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Construction of Efficient Deep Blue AIE Luminogen from Triphenylethene for Non-Doped OLED Applications

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ABSTRACT: Deep blue emitters are crucial for full color displays and organic white lighting. Thanks to the research efforts by scientists, many efficient light emitters with aggregation-induced emission (AIE) characteristics have been synthesized and found promising applications in organic light-emitting devices (OLEDs). However, few AIE emitters with deep blue emissions and excellent electroluminescence (EL) performance have been reported. The contribution here reports a simple but successful molecular design strategy for synthesizing efficient solid-state emitters for non-doped OLEDs with both deep blue and white emissions. This strategy utilizes triphenylethene, a weakly conjugated AIEgen as building block for constructing deep blue emitter, involving no complicated control on emission color through adjustment of the steric hindrance of chromophores and enables a wide selection of partnered functional units. The synthesized AIE luminogen, abbreviated as BTPE-PI, is thermally stable and exhibits high fluorescence quantum efficiency as well as good charge injection capability in the solid state. Non-doped deep blue OLED fabricated from BTPE-PI shows a very high external quantum efficiency of 4.4% with a small roll-off, whose performance is the best among deep blue AIE materials reported so far. An efficient white OLED with Commission Internationale de l'Eclairage (CIE) coordinates of (0.33, 0.33) at theoretical white point was first achieved by using AIE luminogen BTPE-PI as deep blue emitter. Such molecular design strategy opens a new avenue in the development of efficient solid-state deep blue emitters for non-doped OLED applications.

INTRODUCTION

Development of efficient luminescent materials with primary RGB (red, green and blue) colors in the solid state is critical for the applications in organic light-emitting diodes (OLEDs).¹ Among the primary RGB colors, deep blue emission is important for full color displays and organic white lighting.^{2,3} It can also offer a wide-gamut RGB color space, which possesses high saturated and "pure" light colors.⁴ On the other hand, it can be used as an excitation source to activate other color emissions through energy transfer.⁵ However, luminophores with efficient deep blue emission are rare. Despite of their high external quantum efficiency in electroluminescence (EL) devices, organometallic complexes with deep blue emissions are rarely observed because of the consumption of the energy of the high level metal-ligand charge transfer bond by the metal *d* orbitals through non-radiative relaxation channel.⁶⁻⁸ Therefore, considerable

attentions have been placed on fluorophores with this emission color. Even such emitters are prepared, it is challenging to utilize them for fabricating OLEDs with good performance as their wide band-gaps⁹ tend to create large charge injection barrier and unbalanced charge injection and transportation, leading to increased driving voltage and lowered device efficiency.¹⁰ On the other hand, a problem associated with most blue luminogenic materials is the aggregation-caused quenching (ACQ) of light emission in the aggregated state: their intense emission in the solution state is weakened¹¹ and red-shifted to undesirable sky blue color when their molecules are aggregated¹² due to the severe intermolecular interaction.^{13,14}

Several strategies have been adopted to solve the abovementioned problems. To improve the charge balance in OLEDs and hence enhance the EL efficiency, introduction of donor (D) and acceptor (A) units into the luminogenic structure^{15,16} has been proved to be effective

method because of the creation of the intramolecular charge transfer (ICT) effect. However, the extent of ICT effect should be well controlled as strong D-A pair tends to induce excessive bathochromic shift and normally impairs the emission color purity.¹⁷ On the other side, to mitigate the ACQ effect, methods of cross dipole stacking,¹⁸ *J*-aggregate formation^{19,20} and introduction of bulky substituents^{11,21–24} have been employed. However, these methods are far from satisfactory. For example, to achieve a particular stacking mode, careful molecular design and control on the luminogen aggregation are required. Integration of bulky alkyl groups^{25–27} is often accompanied with severe side effects, such as dilution of chromophore concentration and obstruction of the charge injection and transportation in EL devices.

We observed a phenomenon of aggregation-induced emission (AIE) in some organic luminophores with twisted conformations.^{12,28–30} These molecules are non-emissive in dilute solutions because of their active intramolecular motions, which consume the energy of the excitons through non-radiative relaxation pathways. Such motions, however, are restricted in the aggregated state, which enables the excitons to decay radiatively. The twisted conformations of the molecules also prevent the formation of detrimental species such as exciplexes and excimers by strong intermolecular interactions. All these factors make the AIE luminogens to show strong light emission in the aggregated state. Integration of AIE units to conventional ACQ chromophores has been found to be a good strategy to solve their ACQ problem and generates new luminogens with high fluorescence quantum efficiencies and inherited properties from both the parent molecules.^{31,32} Such design strategy thus offers a formulated platform to generate functional luminescent materials with efficient solid-state emissions.

Limited by the long conjugation length of some typical AIE luminogens such as tetraphenylethene and silole,^{33,34} it is difficult to achieve deep blue emission when using them as building blocks. Few such emitters have been prepared but their design involves complicated control on their steric hindrance.^{35,36} On the other hand, as these emitters generally possess highly twisted structures, their EL performances are only moderate due to the poor charge transportation. Thus, a simple and direct method to tackle the above problem is to find an AIE luminogen with efficient emission but low conjugation. With such AIE unit, complicated design and synthesis of luminogens with highly twisted structures or severe steric hindrance are not needed and meanwhile a much wider partnered functional units can be selected for luminogen construction. Following this idea, triphenylethene (TPE) is selected as the main molecular skeleton. Although it possesses a low conjugation, it and its derivatives show splendid AIE characteristics.^{37,38} Compared to tetraphenylethene in the solid state, its emission is much bluer, exhibiting a hypsochromatic shift in the emission maximum of as large as 25 nm (Figure S1 in the Supporting Information). Such property should significantly improve the color purity of its derivatives. On the other side, we utilized phenanthro[9,10-d]imidazole (PI) as a deep blue building block for its good

thermal and carrier injection properties.^{39,40} Last, covalent combination of these two parts produce a luminogen, where one TPE moiety (A) is connected to the phenyl ring of PI at the *para* position to impart good conjugation and hence improved charge transportation.⁴¹ The another TPE unit (B) is linked laterally at the *meta* position to lower the conjugation and hence shorten the emission wavelength (Chart 1). This also endows the molecule a twisted conformation, which can prevent undesirable intermolecular interactions in the solid state.

In this paper, we demonstrate a new molecular design strategy and report the synthesis of an efficient deep blue AIE luminogen constructed from triphenylethene and PI. The synthesized luminogen, abbreviated as BTPE-PI, is thermally stable with balanced carrier injection properties. Non-doped multilayer EL devices fabricated by using BTPE-PI as emitting layer, showed deep blue EL with external quantum efficiency of as high as 4.4% and a small roll-off. In addition, efficient white EL device with Commission Internationale de l'Eclairage (CIE) coordinates of (0.33, 0.33) at the exact theoretical white point was first achieved by employing AIE luminogen (BTPE-PI) as the blue deep emitter. Such good results indicated that such strategy on molecular design shed some light on the development of efficient deep blue luminescent materials for OLED applications.

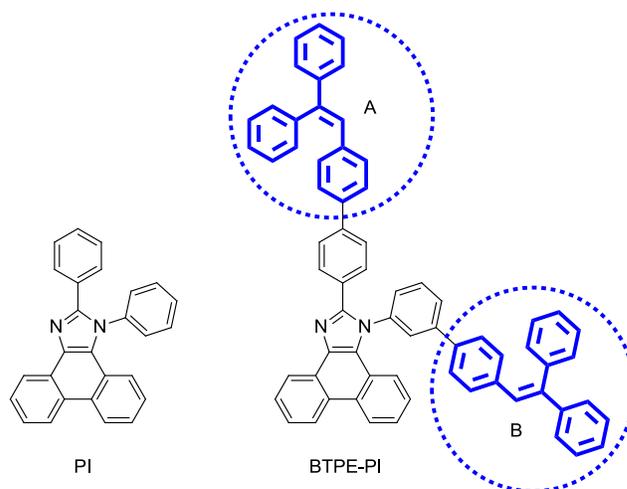


Chart 1. Chemical structures of PI and BTPE-PI.

EXPERIMENTAL SECTION

Materials. THF was distilled from sodium benzophenone ketyl under dry nitrogen immediately prior to use. Intermediates **5**⁴², **6**⁴³, **7**⁴⁴ and tetraphenylethene⁴⁵ were prepared according to the reported experimental procedures. Other chemicals and reagents were purchased from Aldrich and used as received without further purification.

Characterization. ¹H and ¹³C NMR spectra were measured on a Bruker AV 400 spectrometer in CD₂Cl₂ or CDCl₃ using tetramethylsilane (TMS; δ = 0) as internal reference. High resolution mass spectra (HRMS) were recorded on a GCT premier CAB048 mass spectrometer operating in MALDI-TOF mode. Elemental analysis was operated on a ThermoFinnigan Flash EA1112 analyzer. UV

spectra were measured on a Milton Roy 5 Spectronic 3000 Array spectrophotometer. Photoluminescence spectra were recorded on a Perkin-Elmer LS 55 spectrofluorometer. The solution fluorescence quantum yields were determined by using 9,10-diphenylanthracene as standard ($\Phi_F = 90\%$ in cyclohexane). The absorbance of the solutions was kept below 0.1 to avoid the internal filter effect. The emission efficiencies of thin films of PI and BTPE-PI were measured on a calibrated integrating sphere. Their thermal stabilities were measured on TGA Q5000 and DSC Q1000 instruments under nitrogen at a heating rate of 20 °C/min. Cyclic voltammetry (CV) experiments were carried out in dichloromethane solution with 0.1 M tetrabutylammonium hexafluorophosphate as the supporting electrolyte at a scan rate of 100 mV/s, by using Ag/AgNO₃ as the working electrode and saturated calomel electrode (SCE) as the reference electrode.

Synthesis. Preparation of PI: The compound was prepared according to the reported methods⁴⁶ with a slight modification. Into a stirred mixture of 9,10-phenanthrenequinone (624 mg, 3 mmol), benzaldehyde (318 mg, 3 mmol), aniline (1.4 g, 15 mmol) and ammonium acetate (2.87 g, 37.2 mmol), 20 mL of glacial acetic acid was added under nitrogen. The mixture was heated to reflux for 12 h. After cooling to room temperature, a pale yellow mixture was obtained and poured into a methanol solution under stirring. The precipitates were filtered, washed with methanol, and dried to give a pale yellow solid. After filtration and solvent evaporation under reduced pressure, the product was purified by silica-gel column chromatography using hexane/dichloromethane as eluent. It was further purified by stirring in reflux ethanol. After subsequent filtration and drying in vacuum, PI was obtained in 53.2% yield (591 mg) as a white powder. ¹H NMR (400 MHz, CD₂Cl₂, δ): 8.79 (t, $J = 8.0$ Hz, 2H), 8.74 (d, $J = 8.4$ Hz, 1H), 7.85–7.83 (m, 1H), 7.76 (td, $J = 8.0$, 0.8 Hz, 1H), 7.72–7.67 (m, 2H), 7.59–7.46 (m, 7H), 7.35 (td, $J = 8.0$, 0.8 Hz, 1H), 7.22 (d, $J = 8.4$ Hz, 1H). ¹³C NMR (100 MHz, CD₂Cl₂, δ): 150.01, 140.19, 137.80, 133.75, 132.45, 131.97, 131.91, 131.21, 129.72, 129.64, 128.69, 128.62, 128.39, 127.83, 127.45, 126.98, 126.22, 125.62, 124.59, 123.89, 123.84, 123.62, 123.13, 122.89, 121.19. HRMS (MALDI-TOF), m/z : M^+ , calcd. for C₂₇H₁₈N₂, 370.1470; found, 370.1479.

Preparation of BBr-PI: The compound was prepared from 9,10-phenanthrenequinone (624 mg, 3 mmol), 4-bromobenzaldehyde (555 mg, 3 mmol), 3-bromoaniline (2.58 g, 15 mmol, 1.65 mL), ammonium acetate (2.87 g, 37.2 mmol) and 20 mL of glacial acetic acid by following the same procedure described above, affording a white solid in 51.0% yield (808 mg). ¹H NMR (400 MHz, CD₂Cl₂, δ): 8.79 (t, $J = 8.0$ Hz, 2H), 8.74 (d, $J = 8.4$ Hz, 1H), 7.85–7.83 (m, 1H), 7.76 (td, $J = 8.0$, 0.8 Hz, 1H), 7.72–7.67 (m, 2H), 7.59–7.46 (m, 7H), 7.35 (td, $J = 8.0$, 0.8 Hz, 1H), 7.22 (d, $J = 8.4$ Hz, 1H). ¹³C NMR (100 MHz, CD₂Cl₂, δ): 150.01, 140.19, 137.80, 133.75, 132.45, 131.97, 131.91, 131.21, 129.72, 129.64, 128.69, 128.62, 128.39, 127.83, 127.45, 126.98, 126.22, 125.62, 124.59, 123.89, 123.84, 123.62, 123.13, 122.89, 121.19. HRMS (MALDI-TOF), m/z : M^+ , calcd. for C₂₇H₁₆Br₂N₂, 527.9660; found, 527.9673.

Preparation of BTPE-PI: Into a stirred mixture of BBr-PI (264 mg, 0.5 mmol), 4-(2,2-diphenylvinyl)phenylboronic acid (600 mg, 2.0 mmol), Pd(PPh₃)₄ (57 mg) and K₂CO₃ (1.24 g, 9 mmol), 40 mL of THF and 5 mL of water was added under nitrogen. The mixture was heated to 70 °C for 24 h. After cooling to room temperature, the solution was extracted with dichloromethane (150 mL \times 2), washed with water and dried over Na₂SO₄. After filtration and solvent evaporation under reduced pressure, the product was purified by silica-gel column chromatography using hexane/dichloromethane as eluent. BTPE-PI was obtained in 54% yield (237 mg) as a white powder. ¹H NMR (400 MHz, CD₂Cl₂, δ): 8.82–8.78 (m, 2H), 8.73 (d, $J = 8.4$ Hz, 1H), 7.86 (dd, $J = 8.0$, 1.2 Hz, 1H), 7.77–7.74 (m, 2H), 7.69–7.65 (m, 4H), 7.54–7.49 (m, 4H), 7.40–7.27 (m, 22H), 7.24–7.21 (m, 2H), 7.20–7.17 (m, 2H), 7.10–7.06 (m, 4H), 7.02 (s, 1H), 7.0 (s, 1H). ¹³C NMR (100 MHz, CD₂Cl₂, δ): 150.81, 143.70, 143.56, 143.35, 143.04, 140.92, 140.83, 140.70, 139.71, 138.45, 137.89, 137.81, 137.62, 137.36, 130.99, 130.67, 130.60, 130.51, 130.41, 129.99, 129.87, 129.49, 129.14, 128.69, 128.60, 128.42, 128.02, 127.93, 127.89, 127.70, 127.61, 127.50, 126.88, 126.83, 126.75, 126.69, 125.97, 125.31, 124.49, 123.60, 123.47, 122.90, 121.33. HRMS (MALDI-TOF), m/z : M^+ , calcd. for C₆₇H₄₆N₂, 878.3661; found, 878.3678. Anal. calcd (%) for C₆₇H₄₆N₂: C 91.54, H 5.27, N 3.19; found: C 91.44, H 5.37, N, 3.20.

Preparation of nanoaggregates. Stock THF solutions of PI and BTPE-PI were prepared with a concentration of 10⁻⁴ M. Aliquots of the stock solutions were transferred to 10 mL volumetric flasks. After appropriate amounts of THF were added, water was added dropwise under vigorous stirring to prepare 10⁻⁵ M solutions with different water contents (0–90 vol %). The PL spectra of the resultant solutions were immediately measured.

Device fabrication. The devices were fabricated on 80 nm ITO-coated glass with a sheet resistance of 25 Ω per square. Before loading into the pretreatment chamber, the ITO-coated glasses were soaked in ultrasonic detergent for 0.5 h, followed by spraying with deionized water for 10 min, soaking in ultrasonic deionized water for 0.5 h, and oven-baking for 1 h. The cleaned samples were treated by fluorine plasma (CHF₃) with a power of 10 W, gas flow of 50 sccm, and pressure of 0.2 Torr for 10 s in the pretreatment chamber. The solid samples were transferred to the organic chamber with a base pressure of 5 \times 10⁻⁷ Torr for the deposition of NPB, BTPE-PI and TPBi. The samples were then transferred to the metal chamber for the deposition of cathode, which was composed of lithium fluoride (LiF) capped with aluminium (Al). The light-emitting area was 4 mm² as defined by the overlap of the cathode and anode. The current density–voltage–luminance characteristic curves of the devices were determined by a HP4145B semiconductor parameter analyzer and a calibrated UDT PIN-25D silicon photodiode. The luminance and external quantum efficiencies of the devices were inferred from the photocurrent of the photodiode. The EL spectra were measured by a PR650 spectrophotometer. All the measurements were performed at room temperature under air without device encapsulation.

RESULTS AND DISCUSSION

Synthesis and Characterization. BTPE-PI and PI were prepared facilely according to the synthetic routes shown in Scheme S1. The key intermediate BBr-PI was prepared from 9,10-phenanthrenequinone (**1**) according to previously published synthetic procedures.⁴⁶ BTPE-PI was obtained by Suzuki coupling of BBr-PI and 4-(2,2-diphenylvinyl)phenylboronic acid (**7**) using Pd(PPh₃)₄ as a catalyst under basic conditions. All the intermediates and product were characterized by standard spectroscopic techniques, which gave satisfactory analysis data corresponding to their chemical structures and demonstrated their high purity. Detailed synthetic procedures and examples of ¹H NMR and mass spectra were shown in Figures S2 and S3.

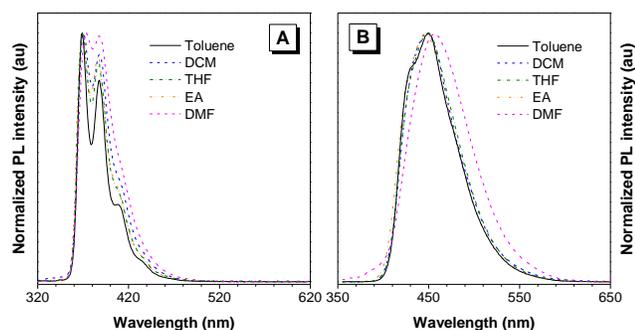


Figure 1. Normalized PL spectra of (A) PI and (B) BTPE-PI in different solvents. Excitation wavelength (nm): 312 (PI) and 340 (BTPE-PI).

Optical properties. BTPE-PI possesses a good solubility in common organic solvents, such as dichloromethane, chloroform and tetrahydrofuran (THF), but is insoluble in water. The UV spectra of PI and BTPE-PI measured in THF are peaked at 312 and 340 nm, respectively (Figure S4). The associated molar absorptivity ($7.2 \times 10^4 \text{ L mol}^{-1} \text{ cm}^{-1}$) for BTPE-PI is 3-fold higher than that of PI due to its more conjugated structure. In THF solution, PI emits a strong purple photoluminescence (PL) at 370 and 387 nm, while the PL spectrum of BTPE-PI was centered in the blue region at ~ 449 nm. The PL spectra of PI and BTPE-PI measured in different solvents are shown in Figure 1. When the solvent is changed from nonpolar toluene to polar *N,N*-dimethylformamide, the emission maximum red-shifts but in a small extent (~ 4 nm for PI and ~ 6 nm for BTPE-PI). Under similar conditions, the emission of 4-(*N,N*-dimethylamino)benzonitrile, a typical strong D-A molecule, shifts bathochromically by ~ 100 nm.⁴⁷ This suggests that both PI and BTPE-PI are insensitive to solvent polarity and show a weak D-A interaction, which is beneficial for the improvement of charge balance in OLEDs but will not cause excessive emission red-shift. Their PL properties are also studied in THF/water mixtures with different water fractions (f_w) with a view to study the effects of solvent polarity and aggregation on their light emission process. When a small amount of water ($f_w < 70$ vol %) was added to the THF solution of PI, its PL intensity was enhanced slightly. Further increment of the water content, however, weakens the light emission. At $f_w = 90\%$, the PL intensity is merely 50% of that in pure

THF solution (Figures 2A and 2B). Since PI is insoluble in water, its molecules should form aggregates in solvent mixtures with high water contents. Clearly, like most conventional chromophores, the PL of PI is quenched by aggregate formation. Similar phenomenon was also observed upon solution thickening. The emission was significantly quenched when the solution concentration was increased from 10 μM to 1 mM (Figure S5). On the other hand, attachment of TPE units to PI has endowed the resulting adduct (BTPE-PI) with AIE attribute. As shown in Figures 2C and 2D, the peak intensity of BTPE-PI was slightly weakened upon addition of a small amount of water ($f_w \leq 50$ vol %). Afterwards, the emission becomes stronger swiftly. The maximum emission enhancement was observed at $f_w = 90\%$, being ~ 5 -fold higher than that in pure THF solution. Evidently, BTPE-PI is AIE-active. Quantitative measurements on their fluorescence quantum yields (Φ_F) in solution and solid states also draw the same conclusion. The Φ_F of PI in pure THF solution is 58.8%, which drops to 38.2% in the thin film state accompanied with a ~ 30 nm red-shift in emission because of strong intermolecular interactions (Table 1). In contrast, the thin film of BTPE-PI emits at a spectral region similar to that in the solution state but exhibits a much higher Φ_F (93.8% vs 14.5%). Thus, this highly blue emissive luminogen is a promising candidate for the fabrication of efficient EL device.

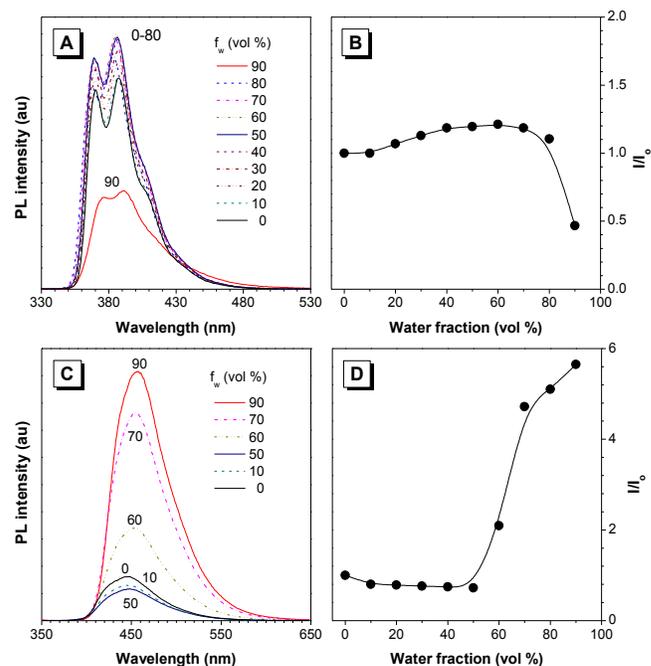


Figure 2. PL spectra of (A) PI and (C) BTPE-PI in THF/water mixtures with different water fractions (f_w). Plots of relative PL intensity (I/I_0) versus the composition of THF/water mixtures of (B) PI and (D) BTPE-PI. I_0 = emission intensity in pure THF solution. Concentration: 10 μM ; excitation wavelength (nm): 312 (PI) and 340 (BTPE-PI).

Electronic properties. To get a further insight into the photophysical properties of PI and BTPE-PI at the molecular level, density functional theory calculations are carried out using a suite of Gaussian 03 program. The nonlocal density functional of B3LYP with 6-31G (d) basis

sets was used for the calculation. The highest occupied molecular orbital (HOMO) of BTPE-PI was dominated by the orbitals from the PI core and TPE (A) unit on the same axis. The electron cloud of the lowest unoccupied molecular orbital (LUMO), however, was located mainly on the PI core and the lateral TPE (B) unit (Figure S6). Such electron distribution imparts BTPE-PI an intrinsic intramolecular charge transfer property. The dipole moment of BTPE-PI is small and equal to 5.29, which is indicative of a weak D-A property and is consistent with the experimental data given in Figure 1. The optimized structures of PI and BTPE-PI are shown in Figure S7 for comparison. PI adopts an almost planar conformation, which favors strong intermolecular interactions that lead to fluorescence quenching. On the contrary, BTPE-PI is non-planar. The torsion angles between the imidazole plane (a) and the adjacent phenyl rings are 18.6° (phenyl ring b) and 88.9° (phenyl ring f). In such case, the PI core is well-conjugated with the TPE (A) unit that lies on the same axis but shows a poor electronic communication with the lateral one (B). The dihedral angles between the PI unit and the TPE moieties are both ~35°, while those between any phenyl ring and the olefin core of the TPE unit fall in the range from 22.7° to 64.9° (Table S1). The non-planar conformation of BTPE-PI is mainly attributed to the propeller-shaped TPE units, which disfavor the close molecular packing in the solid state. Coupled with the restriction of intramolecular motions in the aggregated state, all these made BTPE-PI an efficient solid-state emitter.

Thermal properties. The thermal properties of PI and BTPE-PI are examined by thermogravimetric analysis (TGA) and differential scanning calorimetry (DSC) measurements (Figure 3 and Table 1). BTPE-PI is thermally very stable, exhibiting a 5% weight loss at 460 °C. Such value is much higher than that of PI ($T_d = 262$ °C) (Figure 3A). The glass transition temperature (T_g) of BTPE-PI was detected at 145 °C, while such thermal transition was not detected in PI (Figure 3B). No transition peak associated with crystallization was observed when BTPE-PI was heated up to 200 °C. The non-planar TPE units in BTPE-PI hinder close packing, thus making the luminogen non-crystalline or amorphous in nature. Luminogens with high thermal and morphological stabilities are beneficial for the formation of amorphous thin films by vacuum deposition, which is important for the stability of OLEDs.⁴⁸

Electrochemical properties. The electrochemical properties as well as energy levels of BTPE-PI and PI are investigated by cyclic voltammetry (Figure S8 and Table 1). The HOMO of BTPE-PI is estimated from its oxidation onset potential and is calculated to be -5.50 eV. This suggests a good hole injection property due to the fine contribution from the PI unit ($E_{\text{HOMO}} = -5.48$ eV). Its LUMO can be obtained by addition of the optical band

gap energy (E_g) estimated from the onset absorption wavelength from the HOMO value and is calculated to be -2.45 eV. This value is lower than that of PI (-2.16 eV) but close to that of 1,3,5-tri(phenyl-2-benzimidazolyl)-benzene (TPBi, -2.7 eV), which is widely used as an electron-injection layer in OLEDs.⁴⁹ Clearly, the electron injection ability of BTPE-PI is similar to TPBi but better than PI. This also suggests a good carrier injection property in BTPE-PI due to its suitable HOMO and LUMO energy levels. This property will enable the fabrication of devices with simple configurations, which is an advantage for practical OLED applications.

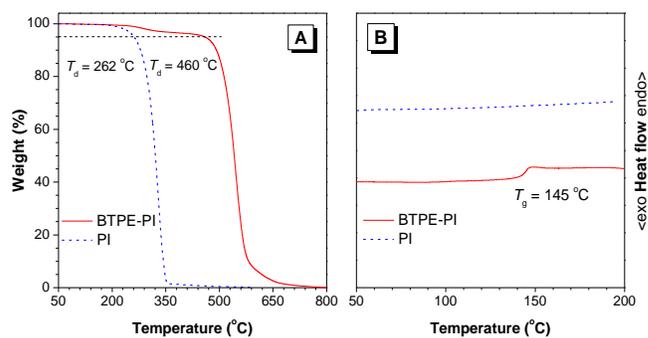


Figure 3. (A) TGA and (B) DSC (second heating cycle) thermograms of PI and BTPE-PI recorded under nitrogen at a heating rate of 20° C/min.

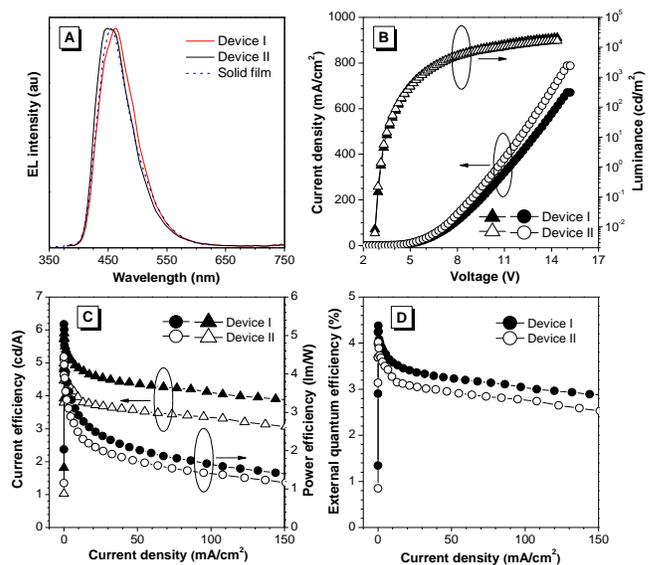


Figure 4. (A) PL and EL spectra of solid thin film of BTPE-PI. (B) Current density–voltage–luminance characteristics of multilayer EL devices of BTPE-PI. Changes in (C) current efficiency and power efficiency and (D) external quantum efficiency with the applied current density in multilayer EL devices of BTPE-PI. Device configuration: ITO/NPB/BTPE-PI/TPBi/LiF/Al.

Table 1. Optical and thermal properties of PI and BTPE-PI^{a)}

Compound	λ_{ab} [nm]	E_{HOMO} [eV]	E_{LUMO} [eV]	E_g [eV]	λ_{em} [nm]			T_d/T_g [°C]
					soln ($\Phi_{F,s}$)	aggr	film ($\Phi_{F,f}$)	
PI	312, 344, 362	-5.48	-2.16	3.32	370, 387 (58.8)	376, 391	397, 415 (38.2)	262/nd
BTPE-PI	340	-5.50	-2.45	3.05	449 (14.5)	456	455 (93.8)	460/145

^{a)} Abbreviation: λ_{ab} = absorption maximum in THF, E_{HOMO} = highest occupied molecular orbitals calculated from the onset oxidation potential, E_g = energy band gap calculated from the onset of the absorption spectrum, E_{LUMO} = lowest unoccupied molecular orbitals estimated by using the equation: $E_{LUMO} = E_{HOMO} + E_g$, λ_{em} = emission maximum in THF solution (soln), THF/water mixture (1:9 by volume) (aggr) and solid thin film spin-coated from THF solution with fluorescence quantum yield (Φ_F) given in the parentheses, T_d = temperature for 5% weight loss, T_g = glass transition temperature, nd = not detected.

Table 2. EL performances of BTPE-PI^{a)}

Device	λ_{EL} [nm]	V_{on} [V]	L_{max} [cd/m ²]	$\eta_{C,max}$ [cd/A]	$\eta_{P,max}$ [lm/W]	$\eta_{ext,max}$ [%]	CIE (x, y)
I	463	3.2	20300	5.9 (4.6)	5.3 (2.5)	4.4 (3.4)	0.15, 0.15
II	450	3.2	16400	4.9 (3.7)	4.4 (3.1)	4.0 (3.0)	0.15, 0.12
III	450, 511, 618, 674	3.2	19200	10.7 (9.8)	8.1 (5.3)	6.4 (5.9)	0.33, 0.33

^{a)} Abbreviation: λ_{EL} = EL maximum, V_{on} = turn-on voltage at 1 cd/m², L_{max} = maximum luminance, $\eta_{P,max}$ = maximum power efficiency, $\eta_{C,max}$ = maximum current efficiency and $\eta_{ext,max}$ = maximum external quantum efficiency. The values in the parentheses are taken at a luminance of 1000 cd/m². CIE (x, y) = Commission Internationale de l'Eclairage chromaticity coordinates collected at 8 V. Device configuration: ITO/NPB(60 nm)/BTPE-PI(20 nm)/TPBi(40 nm)/LiF(1 nm)/Al(100 nm) (Device I), ITO/NPB(40 nm)/BTPE-PI(20 nm)/TPBi(40 nm)/LiF(1 nm)/Al(100 nm) (Device II) and ITO/NPB(40 nm)/Ir(btp)₂(acac)(6 wt %):CBP(10 nm)/Ir(ppy)₃(8 wt %):CBP(4 nm)/CBP(3 nm)/BTPE-PI(5 nm)/TPBi(40 nm)/LiF/Al (Device III).

Electroluminescence. Non-doped deep blue OLED. The efficient solid-state deep blue emission of BTPE-PI as well as its high thermal stability and suitable LUMO and HOMO levels prompts us to investigate its EL properties. Most deep blue EL devices reported previously involve the use of dopants. However, to achieve an optimized device performance, careful control on the dopant concentration is required.^{50,51} On the other hand, performance degradation due to phase separation upon heating is also a problem encountered in these devices.⁵²⁻⁵⁴ Nevertheless, AIE luminogens are free from the ACQ effect and thus require no dopant for EL device fabrication. With such regard, we thus fabricated a non-doped OLED device with a simple configuration of ITO/NPB(60 nm)/BTPE-PI(20 nm)/TPBi(40 nm)/LiF(1 nm)/Al(100 nm) (Device I). In this device, BTPE-PI works as blue emitting layer (EML), *N,N'*-di(1-naphthyl)-*N,N'*-diphenyl-(1,1'-biphenyl)-4,4'-diamine (NPB) functions as a hole-transporting layer (HTL) and TPBi serves as both electron-transporting (ETL) and hole-blocking layer. The EL spectrum of device I was peaked at 463 nm, whose pattern resembles the PL spectrum of the solid thin film. This confirms that the EL is indeed from the emitting layer. The device turns on at a very low bias voltage of 3.2 V and emits brilliant blue light with maximum luminance (L_{max}) of 20300 cd/m², maximum current efficiency ($\eta_{C,max}$) and power efficiency ($\eta_{P,max}$) of 5.9 cd/A, 5.3 lm/W, respectively (Figure 4 and Table 2). The external quantum efficiency ($\eta_{ext,max}$) is

determined to be 4.4%, which is rather high and close to the theoretical limit (5%) for fluorescent materials. Driven at 1000 cd/m², the device also performs well with high η_{ext} of 3.4%. In addition, the roll-off is small, which drops merely 14.4% when the device brightness is increased from 1000 cd/m² to 5000 cd/m². The CIE coordinates are calculated to be (0.149, 0.147), which fall in the deep blue region ($y < 0.15$)^{17, 55-57} (Figures 5A and 5D).

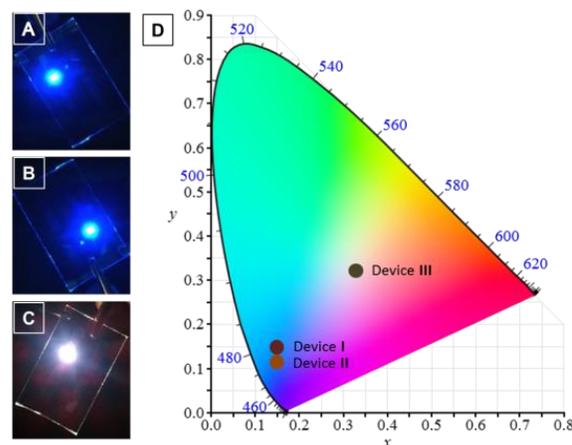


Figure 5. Photos of (A) Device I, (B) Device II and (C) Device III. (D) Commission Internationale de l'Eclairage (CIE) chromaticity coordinates of the three devices.

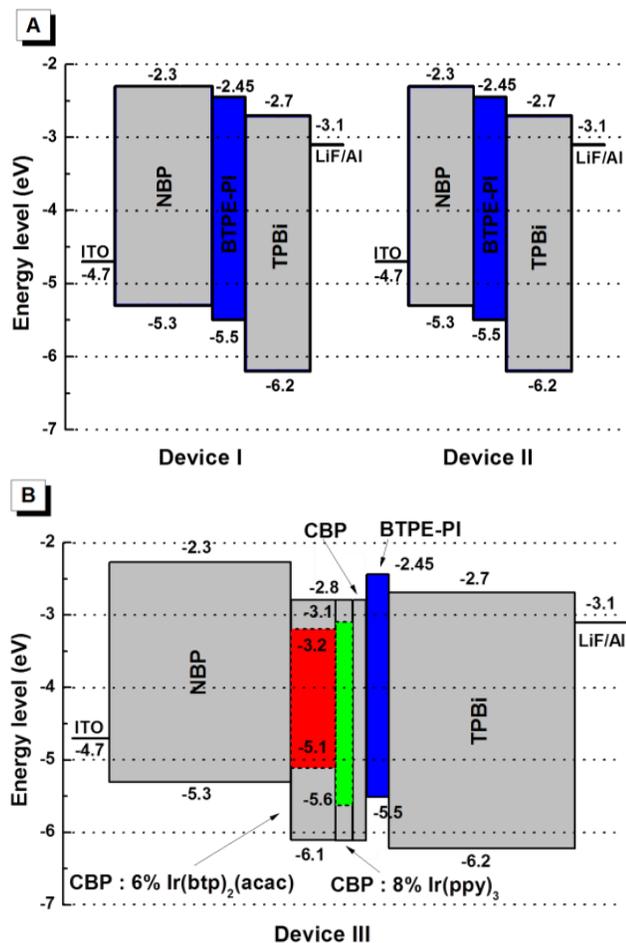


Figure 6. Energy level diagrams and device configurations of BTPE-PI based (A) deep blue (B) white OLEDs.

The excellent EL performance of BTPE-PI is somewhat related to its good charge injection property as well as its weak D-A system. To get a further insight, the energy level diagrams and device configurations of the EL devices are illustrated in Figure 6A. Thanks to the suitable energy levels of BTPE-PI, the electron injection barrier at the TPBi/BTPE-PI junction is only 0.25 eV, while the hole injection barrier is also small (0.20 eV) at the NPB/BTPE-PI junction. These small injection barriers enable efficient injection of both holes and electrons into the emitting layer at low bias. Moreover, the TPBi layer effectively blocks the leakage of holes from the emitting layer. These well matched energy levels are beneficial for charge balance and efficient exciton recombination at the emitting layer, thus improving the EL efficiency. By fine adjustment of the thickness of the hole-transporting layer, we can shift the EL spectrum to the shorter wavelength region due to the interference effect.^{58–60} Device II with a configuration of ITO/NPB(40 nm)/BTPE-PI(20 nm)/TPBi(40 nm)/LiF(1 nm)/Al(100 nm) was also fabricated, which exhibits a bluer light than Device I with CIE coordinates of (0.15, 0.12) (Figures 5B, 5D and 6A). The device shows high performance with L_{\max} , $\eta_{C,\max}$, $\eta_{P,\max}$ and $\eta_{\text{ext},\max}$ of 16400 cd/m², 4.9 cd/A, 4.4 lm/W and 4.0%, respectively (Table 2 and Figure 4). Table S2 compares the EL performances of BTPE-PI with those of

deep blue emitters reported previously.^{17,23,35,36,46,61–72} The EL devices fabricated from BTPE-PI with standard three-component active layers (HTL/EML/ETL) exhibit really high performance and energy conversion as reflected by their high external quantum and power efficiencies. The maximum luminance (> 15000 cd/m²) attained by them are one order of magnitude higher than the previous results. Some EL devices with excellent $\eta_{\text{ext},\max}$ (>5%) have been reported^{62,63,71,72} but their device structures are complicated and involve four-component active layers. When driven at 1000 cd/m² for practical applications, most of these devices show an obvious efficiency roll-off. The BTPE-PI based EL devices, however, still retain very high performances with efficiencies of up to 3.1% and 3.0 lm/W. Such results are impressive and are the best among the deep blue non-doped OLEDs.

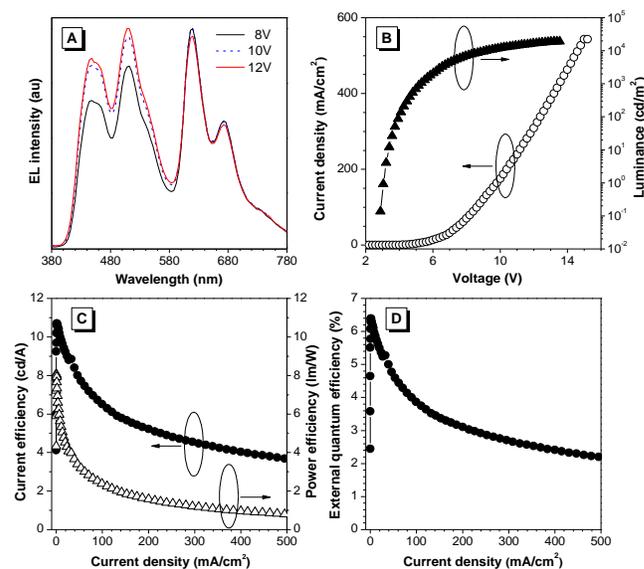


Figure 7. (A) EL spectra of a multilayer WOLED at different applied voltages. (B) Current efficiency–voltage–luminance characteristics of a multilayer WOLED. Changes in (C) current efficiency and power efficiency and (D) external quantum efficiency with the applied current density in a multilayer WOLED. Device configuration: ITO/NPB/Ir(btp)₂(acac):CBP/Ir(ppy)₃:CBP/CBP/BTPE-PI/TPBi/LiF/Al.

White light OLED. One of the most important applications for deep blue emitters is the fabrication of white OLED (WOLED).^{73,74} Many WOLEDs containing primary RGB emitters were fabricated, showing unique advantages of high stability of blue fluorophores and high efficiency of phosphors with longer emission wavelength. Their multilayer structure allows flexible manipulation of each EML as well as the precise control on the exciton distribution in different EMLs. We fabricated a white EL device with a configuration of ITO/NPB(40 nm)/Ir(btp)₂(acac)(6 wt %):CBP(10 nm)/Ir(ppy)₃(8 wt %):CBP(4 nm)/mCP(3 nm)/BTPE-PI(5 nm)/TPBi(40 nm)/LiF/Al, where BTPE-PI functions as blue emitting layer, bis[2-(2'-benzothienyl)pyridinato-N,C₃'](acetylacetonato)iridium(III) Ir(btp)₂(acac) and

1 tris[2-phenylpyridinato-C₂,N]iridium(III) Ir(ppy)₃ work as
2 red and green emitters, respectively, and 4,4'-bis(9H-
3 carbazol-9-yl)biphenyl CBP was used as host to reduce
4 the structural heterogeneity and facilitate charge
5 transport between the adjacent red and green emitting
6 layers. A 3 nm thickness of 1,3-bis(carbazol-9-yl)benzene
7 (mCP) buffer layer is inserted between the fluorescent
8 emitter (BTPE-PI) and the phosphorescent emitter
9 Ir(ppy)₃ to confine the singlet excitons and their
10 recombination in the BTPE-PI region. This configuration
11 is expected to harvest both single and triplet excitons.⁷⁵
12 The device emits intense white light ($L_{\text{max}} = 17900 \text{ cd/m}^2$)
13 through the simultaneous emission from the primary RGB
14 emitters and shows excellent performance with $\eta_{\text{C,max}}$,
15 $\eta_{\text{P,max}}$ and $\eta_{\text{ext,max}}$ of, 12.3 cd/A, 9.5 lm/W and 7.3%,
16 respectively (Figure S9). The CIE coordinates are
17 calculated to be (0.31, 0.32) at 8 V, which are slightly
18 deviated from the values of the theoretical white point
19 (0.33, 0.33)⁷⁴ due to insufficient green and red
20 components in the emission spectrum. To further
21 improve the white color purity, we optimized the device
22 configuration and chose CBP ($E_{\text{LUMO}} = -2.8 \text{ eV}$) as the
23 buffer layer, whose LUMO energy level is 0.4 eV lower
24 than that of mCP ($E_{\text{LUMO}} = -2.4 \text{ eV}$). This buffer layer
25 should facilitate more electrons and thus more
26 recombination in the green and red emission layers due
27 to no injection barrier from the buffer layer to the green
28 and red emitting layers (Figure 6B). Thus, another
29 WOLED device with a configuration of ITO/NPB(40
30 nm)/Ir(btp)₂(acac)(6 wt %):CBP(10 nm)/Ir(ppy)₃(8
31 wt %):CBP(4 nm)/CBP(3 nm)/BTPE-PI(5 nm)/TPBi(40
32 nm)/LiF/Al (Device III) was constructed. The device
33 starts to emit at 3.2 V, showing similar intense white
34 emission with L_{max} , $\eta_{\text{C,max}}$, $\eta_{\text{P,max}}$, and $\eta_{\text{ext,max}}$ of 19200 cd/m²,
35 10.7 cd/A, 8.1 lm/W and 6.4%, respectively (Figure 7 and
36 Table 2). Even at 1000 cd/m², high η_{C} , η_{P} and η_{ext} values of
37 9.8 cd/A, 5.3 lm/W and 5.9%, respectively, are still
38 achieved. More importantly, the CIE coordinates of this
39 device exactly match those of the theoretical white point
40 (0.33, 0.33) at 8 V (Figures 5C and 5D). To the best of our
41 knowledge, it is the first WOLED device with intense
42 emission at theoretical white point fabricated from deep
43 blue AIE fluorophore. In addition, the device shows
44 impressive color stability with CIE coordinates of (0.31,
45 0.32) even at high voltages of 10 V and 12 V (Figure 7A).
46 This suggests that the hole-electron recombination zone
47 is well confined in the emitting layers in a wide range of
48 working voltage. It is noteworthy that all the EL curves in
49 device I, II and III are smooth, thanks to the high stability
50 of the devices at different applied voltages. Thus, the
51 molecular design strategy present here is simple but
52 successful. It makes full use of the intrinsic advantages of
53 both TPE and conventional chromophores to create deep
54 blue AIE luminogens with efficient solid-state emission
55 and prominent thermal stabilities for high performance
56 OLED applications.

57 CONCLUSION

58 In summary, a new efficient deep blue AIE luminogen
59 (BTPE-PI) was designed and synthesized through a simple
60

integration of a weakly conjugated AIE luminogen into a
conventional chromophore. The resulting luminogen
BTPE-PI shows a weak D-A interaction but a typical AIE
characteristic and improved solid-state fluorescence
quantum yield than its parent molecules. It also enjoys
high thermal stability and good carrier injection
capability. Non-doped multilayer EL device fabricated
from BTPE-PI shows a small roll-off and emits a deep blue
emission with high luminance, current efficiency, power
efficiency and external quantum efficiency of 20300 cd/m²,
5.9 cd/A, 5.3 lm/W and 4.4%, respectively. These results
are the best for deep blue AIE emitters reported so far. In
addition, for the first time, an efficient AIE-based
WOLED using BTPE-PI as deep blue emitter was
fabricated and realized pure white light with CIE
coordinates of (0.33, 0.33). It is anticipated that by
following the molecular design strategy present here,
many new efficient deep blue AIE luminophores will be
rationally designed and synthesized in the future.

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ASSOCIATED CONTENT

Supporting Information Available

NMR, MS, UV, PL, CV spectra, and molecular orbital
quantum calculation of PI or BTPE-PI, EL performance of
WOLED device. This material is available free of charge via
the Internet at <http://pubs.acs.org>.

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Construction of Efficient Deep Blue AIE Luminogen from Triphenylethene for Non-Doped OLED Applications

