Effect of Rhodamine-B on the Electrodeposition of Lead on Copper

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ABSTRACT

Rhodamine-B chloride (10 μ M) has been used as a model plating additive in a study of the electrodeposition of Pb from 1M NaClO₄, 0.5 and 5 mM Pb⁺⁺ (pH 3) on Cu. Ellipsometer measurements during cyclic voltammetry have shown that the addition of dye results in a more compact bulk deposit than obtained in its absence. It also prevents complete monolayer coverage during formation of the Pb underpotential deposit and shifts the bulk deposition peak to more cathodic potentials during the first potential cycle. Dye effects on potential and micromorphology disappear during subsequent cycling, but reappear after relaxation periods at open circuit. Depletion and readsorption of dye on the surface have been confirmed by spectroscopic ellipsometry. Different optical film models have been investigated for the interpretation of spectroscopic ellipsometer measurements by use of multidimensional analysis.

Surface active agents (additives, inhibitors) have long been used on an empirical basis to control surface finish in electrolytic metal deposition. Indications are that uncontrolled impurities often cause similar effects. Few fundamental studies have been undertaken to define the role of these agents during the initial stages of electrocrystallization, despite the great technical importance of their use (1). The objectives of this work were to correlate variations in adsorbate coverage on the electrode with differences in overpotential and micromorphology of the deposit. Deposit thicknesses investigated ranged from a monolayer (the underpotential deposit, UPD) to thicknesses of about 100 nm. The principal experimental techniques used were cyclic voltammetry and ellipsometry (both spectroscopic and fixed wavelength). A selfcompensating instrument with rapid spectral scanning capabilities for the visible range and automated data collection has been built for this purpose and is discussed elsewhere (2, 3).

Selection of Electrochemical System

The materials used in this study were selected to satisfy electrochemical and optical criteria. Most of the experimental investigations were conducted with Cu as the electrode substrate, Pb as the deposited metal, and rhodamine-B chloride as the model inhibitor. The supporting electrolyte was composed of 1M sodium perchlorate, acidified to pH 3 by use of perchloric acid. The inhibitor concentration was typically 10 μM and $Pb^{\scriptscriptstyle ++}$ ion concentration was 5.0 mM (as nitrate).

The materials chosen have optical constants which are distinguishable over the spectral range accessible to the optical instrumentation (370-720 nm). Cu shows a distinctive absorption in the visible (around 550 nm), which is believed to be due to electronic transitions from the filled d bands into the sp conduction bands (4-6). In contrast to Cu, Pb shows no such characteristic in the visible and has relatively flat optical constants (7). Dissolved rhodamine-B has a characteristic absorption band at about 555 nm, believed to be due to an electronic transition from a singlet ground state to an excited triplet state (8). The electronic transition moment related to this absorption process lies parallel to the three conjugated rings of the rhodamine-B molecule. Rhodamine-B can be expected to adsorb on the surface with a preferential orientation of the transition moment, resulting in a birefringent (dichroic) adsorbate layer. This birefringence, and its spectral dependence, can be interpreted in terms of molecular orientation of the adsorbate on the surface (9, 10).

Electrochemically, Cu is stable to dissolution and oxidation at the reversible potential for Pb deposition from acidic solutions (11), and Pb forms a well-defined underpotential deposit on Cu substrates (12). In screening studies with over 30 materials, 14 of which were indicators or laser dyes, rhodamine-B was found to induce the largest increase in overpotential (125 mV) for bulk Pb deposition.

Experimental Procedure

The automatic spectroscopic ellipsometer used in this employed the polarizer-compensator-samplestudy analyzer optical configuration. It was self-nulling by use of a magneto-optic polarizer and analyzer, and could be operated in either a spectroscopic or fixed wavelength mode (2, 3). The spectral range was 370-720 nm, the angle of incidence used was 75°. A complete scan of spectroscopic measurements (delta and psi at different wavelengths) could be made in 3s although improved signal-tonoise ratios were obtained with slower scanning (e.g., one scan per 15s) or by averaging multiple scans. Slower scanning was essential to observe dye adsorption. Faster scanning gave sufficient signal quality for resolution of deposit micromorphology. Data acquisition was fully automated with a LSI-11/02 microcomputer system, which was also used for instrument calibrations and to interpret measurements by use of optical models.

The electrochemical cell used for experiments with Cu substrates was made of Teflon. It was designed for clean operation and low light absorption in the solution with a small liquid volume (2 ml) and a short optical path (2.5 cm) through the solution. The cell was pie-shaped, with windows made of polished and annealed optical-quality quartz, oriented normal to incident and reflected light beams for operation at an angle of incidence of 75°.

The round working electrodes were machined from polycrystalline oxygen-free Cu with an exposed surface of 1.27 cm diam. The electrodes were polished mechanically with a final alumina abrasive of 0.05 $\mu m.$ Air-formed oxide on the electrode surface was removed by prepolarization in the acidic electrolyte. The counterelectrode consisted of a Pb wire (99.999% purity) wound into a flat spiral, positioned 0.8 cm from the working electrode and of equal diameter. Lead oxide resulting from the corrosion of the counterelectrode was removed by cathodic prepolarization, using a second Pb wire electrode and an initial electrolyte charge. The initial electrolyte charge was replaced when the Cu working electrode was inserted, without exposing the anode to air.

A double-junction Ag/AgCl reference electrode (Dow-Corning 476067), inserted into a Teflon reference electrode compartment, was connected to the electrochemical cell with a Teflon capillary (0.16 cm od). The reference electrode was connected to the potentiostat via an electrometer probe (PAR).

Two syringes (15 ml each) were used to supply electrolyte to and withdraw from the cell. The electrolyte consisted of 5 mM Pb(NO₃)₂, 1M NaClO₄ at pH 3, with or without 10 μM rhodamine-B addition. It was prepared from deionized, singly distilled (10 M Ω) water and analyticalgrade salts. The electrolyte was nitrogen stripped for about 15 min prior to experiments, and the cell was purged with nitrogen before filling. The syringes and a

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nitrogen purge stream were connected with Teflon capillaries.

The potential of the working electrode was controlled with a potentiostat (PAR 173), and the charge passed was determined with a digital coulometer (PAR 179). The coulombic efficiencies between deposition and dissolution were always greater than 95%. Film thickness d_q based on charge is that given for a compact lead layer. Cell potential and current, together with the measured ellipsometer parameters delta and psi (measured as Faraday cell currents), were recorded by the LSI microcomputer system via an eight-channel analog-to-digital converter (Datel MST-LSI2).

Spectroscopic Ellipsometry

Electrochemical deposition of Pb films.—Thin films of lead on copper substrates were prepared by potentiostatic deposition (-600 mV vs. Ag/AgCl) from an electrolyte consisting of 5 mM Pb(NO₃)₂ and 1M NaClO₄ at pH 3, with deposition times of 1-8 min.

Based upon charge passed, the film thicknesses d_q would have been 31, 60, and 110 nm if the deposit had formed uniformly and had been compact. However, it was found that the deposit was not homogeneous and compact, and multilayer optical models had to be used to explain the spectroscopic ellipsometer measurements.

Single-layer optical models.-Film models investigated for the interpretation of measurements are illustrated in Fig. 1 (2). Examples of interpretations given below illustrate the use of the high information content of spectroscopic ellipsometer measurements for the derivation of wavelength-independent film parameters. Predictions based upon the following three optical models for singlelayer films (Fig. 1a-1c) are compared to experimental spectroscopic ellipsometer measurements in Fig. 2-4. In the compact film model, the entire deposit was treated as bulk Pb and there was only one adjustable parameter, the deposit thickness. Figure 2 shows spectroscopic simulations of delta for various assumed thicknesses (solid lines labeled 0, 1, 5, 10 nm, etc.). Comparison with experimental measurements also shown on the figure (dotted lines for 31, 60, and 110 nm compact thicknesses) illustrates that this model is inadequate.

The second single-layer optical model was that of a porous film in which porosity was taken into consideration by use of an effective refractive index. Pores were treated as microinclusions of electrolyte (n = 1.34) in a Pb host medium ($\hat{n} = 2.1$ -4.2*i*), and the optical properties of an isotropic homogeneous film were computed by use of the Maxwell-Garnett or Bruggeman equations. These equations have been used in the literature for modeling rough



Fig. 1. Schematic of film models investigated for the interpretation of spectroscopic ellipsometer measurements of thin Pb deposits on Cu substrates. Number of adjustable parameters given in parentheses. (a): Compact single film (1). (b): Porous single film, optical constants of effective medium determined according to Maxwell-Garnett (2) or Bruggeman (2). (c): Island single film, reflection coefficient determined by coherent superposition of polarization states, islands compact (2), porous (3) or anisotropic (6). (d): Multilayer films, three layers: compact and porous layers, dendritic islands (6) or two layers: compact and porous films (3).



Fig. 2. Spectroscopic simulations for compact film model (solid lines). Pb deposits of thickness 0, 1, 5, 10, 20, 31, 60, 110 nm. Experimental measurements (dotted lines) for 31, 60, 110 nm equivalent compact Pb deposits (d_{α} , based upon coulometric measurements) on Cu.



Fig. 3. Spectroscopic simulations of porous film model (solid lines). Volume fractions of Pb 0.1-0.9, amount of deposit corresponding to a 31 nm compact Pb layer on a Cu substrate. Simulations based upon the effective medium approximation (Bruggeman equation). Experimental measurements (dotted lines) for 0 and 31 nm equivalent compact film.



Fig. 4. Spectroscopic simulations of island film model (solid lines). Surface coverage 0.1-1.0, amount of deposit corresponding to 31 nm compact layer, Cu substrate. Simulations based on coherent superposition of polarization states. Experimental measurements (dotted lines) for 0 and 31 nm equivalent compact film.

surfaces and particulate films (13-16). This model involved two adjustable parameters: the deposit thickness and porosity. Spectroscopic simulations of the ellipsometer parameter delta for this model are shown in Fig. 3 for an amount of deposit corresponding to a 31 nm thick, compact Pb layer (based upon coulometric measurements) redistributed on a Cu substrate as deposits of various porosities. Simulations are represented by solid lines, and experimental measurements are represented by dotted lines. It is evident that this model alone is also inadequate to explain experimental measurements of delta and deposit thickness based on charge.

The third model was based on the coherent superposition of polarization states resulting from reflection of adjacent film-covered and bare elements of a patchwise covered surface (17-19). The island film may be compact, porous, or anisotropic. In the latter case, at least six unknown film parameters would have to be determined, which results in unacceptable parameter variances. The dimensions of the islands are assumed to be smaller than the transverse and longitudinal coherence of the incident light at the specimen surface. For the present spectroscopic ellipsometer, the longitudinal coherence ranges from 16 μ m at 400 nm to 49 μ m at 700 nm (20); the transverse (lateral) coherence ranges from 10 μ m at 400 nm to 17 μ m at 700 nm (21). These estimates assume (conservatively) a bandwidth twice that of the source (20 nm). Since complementary SEM studies of the films investigated here have shown the dimensions of dendritic islands to be less than 15 μ m, use of the coherent superposition model is justified for the present experiments.

Spectroscopic simulations of delta for films corresponding to a 31 nm thick, uniform, compact Pb deposit (based upon coulometric measurements), redistributed on a Cu substrate as compact islands, are shown in Fig. 4. The island model used for these simulations employed two adjustable parameters: fractional surface coverage and island thickness. Combinations of these two parameters which were consistent with the amount of Pb known to be in the film from coulometric measurements were used. It can be seen that this model is also incapable of explaining the experimental measurements.

For all single-layer optical models, disagreement of the computed spectral dependence of psi was similar to that shown for delta. It is important to note that interpretations of measurements at a single wavelength often are possible, but the entire spectral range cannot be fitted with any of the models. This fact illustrates the greatly increased information content of spectroscopic measurements.

Multiple-layer optical models.—A three-layer model (2) is illustrated in Fig. 1d. The first layer was assumed to be an isotropic film representing the underpotential deposit. Optical properties of this layer were determined in a separate study (22).

The second layer was assumed to be a granular, porous deposit with optical constants computed from the properties of Pb and electrolyte, by use of the Bruggeman theory for a binary mixture, Eq. [1]-[5]. Incorporation of this layer into the optical model was motivated by light scattering measurements (22)

$$\frac{\hat{\epsilon}_{\rm Ph} - \hat{\epsilon}_{\rm f}}{\hat{\epsilon}_{\rm Ph} + 2\hat{\epsilon}_{\rm f}} \theta_{\rm Ph} + \frac{\hat{\epsilon}_{\rm s} - \hat{\epsilon}_{\rm f}}{\hat{\epsilon}_{\rm s} + 2\hat{\epsilon}_{\rm f}} (1 - \theta_{\rm Ph}) = 0 \qquad [1]$$

$$\hat{\epsilon}_{\rm f} = \frac{1}{2} \left[-\hat{A} \pm (\hat{A}^2 - 4\hat{B})^{1/2} \right]$$
 [2]

$$\hat{A} = \frac{1}{2} \left[\hat{\epsilon}_{\rm Pb} (1 - 3\theta_{\rm Pb}) + \hat{\epsilon}_{\varsigma} (3\theta_{\rm Pb} - 2) \right]$$
[3]

$$\hat{B} = -\frac{1}{2} \left[\hat{\epsilon}_{\rm Pb} \hat{\epsilon}_{\rm s} \right]$$
[4]

$$\hat{n}_{\rm f} = \hat{\epsilon}_{\rm f}^{1/2}$$
 [5]

The root with the largest modulus was used as solution of Eq. [1] (23).

The third layer represents dendritic Pb islands, visible in scanning electron micrographs, and was modeled by an island film and coherent superposition of polarization states (Eq. [6]) resulting from reflection on adjacent surface elements with and without island coverage

$$\hat{r}_{
u} = heta_{\mathrm{f}}\hat{r}_{
u,\mathrm{f}} + (1- heta_{\mathrm{f}})\hat{r}_{
u,\mathrm{s}}$$

Use of coherent superposition implies that the diameter of islands is smaller than the spacial coherence of the illuminating light. The total amount of Pb contained in the three layers was adjusted to agree with the coulometric measurements.

Although the three-layer model provides very good agreement between predictions and measurements, the computations have shown that the optical effect of the third (dendritic island) layer is often negligible, because the surface coverage by islands is small (5-10%). A simplified two-layer model was therefore used for interpretations. Only three unknown parameters have to be determined with this model (thickness of the first, compact layer, and thickness and porosity of the second, porous layer), as opposed to six parameters required for the three-layer model. Despite the insignificant optical effect of the dendritic island layer, it usually contained about two-thirds of the Pb deposit known to be present from coulometric measurement. All optical constants were determined experimentally from independent measurements, and were not treated as adjustable parameters.

Optimization of the two-layer model.—Optimized values of the wavelength-independent parameters of the twolayer model (thicknesses and porosity) were obtained by minimizing the sum-of-squares error between measurements and model predictions over the entire spectral range. As outlined in a previous paper (22), parameter confidence intervals are calculated from the variance of the measurements by use of the student t-statistic at a 95% confidence level for 2N-P degrees of freedom, where N is the number of delta-psi measurements over the spectral range (from a single spectroscopic scan) and P is the number of adjustable parameters. Wavelengths of individual measurements are spaced at intervals greater than the source bandwidth; measurements at each wavelength are treated as independent observations of the same surface.

Optimization of the wavelength-independent parameters required to specify only the first two layers resulted in smaller parameter variances than if all three layers were included, because the variance for any parameter increases dramatically as the total number of model parameters P approaches the number of data points (2N) used in the optimization. This strategy is based on the results of optical computations which had shown that the third (island) layer has an insignificant optical effect. All deposited Pb not accounted for in the first two layers after optimization was assumed to be contained in the dendritic islands.

Results of the optimization are shown in Fig. 5 and 6 (delta and psi vs. wavelength). The solid lines represent experimental measurements, and the circles represent points calculated by a multidimensional optimization routine. Very good agreement was obtained by optimization of the thickness and porosity of the second layer, which are wavelength-independent adjustable parameters. Table I summarizes the results of the optimization. Measured and calculated values of delta and psi at particular wavelengths are tabulated for three thicknesses. The amount of material, based on coulometric measurements,

Table I. Spectroscopic ellipsometry of Pb deposits: optimization of twolayer model. Confidence intervals for model parameters given for 95% confidence limits

Thickness of compact deposit d _g based on charge	30 nm	60 nm	110 nm
Thickness of porous layer	31 ± 4.1	50 ± 16.2	87 ± 266.4
Vol. fraction Pb in porous laver	0.585 ± 0.139	0.775 ± 0.165	0.968 ± 0.274
Thickness of compact (UPD) laver (nm)	0.48 ± 0.11	0.48 ± 0.11	0.48 ± 0.11

[6]



Fig. 5. Spectroscopic simulation for optimum fit of ellipsometer parameter delta for a two-layer film model (circles) and measurements (solid lines) for equivalent deposit thicknesses of 0, 31, 60, and 110 nm. Film parameters derived from aptimum fit of model predictions to measurements given in Table 1.



Fig. 6. Spectroscopic simulation for optimum fit of ellipsometer parameter psi for a two-layer film model; other comments as in Fig. 5.

present in these films corresponds to compact layers of thickness 31, 60, and 110 nm. By minimizing the sum-ofsquares error (24) between the model predictions and measured values of delta and psi over the entire spectral range, one determines the optimum values of the wavelength-independent adjustable parameters (thickness and porosity). Table I shows the values of the thickness and porosity of the porous layer with their statistically determined confidence interval and the thickness of the compact layer which corresponds to that of the underpotential deposits (19, 22). Thicknesses and porosities of the second layer are thus found to be 31 nm and 41%, 50 nm and 22%, and 87 nm and 3%, respectively.

The large error limits given for the film thickness in the last column of Table I (87 ± 266 nm) are caused by the fact that a layer of that thickness is just barely penetrated by the light (conventional penetration depth for normal incidence 8 and 13 nm at wavelengths 400 and 700 nm, respectively) and behaves optically almost like bulk lead. The measurements are therefore insensitive to deposit thickness, resulting in a wide confidence interval.

Inhibitor Effects

Conclusions about the effect of rhodamine-B on deposit porosity were derived from spectroscopic ellipsometer measurements by the methods discussed in the preceding section. Figures 7 and 8 show, for a thin deposit (31 nm, based on charge), how the distinctive spectral properties of the Cu substrate, which are still visible through the deposit obtained without dye, were completely masked by a deposit of equal lead content obtained in the presence of the dye. The large optical difference is due to differences

Table II. Effect of rhodamine-B on the morphology of bulk Pb deposits during cyclic voltammetry. Comparison of thickness d_q of bulk deposit derived from charge passed with thickness d and volume fraction θ_{Pb} obtained from ellipsometer measurements. Points B to F selected in first and second potential sweep identified in Fig. 11 and 12

					$d \times \theta_{Pb}$
Sweep no., point	<i>E</i> (mV)	$d_{ m q}$ (nm)	<i>d</i> (nm)	$ heta_{\mathrm{Pb}}$	$\frac{d_q}{d_q} \times 100(\%)$
		No d	ye added		
1, B	-582	16.3	7.5	0.38	17
1, C	-430	37.9	50.6	0.56	75
2, E	-573	16.3	8.6	0.35	18
2, F	-400	37.9	50.9	0.55	74
		$10 \ \mu M$ rh	nodamine	e-B	
1, B	-800	21.7	11.3	0.89	46
1, C	-439	37.9	29.6	0.85	66
2, E	-800	27.1	24.4	0.61	55
2, F	-388	43.3	42.0	0.66	64

in porosity (45% without dye, < 10% with dye). Spectral scans of delta for a thicker deposit (110 nm, based on charge) are shown in Fig. 9. Here, deposits made with and without rhodamine-B show similar optical characteristics, which can be interpreted as Pb layers too thick to be penetrated by the light, even for deposits obtained without dye. The optical interpretations were similar for deposits obtained with constant or cycling potential. The underpotential deposit can be represented as a compact monolayer in the absence of the dye and an incomplete monolayer in its presence.



Fig. 7. Spectroscopic measurement of ellipsometer parameter delta for thin Pb deposit on Cu (corresponding to 31 nm compact thickness); effect of presence of dye, bare Cu substrate shown for comparison. Potentiostatic deposition, -600 mV vs. Ag/AgCl, electrolyte as in Fig. 1, except for dye. Substrate visible through porous deposit obtained without dye.







Fig. 9. Spectroscopic measurement of ellipsometer parameter delta for thick Pb deposit on Cu (corresponding to 110 nm compact thickness). Substrate obscured.



SCALE: 10 MICRONS

Fig. 10. Effect of rhodamine-B on the micromorphology of thin (1 min, 30 nm) and thick (8 min, 110 nm) Pb deposits; scanning electron micrographs. Potentiostatic deposition at -600 mV vs. Ag/AgCl.

The scanning electron micrographs shown in Fig. 10 illustrate the fine-grained initial (1 min, corresponding to 30 nm compact thickness) deposit obtained with the dye and the dendritic islands obtained without it. At a later stage of deposition (8 min, corresponding to 110 nm compact thickness) the effect of the dye begins to wear off and patches of porous deposit appear, while without dye the dendrites continue to grow into a matted layer. Energy dispersive x-ray analysis has shown that the dendrites are composed of lead, and that, without dye, the deposit between dendrites is very thin, while it is thick and uniform with the dye, as had been inferred from the ellipsometer measurements.

Ellipsometry and Cyclic Voltammetry

Cyclic voltammetry and ellipsometry measurements (at 550 nm wavelength) were conducted simultaneously. A potential ramp (+100 to -800 mV vs. 4M Ag/AgCl reference, 2.8 V/min) and current response are shown in Fig. 11. The corresponding ellipsometer measurements are given in Fig. 12. They show a response which coincides with the onset of a cathodic deposition current (points A and D for first and second sweep). The deposit thickness and porosity $(1 - \theta_{Ph})$ were determined by fitting the ellipsometer measurements to the two-layer optical model discussed above.

Interpretations for selected measured points B-F, Fig. 11 and 12, are given in Table II. It can be seen that, during the first potential cycle, the presence of the dye results in



Fig. 11. Potential cycles (+100 to -800 mV vs. Ag/AgCl, sweep rate 2.8 V/min) applied to working electrode (1.13 cm²) and resulting current response. Pb deposition on Cu from 5 mM·Pb(NO₃)₂, 1M NaClO₄, pH 3, 10 μ M rhodamine-B. Evaluation of measurements at points B to F shown in Table II.



Fig. 12. Ellipsometer measurements collected during potential cycles shown in Fig. 11; wavelength 550 nm (10 nm bandwidth), angle of incidence 75°.

a higher volume fraction (lower porosity) of lead in the bulk deposit. The effect of the dye gradually disappears in the second and subsequent cycles, where volume fractions close to those without dye are obtained (22). Thus, a 15% porosity obtained in the first potential sweep for a layer 29.6 nm thick compares to a 34% porosity obtained in the second sweep at 42 nm thickness. This increase in porosity is responsible for the change from a positive to a negative response in delta to deposition between the first and subsequent sweeps. Depletion of dye at the electrode surface is the cause for this change in behavior. The effect of rhodamine-B on deposit porosity reappears after relaxation periods at open circuit. The optical constants of compact lead ($\hat{n} = 2.1-4.2i$) were combined with the refractive index of the electrolyte (n = 1.34) to compute the optical properties of the porous deposits by use of the Bruggeman equation. The optical constants of the copper substrate were measured as $\hat{n} = 0.7-3.4i$.

The amount of lead in the bulk deposit derived from the ellipsometer measurements (product of thickness *d* and volume fraction θ_{Ph}) is always less than the amount based on the charge passed (thickness d_{q}), as shown in the last column of Table II. The ratio is lower for deposits without dye, in agreement with the formation of more dendritic islands, which are not measured optically (2).

Measurement of the Inhibitor Layer

Underpotential and bulk peaks in the cyclic voltammograms are shifted to more cathodic potentials by adsorbed rhodamine-B. Due to depletion of the dye, this potential shift disappears, however, after several uninterrupted potential cycles, but it reappears after relaxation periods (10-60 min) at open circuit, during which



Fig. 13. Readsorption of rhodamine-B on Cu during 5-60 min relaxation at open circuit after removal by multiple potential sweeps; effect on cyclic voltammograms, 2 V/min, 1M NaClO₄, 5 mM Pb⁺⁺, 10 μ M dye, p H 3, 1.27 cm².

adsorption can take place by diffusion from the electrolyte (Fig. 13). The repopulation of the electrode surface with rhodamine-B adsorbate has also been observed by spectroscopic ellipsometry; differential measurements with and without dye (Fig. 14 and 15) show the development of spectral features due to the adsorbed dye. These spectral scans were taken immediately preceding the corresponding cyclic voltammograms of Fig. 13, the relaxation times in minutes are indicated along the curves. A study of the adsorbed dye layer (25) which will be presented elsewhere, has shown that molecules oriented normal to the electrode surface, forming an adsorbate layer 32Å thick, appear to be responsible for the inhibition of Pb deposition.

The long relaxation times required to restore rhodamine-B coverage of the surface are due to the slow diffusion from the very dilute (10 μ M) solution (26). For a coverage between 4 × 10¹³ and 2 × 10¹⁴ molecules/cm², assuming a linear adsorption isotherm with an equilibrium constant of 0.01-0.03 cm (27), a 30 min time to halfcoverage would require a diffusion coefficient between 5.6×10^{-8} and 5.0×10^{-7} cm² s⁻¹, which appears reasonable.

Conclusions

Rhodamine-B acts as a deposition inhibitor for the electrodeposition of Pb on Cu and Ag. Potentials required for deposition are more cathodic in the presence of rhodamine-B than in its absence.



Fig. 14. Differential spectroscopic measurements of ellipsometer parameter Δ for readsorption of rhodamine-B on Cu after multiple potential sweeps. (Spectra for bare Cu in electrolyte without dye subtracted.) Relaxation periods at open circuit in minutes given along curves. Experimental conditions as in Fig. 13.



Fig. 15. Differential spectroscopic measurements of ellipsometer parameter ψ for readsorption of rhodamine-B on Cu. Experimental conditions as in Fig. 13.

Less porous deposits are formed in the presence of the dye and the underpotential deposit is less than a complete monolayer. The observed decrease in bulk deposit porosity resulting from the addition of rhodamine-B can be attributed to an increase in nucleation density on the surface and inhibition of dendrite growth.

If nucleation of the bulk deposit is favored at imperfections of the first monolayer of the underpotential deposit, the partial monolayer formed in the presence of rhodamine-B may provide a higher density of nucleation sites. Also, because deposition occurs at more cathodic potentials in the presence of the dye, the equilibrium size of nuclei on the surface is smaller and their number larger. Kinetic effects of the adsorbed dye molecules may involve a decrease in the surface diffusion of Pb adatoms, which would also result in a higher nucleation density.

The selective reduction of rhodamine-B at easily accessible (high activity) sites, such as tips of dendrites, in favor of Pb^{++} reduction, could shift the deposition to less accessible sites close to the surface which would preferentially be supplied by Pb^{++} because of its higher diffusion coefficient.

The dye is reduced at the surface and inhibitory effects on the deposit micromorphology disappear until the surface is repopulated with adsorbate during relaxation periods at open circuit.

Spectroscopic ellipsometry allows one to calculate confidence intervals of wavelength-independent parameters for micromorphological optical models and to justify the use of more sophisticated optical models on the basis of the greater degrees of freedom.

For three electrolytically formed Pb deposits (compact thickness 31, 60, and 110 nm) the best agreement between measurements and model predictions was obtained for a three-layer model or a two-layer simplification of it. The distribution of deposited material between compact, porous (granular), and dendritic island layers could be determined.

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LIST OF SYMBOLS

- *d* thickness of deposit derived from ellipsometer measurements
- $d_{\rm q}$ \$ thickness of deposit for a compact layer, based on charge

- k imaginary part of complex refractive index
- real part of complex refractive index n
- complex refractive index of porous film $\hat{n}_{\rm f}$
- complex refractive index of substrate \hat{n}_{s}
- complex Fresnel reflection coefficient for polariza- \hat{r}_{ν} tion ν (s or p) complex Fresnel reflection coefficient for film-
- $\hat{r}_{\nu,\mathrm{f}}$ covered surface
- complex Fresnel reflection coefficient for bare $\hat{r}_{v,\mathrm{s}}$ surface
- \hat{A}, \hat{B} complex parameters for quadratic equation
- potential vs. 4M Ag/AgCl number of pairs of Δ and ψ measurements Ε
- \tilde{N} P
- number of model parameters to be fitted
- Q charge passed
- level of confidence $(1-2\alpha)$ α
- complex dielectric function of porous film $\hat{\boldsymbol{\epsilon}}_{\mathrm{f}}$
- complex dielectric function of metallic Pb $\hat{\epsilon}_{\mathrm{Pb}}$
- complex dielectric function of electrolyte solution $\ddot{\theta_{\mathrm{f}}}$ fraction of surface covered by film of refractive in $dex \hat{n}_{f}$
- $heta_{\mathrm{Pb}}$ volume fraction of Pb in porous film derived from ellipsometer measurements
- ellipsometer parameter, phase difference between pΔ and s electric field components after reflection, relative to the incident (degrees)
- calculated value of Δ $\Delta_{\rm C}$
- measured value of Δ $\Delta_{\rm M}$
- ψ ellipsometer parameter, amplitude ratio of p and selectric field components after reflection (tan ψ), relative to the incident (degrees)
- calculated value of ψ $\psi_{\rm C}$
- measured value of ψ $\psi_{\rm M}$

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Electrochemical Deposition of Cadmium Sulfide Thin Films from Organic Solution

I. Sulfur Reduction and Cadmium-Polysulfide Complex Formation

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ABSTRACT

The reduction of sulfur in an aprotic solvent (DMSO) is known to be a two-stage process; each stage involves a bielectronic electrochemical step followed by a chemical dismutation reaction. When reduction is performed at a rotating electrode, it is shown that the electrochemical steps alone are involved in the electrode process. The voltammetric waves then correspond to the simple scheme

1st wave: $S_8 + 2e^- \rightarrow S_8^{2-}$ 2nd wave: $S_8^{2-} + 2e^- \rightarrow 2S_4^{2-}$

The slower dismutation reactions appear only in the bulk of solution. They lead to electron numbers which differ from 2 when S_8 is electrolyzed. The addition of Cd^{2+} ions to a polysulfide ion solution yields soluble polysulfide-cadmium 2:1 complexes $[Cd(S_x)_2]^{2-}$ with x = 8, 6, and 4. The stability of the complex is increased when x decreases from 8 to 4, which leads to dismutation and exchange reactions between complexes, polysulfide ions, and sulfur.

Cadmium sulfide CdS is a potential material for fabrication of low-cost solar cells based on thin polycrystalline films (1, 2). Besides several chemical or physical processes (3-5), thin cadmium sulfide films are obtained by means of electrochemical reduction of an organic solution containing molecular sulfur S_8 and a cadmium salt (6-9). Although it is sometimes written as a simple reaction $(Cd^{2+} + 1/4 S_{e} + 2e^{-} \rightarrow CdS)$ (7), the electrochemical deposition of CdS soon proves to be a rather intricate process. The purpose of this paper (and of our next one on the same topic) is to demonstrate how sulfur reduction is implied with further chemical and electrochemical reactions to yield CdS deposits.

Chemical reaction between molecular sulfur S₈ and monosulfide ion S^{2-} in various solvents yields a mixture of different polysulfide ions S_{r}^{2-} . The sulfuration index