Magnetic Materials

Fe₃O₄ Core/Layered Double Hydroxide Shell Nanocomposite: Versatile Magnetic Matrix for Anionic Functional Materials**

Liang Li,* Yingjun Feng, Yongsheng Li, Wenru Zhao, and Jianlin Shi*

Recently, micro- or nanospheres with magnetic cores and functional shell structures have received much attention because of their potential applications in catalysis, drug storage/release, selective separation, chromatography, and chemical or biosensors.^[1-10] Until now, different shell frameworks have been developed to pursue these aims. Some groups fabricated mesoporous silica shell to load metal oxides and fluorescence imaging materials and used them in catalysis and drug storage/release systems.^[1-3] Diat et al. discovered that monodisperse and compactly packed onion phases could be obtained in concentrated solutions of surfactants under controlled mechanical shear. Their applications as carriers for drug delivery or nanoreactors for the synthesis of metal nanoparticles have been explored.^[8] Wang et al. used dimercaptosuccinimide acid modified Fe₂O₃ nanoparticles and combined them with CdSe/ZnS quantum dots (QDs) to prepare water soluble magnetic luminescent composites in an organic/water two-phase mixture.^[9] Bawendi and co-workers simultaneously encapsulated both CdSe QDs and y-Fe₂O₃ in silica microspheres to prepare a magnetic luminescent composite.^[10] All these exploitations of the shell structure greatly expand the realm of applications of magnetic core/ shell materials. However, in all of these applications, the method for loading functional materials into the shell structure is largely different from each other, even for the same shell structures. Is there a facile and versatile approach to synthesize magnetic core/functional shell structures?

Layered double hydroxides (LDHs), consisting of stacked brucite-type octahedral layers with anions and water molecules occupying the interlayer space, are currently attracting intense research interest.^[11-13] Recently, we succeeded in total delamination of LDH crystals in nitrate form upon treatment with formamide, and used this product as a building block to

Microstructure, Shanghai Institute of Ceramics, Chinese Acade of Sciences, Shanghai 200050 (China)

- [**] This work was supported by the National Nature Foundation of China (grant no. 20633090), Shanghai Pujiang Program (grant no. 08PJ14035), and Shanghai Nature Science Foundation (grant no. 07ZR14028).
- Supporting information for this article is available on the WWW under http://dx.doi.org/10.1002/anie.200901730.

fabricate the LDH hollow shell structure.^[14] The LDH structure collapsed upon heat treatment at about 500 °C to yield mixed oxides of constituent elements, which underwent restoration of the LDH structure in the presence of water and anions.^[15] This so-called memory effect property of LDHs has been employed as an effective synthetic route for inserting various functional inorganic and organic anions into LDHs galleries.^[16,17]

Herein, we report the synthesis and characterization of a Fe_3O_4 core/oxide shell nanocomposite decomposed from a layered double hydroxide, which combines both versatile anion loading capability and high separation efficiency, making it an ideal support for recoverable anionic functional materials. As shown in Figure 1, the center of the structure is a



Figure 1. Synthesis of the magnetic core/anionic functionalized LDH shell composite structure. The Fe₃O₄ core is first coated with a thin layer of silica, then further coated with a multilayer CO_3^{2-} -LDH composite shell. Afterwards, a calcination process is performed to remove CO_3^{2-} , and the shell become amorphous. Finally, the LDH structure can be restored and anions can be simultaneously intercalated in between the LDH galleries upon immersion in an aqueous solution containing the desired anions.

silica-coated magnetite (Fe₃O₄) core, which is composed of many small Fe₃O₄ crystallites. The Fe₃O₄ cores strongly interact with external magnetic fields and can be easily separated from solution under a moderate magnetic field. The outer surface of the structure is layered double hydroxide. It can be expected that recoverable anionic functional materials will be easily formed through the reconstruction of this thermally decomposed Fe₃O₄ core/LDH shell composite.

Mg–Al LDH powder (mole ratio Mg/Al = 3:1) in carbonate form was obtained from Kyowa Chemical Industry Co., Ltd. Mg–Al LDH was converted into a nitrate form by anion exchange in a typical salt/acid treatment, and subsequently



© 2009 Wiley-VCH Verlag GmbH & Co. KGaA, Weinheim

^[*] Prof. Dr. L. Li, Dr. Y. Feng, Dr. Y. Li, Dr. W. Zhao, Prof. Dr. J. Shi Key Laboratory for Ultrafine Materials of Ministry of Education, School of Materials Science and Engineering, East China University of Science and Technology, Shanghai 200237 (China) Fax: (+ 86) 21-6425-0740 E-mail: liliang@ecust.edu.cn
Prof. Dr. J. Shi State Laboratory of High Performance Ceramic and Superfine Microstructure, Shanghai Institute of Ceramics, Chinese Academy

E-mail: jlshi@sunm.shcnc.ac.cn

delaminated in formamide.^[14a] A typical synthetic procedure of versatile magnetic matrix for anionic functional materials is as follows. Magnetic Fe₃O₄ crystalline particles (ca. 400 nm) were synthesized using a solvent-thermal method.^[18] After being coated by a layer of silica following a sol-gel process reported previously by Shi et al.,^[3] silica-coated magnetite core composites (0.5 g) were dispersed in a formamide suspension (100 mL) containing LDH nanosheets (0.05 g) and then ultrasonically agitated for 20 minutes to promote the adsorption of LDH nanosheets onto the silica surface. The sample was recovered by centrifugation (6000 rpm, 10 min) and washed with ultrapure water. In the next step, the sample was redispersed in an aqueous solution of Na₂CO₃ (100 mL, 2 gL^{-1}). The product was recovered by centrifugation. Fe₃O₄/ SiO₂ cores coated with 20 layer pairs of carbonate and LDH nanosheets $((CO_3^{2-}/LDH)_{20})$ were synthesized by repeating the above procedures 20 times. The obtained sample was heated to 480°C at a ramp rate of 20°Ch⁻¹ under N₂ atmosphere and kept at this temperature for 4 h to remove CO_3^{2-} and water. Finally, the calcined material was dispersed into aqueous solution to recover its original LDH structure and in the meantime absorb the functional anions into the LDH galleries.

Figure 2 depicts X-ray diffraction (XRD) data for the sample at various stages of the fabrication of Fe_3O_4 core/LDH shell nanocomposite. Characteristic peaks from the Fe_3O_4



Figure 2. XRD patterns of Fe₃O₄ core alone (a); after deposition of 20 LDH/CO₃²⁻ layer pairs (b); after calcination at 480 °C for 4 h (c); and after treatment in $W_7O_{24}^{6-}$ aqueous solution (d). (Diffraction peaks of \bullet :Fe₃O₄; \blacktriangle : carbonate–LDH; \blacksquare : $W_7O_{24}^{6-}$ –LDH; \star : polyoxometalate–LDH).

core in an angular range of 18–65° remained intact during the complete process. The sample coated with 20 layer pairs of LDH nanosheets/CO₃^{2–} shows two major peaks at 2 θ values of 11.3 and 22.7° (indicated by triangles), which can be ascribed to the formation of carbonate LDH shell nanostructure with a repeating distance of approximately 0.78 nm; a similar value has been reported many times for carbonate LDH in power form.^[19] One additional new peak at 60.5° can be assigned to intrasheet reflections of 110 peaks from a twodimensional hexagonal cell (a = 0.31 nm),^[20] confirming that the LDH nanosheet architecture remained intact in the layerby-layer assembly process.

This kind of LDH core/shell structure possesses versatile anion loading capability after calcination because of its memory effect, which makes them ideal supports for recoverable anionic functional materials. Herein, we chose $W_7O_{24}^{6-}$ as an example and investigated the photodegradation behavior of aqueous organochlorine pesticide by using this Fe₃O₄ core/W₇O₂₄⁶⁻ LDH as a model system to demonstrate the use of our core/shell structure as a versatile recoverable catalyst support.

The original carbonate LDH shell structure was destroyed after heating at 480 °C under N2 atmosphere for 4 h. The XRD pattern (Figure 2c) does not show any peak other than that from the Fe₃O₄ core, suggesting the transformation of the LDH shell to amorphous metal oxides such as MgO and Al₂O₃. When this Fe₃O₄ core/amorphous shell material was dispersed into the prepared aqueous solution at 50°C to absorb the $W_7O_{24}^{6-}$ for 24 h, the sample became crystalline again (Figure 2 d). The four major peaks (at 2θ values of 7.2°, 14.5°, 21.6°, and 29.0°) can be ascribed to a basal diffraction series with an interplanar spacing of 1.2 nm. The additional peak at 60.5° can be identified as two-dimensional diffraction peaks of the 110 plane for a hexagonal cell with a = 0.31 nm. The XRD data strongly suggest an anion inserting process and reconstruction of the LDH structure by the layer memory effect. The basal spacing of 1.2 nm is characteristic of LDHs in $W_7O_{24}^{6-}$ form.^[17] Furthermore, compared with the original calcined sample, there is also a broad diffraction peak around a 2θ value of 10° (indicated by the asterisk), which is the fingerprint of polyoxometalates pillared LDH.^[20] This result is further confirmation that $W_7 O_{24}^{6-}$ ions have been successfully inserted into the LDH gallery. In addition, the average shell thickness of the reconstructed Fe₃O₄ core/W₇O₂₄⁶⁻ LDH shell composite is calculated by using Sherrer's equation on the (003) peak to be around 20 nm.

Figure 3c shows a scanning electron microscopy (SEM) image of the obtained Fe_3O_4/SiO_2 core/ CO_3^{2-} LDH shell composite. The spherical morphology of monodisperse Fe_3O_4 core/silica shell beads was preserved well after the deposition of the 20 layers of carbonate LDH shell. The LDH nanosheets can rarely be identified in SEM images because of their high flexibility. The only noticeable difference between the spheres with or without shells was the surface roughness. The spherical morphology was well preserved and there is no visible unwrapped core or separate irregular particles in the SEM image. The CO_3^{2-} -LDH shell thickness is estimated from Figure 3 d to around 15 nm.

When calcined at 480 °C, the Fe₃O₄ core/carbonate shell sample lost 4.2 % of its weight in two steps (Figure 4). These weight losses were accompanied by huge endothermic peaks, suggesting the decomposition of the carbonate LDH shell. The endothermic peak at 214 °C can be assigned to dehydration of the carbonate LDH layer, and two peaks at about 320 °C and 412 °C can be assigned as dehydroxylation or the collapse of the hydroxide layers, which overlaps the decomposition of CO_3^{2-} to CO_2 , as shown in Figure 5. These data are consistent with those in the literature.^[18] Slow heating at a rate of 2 °C min⁻¹ was essential to retain the shell structure. Rapid

Communications



Figure 3. SEM images of Fe₃O₄ core (a), Fe₃O₄ core covered with SiO₂ layer (b), Fe₃O₄/SiO₂ core/CO₃²⁻-LDH shell composite (c), the core shell structure after calcination (e), and the Fe₃O₄/SiO₂ core/W₇O₂₄⁶⁻-LDH shell structure after reconstruction (g); TEM images of Fe₃O₄/SiO₂ core/carbonate LDH shell composite (d), the core shell structure after calcination (f), and Fe₃O₄/SiO₂ core/W₇O₂₄⁶⁻-LDH shell structure after reconstruction (g); TeM images of Fe₃O₄/SiO₂ core/W₇O₂₄⁶⁻-LDH shell structure after reconstruction (h).

heating of the sample could disrupt the shell structure as a result of violent evolution of steam and carbon dioxide.

The FTIR data shown in Figure 5 are additional evidence for the chemical and structural modifications of the sample described above. The as-prepared silica covered Fe₃O₄ core/ carbonate LDH shell nanocomposite showed sharp adsorption bands attributable to CO_3^{2-} and SiO₂. A strong peak at 1350 cm⁻¹ can be assigned as a vibration mode of carbonate ions and a peak at 1090 cm⁻¹ can be assigned as a stretching mode of Si-O-Si, indicating the formation of carbonate LDH shell on the surface of Fe₃O₄/SiO₂ core after the layer-by-layer



Figure 4. Thermogravimetry/differential thermal analysis (TG-DTA) diagram of Fe $_3O_4$ core/carbonate LDH shell composite.



Figure 5. FTIR spectra of Fe₃O₄/SiO₂ core initially (a), after layer-bylayer coating with carbonate LDH (b), after calcination at 480 °C for 4 h (c), and after reconstruction in $W_7O_{24}^{6-}$ aqueous solution (d). •: adsorption bands of water molecules; •: $W_7O_{24}^{6-}$ ions.

process. In addition to these peaks, there were strong adsorption bands at 3700–3000 and at 1630 cm⁻¹, which are attributable to the stretching and bending vibrations of water molecules. This result suggests that the LDH/CO₃^{2–} shell has been hydrated. In contrast, the spectrum of the calcined sample was rather featureless as a consequence of the removal of interlayer ions and the collapse of LDH layers. The reconstruction of the LDH structure after mixing with the W₇O₂₄^{6–} aqueous solution is clear from the bands at 3400 and 1630 cm⁻¹ (labeled with black dots). Furthermore, some additional peaks at 950, 905, 833, 661, and 581 cm⁻¹ (indicated by triangles) appeared. All these bands can be ascribed to the vibration mode of W₇O₂₄^{6–} form.

After the shell structure recovered from the amorphous metal oxides to the LDH structure, the spherical morphology is still apparent (Figure 3e,f). Compared with the sample before reconstruction, the shell thickness became somewhat larger and the surface became rougher. The high-resolution TEM (HRTEM) image shows a lamellar structure with a repeating fringe of 1.2 nm (Figure 3h), and a shell thickness about 20 nm, which is consistent with the XRD results.

The content of $W_7O_{24}^{6-}$ in the resulting catalyst is about 8.0 wt %, as determined by chemical analysis. The activity of this magnetic recoverable catalyst was tested with the photo-degradation of trace hexachlorocyclohexane (HCH) in aqueous solution (Figure 6). After the suspension was stirred in the



Figure 6. Mineralization of HCH as a function of irradiation time in the presence of magnetic $W_7O_{24}^{6-}$ catalyst.

dark for 12 h or irradiated with a high pressure mercury lamp (HPML) for 12 h in the absence of catalyst, there was no apparent change in the concentration of HCH, and Cl- and CO₂ were not detected. However, a significant transformation of HCH into Cl⁻ and CO₂ was observed after irradiating the suspension containing synthesized catalyst with the HPML. In the reaction system, the concentration of the product Cl⁻ ions increased with the reaction time, suggesting that the degradation of aqueous HCH was the result of photoexcited catalysis rather than direct photolysis. Complete mineralization of HCH was demonstrated by 96.3 % chlorine recovered as Cl⁻ for 10 mgL⁻¹ HCH after 12 h irradiation of the suspension. Photocatalytic activity was not observed with the Fe₃O₄ core/carbonate LDH shell nanocomposite. The magnetic separation and recycling of the catalyst were also investigated with the same reaction (Figure 7). The catalyst could be effectively recycled and reused six times without any apparent decrease in its catalytic activity (Table 1). This catalyst also shows high catalytic activities in oxidative



Figure 7. Photodegradation of HCH and magnetic separation of the catalyst.

Angew. Chem. Int. Ed. 2009, 48, 5888-5892

Table 1: Magnetic separation and recycling of the catalyst in the photodegradation of HCH.

Cycle	1	2	3	4	5	6
Yield [%] of photodegradation	96.3	95.8	96.1	96.5	95.6	96.4

alkoxylation of mesitol with various alcohols (see the Supporting Information). $^{\left[21-23\right] }$

In summary, we have demonstrated the fabrication of magnetic core/LDH shell structure and its use as recoverable support for nanocatalysts. This composite structure possesses high anion loading capacity and can be conveniently separated. The success in the assembly of the $W_7O_{24}^{6-}$ catalyst may provide a general route to the facile and direct fabrication of magnetic core/various anion-functionalized shell composite structures. The magnetic nanocomposite catalyst system demonstrated herein is expected to find many important applications in catalysis. For example, the oxide shell could be used as an absorbent to carry anionic drugs and used in a drug storage/release system.

Experimental Section

X-ray diffraction (XRD) data were collected using a Rigaku Rint-2000 powder diffractometer with graphite monochromated Cu KR radiation ($\lambda = 0.15405$ nm) Transmission electron microscopy (TEM) observations were performed on a field emission JEM-3000F (JEOL) electron microscope operated at 300 kV equipped with a Gatan-666 electron energy loss spectrometer and energy-dispersive X-ray spectrometer. Scanning electron microscopy (SEM) images were obtained using a JEOL JSM-6700F electron microscope (accelerating voltage of 10 kV). FTIR spectra were obtained on Nicolet 7000-C spectrometer.

Photocatalytic procedure: The temperature of the system was maintained at 28 °C. The light source was a 125 W high-pressure mercury lamp. A general photocatalytic procedure was carried out as follows. The Fe₃O₄ core/layered double hydroxide shell nanocomposite (0.5 g) were suspended in a fresh aqueous solution of HCH. The suspension was ultrasonicated for 5 min, stirred in the dark for 30 min, and then irradiated with the lamp. The suspension was vigorously stirred during the irradiation process.

Received: March 31, 2009 Revised: May 20, 2009 Published online: July 2, 2009

Keywords: heterogeneous catalysis · layered materials · magnetic properties · nanostructures

- M. Shokouhimehr, Y. Piao, Y. Kim, Y. Jang, T. Hyeon, Angew. Chem. 2007, 119, 7169–7173; Angew. Chem. Int. Ed. 2007, 46, 7039–7043.
- J. Kim, H. S. Kim, N. Lee, T. Kim, H. Kim, T. Ku, I. C. Song, W. K. Moon, T. Hyeon, Angew. Chem. 2008, 120, 8566–8569; Angew. Chem. Int. Ed. 2008, 47, 8438–8441.
- [3] W. Zhao, J. Gu, L. Zhang, H. Chen, J. Shi, J. Am. Chem. Soc. 2005, 127, 8916–8917.
- [4] H. Zhao, J. F. Chen, Y. Zhao, L. Jiang, J. W. Sun and J. Yun, Adv. Mater. 2008, 20, 3682–3686.
- [5] X. Hong, J. Li, M. Wang, J. Xu, W. Guo, J. Li, Y. Bai, T. Li, *Chem. Mater.* 2004, 16, 4022–4027.
- [6] J. Ge, T, Huynh, Y. Hu, Y. Yin, Nano Lett. 2008, 8, 931-934.

Communications

- [7] E. V. Shevchenko, M. I. Bodnarchitk. M. V. Kovalenko, D. V. Talapin, R. K. Smith, S. Aloni, W. Heiss, A. P. Alivisatos, *Adv. Mater.* 2008, 20, 4323–4329.
- [8] O. Diat, D. Roux, F. Nallet, J. Phys. II 1993, 3, 1427-1452.
- [9] D. Wang, J. He, N. Rosenzweig, Z. Rosenzweig, *Nano Lett.* 2004, 4, 409–413.
- [10] N. Insin, J. B. Tracy, H. Lee, J. P. Zimmer, R. M. Westervelt, M. G. Bawendi, ACS Nano 2008, 2, 197–202.
- [11] B. Sels, D. De Vos, M. Buntinx, F. Pierard, A. K. Mesmaeker, P. Jacobs, *Nature* **1999**, 400, 855–857.
- [12] J.-H. Choy, J.-M. Oh, M. Park, K.-M. Sohn, J.-W. Kim, Adv. Mater. 2004, 16, 1181–1184.
- [13] M. Darder, M. López-Blanco, P. Aranda, F. Leroux, E. Ruiz-Hitzky, *Chem. Mater.* 2005, 17, 1969–1977.
- [14] a) L. Li, R. Ma, Y. Ebina, N. Iyi, T. Sasaki, *Chem. Mater.* 2005, *17*, 4386–4391; b) L. Li, R. Ma, N. Iyi, Y. Ebina, K. Takada, T. Sasaki, *Chem. Commun.* 2006, 3125–3127.

- [15] S. Miyata, Clays Clay Miner. 1980, 28, 50-55.
- [16] T. Hibino, A. Tsunashima, Chem. Mater. 1998, 10, 4055-4061.
- [17] Y. Guo, D. Li, C. Hu, Y. Wang, E. Wang, Y. Zhou, S. Feng, *Appl. Catal. B* 2001, *30*, 337–349.
- [18] H. Deng, X. Li, Q. Peng, X. Wang, J. Chen, Y. Li, Angew. Chem. 2005, 117, 2842–2845; Angew. Chem. Int. Ed. 2005, 44, 2782– 2785.
- [19] Handbook of Layered Materials (Eds.: S. M. Auerbach, K. A. Carrado, P. K. Dutta), Marcel Dekker, New York, 2004.
- [20] Intercalation Chemistry (Eds.: M. S. Whittingham, A. J. Jacobson), Academic Press, New York, **1982**.
- [21] B. F. Sels, D. E. De Vos, P. A. Jacobs, J. Am. Chem. Soc. 2007, 129, 6916-6926.
- [22] B. F. Sels, D. E. De Vos, M. Buntinx, P. A. Jacobs, J. Catal. 2003, 216, 288–297.
- [23] B. F. Sels, D. E. De Vos, P. A. Jacobs, Angew. Chem. 2005, 117, 314–317; Angew. Chem. Int. Ed. 2005, 44, 310–313.