CeO₂ Nanorods and Gold Nanocrystals Supported on CeO₂ Nanorods as Catalyst

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The formation mechanism of uniform CeO₂ structure at the nanometer scale via a wet-chemical reaction is of great interest in fundamental study as well as a variety of applications. In this work, large-scale wellcrystallized CeO₂ nanorods with uniform diameters in the range of 20–30 nm and lengths up to tens of micrometers are first synthesized through a hydrothermal synthetic route in 5 M KOH solution at 180 °C for 45 h without any templates and surfactants. The nanorod formation involves dehydration of CeO₂ nanoparticles and orientation growth along the $\langle 110 \rangle$ direction in KOH solution. Subsequently, gold nanoparticles with crystallite sizes between 10 and 20 nm are loaded on the surface of CeO₂ nanorods using HAuCl₄ solution as the gold source and NaBH₄ solution as a reducing agent. The synthesized Au/CeO₂ nanorods demonstrate a higher catalytic activity in CO oxidation than the pure CeO₂ nanorods.

I. Introduction

As an important class of nanomaterials, the nanorod has attracted enormous technological and scientific interest.¹ In synthesizing metal oxide nanorods, such as ZnO,^{2,3}Al₂O₃,^{4,5} MgO,⁶ CuO,^{7,8} SnO₂,⁹ MnO₂,¹⁰ Co₃O₄,¹¹ V₂O₅,¹² and Eu₂O₃,¹³ many approaches have been applied in the past several years, including thermal evaporation, microemulsion method, hydrothermal synthesis, template methods, and electrochemical deposition technique. Hydrothermal reaction under moderate conditions is an effective approach in synthesizing nanosize crystals of inorganic oxides and hydroxides. In many cases, alkaline solution is used in the hydrothermal synthesis in which the shape and size of the crystals are well-controlled. For example, anatase and titanate nanotubes and nanorods were obtained hydrothermally in NaOH solution,14-17 and lanthanide hydroxide nanowires were synthesized successfully via hydrothermal reaction in KOH solution.¹⁸ The hydroxide nanorods or titanate nanorods can be converted into the corresponding oxide nanorods through a subsequent dehydration process at high temperature.

Ultrafine CeO₂ particles of regular shape and uniform size are desired in a variety of applications, i.e., ceramic ingredient, fluorescent powder, polishing agent, catalyst carrier, catalyst for the exhaust gas treatment from automobiles, and solid electrolyte for solid oxide fuel cells.^{19–22} Recently, it has been reported that the polycrystalline CeO₂ nanowires can be fabricated in anodic alumina membranes (AAM) as templates,^{23,24} and crystalline CeO₂ nanorods can be formed through a surfactant assemblage because the surfactant in water can form a chain structure.^{25–27} The formation mechanism of the uniform CeO₂ nanorod, via a wet chemical reaction, is of great interest in fundamental study. Also, it has been reported that CeO_2 and rare earth oxide nanocrystals can significantly increase the catalytic activity of dispersed Au nanoparticles for CO oxidation.^{28–32} The decoration of Au nanoparticles on the support of stable CeO₂ nanorods could efficiently prevent Au nanoparticles from the agglomeration. In the present study, we report a simple approach for the synthesis of large-scale CeO₂ nanorods with single crystalline structure in alkaline solution, without any templates and surfactants. Subsequently, gold nanoparticles are loaded on the so-fabricated CeO₂ nanorods by using HAuCl₄ solution as a gold source and NaBH₄ solution as a reducing agent. The catalytic activity of Au/CeO₂ nanorods in CO oxidation is measured.

II. Experimental Section

1. Sample Preparation and Characterization. In a typical synthesis, metallic cerium was first dissolved in concentrated hydrochloride acid to obtain cerium chloride solution, and 5 M KOH solution was added gradually into the cerium chloride solution under stirring until the cerium ions precipitate completely. The precipitates ($CeO_2 \cdot nH_2O$) were aged for 30 min in air and then washed repeatedly with deionized water to remove Cl⁻ anions in the precipitates. The wet precipitates were then mixed with 5 M KOH solution (40 mL) using an ultrasonic bath. The obtained mixture was transferred in a stainless autoclave with a poly(tetrafluoroethylene) (PTFE) container and maintained at 180 °C for different reaction durations. The solid was recovered from the autoclaved mixture, rinsed with water until a value of about pH 7 was reached. Gold nanoparticles were loaded on the CeO₂ nanorods by dispersing 0.1 g of the nanorods in 40 mL of 0.1 M HAuCl₄ solution. The impregnated nanorods were separated by centrifugation, and NaBH₄ solution (10 mL, 0.1 M) was added to the solid. The solid was then recovered, rinsed with deionized water to remove Cl⁻ anions, and dried at 100 °C. The so-obtained product (CeO₂ nanorods) was further dried at 60 °C in air for 1 day. The microstructure of the prepared CeO₂ powders was characterized by transmission

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Figure 1. SEM images of CeO₂ nanorods synthesized at different reaction conditions: (a) before the hydrothermal treatment, (b) 5 M KOH at 180 °C for 4 h, (c) 5 M KOH at 180 °C for 24 h, (d) 5 M KOH at 180 °C for 45 h, (e) 5 M KOH at 150 °C for 45 h, (f) 10 M KOH at 150 °C for 45 h, and (g) 10 M KOH at 180 °C for 45 h.

electron microscopy (TEM; FEI Tecnai 20) and X-ray diffraction (XRD, Rigaku D/max-2500) with a Cu K α radiation. Since the as-synthesized nanoparticles were suspended in solution, a droplet of it was also deposited on a Cu grid covered with a carbon film for TEM analysis. X-ray photoelectron spectra (XPS) analyses were investigated using PHI-5300 ESCA in order to assess the chemical state and element distribution on the catalyst surface.

2. Catalytic Activity. The catalytic activity measurements of the catalysts for CO oxidation were carried out in a fixed bed flow microreactor (7 mm i.d.) under atmospheric pressure using 50 mg of catalyst powder doped with 2 g of fine quartz

sand. The reaction gas mixture consisting of 1% CO balanced with air is passed through the catalyst bed at a total flow rate of 33.6 mL/min. The reactant and product composition were analyzed on-line by a GC-508A gas chromatograph equipped with a thermal conductivity detector (TCD).

III. Results and Discussion

Scanning electron microscopy (SEM) images of CeO_2 nanorods synthesized at different hydrothermal reaction conditions are shown in Figure 1. The big agglomerated particles are obtained before the hydrothermal treatment. It can be seen that



Figure 2. XRD patterns of the CeO₂ nanorods synthesized at 180 °C for 45 h and CeO₂ $\cdot nH_2O$ nanoparticles.

CeO₂ nanorods can be formed at the hydrothermal temperature of 180 °C for 4 h though some nanoparticles still coexisted. Longer reaction duration of the hydrothermal treatment improves of the purity of CeO2 nanorods and enhances the aspect ratio of CeO₂ nanorods. The alkaline concentration and hydrothermal temperature are observed to have a strong effect on the structure and morphological features of the resulting CeO₂ nanorods. The morphological transformation from particles to nanorods is not complete when the hydrothermal temperature is below 150 °C at 5 or 10 M KOH solution, as shown in Figure 1e,f. It reveals that the hydrothermal temperature is a key parameter that influences the nucleation and crystal growth of CeO₂ nanorods, which is also found for the preparation of titanate nanorods under the hydrothermal condition in NaOH solution.¹⁷ However, the shape of the hydrothermal products is irregular when the alkaline concentration is higher than 5 M at 180 °C. In particular, the large quantity of thicker rods attached with thin nanorods is formed in 10 M KOH solution at 180 °C, indicating that some nanorods further grow and become thicker and longer at the expense of thinner nanorods being dissolved in the higher alkaline concentration.

The XRD patterns of CeO2 nanorods synthesized at 180 °C for 45 h and CeO₂•nH₂O nanoparticles are illustrated in Figure 2. All the diffraction peaks for the products before and after hydrothermal treatment in alkaline solution can be indexed to (111), (200), (220), (311), (222), (400), (331), and (420) phases, corresponding to a face-centered cubic (fcc) fluorite structure for CeO₂ (JCPDS Card No. 43-1002). There are not any other impurities of the hexagonal structure of Ce(OH)₃ or Ce(OH)-CO₃ detected. It is well-known that the Ce(III) oxidation state is unstable as compared with the Ce(IV) oxidation state in alkaline solution and in the presence of air. From the Ce3d core level peak in the XPS spectra shown in Figure 3, it is clear that the cerium exists as the Ce(IV) oxidation state (882.8 eV), without any impurity of the Ce(III) oxidation state. Also, the XPS peak centered at 529.6 eV corresponds to the O2contribution.³³ Thus, the XPS-detected binding energies of Ce3d and O1s are in agreement with the standard CeO2. Thermal gravimetrical analysis (TGA) of CeO₂•nH₂O nanoparticles indicates a mass loss of about 10.7 wt % below 600 °C, due to the loss of water from the sample. Both nanoparticles and nanorods exhibit a light yellow color, similar to the standard CeO₂. Therefore, the nanoparticles are deduced to be



Figure 3. Ce3d (a) and O1s (b) core level spectra observed for CeO_2 nanorods synthesized at 180 °C for 45 h.

 CeO_2 •1.1H₂O, and the synthesis process of CeO_2 nanorods can be described as follows:

$$4\operatorname{Ce}^{3+} + 12\operatorname{OH}^{-} + \operatorname{O}_{2} + (4n - 6)\operatorname{H}_{2}\operatorname{O} \rightarrow 4(\operatorname{CeO}_{2} \cdot n\operatorname{H}_{2}\operatorname{O})$$
$$\operatorname{CeO}_{2} \cdot n\operatorname{H}_{2}\operatorname{O} \rightarrow \operatorname{CeO}_{2} + n\operatorname{H}_{2}\operatorname{O}$$

The XRD peaks for CeO2 • 1.1H2O nanoparticles are obviously broader, compared with those of the nanorods, indicating that the crystallites of the nanoparticles are smaller than those in the CeO₂ nanorods. The hydrothermal treatment in alkaline solution at 180 °C results in a remarkable improvement in crystallinity. As can be seen clearly from TEM images of the nanorods synthesized at 180 °C for 45 h and nanoparticles (Figure 4), we obtained the straight CeO_2 nanorods of a high purity with uniform diameters in the range of 20-30 nm and lengths up to tens of micrometers (Figure 4a). It is also confirmed by energy-dispersive X-ray spectra (EDS, Figure 4b) that there are not any impurities, indicating that sodium ions and chlorides are completely removed during washing. Cu and C signals are coming from the Cu grid and carbon film supporting the specimen in TEM observation. The selected area electron diffraction (SAED) pattern (insert in Figure 4c) indicates the single crystalline nature of the CeO₂ nanorods with a face-centered cubic structure, and this is consistent with the XRD results. The calculated interference fringe spacing in the HRTEM image is about 0.31 nm, which is consistent with the



Figure 4. TEM images (a) and EDS (b) of the CeO₂ nanorods synthesized at 180 °C for 45 h. The HRTEM images of the CeO₂ nanorods and CeO₂ $\cdot n$ H₂O nanoparticles are shown in c and d. The inserts in c and d are SAED of individual CeO₂ nanorod and CeO₂ $\cdot n$ H₂O nanoparticles, respectively.

interplanar distance of the (111) plane in the cubic fluorite structure. This suggests that the preferred orientation growth of the individual CeO₂ nanorod is along the $\langle 110 \rangle$ direction,³⁴ similar to that of the CeO₂ nanorods obtained by PEG surfactant-assisted preparation.²⁶ The HRTEM image and SAED pattern also confirm that the precursor nanoparticles have a polycrys-talline structure with smaller crystallite sizes of 6–9 nm (Figure 4d). Therefore, the hydrothermal treatment in alkaline solution is essential to the formation of large-scale CeO₂ nanorods via the orientation growth.

The nanorods of titanate,^{16,17} lanthanide hydroxide,¹⁸ and CuO⁸ have been synthesized hydrothermally in a concentrated alkaline solution without any templates and surfactants. It appears that there are common features and a probably common mechanism for the formation of these metal oxides and hydrous oxides. We believe that the Ostwald ripening process, in which larger crystallites grow at the expense of smaller crystallites being dissolved,³⁵ is involved in the structure transformation from the polycrystalline CeO2 ·1.1H2O nanoparticles to the single crystalline CeO₂ nanorods during the hydrothermal treatment at 180 °C for 45 h. The above formation mechanism is different from the polycrystalline or nanocrystalline CeO₂ nanowires prepared by the AAM method.^{24,25} Also the OH⁻ concentration in the reaction system could be the key factor controlling the growth rate of different crystal faces and leading to the formation of a rodlike morphology. In the hydrothermal process, the nanosize CeO₂ crystallinates grow along the $\langle 110 \rangle$ direction in the concentrated KOH solution by consuming the CeO₂•1.1H₂O precipitates, yielding the rodlike structure.

As shown in Figure 5a, the gold nanoparticles are dispersed effectively on the surface of the CeO₂ nanorods synthesized at 180 °C for 45 h and are not agglomerated. The gold nanoparticles on the CeO₂ nanorods have crystallite sizes between 10 and 20 nm. The interplanar spacing of the gold nanoparticle in HRTEM images (Figure 5b) is ca. 0.24 nm, corresponding to the interplanar distance of the (111) plane of fcc Au nanocrystal as found in Au/TiO₂ nanotubes.^{36,37} The energy-dispersive X-ray spectrum (EDS, as inserted in Figure 5a) confirms the presence of Au element. The observation of the diffractions peaks of metallic gold (JCPDS Card No. 04-0784) in the XRD pattern (Figure 6) confirms that gold exists as metallic particles. It is noted that the surface of the CeO₂ nanorods, after the decoration of gold nanoparticles, is roughened as compared to that of the pure CeO₂ nanorods, indicating a strong interaction at the interface of Au nanoparticles and CeO₂ nanorods during the reduction in HAuCl₄ and NaBH₄ solution. Moreover, the surface of a gold nanoparticle is covered by a thin discrete layer of the cerium oxides (1-2 nm) attributed to migration of the oxide onto the gold surface.³⁸ The interference fringe of the (111) plane of inner CeO2 nanorods is clearly observed in the HRTEM image (Figure 5b), demonstrating a good crystallinity and a high stability of CeO2 nanorods after the decoration of Au nanoparticles. Similarly, gold nanoparticles supported on CeO2 nanoparticles are also found with the preferential (111) plane of fcc Au nanocrystal (interplanar spacing is ca. 0.24 nm as shown in Figure 5d). However, the dispersed state of gold nanoparticles on CeO₂ nanoparticles is relatively poor as compared with that on CeO₂ nanorods.



Figure 5. TEM and HRTEM images of Au/CeO_2 nanorods (a and b) and Au/CeO_2 nanoparticles (c and d). The EDS spectrum of Au/CeO_2 nanorods is inserted in a.



Figure 6. XRD pattern of Au/CeO₂ nanorods.

Catalytic activities of CeO₂ nanorods and gold nanoparticles on the CeO₂ nanorods as a function of reaction temperature are shown in Figure 7. It is noted that the CO conversion increases with increasing reaction temperature for all samples. An 81% CO conversion is achieved at 220 °C for Au/CeO₂ nanorods as catalyst, while only 20–22% CO conversion is obtained at the same temperature for pure CeO₂ nanorods and nanoparticles as catalyst. The catalytic activity of Au/CeO₂ nanorods and Au/ CeO₂ nanoparticles is much higher than that of pure CeO₂ nanorods and nanoparticles. There could be a synergistic interaction between gold and CeO₂ nanorods, which is responsible for the high activity. It should be noted that the catalytic



Figure 7. Catalytic activities of CeO_2 nanorods and Au/CeO_2 nanorods (solid) as compared with CeO_2 nanoparticles and Au/CeO_2 nanoparticles (open) at different reaction temperatures.

activity of Au/CeO₂ nanoparticles is measured to be superior in all samples. Recently, Venezia found that the reactive oxygen on the surface of nanocrystalline CeO₂ and the strong interaction between cationic gold and ceria may determine the high activity for CO conversion.^{32,39} The smaller CeO₂ nanoparticles may supply more active sites and contribute to the strong interaction between gold and nanocrystalline CeO₂. Therefore, the catalytic activity of Au/CeO₂ nanoparticles is better as compared to that of Au/CeO₂ nanorods. However, the thermal stability of the Au/ CeO_2 catalyst is also an important factor for application. The CeO_2 nanorods, with the higher crystallinity and larger particle size, have high thermal stability,¹ which enhances the durability of the catalyst.

IV. Conclusion

In summary, uniform single-crystal CeO₂ nanorods have been synthesized through a hydrothermal synthesis in KOH solution at 180 °C for 45 h, without any templates and surfactants. The nanorod formation involves the dehydration of CeO₂•1.1H₂O nanoparticles, while the orientation grows along the $\langle 110 \rangle$ direction in KOH solution. The hydrothermal synthesis in alkaline solution is an efficient method for producing largescale metallic oxide nanorods and can be engineered to produce functional nanostructures under moderate conditions. Subsequently, Au nanoparticles supported on the surface of CeO₂ nanorods were prepared by reducing HAuCl₄ impregnated on the nanorods with NaBH₄ solution as CO oxidation catalyst, which exhibits a much higher catalytic activity than the pure CeO₂ nanorods.

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