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Effect of incorporated nitrogen on the kinetics of thin rapid thermal N_2O oxides

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We have grown ~10 nm O₂ and N₂O-oxides on Si(100) by RTO (rapid thermal oxidation) over the temperature range 800–1200 °C. Although the growth rates of both oxides exhibit Arrhenius behavior over the entire temperature range, the N₂O-oxides exhibit a large change in the Arrhenius preexponential factor for oxidation temperatures greater than 1000 °C. Above this temperature, N₂O-oxides grow a factor of 5 slower than O₂ oxides. Below this temperature, N₂O-oxide growth rates approach those of O₂-oxides. This growth rate inflection can be explained in terms of N incorporation, which increases with increasing oxidation temperature. The equivalent of one monolayer of N coverage is achieved at about 1000 °C, coincident with the inflection. The incorporated N retards the linear growth of the thin N₂O-oxides either by occupying oxidation reaction sites or inhibiting transport of oxidant species to the vicinity of the interface. © 1995 American Institute of Physics.

The incorporation of N (nitrogen) into SiO₂, i.e., oxynitridation, significantly retards boron penetration through the resultant Si-O-N dielectric.^{1,2} Therefore, oxynitrides are potentially important gate dielectrics for ultralarge scale integration, where dielectric thicknesses will be ≤ 7 nm. In the absence of N, boron, the P^+ polycrystalline silicon gate electrode dopant can diffuse through the thin SiO₂ layer during postgate dielectric growth processing, causing unacceptable shifts in device threshold voltage.^{3,4} The simplest and most manufacturable of the many oxynitridation chemistries is the direct oxynitridation of Si in N2O, which has been extensively studied with respect to processing,^{5,6} electrical properties,^{3,7,8} and reliability.³ In N₂O-oxides, the incorporated N is closely confined to the Si/SiO₂ interface.^{3,6,9} Further, the amount of N incorporated in RTO (rapid thermal oxidation) N2O-oxides is small, but has been found to increase with increasing oxidation temperature^{3,10,11} and oxide thickness.^{5,9,12} For example, the equivalent of about one monolayer of N on Si(100), 7×10^{14} N/cm², is incorporated in a 10 nm N₂O-oxide grown at 1000 °C.¹⁰ Other N incorporation data from the literature has been summarized in Ref. 10. The small amount of incorporated N has a large effect in retarding the kinetics of N₂O-oxide growth.¹³⁻¹⁵ In this letter we report the relationship between oxidation kinetics of RTO O₂ and N₂O-oxides and incorporated N content for oxides of current technological importance, i.e., ~10 nm thick. We show that retardation of N₂O-oxide kinetics is due to the effect of incorporated N content on the Si/SiO₂ interfacial oxidation reaction constant.

RTO O₂ and N₂O-oxides ~10 nm thick (actual range 8.5–11.2 nm), were grown at temperatures ranging from 800 to 1200 °C in pure ambients and at atmospheric pressure. All oxides were grown on *p*-type (10–20 Ω cm, boron), 125 mm, Si(100) substrates. The N₂O-oxides are the same sample set whose incorporated N contents were reported in Fig. 1 of

Ref. 10. Oxide thickness was determined by ellipsometry, with the index of refraction of the N₂O-oxides assumed to be that of the O₂-oxides, 1.459. The validity of this assumption has been confirmed by independent ion scattering measurements, and is consistent with the small amount of N in the oxides.

Figure 1(a) is an Arrhenius plot of average oxidation rate R_{O_2} or R_{N_2O} , where R_{O_2} and R_{N_2O} are defined as the oxide thickness divided by the time at oxidation temperature to reach ~10 nm thickness, for O₂ and N₂O-oxides, respectively. The O₂-oxide data obey an Arrhenius relationship that yields an activation energy of 2.1 ± 0.1 eV. In contrast, the N₂O-oxide data appear to only obey an Arrhenius relationship between about 1000 and 1200 °C, with activation energy equal to 1.9 ± 0.1 eV. In this temperature range, the N₂O-oxide growth rate is about five times slower than the O₂-oxide growth rate. Below 1000 °C, the N₂O-oxide growth rate, nearly equaling it below 850 °C.

The N₂O-oxide growth rate inflection can be seen more clearly in Fig. 1(b), where the ratio of the N₂O to O₂-oxide average growth rates, R_{N_2O}/R_{O_2} , is plotted as a function of oxidation temperature. The presence of the inflection in $R_{N_2O}R_{O_2}$, which can be clearly observed, is indicative of an abrupt transition in N₂O-oxide growth rate with respect to O₂-oxide growth rate. It has been shown,¹⁰ for the present N₂O-oxide sample set, that incorporated N content increases approximately linearly in the oxidation temperature range 800–1200 °C, and that at 1000 °C, its value is 7×10^{14} N/cm², the equivalent of about one monolayer. Therefore, we can correlate the abrupt transition in R_{N_2O}/R_{O_2} , Fig. 1(b), with the increase in N content previously observed in Ref. 10. In Fig. 2 we have plotted R_{N_2O}/R_{O_2} as a function of N content, and it can be seen that the incorporation of a mono

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FIG. 1. (a) Arrhenius plot of average growth rate for ~ 10 nm thick O_2 and N_2O -oxides, over the temperature range 800–1200 °C, and (b) the ratio of the N_2O to O_2 -oxide average growth rates as a function of oxidation temperature.

layer equivalent of N saturates the retardation of N_2O -oxidation kinetics.

Figure 1(a) also suggests that the growth of ~ 10 nm O₂ oxides occurs in the linear regime (as defined by Deal–Grove),¹⁶ since the derived activation energy, 2.1 ± 0.1 eV, is in excellent agreement with the 2.0 eV Deal-Grove value for their linear growth rate parameter. Other researchers have observed activation energies of 2.26 eV¹⁷ and 2.0 eV¹⁸ for the linear rate parameter for thin RTO O₂-oxides. Further, since the N₂O-oxides of Fig. 1(a) also show an activation energy of 1.9 ± 0.1 eV for T > 1000 °C, within error equal to the Deal-Grove value, we assume that in this temperature range the N₂O-oxides grow in the linear regime. This must also be true for the N₂O-oxides grown below 1000 °C, since in that temperature range the N₂O-oxide growth rates approach the O₂-oxide growth rates. The apparent deviation from Arrhenius behavior exhibited by the N₂O-oxides is therefore probably due to a change in the preexponential factor. The activation energy, reflective of the oxide growth mechanism, appears to be unchanged for oxidation in either O₂ or N₂O ambient.

The N incorporation/growth rate ratio data of Fig. 2 can



FIG. 2. The ratio of the N_2O to O_2 -oxide average growth rates as a function of nitrogen content of the N_2O oxides. The nitrogen data come from Fig. 1 of Ref. 10.

be rationalized with linear, i.e., reaction controlled, oxide growth. It is known that the N is concentrated very close to the Si/SiO₂ interface.^{3,6,9} We propose that as the N concentration increases with increasing temperature up to one monolayer equivalent, it continuously retards further oxidation by one of two mechanisms. If one assumes a pure interface oxidation reaction model,¹⁶ then the N might be occupying oxidation reaction sites at the interface, as has been previously suggested.¹⁹ This model suggests that N exists as a monolayer specifically bound to Si/SiO₂ interface sites. Alternatively, in a reactive layer model,²⁰ the N may be inhibiting transport of the oxidant species through the reactive layer, thereby retarding the interfacial reaction. In this way, the N acts to mask the approximately one monolayer of Si available to react at the interface. Either model is consistent with our finding of linear growth.

In summary, we have grown $\sim 10 \text{ nm O}_2$ an N₂O-oxides by RTO over the temperature range 800–1200 °C and have found that growth is dominated by linear, i.e., Si/SiO₂ interfacial reaction kinetics. The N₂O-oxide growth rate decreases as N incorporation increases, due either to oxidation site occupation by N, or inhibition of oxidant transport through the reactive layer, again by N. At an N content equivalent to one monolayer on Si(100), the effect of the incorporated N on the N₂O-oxide kinetics saturates.

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