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New tetraphenylethylene-containing conjugated polymers: Facile synthesis, aggregation-induced emission enhanced characteristics and application as explosive chemsensors and PLEDs

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1. Introduction

ABSTRACT

In this paper, tetraphenylethylene (TPE) units, one of the typical aggregation-induced emission (AIE) moieties, are utilized to construct a new functional polyfluorene (PF) **P1**, which exhibited the exciting property of the aggregation-induced emission enhancement (AIEE), instead of the aggregation-caused quenching (ACQ) of normal PFs, and could probe the explosive with high sensitivity both in the nano-particles and solid state. Three other TPE-containing polymers, **P2–P4**, were also successfully prepared, and demonstrated good performance as explosive chemosensors and light-emitting materials. **P3**, bearing carbazole as hole-transporting units showed the best performance with a maximum luminance efficiency of 1.17 cd/A and a maximum brightness of 3609 cd/m² at 12.9 V in its light-emitting diode device.

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In the past decades, considerable interests have been attracted in the development of light-emitting diodes (LED) based on conjugated polymers, due to their wide-ranging applications [1]. Among them, polyfluorene (PF) and its derivatives are considered as promising light-emitting materials for their exceptionally high efficiencies both in photoluminescence (PL) and electroluminescence (EL) [2]. The high PL efficiencies also make some specific functional PFs widely used in other fields, for example, chemosensors [3]. However, PFs are likely to aggregate and form excimers in solid states, directly leading to the fluorescence quenching, namely aggregation-caused quenching (ACQ) [4], hence inhibiting their prospective utilizations in a large degree. Although many different approaches including chemical, physical, and engineering ones have been attempted to hamper these luminophore aggregations, they are still a little far from ideal, because the ACQ effect is alleviated at the expense of other useful properties [5]. Thus, it is badly needed to change the ACQ property of PFs, without sacrificing

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other advantages. For chemists, the general approach is to modify the structure at the molecular level.

Tetraphenylethene (TPE) is characterized by its aggregationinduced emission (AIE) attribute [6], which was first reported by Tang's group [7a]: it is virtually nonluminescent when molecularly dissolved in its good solvents, but emits intensely when aggregated in its poor solvents or fabricated into thin solid film [7]. This is exactly opposite to the ACQ effect, and has recently been given much attention due to the huge potential high-tech applications in many areas [8]. Thus, if the TPE unit was introduced into the PF system, perhaps, the original ACQ behavior might be converted to AIE(E) phenomena. However, the related reports were very scarce. Inspired by this idea, in this paper, a new functional PF P1 (Chart 1), containing the TPE moieties in the main chain, was therefore synthesized through typical Pd-catalytic Suzuki polymerization. As expected, the light-emitting behavior of P1 was not ACQ any longer, but aggregation-induced emission enhancement (AIEE). That was to say, in solution, P1 could emit weak fluorescence, however, the emission was enhanced largely in solid state including nanoparticles and thin films. And it could be used as chemsensors for the detection of explosives. Furthermore, other three TPE-containing polymers, P2-P4 (Chart 2), were successfully prepared, in which the spirobifluorene, carbazole and 1,3,4-oxadiazole moieties were



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Chart 1. The sketch map from normal PF to TPE-containing P1.

used as comonomers, to further investigate the above idea and improve the LED performance. All these polymers possessed AIEE characteristic, and act as good chemosensors and light-emitting materials. Herein, we would like to present their synthesis, characterization, AIEE properties, sensing behavior, and LED device performance in detail.

2. Experimental section

2.1. Materials

Tetrahydrofuran (THF) was dried over and distilled from K-Na alloy under an atmosphere of dry nitrogen. Pyridine was dried over from CaH₂ and distilled under normal pressure before use. 9,9-Dihexylfluorene-2,7-bis(trimethyleneborate) (**S4**) was purchased from Aldrich while the other two bronic esters **S5** and **S6** were prepared through the same synthetic routes reported in the literature [9]. 2,5-Di(4-bromobenzoyl)-1,3,4-oxadiazole (**S5**) was also synthesized according to our previous work [1f]. 2-Isopropoxy-4,4,5,5-tetramethyl-1,3,2-dioxaborolane was purchased from Aldrich, while *n*-butyllithium (2.6 M in hexane) and 4-bromo

benzophenone (**S1**) were bought from Alfa-Aesar. All other reagents were used as received.

2.2. Instrumentation

¹H and ¹³C NMR spectra were measured on a Varian Mercurv300 spectrometer or Varian Mercury600 spectrometer using tetramethylsilane (TMS: $\delta = 0$ ppm) as internal standard. The Fourier transform infrared (FT-IR) spectra were recorded on a Perkin Elmer-2 spectrometer in the region of 4000-400 cm^{-1} on KBr pellets. UV-visible spectra were obtained using a Shimadzu UV-2550 spectrometer. FAB-MS spectra were recorded with a VI-ZAB-3F-Mass spectrometer. Elemental analyses were performed by a CAR-LOERBA-1106 micro-elemental analyzer. Photoluminescence spectra were performed on a Hitachi F-4500 fluorescence spectrophotometer. Gel permeation chromatography (GPC) was used to determine the molecular weights of polymers. GPC analysis was performed on a Waters HPLC system equipped with a 2690D separation module and a 2410 refractive index detector. Polystyrene standards were used as calibration standards for GPC. THF was used as an eluent and the flow rate was 1.0 mL/min. Thermal analysis was performed on NETZSCH STA449C thermal analyzer at a heating rate of 10 °C/min in nitrogen at a flow rate of 50 cm³/min for thermogravimetric analysis (TGA). The thickness of the films was measured with an Ambios Technology XP-2 profilometer. Electrochemical Workstation with a Pt disk, a Pt plate, and a Ag/Ag⁺ electrode as the working electrode, counter electrode and reference electrode, respectively, in a 0.1 mmol/mL tetrabutylammonium hexafluorophosphate (Bu₄NPF₆) acetonitrile solution. Scanning electron microscopy (SEM) was performed on a QUANTA scanning electron microscope (FEI Co., Netherlands), and the nanoaggregate thin films were prepared by its THF/water (1/9 v/v) suspension (0.02 mg/mL in THF) dropped on the slides and dried at about 60 °C.

2.3. Synthesis of 1,2-bis(4-bromophenyl)-1,2-diphenylethene (S2)

To a mixture of Zn (3.90 g, 60.0 mmol) and 40 mL of dry THF was added TiCl₄ (3.3 mL, 30.0 mmol) under an atmosphere of argon at 0 °C. The mixture was stirred for 30 min at room temperature and another 2 h at 60 °C, then 4-bromobenzophenone (**S1**) (0.85 mg, 2.5 mmol) in 30 mL of THF with pyridine (1 mL) was added dropwise in 1 h at 0 °C. The mixture was refluxed overnight. After



Chart 2. The different chemical structures from P1 to polymers P2-P4.



Fig. 1. The ¹H NMR spectra of polymers P1-P4 and monomer S3 conducted in chloroform-d.

cooling to room temperature, 10% K₂CO₃ (aq) was added and the mixture was filtrated. The filtrates were extracted with CH₂Cl₂ and the extracts were collected, combined, and washed with brine and dried over anhydrous Na₂SO₄. After removal of the solvents under reduced pressure, the crude product was purified by column chromatography on silica gel using chloroform/petroleum ether (1/ 5, V/V) as eluent to afford white solid (1.12 g, 76.2%). IR (KBr), v (cm⁻¹): 1584 (C=C). ¹H NMR (300 MHz, CDCl₃, 298 K), δ (TMS, ppm): 6.87 (d, J = 5.1 Hz, 4H, ArH), 6.99 (s, br, 4H, ArH), 7.10 (s, br, 6H, ArH), 7.20 (d, J = 5.1 Hz, 4H, ArH). ¹³C NMR (75 MHz, CDCl₃,

298 K), δ (TMS, ppm): 120.6, 126.9, 127.7, 128.0, 130.8, 131.2, 132.8, 140.2, 142.3, 142.7. MS (FAB), m/z [M⁺]: 487.97, calcd: 487.98. C₂₆H₁₈Br₂ (EA) (%, found/calcd): C, 62.71/63.70; H, 3.95/3.70.

2.4. Synthesis of 1,2-diphenyl-1,2-bis(4-(4,4,5,5-tetramethyl-1,3,2-dioxaborolan-2-yl)phenyl)ethane (**S3**)

To a solution of **S2** (980 mg, 2.0 mmol) in 40 mL dry THF at -78 °C, 2.2 mL of *n*-butyllithium (2.6 M in hexane, 5.7 mmol) was



Fig. 2. A: Photoluminescence (PL) spectra of **P1** in THF and THF/water mixtures ($2.0 \times 10^{-3} \text{ mg/mL} \lambda_{ex} = 377 \text{ nm}$); B: Normalized PL spectra of **P1** in the solution and aggregate states.



Fig. 3. Variations in the fluorescence quantum yields (Φ_F) of **P1–P4** with water fractions in the THF/water mixtures. 9,10-diphenylanthracene in cyclohexane ($\Phi_F = 90\%$) as reference.



Fig. 4. A: PL spectra of **P1** in THF/water mixture $(1:9 v/v 2.0 \times 10^{-3} mg/mL)$ containing different amounts of picric acid (PA); B: PA concentration effect on the PL peak intensity of **P1** in THF/water mixture $(1:9 v/v 2.0 \times 10^{-3} mg/mL)$, where I = peak intensity and $I_0 =$ peak intensity at [PA] = 0 mg/mL; Inset of B: The structure of PA; C: fluorescence images of **P1–P4** adsorbed in the filter papers before (the left one), after (the middle one) being partially dipped into pure toluene and after (the right one) being partially dipped into a toluene solution of PA (**P1** and **P2**: 60 µg/mL; **P3** and **P4**: 50 µg/mL).

added dropwise. The mixture was stirred at -78 °C for 2 h. Then, 2.2 mL of 2-isopropoxy-4,4,5,5,-tetramethyl-1,3,2-dioxaborolane (9.6 mmol) was added rapidly to the above solution and the mixture was stirred for 2 h. The resulting mixture was warmed up to room temperature and stirred for 24 h, then poured into water, and extracted with chloroform. The combined extracts were washed with brine, dried over anhydrous Na₂SO₄, and concentrated. The crude product was purified by column chromatography on silica gel using ethyl acetate/petroleum ether (1:10) as eluent to afford white powder **S3** (438 mg, 37.8%). IR (KBr), v (cm⁻¹): 1606 (C=C).¹H NMR (300 MHz, CDCl₃, 298 K), δ (TMS, ppm): 1.32 (s, 24H, -CH₃), 7.0-7.1 (m, 14H, ArH), 7.53 (d, I = 7.2 Hz, 4H, ArH). ¹³C NMR (75 MHz, CDCl₃, 298 K), δ (TMS, ppm): 25.1, 83.4, 126.7, 127.8, 127.9, 130.9, 131.5, 134.2, 124.3, 141.4, 143.5, 146.9. MS (FAB), m/z [M⁺]: 584.35, calcd: 584.33. C₃₈H₄₂B₂O₄ (EA) (%, found/calcd): C, 78.83/ 78.10; H, 7.24/7.24.

2.4.1. Synthesis of P1

A mixture of 9,9-dihexylfluorene-2,7-bis(trimethyleneborate) (S4) (125.6 mg, 0.25 mmol), monomer S2 (122.6 mg, 0.25 mmol), sodium carbonate (265 mg, 2.5 mmol), THF (7.5 mL)/water (1.25 mL), and Pd(PPh₃)₄ (4.0 mg) was carefully degassed and charged with argon. Then the reaction mixture was stirred at 60 °C for 3 days. Then, the trace end groups were capped by refluxing for 12 h with phenylboronic acid and bromobenzene sequentially. A lot of methanol was poured into the mixture, then filtered. The obtained solid was dissolved in THF. and the insoluble solid was filtered out. After removal of the solvent, the residue was further purified by several precipitations from THF into acetone, and the obtained solid was then washed with a lot of acetone and dried in a vacuum at 40 °C to a constant weight. The resultant polymer was obtained as dark green powder (113.2 mg, 68.4%). $M_w = 10,900, M_w/M_n = 1.76$ (GPC, polystyrene calibration). IR (KBr), v (cm⁻¹): 1604 (C=C). ¹H NMR (300 MHz, CDCl₃, 298 K), δ (TMS, ppm): 0.6–1.0 (-CH₃), 1.0–1.4 (-CH₂-), 1.8–2.2 (–CCH₂–), 6.8–7.3 (ArH), 7.3–7.8 (ArH). 13 C NMR (75 MHz, CDCl₃, 298 K), δ (TMS, ppm): 14.3, 22.8, 24.0, 30.0, 31.7, 40.7, 55.5, 110.4, 114.3, 120.2, 121.3, 126.5, 128.2, 131.1, 131.6, 132.0, 133.3, 139.6, 140.3, 132.0, 151.9.

2.4.2. Synthesis of P2

The synthesis of **P2** was similar as **P1. S2** (62.1 mg, 0.127 mmol), bronic ester **S5** (72.0 mg, 0.127 mmol), sodium carbonate (134.3 mg, 1.27 mmol), THF (3.9 mL)/water (0.65 mL), and Pd(PPh₃)₄ (2.0 mg). The resultant polymer was obtained as pale green powder (57.1 mg, 69.9%). $M_w = 5700$, $M_w/M_n = 1.53$ (GPC, polystyrene calibration). IR (KBr), v (cm⁻¹): 1601 (C=C). ¹H NMR (300 MHz, CDCl₃, 298 K), δ (TMS, ppm): 6.5–6.7 (ArH), 6.7–7.4 (ArH), 7.4–7.6 (ArH), 7.6–7.8 (ArH). ¹³C NMR (150 MHz, CDCl₃, 298 K), δ (TMS, ppm): 120.2, 120.4, 122.5, 124.2, 126.2, 126.3, 126.4, 126.5, 126.6, 126.9, 127.0, 127.3, 127.8, 127.9, 128.8, 130.9, 131.5, 131.6, 131.8, 131.9, 138.6, 140.5, 140.6, 140.8, 141.1, 141.6, 142.0, 142.8, 142.9, 143.8, 148.8, 149.3, 149.5.

2.4.3. Synthesis of P3

The synthesis of **P3** was similar as **P1**. **S2** (122.6 mg, 0.25 mmol), bronic ester **S6** (132.8 mg, 0.25 mmol), sodium carbonate (265 mg, 2.5 mmol), THF (7.5 mL)/water (1.25 mL), and Pd(PPh₃)₄ (4.0 mg). The resultant polymer was obtained as yellow powder (122.0 mg, 80.3%). $M_w = 11,000$, $M_w/M_n = 1.91$ (GPC, polystyrene calibration). IR (KBr), v (cm⁻¹): 1601 (C=C). ¹H NMR (300 MHz, CDCl₃, 298 K), δ (TMS, ppm): 0.6–1.0 (-CH₃), 1.0–1.4 (-CH₂-), 1.5–2.1 (-CH₂-), 3.8–4.3 (-NCH₂-), 6.9–7.2 (ArH), 7.2–7.6 (ArH), 7.9–8.2 (ArH). ¹³C NMR (150 MHz, CDCl₃, 298 K), δ (TMS, ppm): 11.3, 14.3, 23.3, 24.7, 29.0, 31.2, 39.6, 47.5, 107.4, 118.7, 120.7, 122.1, 126.8, 127.1, 128.0, 128.1, 129.0, 131.8, 132.2, 138.6, 140.1, 141.0, 142.3, 142.9, 144.2.

2.4.4. Synthesis of P4

The synthesis of **P4** was similar as **P1. S3** (116.9 mg, 0.20 mmol), **S7** (76.0 mg, 0.20 mmol), potassium carbonate (276 mg, 2.0 mmol), THF (6 mL)/water (1 mL), and Pd(PPh₃)₄ (3.2 mg). The resultant polymer was obtained as yellow powder (65.1 mg, 59.1%). $M_w = 5600, M_w/M_n = 1.66$ (GPC, polystyrene calibration). IR (KBr), v(cm⁻¹): 1609 (C=C). ¹H NMR (300 MHz, CDCl₃, 298 K), δ (TMS, ppm): 7.0–7.3 (ArH), 7.4–7.5 (ArH), 7.6–7.8 (ArH), 8.1–8.3 (ArH). ¹³C



Fig. 5. A: The EL spectra of P1-P4; B: Luminance-voltage characteristics of the PLED devices; C: Current-voltage characteristics of the PLED devices; D: The luminescence efficiencycurrent characteristics of PLED devices.

NMR (150 MHz, CDCl₃, 298 K), δ (TMS, ppm): 122.7, 126.8, 127.7, 127.9, 128.1, 131.7, 132.4.

2.4.5. Fabrication and characterization of PLEDs

In a general procedure, indium-tin oxide (ITO)-coated glass substrates were etched, patterned, and washed with detergent, deionized water, acetone, and ethanol sequentially. We fabricated PLEDs using these polymers P1-P4 as the emissive layer with a structure of ITO/PEDOT:PSS (25 nm)/Poly-TPD (25 nm)/EML (32 nm)/TPBI (35 nm)/Cs₂CO₃ (8 nm):Ag (100 nm), (ITO was indium-tin oxide; PEDOT was poly(3,4-ethylendioxythiophene); poly(styrenesulfonate); TPD PSS was was N,N'-Bis(3methylphenyl)-N,N'-bis(phenyl)-benzidine and TPBI was 1,3,5tris(N-phenylbenzimidazol-2-yl)benzene), where PEDOT:PSS, Poly-TPD and TPBI were used as hole injection, electron-blocking/ hole-transporting and hole-blocking/electron-transporting layer, respectively. The active layer was spin-coated from chlorobenzene solution and TPBI layer was deposited by means of conventional vacuum deposition onto the ITO coated glass substrates at a pressure of 8 \times 10⁻⁵ Pa. The active area of device was 5 mm². A quartz crystal oscillator placed near the substrate was used to monitor the thickness of each layer, which was calibrated ex situ using an Ambios Technology XP-2 surface profilometer. UV-vis absorption and fluorescence spectra were collected with a Hitachi U-3010 and Hitachi F-4500 spectrophotometer, respectively. EL spectra and chromaticity coordinates were measured with a SpectraScan PR650 photometer. Current density-voltage-luminance (J-V-L) measurements were made simultaneously using a Keithley 4200 semiconductor parameter analyzer and a Newport multifunction 2835-C optical meter, with luminance measured in the forward direction. All device characterizations were carried out under ambient laboratory air at room temperature.

3. Results and discussion

3.1. Synthesis and characterizations

As shown in Scheme 1, the bromized TPE S2 was prepared via McMurry Olefination from 4-bromobenzophenone (S1). The reaction underwent smoothly, and the desired compound was obtained in the yield of 76.2%. Then, during normal procedure for the preparation of bronic ester [9], S3 was obtained as white solid. Other comonomers were obtained by following the reported procedures [1f,9], and the synthetic routes were listed in Scheme S1 in supporting information. The synthetic route of polymers P1-P4 was shown in Scheme 2, and the polymerization procedure was proceeded smoothly through the typical Suzuki coupling reaction using $Pd(PPh_3)_4$ as a catalyst, sodium carbonate (aq) as a base, and THF as the solvent. After three days, the end-capped groups (phenylboronic acid and bromobenzene) were added to react with the bromo- and bronic ester end groups (which might quench the fluorescence of the resultant polymers) [1]. P1-P4 were obtained with satisfied yields (Table 1), and they were readily soluble in common organic solvents, such as THF, CHCl₃, etc.

The molecular structures of these four polymers **P1–P4** were characterized by "wet" spectroscopic techniques, and the detailed spectral analysis data for them were given in the experimental section and supporting information. Fig. S1 (in the supporting information) showed their FT-IR spectra. The TPE-containing compound **S2** exhibited an absorption band at around 1605 cm⁻¹, which could be assigned to the stretching vibrations of C=C. The appearance of the stretching vibration of C=C in the IR spectra of the polymers indicated that the TPE moieties were successfully introduced.

Fig. 1 showed the ¹H NMR spectra of polymers **P1–P4**, which were conducted in the solvent of chloroform-d. All the peaks could be readily assigned to the resonances of appropriate protons of these polymers (Scheme 2), and no unexpected peaks appeared. Except the resonance peaks of TPE unit, there were still some characteristic peaks assigned to the resonances of protons of the other comonomers, which were marked in Fig. 1, indicating the Suzuki polymerization were successful. Taking P3, containing carbazole as comonomer, as an example: the ArH signals of TPE unit were at about δ = 7.0 and 7.5 ppm; except that, there were also some new peaks assigned to the protons of ArH in carbazole (typically at about δ = 8.3 ppm) and the NH₂ protons at about δ = 4.3 ppm, as well as some CH₂ peaks in the range of δ = 0.8–2.0, indicating this polymer was derived from TPE moieties and carbazole moieties. And due to the polymerization, all the peaks showed an inclination of signal broadening. In their ¹³C NMR spectra (see supporting information and experimental section), the similar phenomenon was also observed. In P3, except the signals of TPE unit, some alkyl carbon signals and aromatic carbon signals. which should be derived from carbazole moieties, appeared. In addition, the peaks at about 25 and 83 ppm, typical signals of the boronic ester group, disappeared completely, also indicating the successful polymerization.

The molecular weights of **P1–P4**, determined by gel permeation chromatography (GPC) with THF as an eluent and polystyrene standards as calibration standards, were listed in Table 1 and experimental section. Here, P2 and P4, only constructed by rigid aromatic rings, demonstrated lower molecular weights than those of P1 and P3, possibly due to their relative poor solubility. Actually, during the polymerization procedure of P4, some insoluble polymer was also vielded, indicating that the soluble fractions with high molecular weights could not be obtained. Their thermal gravimetric analysis (TGA) thermograms were shown in Fig. S10, with the 5% weight loss temperature (T_d) summarized in Table 1. All the polymers were thermolytically resistant, and the T_d values were all higher than 360 °C. Here, the rigid structure demonstrated the advantage of good stability, for example, the T_d value of **P2** was as high as 473 °C. Also, the glass transition temperature (T_g) were investigated by using a differential scanning calorimeter (DSC) technology. It was a pity that the glass transition temperatures of nearly all the polymers were not obvious in the DSC curve except P1 (143 °C). Since the structure of P2 and P4 were much more rigid than **P1**, we could conjecture that their T_g values may be even higher than 143 °C, while **P3** might have a similar T_g value as **P1**.

3.2. Optical properties and AIEE effects

As mentioned above, the original goal of the introduction of TPE moieties to PF, was to change its properties from ACQ to AIE effect. Thus, we attempted to investigate the photoluminescence (PL) behaviors of P1-P4 in the solution and aggregate states. The aggregates were prepared by adding distilled water into their THF solutions under vigorous stirring. The resultant mixtures were visually transparent and macroscopically homogenous, suggesting that the polymer aggregates were nanometer-sized [10]. When illuminated with a UV lamp ($\lambda = 365$ nm), there was nearly none fluorescence emitted from the dilute THF solution of P1, while a very weak sky blue florescence observed from its concentrated solutions. But the emission became much stronger in its aggregate states and solid films, and the fluorescence became green light. To verify the visual observations, the UV-vis and PL spectra were studied. As shown in Fig. S11 (Supporting Information), the UV-vis spectra of **P1** in pure THF and THF/water mixtures exhibited similar profiles. However, the PL behaviors were quite different. Upon excitation at 377 nm, P1 was a very weak emitter when molecularly dissolved in pure THF (Fig. 2A). In contrast, when a large amount of water was added into the solution, an intense PL peak centered at 503 nm was recorded under the same measurement conditions (Fig. 2A). Meanwhile, its normalized PL spectra in the solution and aggregate state were showed in Fig. 2B. It was easily seen that in its PL spectra in aggregate state, there was only one main peak centered at 503 nm, while in the solution, except this peak, there were still some small ones. It was understandable. First, we could confirm that the peak around 503 nm was the emission of the aggregated particles, since only this peak remained in the aggregate state and dramatically enhanced in comparison with that in the solution, similar to the optical properties of TPE. As well known, the



Scheme 1. Synthesis of monomers S2 and S3.



Scheme 2. Synthesis of TPE-containing polymers P1-P4.

AIE effect was considered to be induced by the restriction of the intramolecular rotation (IMR) process of the luminophore [7a,7b,11]. In the case of TPE, its four phenyl rings underwent an active IMR process in the solution state, thus, quenched its emission, while in the aggregate state, the IMR process was impeded, which blocked the non-radiative decay channel and hence made TPE emissive. However, in the conjugated polymer of P1, the IMR process of two of the four phenyl rings in the TPE moieties should be limited in some degree, and this peak was therefore still present in the solution state, but very weak. This point, from another side, confirmed that P1 possessed only AIEE characteristic, not AIE. Thus, it was also explained that this emission of 503 nm was the enhanced TPE-centered emission, however, due to the more conjugative system, the emission was red-shifted than TPE (465 nm). On the other hand, just like normal PF, the conjugated system should give some emission, thus, some blue emissions were observed in the concentrated solutions. Coupled with some intermolecular energy transfer effect, these small peaks in the blue region were completely disappeared in the aggregate state or solid state. The same phenomena were also observed in P2-P4 (Figs. S12–S16 in supporting information and Table 2): in their solutions, there were two types of very weak fluorescence, while only one peak in their aggregate states accompanying with the enhanced emission. To confirm the presence of the nanoparticles in the mixture solvents, SEM picture of P1 and P2 were taken as the examples. As shown in Fig. S17, in the mixture solvent of THF/water (1/9 v/v), both of them formed nano-sized particles.

Table 1
Polymerization results of polymers

No.	Yield (%)	M_w^{a}	M_w/M_n^a	$T_g^{\mathbf{b}}(^{\circ}C)$	$T_d^{c}(^{\circ}C)$
P1	68.4	10900	1.76	143	363
P2	69.9	5700	1.53	$(-)^{d}$	473
P3	80.3	11000	1.91	$(-)^{d}$	403
P4	59.1	5600	1.66	$(-)^{d}$	420

^a Determined by GPC in THF on the basis of a polystyrene calibration.

^b The glass transition temperature (T_g) of polymers detected by the DSC analyses

at a heating rate of 10 °C/min under nitrogen. ^c The 5% weight loss temperature of polymers detected by the TGA analyses at a heating rate of 10 °C/min under nitrogen.

^d Not obtained.

To have a quantitative picture, we estimated the Φ_F values of **P1** using 9,10-diphenylanthracene in cyclohexane ($\Phi_F = 0.9$) as reference. In pure THF, the polymer exhibited negligibly Φ_F value of 9.6×10^{-3} , while at a water content of 99%, the Φ_F value of the polymer dramatically increased to 0.194, 21 fold higher than that in THF solution (Fig. 3 and Table 2). All of the above phenomena showed that the PL behavior of **P1**, using TPE units as important construction block, was changed from ACQ to AIEE active as expected. Similar to **P1**, **P2–P4** were also AIEE active, and the fluorescence quantum yields in aggregate states were enlarged in a large degree in comparison with **P1**.

Fluorescent stability of the polymers was another important parameter with regard to real-world applications. Thus, the annealing experiments of these four polymers were conducted to investigate this property. Taking **P1** as an example: the solid film of **P1** was first baked at 100 °C for 30 min in air, then at 150 °C for another 30 min, followed by 200 °C and so on. Fig. S18 showed the PL emission spectra of **P1** after annealing in air. Because of the aggregation effect or keto formation, the stability of normal PF was poor, and it was reported that thermal treatment of the film of poly(dihexylfluorene) at 100 °C would be unstable in air [12]. However, once TPE was introduced, **P1** was still very stable after

Table 2Optical properties of polymers in solution and solid state.

No.	λ _{Abs,sol} a (nm)	$\lambda_{PL,sol}^{b}$ (nm)	λ _{PL,agg} c (nm)	$\lambda_{PL,fim}^{d}$ (nm)	$\Phi_{F,S}^{e}$	$\Phi_{F,A}{}^{\rm f}$	$\Phi_{F,A}/\Phi_{F,S}{}^{\mathrm{g}}$
P1	346	408, 432, 501	503	504	$9.6 imes 10^{-3}$	0.194	21
P2	366	415, 434, 502	506	510	8.5×10^{-3}	0.334	39
P3	352	408, 434, 507	508	507	8.1×10^{-3}	0.282	35
P4	354	408, 432, 466, 504	506	515	$9.8 imes 10^{-3}$	0.302	31

 $^a\,$ The absorption wavelength of polymer solutions in pure THF, the concentration was $2\,\times\,10^{-3}$ mg/mL.

 b The emission wavelength of polymer solutions in pure THF, the concentration was 2 \times 10 $^{-3}$ mg/mL

 c The emission wavelength of polymer solutions in THF/methanol = 1/9 (V/V), the concentration was 2 \times 10 $^{-3}$ mg/mL.

^d The emission wavelength of polymer in films.

 $^{\rm e}\,$ Quantum yields in THF solutions using 9,10-diphenylanthracene in cyclohexane ($\Phi_{\rm F}=90\%)$ as standard.

^f Quantum yields in THF/methanol = 1/9 (V/V) using 9,10-diphenylanthracene in cyclohexane (Φ_F = 90%) as standard.

^g Ratio of fluorescence quantum yields of aggregate and solution.

baked at 150 °C for half an hour, and the PL spectra were almost the same as those tested before annealing. To our surprised, the fluorescent stability of P2 and P4 was even better than P1, possibly due to their much more rigid structure (Figs. S19–S21).

3.3. Electrochemical characterization

The electrochemical properties of the polymers were investigated by using cyclic voltammetry (CV). The onset oxidation potential of P1 occur at 0.87 V. Based on the onset potential and according to the equation of $E_{\text{HOMO}} = -(E_{\text{onset}(\text{ox}),\text{FOC}} + 4.8)$ eV, the HOMO energy level was determined to be -5.67 eV. As the onset reduction potential could not be observed clearly, the LUMO energy level was calculated by the equation of $E_{LUMO} = (E_{HOMO} + E_g^{opt}) \text{ eV}$, where E_{g}^{opt} was the band gap estimated from the absorption spectra of the polymer film. Thus, the LUMO energy level was estimated to be -2.81 eV. The electrochemical data of the other three polymers were also tested and summarized in Table 3. The different copolymer units resulted in different HOMO and LUMO values, further confirming that it was feasible to use different comonomers with different characteristics to adjust the emitting behavior.

3.4. Explosive detection by polymer nanoaggregates

Photoinduced electron transfer (PET) quenching process was usually used to detect nitro aromatic explosives, in particular 2,4,6trinitrotoluene (TNT) and 2,4,6-trinitrophenol (picric acid, PA), the common toxic and explosive pollutants [13]. For the sake of antiterrorism and homeland-security implications, sensitive sensors were badly needed. Here, the strong fluorescence of the nanoaggregates of P1-P4 suspended in the aqueous mixtures made it possible to be utilized as chemosensors toward TNT and PA. Unlike 2,4-dinitrotoluene (DNT) and TNT, PA was commercial availability, thus, it could be used as a model compound to evaluate the sensing property of these polymers.

The explosive probe was prepared as nanoaggregates in the THF/water mixture with 90% water content. After the addition of PA into the aggregate suspension, the fluorescent intensity of P1 decreased rapidly (Fig. 4A), even at a PA concentration as low as 1.0 µg/mL (1.0 ppm). The fluorescence could be quenched completely at a PA concentration of 50 μ g/mL (50 ppm), confirming that the nanoparticle suspension of **P1** could detect nitro aromatic explosives with high sensitivity. The Stern–Volmer plot was nearly linear with a correlation coefficient (R^2) of 0.9855 at the PA concentrations lower than 16 μ g/mL (Fig. 4B). Afterward, the curve deviated from linearity and bent upward, suggesting the occurrence of "super-quenching" due to the involvements of selfabsorption and/or energy transfer [14]. Also, P2-P4 could be used as chemosensors similar to P1, due to their similar AIEE characteristics. Thanks to their much higher quantum yields than P1, the quenching efficiencies of P2–P4 were much better (Table 4), and P3 bearing carbazole moieties possessed the best comprehensive performance. In comparison with other TPE-containing

Tabl	le 3	
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	Electrochemical	properties	of p	olymers
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No.	$E_g^{\text{opt}} (\text{eV})^{\text{a}}$	$E_{\text{onset},(\text{ox})}$ (V) vs FOC ^b	$E_{\rm HOMO}~({\rm eV})^{\rm c}$	$E_{\rm LUMO} ({\rm eV})^{\rm d}$
P1	2.86	0.87	-5.67	-2.81
P2	2.67	0.75	-5.55	-2.38
P3	2.63	0.72	-5.58	-2.95
P4	2.81	0.93	-5.73	-2.92

^a Band gaps obtained from absorption edge ($E_g = 1240/\lambda_{onset}$).

^b $E_{\rm FOC} = 0.48$ V vs Ag/AgCl.

^c $E_{\text{HOMO}} = -(E_{\text{onset}(ox),\text{FOC}} + 4.8) \text{ eV.}$ ^d $E_{\text{LUMO}} = (E_{\text{HOMO}} + E_g^{\text{opt}}) \text{ eV.}$

Table 4

Sensing properties of p	olymers P1	–P4 in ag	gregate state.
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No.	<i>c</i> _{discerned} ^a (ppm)	<i>c</i> _{quench} ^b (ppm)	Linear range ^c (ppm)	$R^{2 d}$
P1	1.0	55	<16	0.9855
P2	0.33	52	<16	0.9570
P3	0.17	40	<20	0.9858
P4	0.17	37	<14	0.9662

^a The lowest detectable PA concentration.

^b The lowest PA concentration for the completely quenched fluorescence.

^c The linear detection range of PA concentration.

^d The correlation coefficient of the linear regression equation.

polymers reported in the literature, **P1–P4** demonstrated higher sensitivity, which should be ascribed to the "molecular wire effect" of the conjugated polymers [3,15].

However, for practical applications, using nanoaggregates as probes was not good enough, and it was more convenient to use in the solid state. Thus, test strips were prepared by immersing filter paper into their THF solutions (0.2 mg/mL) and then dried in air, to investigate if these polymers could be direct used in solid state [16]. The obtained papers were dipped into a toluene solution of PA as well as pure toluene. As shown in Fig. 4C, the paper film displayed strong PL upon excitation, however, became nearly nonemissive after dipping into the PA solution. As to P3 and P4, only 50 μ g/mL of PA was enough to quench the fluorescence completely. These demonstrated a prototype device using the AIEE polymers for detecting explosives in real-world applications.

3.5. Electroluminescence (EL) properties

As mentioned above, the introduction of the TPE unit to PF main chain was to change the fluorescent behavior of PFs, with aim to avoid the ACQ effect. As well known, PF and its derivatives were promising light-emitting materials. Thus, how about the EL properties of the PLED based on P1-P4 with AIEE characteristics? To answer this question, we fabricated PLED devices using these polymers as the emitting layers (EML) according to the "standard recipe" for screening tests: ITO/PEDOT:PSS (25 nm)/Poly-TPD (25 nm)/EML (32 nm)/TPBI (35 nm)/Cs₂CO₃ (8 nm):Ag (100 nm). The EL data of PLED devices were summarized in Table 5 and Fig. 5.

The efficiency of P1-based PLED was not satisfied with a maximum luminance efficiency of 0.20 cd/A and a maximum brightness of 574 cd/m². If using a twisted structure, spirobifluorene, as comonomer instead of normal fluorene, the efficiency of PLED device has a large improvement: the turn-on voltage was 1.8 V decreased, and the maximum luminescence was up to 1761 cd/m². Furthermore, the luminescence efficiency was more than four times than before. On the other hand, to further improve the EL efficiency, it was critical to achieve both efficient charge injection and balanced mobility of both charge carriers inside the

Table 5		
EL data	of PLEDs	devices

No.	$\lambda_{\rm EL}^{a}(nm)$	$V_{\mathrm{on}}{}^{\mathrm{b}}(\mathrm{V})$	Lumin ^c L (cd/m ²)	CD ^d J (mA/cm ²)	CE ^e (cd/A)
P1	494	8.7	574 (16.8)	376	0.20
P2	515	6.9	1761 (15.0)	380	0.82
P3	501	5.7	3609 (12.9)	434	1.17
P4	509	4.8	3109 (12.6)	399	1.14

^a The EL spectra of the device.

^b Turn-on voltage.

^c Maximum luminescence, while the corresponding operating voltage was given in the parentheses.

^d Current density at the maximum luminescence, also the maximum current density.

^e Maximum luminescence efficiency.

electroluminescent polymers. Carbazole, which was very famous for its good hole-transporting and electroluminescent activities, has been already widely used as important block to construct LED materials [17]. 1,3,4-Oxadiazole derivatives were the famous electron deficient materials with good thermal and chemical stabilities as well as high photoluminescence quantum yields, and generally used as electron-transport materials in PLEDs [18]. Thus, in this paper, we also used these two moieties instead of fluorene unit to construct conjugated polymers P3 and P4 to achieve the balance of charge transport, and the results were really better than P2 as expected. For P3-based PLED device, the maximum luminescence was two folds of P2 with a maximum luminance efficiency of 1.17 cd/A. These results suggested that the incorporation of holetransporting carbazole units or electron-transporting oxadiazole units was also effective in improving EL performance of the AIEEbased PLED devices, similar to normal PLED devices. However, to know more information about polymers with AIE or AIEE characteristics, more researches were still needed. Also, it should be pointed out that the EL devices of these polymers were not optimized, thus, better performance might be achieved. Anyhow, these preliminary EL results, especially those of P3 and P4, demonstrated the applicability of AIE or AIEE characteristics polymers.

4. Conclusion

In this paper, for the first time, a new series of TPE-containing conjugated polymers P1-P4 were prepared successfully through simple palladium-catalyzed Suzuki polycondensation reaction, and all polymers exhibited good thermal stabilities ($T_d > 363 \,^{\circ}$ C) and excellent fluorescent stabilities. As expected, the obtained polymers were AIEE active, thanks to the AIE property of TPE moieties. Their AIEE features made them acting as explosive chemosensors both in the aggregate and solid states with high sensitivity. The PLED devices (ITO/PEDOT:PSS (25 nm)/Poly-TPD (25 nm)/P1-P4 (32 nm)/TPBI (35 nm)/Cs₂CO₃ (8 nm):Ag (100 nm)) have been fabricated to investigate their electroluminescent properties. Interestingly, owing to the introduced hole-transporting or electron-transporting groups, the EL performance of P3 and P4 was improved dramatically, with the maximum luminescence of 3609 cd/m^2 (P3), and the maximum luminescence efficiency as high as 1.17 cd/A (**P3**).

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Appendix A. Supplementary material

Supplementary data related to this article can be found online at doi:10.1016/j.polymer.2012.05.035.

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