-21]

Nitridosilicates show a great diversity of interesting materials properties, e.g. high mechanical strength and high thermal stability of silicon nitride ceramics, lithium ion conductivity (e.g. $Li_2SiN_2$)[22-23] or luminescence when doped with $Eu^{2+}$ or $Ce^{3+}$.[22,24-28] $(Sr,Ba)_2Si_5N_8:Eu^{2+}$,[29-31] $CaAlSiN_3:Eu^{2+}$,[19] and $SrMg_3SiN_4:Eu^{2+}$[16] represent prominent examples of rare-earth doped nitrides, which show strong broadband emission in the red spectral region after irradiation with blue light, originating from parity allowed $4f^6 5d^1 \rightarrow 4f^7$ transitions in $Eu^{2+}$ and spin and parity allowed $4f^0 5d^1 \rightarrow 4f^1$ transitions in $Ce^{3+}$. Warm white light can be generated by combination of red and green emitting phosphors, both covered on a blue LED chip, which is usually an (In,Ga)N semiconductor chip.[32-33] The so-called phosphor-converted light-emitting diodes (pc-LEDs) are highly efficient and, over the last decade, developed into an energy-saving replacement for inefficient incandescent light bulbs. Since most commercially available warm white pc-LEDs still show a large portion of its emission beyond the sensitivity of the human eye,[34-35] research for narrow-band red emitting luminescent materials is of particular interest for general illumination purposes.

During our research for innovative red emitting compounds, we came across $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$, a new nitridolithosilicate with outstanding emission properties at fairly long wavelengths. In this contribution, we report on synthesis, structural elucidation, and optical properties of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$. Unprecedented luminescence properties differ essentially from previously described $Eu^{2+}$-doped nitridosilicates and are discussed in detail, also in

comparison to recently reported narrow-band red emitting $Eu^{2+}$-doped nitridosilicates.

## EXPERIMENTAL SECTION

**Synthesis.** Due to the sensitivity of the starting materials and the product towards air and moisture, all manipulations were performed under inert-gas conditions using argon filled glove boxes (Unilab, MBraun, Garching, $O_2 < 1$ ppm, $H_2O < 1$ ppm) and combined Schlenk/vacuum lines. Ar was dried and purified by passing through glass tubes filled with silica gel (Merck), molecular sieve (Fluka, 4 Å), $P_4O_{10}$ (Roth, ≥99%) and titanium sponge (Johnsen Matthey, 99.5%). $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ was synthesized by solid-state metathesis reaction using $Ca(NH_2)_2$ (17.9 mg, 0.25 mmol, synthesized according to Brokamp),[36] $CaH_2$ (8.3 mg, 0.20 mmol, Cerac, 99.5%), $MgF_2$ (15.5 mg, 0.25 mmol, ABCR, 99.99%), $Si_3N_4$ (6.9 mg, 0.05 mmol, UBE, 99.9%), 1.7 mol% $EuF_3$ (0.9 mg, $4.3 \cdot 10^{-3}$ mmol, Sigma-Aldrich, 99.99%), and Li (9.8 mg, 1.41 mmol, Alfa Aeser, 99.9%) as starting materials. The powdery reactants were finely ground and placed with metallic Li in a tantalum ampule, which was weld shut in an arc furnace. The reaction mixture was heated up to 950 °C with a rate of 300 °C·h$^{-1}$, maintained for 24 h, subsequently cooled down to 500 °C with a rate of 5 °C·h$^{-1}$ and finally quenched to room temperature by switching off the furnace. Orange colored single crystals of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ were obtained embedded in an amorphous powder. Additionally, red rod-shaped crystals of $Ca[Mg_3SiN_4]:Eu^{2+}$, colorless crystals of LiF, and some metallic residues were found as side phases of the reaction. The sample is sensitive towards air and moisture and partially shows intensive red luminescence under irradiation with ultraviolet (UV) to green light.

**Single-Crystal X-ray Diffraction.** Single crystals of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ were isolated from the reaction product, washed with dry paraffin oil and enclosed in glass capillaries, which were sealed under argon to avoid hydrolysis. X-ray diffraction data were collected on a Stoe IPDS I diffractometer (Mo-$K_\alpha$ radiation, graphite monochromator). Structure solution was done in SHELXS-97,[37] by using direct methods. The crystal structure was refined by full-matrix least-squares calculation on $|F|^2$ in SHELXL-97 [38-39] with anisotropic displacement parameters for all atoms. Further details of the crystal-structure refinement may be obtained from the Fachinformationszentrum Karlsruhe, 76344 Eggenstein-Leopoldshafen, Germany (Fax: +49-7247-808-666; E-mail: crysdata@fiz-karlsruhe.de), on quoting the depository number CSD-432605.

**Scanning Electron Microscopy.** The chemical composition of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ was confirmed by energy dispersive X-ray spectroscopy (FEI Helios G3 UC scanning electron microscope equipped with an EDX detector, scanning transmission detector and focused ion beam). The reaction product was placed on an adhesive conductive carbon pad and coated with a conductive carbon film (BAL-TEC MED 020, Bal Tec AG).

**Luminescence.** Luminescence investigations on single crystals of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ were performed using a luminescence microscope, consisting of a HORIBA Fluorimax4 Spectrofluorimeter system, which is attached to an Olympus BX51 microscope via fiber optics. The single crystals were measured in sealed glass capillaries with an excitation wavelength of 440 nm. The emission spectrum was measured between 460 and 800 nm; the excitation spectrum was measured between 380 and 680 nm, with a step size of 2 nm, respectively. Since $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ shows emission beyond 800 nm and the spectrometer is only calibrated for wavelengths up to 800 nm, emission peak in the near infrared was simulated using a Gauss-fit.

## RESULTS AND DISCUSSION

**Synthesis and Chemical Analysis.** Solid-state metathesis reaction as described above yielded a heterogeneous reaction mixture containing single-crystals of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$. The crystals show orange body color and undergo rapid hydrolysis when exposed to air. The elemental composition was determined by energy-dispersive X-ray spectroscopy (EDX). Due to the method, light elements like N cannot be determined with a reasonable degree of accuracy, also Li is not determinable. The atomic ratio Ca/Mg/Si of 3.0/1.1/2.0 (six measurements on different crystals), normalized to the Ca content, is in good agreement with the sum formula obtained from single-crystal structure analysis. Besides a small amount of oxygen, which can be attributed to superficial hydrolysis of the product, no further elements were detected.

**Single-Crystal Structure Analysis.** The crystal structure of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ was solved and refined in the monoclinic space group $C2/m$ (no. 12) with $a$ = 5.966(1), $b$ = 9.806(2), $c$ = 11.721(2) Å, and $\beta$ = 99.67(3)°. The crystallographic data are summarized in Table 1. Atomic coordinates, Wyckoff positions, and isotropic displacement parameters are listed in Table 2. Selected bond lengths and angles as well as anisotropic displacement parameters are given in the Supporting Information (Tables S1-S3). Due to the small dopant concentration and, therefore, the insignificant scattering intensity, $Eu^{2+}$ was neglected during structure refinement.

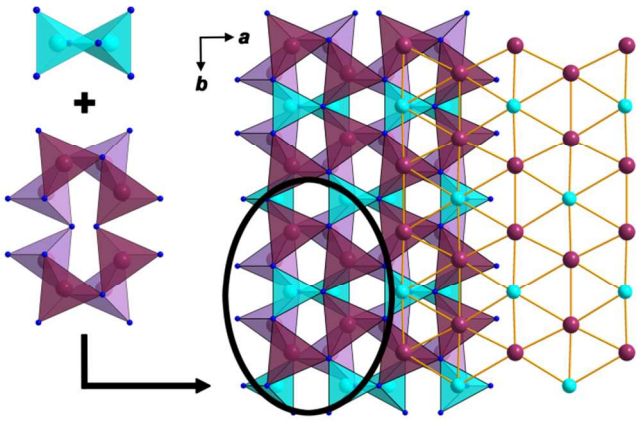

**Figure 1.** Details of layers in $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$. Left: $[Si_2N_6]^{10-}$ bow-tie units (turquoise) and *achter* rings of $LiN_4$ tetrahedra (violet). Right: Layer made up of bow-tie units and *achter* rings forming two sublayers of condensed *dreier* rings, only one of them shown for reasons of clarity.

The crystal structure of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ is closely related to that of $Ca_2Mg[Li_4Si_2N_6]$,[12] which has already been reported in literature. Both structures contain identical layers of $LiN_4$ and $SiN_4$ tetrahedra as well as Ca and Mg ions, which are located between the layers. Since Li as well as Si are part of the tetrahedral network, the compounds can be more precisely classified as nitridolithosilicates. The arrangement of tetrahedra within one layer is shown in Figure 1. It consists of vertex- and corner-sharing $LiN_4$ tetrahedra forming *achter* rings,[40] intercalated by edge-sharing $[Si_2N_6]^{10-}$ "bow-tie" units. This results in layers of condensed *dreier* rings in the *ab* plane. The differences between the structurally related compounds $Ca_2Mg[Li_4Si_2N_6]$ and $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ are additional $[Si_2N_6]^{10-}$ bow-tie units in the crystal structure of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$. Together with the $Ca^{2+}$ and $Mg^{2+}$ ions, these bow-tie units are located between layers of *dreier* rings. A direct comparison of both crystal structures is illustrated in Figure 2.

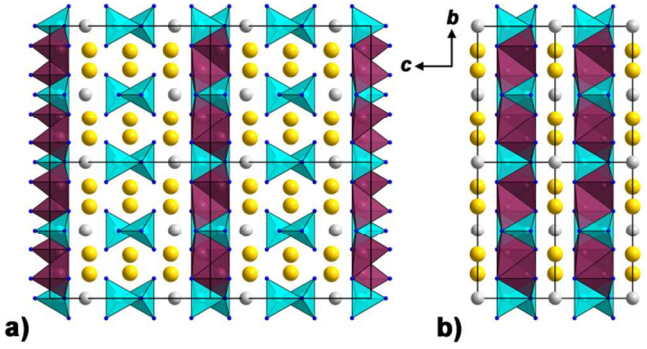


**Figure 2.** Crystal structures of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ (a) and $Ca_2Mg[Li_4Si_2N_6]$ (b).

$[Si_2N_6]^{10-}$ bow-tie units were first described in $Ba_5Si_2N_6$,[2] and are also known in other nitridosilicates, e.g. $Ca_2Mg[Li_4Si_2N_6]$, $Li_2Ca_2[Mg_2Si_2N_6]$,[12] and $Ca_3[Li_4Si_2N_6]$.[41-42] Si–N bond lengths in these bow-tie units ($Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$: 1.719(5)-1.843(7) Å) are slightly larger than distances found in other nitridosilicates.[1,5,16,43-44] This can be attributed to the repulsion of the Si atoms in the pairs of edge-sharing $SiN_4$ tetrahedra (Si–Si: 2.363(4)-2.477(4) Å), which in turn is related to a reduced angle N-Si-N. Li–N distances range from 2.088(11) to 2.225(11) Å and are in good agreement with distances known from other nitrides and with the sum of the ionic radii.[11,44-46]

$Ca^{2+}$ and $Mg^{2+}$ ions balance the charges of the anionic framework. Its coordination spheres are displayed in Figure 3. $Ca^{2+}$ occupies two different sites with distorted octahedral coordination by N and with distances varying from 2.310(4)-2.744(5) Å. $Mg^{2+}$ exhibits fourfold rectangular surroundings by N. Mg–N bond lengths are in a range of 2.159(5)-2.199(5) Å. Additionally, there is one further N atom at a distance of 2.693(7) Å, its coordination to Mg1 is illustrated in Figure 3 by a dashed line. Ca–N as well as Mg–N distances correspond to the sums of the ionic radii (2.55 and 2.10 Å, respectively)[45] as well as to distances known from other nitridosilicates.[1,12,16] The Madelung part of the lattice energy has been calculated to confirm the crystal structure of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$.[45,47-49] Therefore, the charge, distance, and coordination spheres of constituting ions were taken into account. MAPLE values for $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ and its constituting binary and ternary nitrides are given in Table 3. The resulting deviation of 0.31% verifies the electrostatic consistency of the refined structure.

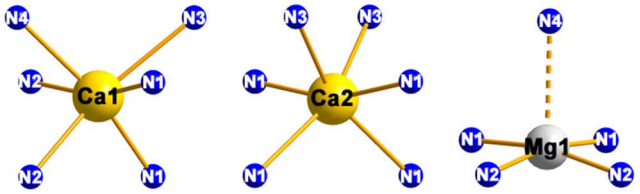


**Figure 3.** Coordination spheres of counter ions $Ca^{2+}$ and $Mg^{2+}$.

**Table 1. Crystallographic data of the single-crystal structure determination of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$.**

| | |
|---|---|
| formula | $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ |
| crystal system | monoclinic |
| space group | *C*2/*m* (no. 12) |
| lattice parameters / Å,° | *a* = 5.966(1) |
| | *b* = 9.806(2) |
| | *c* = 11.721(2) |
| | *β* = 99.67(3) |
| cell volume / Å$^3$ | 675.9(2) |
| formula units / unit cell | 4 |
| density / g·cm$^{-3}$ | 2.935 |
| μ / mm$^{-1}$ | 2.827 |
| T / K | 293(2) |
| diffractometer | Stoe IPDS I |
| radiation | Mo-$K_\alpha$ ($\lambda$ = 0.71073 Å), graphite monochromator |
| F(000) | 592 |
| profile range | $4.04 \le \theta \le 29.90$ |
| index ranges | $-8 \le h \le 8$ |
| | $-13 \le k \le 13$ |
| | $-16 \le l \le 16$ |
| independent reflections | 1028 [*R*(int) = 0.0722] |
| refined parameters | 72 |
| Goodness of fit | 1.323 |

| | |
|---|---|
| $R_1$ (all data); $R_1$ ($F^2 > 2\sigma(F^2)$) | 0.0770, 0.0618 |
| $wR_2$ (all data); $wR_2$ ($F^2 > 2\sigma(F^2)$) | 0.1118, 0.1080 |
| $\Delta\rho_{max}$, $\Delta\rho_{min}$ / e·Å$^{-3}$ | 0.826, –1.176 |

**Table 2. Atomic coordinates, Wyckoff positions and equivalent isotropic displacement parameters of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$.**

| atom | Wyck. | x | y | z | $U_{eq}$/Å$^2$ |
|---|---|---|---|---|---|
| Ca1 | 8*j* | 0.0500(2) | 0.32384(11) | 0.24678(9) | 0.0068(2) |
| Ca2 | 4*h* | 0 | 0.1840(2) | ½ | 0.0085(3) |
| Si1 | 4*i* | 0.3521(3) | 0 | 0.4192(2) | 0.0050(4) |
| Si2 | 4*i* | 0.6885(3) | 0 | 0.0632(2) | 0.0048(4) |
| Mg1 | 4*i* | 0.0623(5) | 0 | 0.2220(2) | 0.0094(5) |
| Li1 | 8*j* | 0.1731(18) | 0.1767(11) | 0.0594(8) | 0.015(2) |
| N1 | 8*j* | 0.2335(7) | 0.1447(4) | 0.3513(4) | 0.0070(8) |
| N2 | 8*j* | 0.3413(7) | 0.3566(5) | 0.1179(4) | 0.0064(8) |
| N3 | 4*i* | 0.3418(10) | 0 | 0.5703(5) | 0.0084(12) |
| N4 | 4*i* | 0.3943(11) | 0 | 0.0921(5) | 0.0068(11) |

**Table 3. MAPLE values and MAPLE sums /kJ·mol$^{-1}$ for $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$. Δ is the deviation between the MAPLE sum of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ and the MAPLE sum of constituting binary and ternary nitrides.**[a]

| atom | MAPLE | atom | MAPLE | ∑, Δ |
|---|---|---|---|---|
| $Ca1^{2+}$ | 2083 | $Li1^{+}$ | 717 | |
| $Ca2^{2+}$ | 1812 | $N1^{3-}$ | 5056 | |
| $Si1^{4+}$ | 9201 | $N2^{3-}$ | 4951 | |
| $Si2^{4+}$ | 9202 | $N3^{3-}$ | 5440 | ∑ = 58949 |
| $Mg1^{2+}$ | 2241 | $N4^{3-}$ | 5439 | Δ = 0.31% |

Total MAPLE: $Ca_3Mg[Li_2Si_2N_6] = Ca_3N_2 + ⅓\cdot Mg_3N_2 + ⅔\cdot Li_2SiN_2 + ⅔\cdot LiSi_2N_3$ = 59131 kJ·mol$^{-1}$

[a] Typical partial MAPLE values /kJ·mol$^{-1}$: $Ca^{2+}$: 1700-2200;; $Mg^{2+}$: 2100-2500; $Li^{+}$: 550-860; $Si^{4+}$: 9000-10200; $N^{3-}$: 4300-6200.[9,16,50-51]

**Luminescence.** Addition of $EuF_3$ to the starting materials yielded a gray microcrystalline sample containing orange colored crystals of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$. Under blue irradiation, the sample shows luminescence in the red spectral region. Due to inhomogeneity of the sample, luminescence investigations have been performed on single crystals of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ in sealed silica glass capillaries. All crystals show comparable red emission under blue irradiation. Excitation of the title compound with a nominal dopant concentration of 1.7 mol% at 440 nm yields an emission band with a maximum at 734 nm and a full width at half maximum (fwhm) of 2293 cm$^{-1}$ (124 nm). The $Eu^{2+}$-doped compound exhibits a maximum absorption around 450 nm and is therefore efficiently excitable with UV to blue light, as provided, for example, by (In,Ga)N-LEDs. Excitation and emission spectra are illustrated in Figure 4.

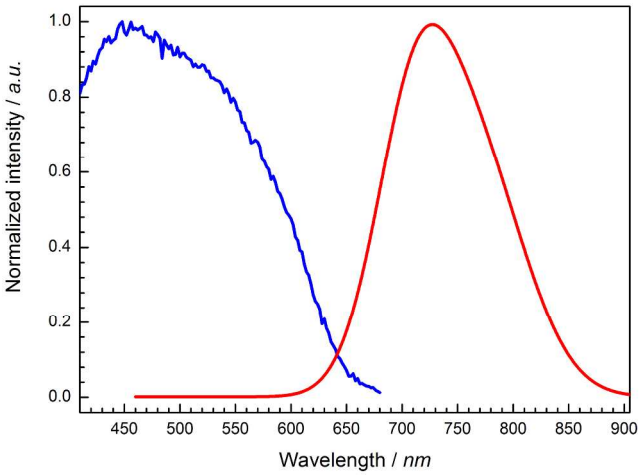


**Figure 4.** Excitation (blue) and emission (red) spectra of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$.

The broad emission band of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ is assigned to parity allowed $4f^6 5d^1 \rightarrow 4f^7$ transitions in $Eu^{2+}$.[33,52-53] Since $Eu^{2+}$ is expected to occupy the Ca site, we assume that luminescence originates from octahedrally coordinated $Eu^{2+}$ ions. The presence of two crystallographically different Ca sites with slightly differing distortion of the octahedra and, therefore, different interatomic distances, influences the width of the emission band by broadening. A comparable 6-fold coordination can also be found in $Ba_3Si_6O_{12}N_2$ [54] and its solid-solution series $Ba_{1-x}Sr_xSi_6O_{12}N_2$ with x≈0.4 and 1. These compounds doped with 2 mol% $Eu^{2+}$ exhibit broad emission bands with maxima between 523 and 549 nm (green luminescence) when irradiated with near-UV to blue light. In general, both the spectral position and the band width of the emission strongly depend on the energy level of 5*d* states of the activator ion, which in turn is influenced by the host lattice. The stronger the covalent interactions between $Eu^{2+}$ and its surroundings, the smaller is the transition energy between 4*f* and 5*d* orbitals of the activator ion (nephelauxetic effect). For this reason and because of the higher formal charge of $N^{3-}$ compared to $O^{2-}$, the 5*d* states of $Eu^{2+}$ in $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ are at lower energies than those in $Ba_3Si_6O_{12}N_2:Eu^{2+}$ and $Ba_{1-x}Sr_xSi_6O_{12}N_2:Eu^{2+}$, respectively, and emission of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ occurs at longer wavelengths.

Typical $Eu^{2+}$ emission of other compound classes are in a range of 356-461 nm (fluorides), 360-540 nm (oxosilicates), 420-555 nm (aluminates and gallates), 470-660 nm (sulfides) and 529-670 nm (nitrides).[55-56] The here presented nitridolithosilicate $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ is, to the best of our knowledge and except for $CaO:Eu^{2+}$ ($\lambda_{em}$ = 733 nm),[57-58] and $MH_2:Eu^{2+}$ (*M* = Ca,Sr,Ba; $\lambda_{em}$ = 728-764 nm)[59] the first and so far the only example of a $Eu^{2+}$-

doped compound that emits at fairly long wavelengths of >700 nm.

Recently, we reported on luminescence investigations of the nitridomagnesosilicate $Li_2(Ca_{1-x}Sr_x)_2[Mg_2Si_2N_6]:Eu^{2+}$ (x = 0 and 0.6),[60] with a nominal dopant concentration of 1 mol%. Just as $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$, the crystal structure of $Li_2(Ca_{1-x}Sr_x)_2[Mg_2Si_2N_6]:Eu^{2+}$ contains $[Si_2N_6]^{10-}$ bow-tie units. In contrast to the here presented nitridolithosilicate with $LiN_4$ and $SiN_4$ tetrahedra as network-building structural motifs and $Ca^{2+}$ and $Mg^{2+}$ as counter ions, $Li^+$ and $Ca^{2+}$ compensate the negative charge of the network, which, in $Li_2(Ca_{1-x}Sr_x)_2[Mg_2Si_2N_6]:Eu^{2+}$, is made up of $MgN_4$ and $SiN_4$ tetrahedra. In the latter compound, chains of edge-sharing $MgN_4$ tetrahedra are interconnected by bow-tie units, forming *vierer* and *sechser* ring channels along [100]. Whereas $Li^+$ ions are located within *vierer* ring channels, $Ca^{2+}$ ($Sr^{2+}$) ions and, therefore, also $Eu^{2+}$ ions, show distorted octahedral coordination by N within *sechser* ring channels. When irradiated with blue light, narrow-band red emission also occurs at rather long wavelengths of 638 nm (x = 0) and 634 nm (x = 0.6) with fwhm = 1513-1532 $cm^{-1}$, respectively.

Both $Li_2(Ca_{1-x}Sr_x)_2[Mg_2Si_2N_6]:Eu^{2+}$ and $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ exhibit similar body colors and show similarities in their absorption bands. The main difference between both nitridosilicates is the Stokes shift, which is larger for $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$. This can be attributed to differences in the chemical surroundings of $Eu^{2+}$, which in each case is expected to occupy the $Ca^{2+}$ ($Sr^{2+}$) site, and also to differences in the phonon frequencies of respective host lattices. $Li_2(Ca_{1-x}Sr_x)_2[Mg_2Si_2N_6]:Eu^{2+}$ exhibits only one crystallographic Ca site (Wyckoff position 4*g* of space group *C*2/*m*) with three different Ca–N distances between 2.49 and 2.75 Å. In $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$, nine interatomic distances Ca–N of two Ca sites (Wyckoff positions 8*j* and 4*h* of space group *C*2/*m*) vary between 2.31 and 2.74 Å. This is, in comparison, a significantly larger spread and entails chemical differentiation of both sites, which eventually leads to a composed emission band of $Eu^{2+}$ on either of the crystallographic sites. Moreover, the shorter bond lengths Ca–N in $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$, especially occurring on the lowest symmetry Ca1 site, may entail efficient energy transfer from $Eu^{2+}$ incorporated on the larger higher symmetry Ca2 site and, thus, also lead to a more pronounced red shift of the emission band for the Eu doping concentration investigated in this report. Whereas in $Li_2(Ca_{1-x}Sr_x)_2[Mg_2Si_2N_6]:Eu^{2+}$, infinite chains of edge-sharing $MgN_4$ tetrahedra are interconnected by $[Si_2N_6]^{10-}$ bow-tie units leading to a three-periodic network, $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ consists of layers made up of $LiN_4$ and $SiN_4$ tetrahedra and isolated bow-tie units. Consequently, due to the lower degree of condensation and also due to lower symmetry of the Ca1 site with its short Ca-N contact lengths, local lattice relaxation of $Eu^{2+}$ in its excited state in $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ is expected to be larger than in $Li_2(Ca_{1-x}Sr_x)_2[Mg_2Si_2N_6]:Eu^{2+}$,[61] resulting in a larger Stokes shift of the title compound.

Red emission and extremely narrow band widths with fwhm <1200 $cm^{-1}$ have been reported for $Sr[Mg_3SiN_4]:Eu^{2+}$ (1170 $cm^{-1}$), and $Sr[LiAl_3N_4]:Eu^{2+}$ (1180 $cm^{-1}$).[16,34] Both compounds show a highly symmetric cube-like coordination of the alkaline earth ion by N.

## CONCLUSION

In this contribution, we report on $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$, a new nitridolithosilicate with exceptional luminescence properties. The compound was synthesized by solid-state metathesis reaction and could be obtained as orange colored single crystals within an amorphous powder. The crystal structure of $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ is related to that of $Ca_2Mg[Li_4Si_2N_6]$. It consists of corner- and edge-sharing $LiN_4$ tetrahedra, intercalated by $[Si_2N_6]^{10-}$ bow-tie units forming anionic layers, which are separated by isolated $[Si_2N_6]^{10-}$ bow-tie units and $Ca^{2+}$ and $Mg^{2+}$ ions. The latter two act as counter ions and compensate the negative charge of the anionic framework. Up to now, only a small number of nitridosilicates containing $Li^+$ and $Mg^{2+}$ ions are known from literature, namely $Ca_2Mg[Li_4Si_2N_6]$, $Li_2Ca_2[Mg_2Si_2N_6]$, and $Li_2(Ca_{1.86}Sr_{0.14})[Mg_2Si_2N_6]$.[12,60] Thereby, $Li^+$ and $Mg^{2+}$ ions may either be part of the tetrahedral network or act as counter ions. This behavior may significantly increase the variety of possible structures within the class of nitridosilicates.

Recently, the quest for narrow-band red emitting phosphors has been triggered by attempts to improve luminescence properties of commercially available pc-LEDs for general illumination and signaling purposes.[16,34,62] Meanwhile, LED technology for other specific applications is on the advance and will increasingly be pursued. This includes, for example, LED technology for plants, in order to replace currently used lamps to reduce costs and, seen in the long term, improve productivity by optimizing growth environments.[63] Similar to requirements for LEDs applied for general illumination, LEDs with regard to application in horticulture have to be cheap, environmentally friendly, highly efficient and reliable. Luminescence investigations on $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ show red emission peaking at 734 nm and a fwhm of 2293 $cm^{-1}$ (124 nm). This represents an unexpected emission at fairly long wavelengths, not yet known for $Eu^{2+}$-doped nitrides. Despite emission >700 nm and, therefore, too high energy loss beyond the sensitivity of the human eye when applied in pc-LEDs, $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ may find application in more specialized fields like horticultural lighting.

## ASSOCIATED CONTENT

**Supporting Information**. Selected bond lengths in $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ (Table S1), selected bond angles in $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ (Table S2), and anisotropic displacement parameters for $Ca_3Mg[Li_2Si_2N_6]:Eu^{2+}$ (Table S3). This material is available free of charge via the Internet at http://pubs.acs.org.

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### Notes

The authors declare no competing financial interest.

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