

## THERMAL DÉCOMPOSITION OF AMMONIUM METAVANADATE SUPPORTED ON $\text{Al}_2\text{O}_3$

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### ABSTRACT

The thermal decomposition of ammonium metavanadate supported on aluminium oxide was investigated using DTA, TG and X-ray diffraction techniques.

The results obtained revealed that ammonium vanadate decomposed at 225–250°C giving an intermediate compound  $((\text{NH}_4)_2\text{V}_6\text{O}_{16})$  which decomposed readily at 335–360°C producing  $\text{V}_2\text{O}_5$ . Alumina was found to enhance the formation of the intermediate compound and retard its decomposition. Some of the  $\text{V}^{5+}$  ions of  $\text{V}_2\text{O}_5$  lattice seemed to be reduced into  $\text{V}^{4+}$  and  $\text{V}^{3+}$  ions by heating in air at 450°C in the presence of  $\text{Al}_2\text{O}_3$ . Such a reaction was attributed to dissolution of some  $\text{Al}^{3+}$  ions in the  $\text{V}_2\text{O}_5$  lattice via location in interstitial positions and/or in cationic vacancies.  $\text{Al}_2\text{O}_3$  was found to interact with  $\text{V}_2\text{O}_5$  at 650°C giving well-crystalline  $\text{AlVO}_4$  which decomposed at about 750°C forming well-crystalline  $\delta\text{-Al}_2\text{O}_3$  and  $\text{V}_2\text{O}_5$ . Pure  $\text{Al}_2\text{O}_3$ , heated in air at 1000°C, existed in the form of the  $\kappa$ -phase which, on mixing with  $\text{V}_2\text{O}_5$  (0.5  $\text{V}_2\text{O}_5$ :1  $\text{Al}_2\text{O}_3$ ) and heating in air at 1000°C, was converted entirely to the well-crystalline  $\alpha\text{-Al}_2\text{O}_3$  phase.

### INTRODUCTION

The thermal decomposition of ammonium metavanadate made the object of several investigations [1–9]. The decomposition process takes place according to different mechanisms depending, mainly, upon the atmosphere in contact with the solid. In an  $\text{N}_2$  atmosphere or under a reduced pressure of  $10^{-6}$  Torr,  $\text{NH}_4\text{VO}_3$  decomposes readily giving  $\text{V}_3\text{O}_7$  [2,4,6] while in an  $\text{O}_2$  atmosphere  $\text{V}_2\text{O}_5$  represents the thermal product of the solid metavanadate [1,3,7–9]. Indeed, the thermal decomposition in  $\text{O}_2$ , or in dry air, proceeds through the formation of an intermediate compound  $((\text{NH}_4)_2\text{V}_6\text{O}_{16})$  [4,5].

The effect of a support on the thermal decomposition of ammonium metavanadate, to our knowledge, has not yet been investigated. The present work reports a study on the thermal decomposition of ammonium metavanadate supported on  $\gamma\text{-Al}_2\text{O}_3$ . The techniques employed were DTA, TG, DTG and X-ray diffraction.

## EXPERIMENTAL

*Materials*

Gamma-alumina, precalcined in air at 500°C, was impregnated with different proportions of ammonium metavanadate dissolved in dilute  $\text{NH}_4\text{OH}$  using a pore-filling method. The composition of the impregnated solids was  $0.1 \text{ NH}_4\text{VO}_3 \cdot \text{Al}_2\text{O}_3$ ,  $0.4 \text{ NH}_4\text{VO}_3 \cdot 1 \text{ Al}_2\text{O}_3$  and  $1.0 \text{ NH}_4\text{VO}_3 \cdot 1.0 \text{ Al}_2\text{O}_3$ , and the solids were dried at 110°C to constant weight.

*Techniques*

Differential thermal analysis of various impregnated solids was carried out using a Netzsch-Gerätebau simultaneous thermal analysis apparatus (STA 409, type 6.223). The rate of heating was kept at  $10^\circ\text{C min}^{-1}$ . A 200-mg sample of each solid specimen was employed in each case.

An X-ray investigation of the thermal products of pure ammonium metavanadate and the metavanadate supported on aluminium oxide was performed with a Philips diffractometer (type PW 1390). The patterns were run with nickel-filtered copper radiation,  $\lambda = 1.5405 \text{ \AA}$  at 40 kV and 25 mA with a scanning speed of  $2^\circ$  in  $2\theta \text{ min}^{-1}$ .

## RESULTS AND DISCUSSION

*Thermal decomposition of ammonium metavanadate supported on  $\text{Al}_2\text{O}_3$  ( $0.1 \text{ NH}_4\text{VO}_3 \cdot 1 \text{ Al}_2\text{O}_3$ )*

Figure 1 represents the DTA, TG and DTG curves of  $0.1 \text{ NH}_4\text{VO}_3 \cdot 1 \text{ Al}_2\text{O}_3$ . Three endothermic peaks are observed, the first is very strong, the second is weak and rapid and the third is sharp and relatively strong. The maxima of these peaks are located at 130, 355 and 655°C, respectively. The first peak, which was followed by a 5.3% loss in weight, indicated desorption of physisorbed water, retained in the  $\text{Al}_2\text{O}_3$  solid. The second peak was accompanied by a loss in weight of 5.3% and corresponded to the decomposition of  $\text{NH}_4\text{VO}_3$ . The peak at 655°C was not followed by any change in weight and may characterize a phase transformation of  $\text{Al}_2\text{O}_3$  solid or a solid-solid interaction between vanadium and aluminium oxides. The identification of such a change will be made by X-ray diffraction, in the next part of the present investigation. The total recorded weight loss, from room temperature to 800°C, was 11.7%.

 *$0.4 \text{ NH}_4\text{VO}_3 \cdot 1 \text{ Al}_2\text{O}_3$* 

Figure 2 represents the DTA, TG and DTG curves of  $0.4 \text{ NH}_4\text{VO}_3 \cdot 1 \text{ Al}_2\text{O}_3$ . Four endothermic peaks are found: the first is very strong; the

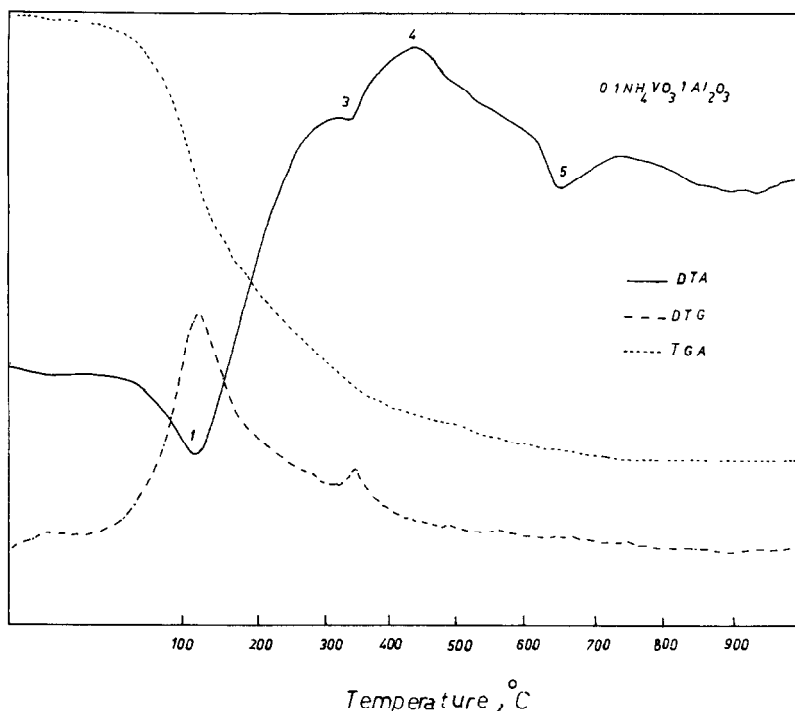


Fig. 1. DTA, TG and DTG curves of 0.1  $\text{NH}_4\text{VO}_3$ :1  $\text{Al}_2\text{O}_3$ .

second and the third are weak and rapid; the fourth is strong. The maxima of these peaks are located at 130, 225, 360 and 660°C, respectively. A possible exothermic peak at 420°C can be also observed from Fig. 2. The weight loss corresponding to the four endothermic peaks attains 5.7, 3.6, 2.3 and 0.0%, respectively. The exothermic peak was followed by 0.8% weight loss and the total weight loss, up to 800°C, was 13.3%. The second and third endothermic peaks at 225 and 360°C indicated the decomposition of  $\text{NH}_4\text{VO}_3$  to an intermediate compound which underwent thermal decomposition giving  $\text{V}_2\text{O}_5$ . This speculation will be confirmed by the X-ray diffraction study presented in the next part of this work. The last endothermic peak at 660°C, which was not followed by any change in weight, indicated a phase transformation or a solid-solid interaction between aluminium and vanadium oxides.

#### 1.0 $\text{NH}_4\text{VO}_3$ :1 $\text{Al}_2\text{O}_3$

Figure 3 represents the DTA, TG and DTG curves of ammonium metavanadate and  $\text{Al}_2\text{O}_3$  in equimolar ratio. Four endothermic peaks are observed, the first two and the last one are sharp and strong while the third is weak and rapid. The maxima of these peaks are located at 130, 250, 235 and 665°C, respectively. An exothermic peak at 430°C, similar to that found in

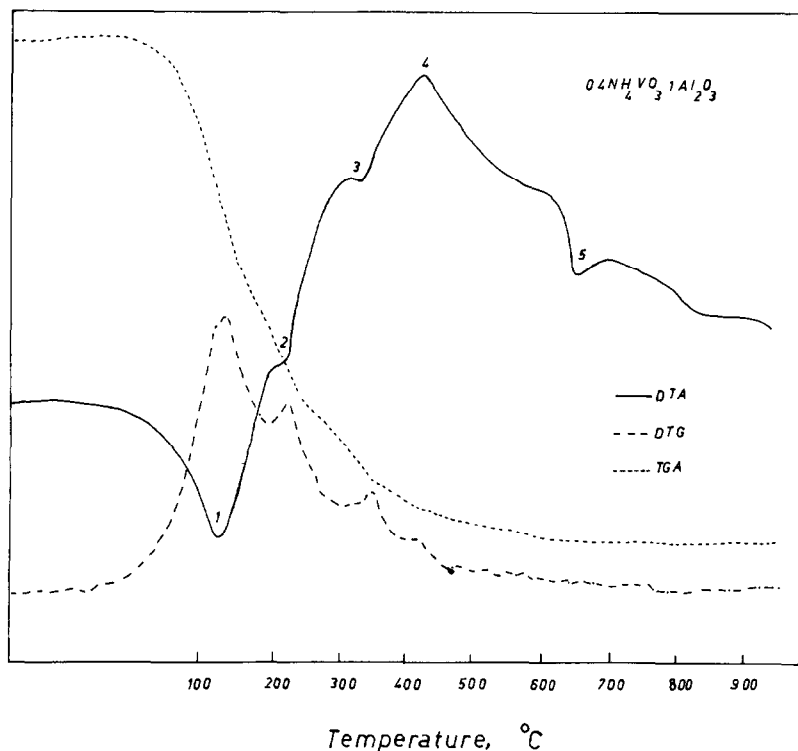


Fig. 2. DTA, TG and DTG curves of 0.4  $\text{NH}_4\text{VO}_3$ :1  $\text{Al}_2\text{O}_3$ .

the case of 0.4  $\text{NH}_4\text{VO}_3$ :1  $\text{Al}_2\text{O}_3$ , is also observed. However, this peak was accompanied by a weight loss of 1.5%. The weight losses corresponding to the four endothermic peaks are 4.1, 6.2, 3.8 and 0.0%, respectively. The total weight loss, up to 800°C, was 15%. The second and third peaks indicated the decomposition of ammonium vanadate to an unstable intermediate compound which readily decomposed producing  $\text{V}_2\text{O}_5$ .

The comparison between Figs. 1–3 reveals that ammonium metavanadate at the smallest concentration, 0.2  $\text{NH}_4\text{VO}_3$ :1  $\text{Al}_2\text{O}_3$ , decomposed in one step and in two distinct steps in the case of the other concentrations. The percentage weight loss corresponding to the decomposition of ammonium metavanadate increased on increasing its concentration. Furthermore, the area of the last endothermic peak, not accompanied by any weight loss, also increases on increasing the extent of ammonium metavanadate. This indicates that the last endothermic peak (peak 5) is more likely to be related to a solid–solid interaction between  $\text{V}_2\text{O}_5$  and  $\text{Al}_2\text{O}_3$ .

The analysis of the data of thermal behaviour of different solids indicates that the weight loss corresponding to the decomposition of ammonium vanadate in different specimens attained 5.3, 5.9 and 10% for the solids containing 0.1, 0.4 and 1.0 mol  $\text{NH}_4\text{VO}_3$ /mol  $\text{Al}_2\text{O}_3$ , respectively. Further-

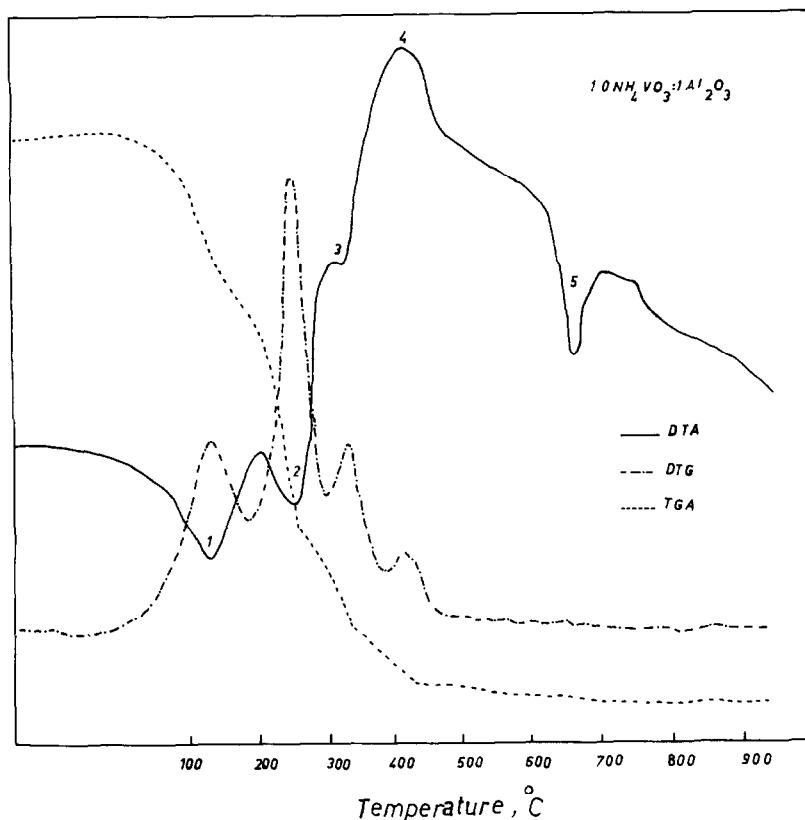
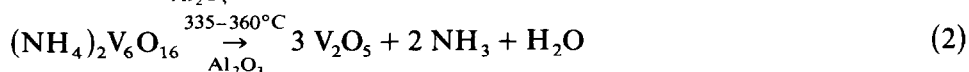
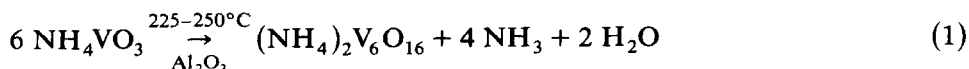


Fig. 3. DTA, TG and DTG curves of 1  $\text{NH}_4\text{VO}_3$ :1  $\text{Al}_2\text{O}_3$ .

more, the decomposition of ammonium metavanadate occurred in two distinct steps (except in the solid containing 0.1  $\text{NH}_4\text{VO}_3$ ). The first step is probably related to the formation of an intermediate compound while the second corresponded to its decomposition to produce  $\text{V}_2\text{O}_5$ . The maximum of the endothermic peak corresponding to the formation of the intermediate compound increased from 225 to 250°C on increasing the vanadate content from 0.4 to 1.0 mol/mol  $\text{Al}_2\text{O}_3$ . In contrast, the maximum related to the decomposition of the intermediate compound decreased from 360 to 335°C on increasing the vanadate content from 0.4 to 1.0 mol. These results indicate that the  $\text{Al}_2\text{O}_3$  support enhanced the formation of the intermediate compound and retarded its decomposition.

Adopting the mechanism of the thermal decomposition of  $\text{NH}_4\text{VO}_3$  [5,6] via the formation of  $(\text{NH}_4)_2\text{V}_6\text{O}_{16}$  as an intermediate compound according to



The weight losses corresponding to reactions (1) and (2) are 14.82 and 8.70%, respectively. The relative weight loss corresponding to reaction (1) is equivalent to 63% and that relative to reaction (2) is 37%. The observed relative weight loss for the thermal decomposition of  $\text{NH}_4\text{VO}_3$  supported on  $\text{Al}_2\text{O}_3$  attained 61:39% for the solid containing 0.4 mol  $\text{NH}_4\text{VO}_3$  and 62:38% for  $\text{Al}_2\text{O}_3$  treated with ammonium vanadate in equimolar ratio. The concordance between the theoretical and the observed data for the weight loss occurring during the thermal decomposition of ammonium metavanadate according to the mechanism represented by reactions (1) and (2) could be taken as evidence for the validity of such a mechanism in the present case.

### *X-ray investigation of the thermal products of various solids*

#### *Pure ammonium metavanadate*

Different specimens of ammonium metavanadate were subjected to heating in air for 5 h at 300, 400, 500 and 600°C. The diffraction lines of the  $\text{V}_2\text{O}_5$  phase were only detected in the X-ray diffraction patterns of different solids. However, the intensity of the diffraction lines was found to increase on increasing the calcination temperature from 300 to 400°C.

#### *Pure aluminium oxide*

Various samples of  $\text{Al}_2\text{O}_3$ , employed in the present work, were heated in air at various temperatures between 500 and 1000°C. The X-ray investigation revealed that  $\text{Al}_2\text{O}_3$  heated at 500°C was poorly crystalline  $\gamma$ -alumina and its degree of crystallinity increased on increasing the calcination temperature to 900°C.  $\text{Al}_2\text{O}_3$  heated at 1000°C existed in the form of  $\kappa$ -alumina [10].

#### *Aluminium oxide treated with different proportions of $\text{NH}_4\text{VO}_3$*

Different specimens of  $\text{Al}_2\text{O}_3$ , preheated at 500°C, were impregnated with various proportions of ammonium metavanadate and the solids produced were heated in air for 5 h at 500, 650, 750 and 1000°C. The X-ray investigation of these solids showed that all the solids heated at 500°C, except that treated with 1 mol  $\text{NH}_4\text{VO}_3$ /mol  $\text{Al}_2\text{O}_3$ , were composed of well-crystalline  $\text{V}_2\text{O}_5$  phase and poorly crystalline  $\gamma$ -alumina. These results indicated the absence of a solid–solid interaction between  $\text{Al}_2\text{O}_3$  and  $\text{V}_2\text{O}_5$  at 500°C, giving new compound(s). Possible interaction between these two oxides could proceed at temperatures above 500°C. Figure 4 represents the X-ray diffraction patterns of the thermal products of  $\text{Al}_2\text{O}_3$  treated with an equimolar ratio of  $\text{NH}_4\text{VO}_3$ . It is observed from Fig. 4 that, in the case of the solid heated at 500°C, all the characteristic diffraction lines of  $\text{V}_2\text{O}_5$  together with few lines of  $\delta$ - $\text{Al}_2\text{O}_3$  [10] of small intensity persisted in the diffraction patterns. It can be concluded that, similar to the case of the  $\text{Al}_2\text{O}_3$  specimens treated with smaller amounts of  $\text{NH}_4\text{VO}_3$ , alumina could

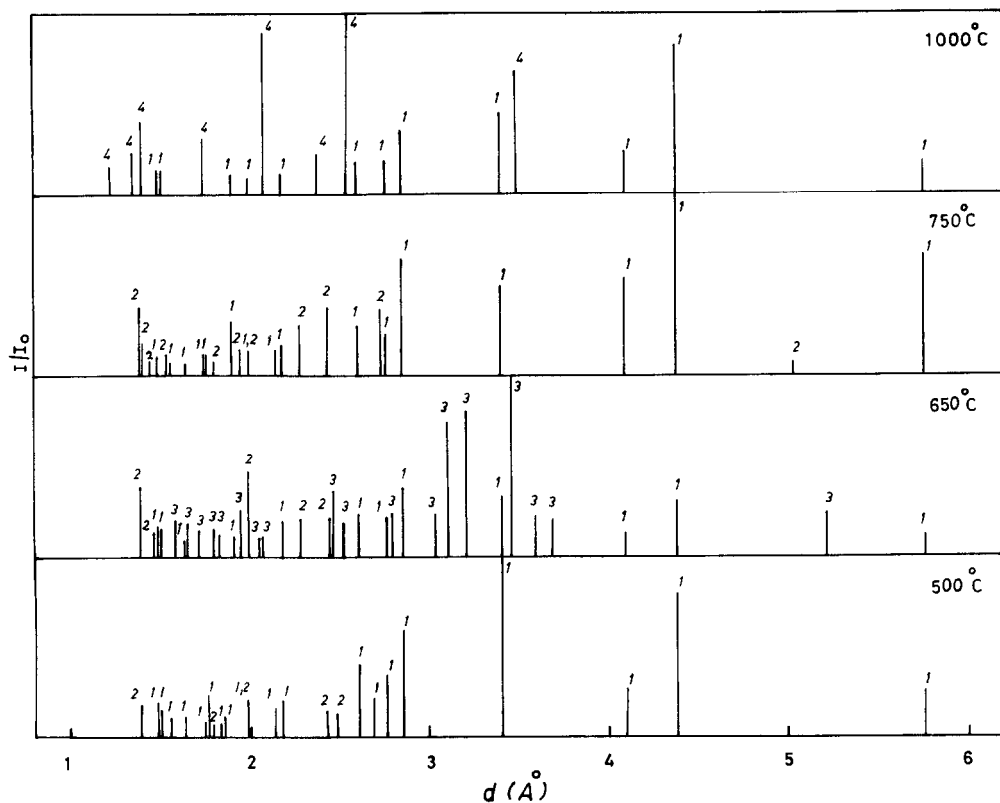


Fig. 4. X-ray diffraction patterns of the thermal products of 1  $\text{NH}_4\text{VO}_3$ :1  $\text{Al}_2\text{O}_3$ . (1)  $\text{V}_2\text{O}_5$ ; (2)  $\delta\text{-Al}_2\text{O}_3$ ; (3)  $\text{AlVO}_4$ ; (4)  $\alpha\text{-Al}_2\text{O}_3$ .

not interact with  $\text{V}_2\text{O}_5$  at 500°C to produce a new phase. However,  $\gamma$ -alumina was converted to  $\delta$ -alumina in the presence of  $\text{V}_2\text{O}_5$  (0.5 mol/mol  $\text{Al}_2\text{O}_3$ ). Increasing the calcination temperature of the solid of composition 0.5  $\text{V}_2\text{O}_5$ :1  $\text{Al}_2\text{O}_3$  to 650°C led to the formation of well-crystalline aluminium vanadate,  $\text{AlVO}_4$  [10] (cf. Fig. 4) together with a small portion of unreacted  $\text{V}_2\text{O}_5$  and  $\delta$ -alumina. On this basis, the strong endothermic peak at 665°C observed in the DTA curves of various solids (peak 5, Figs. 1–3) indicated the solid–solid interaction between  $\text{Al}_2\text{O}_3$  and  $\text{V}_2\text{O}_5$  giving  $\text{AlVO}_4$ . This reaction, which proceeds according to

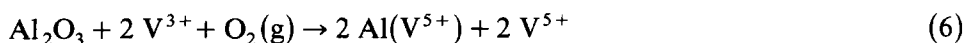
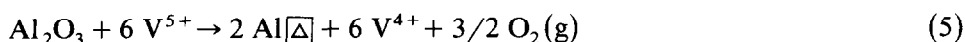
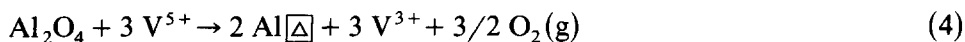


is, in fact, not accompanied by any weight loss of the reacting species. The thermal treatment of the solid at 750°C caused the complete decomposition of aluminium vanadate into well-crystalline  $\delta$ -alumina and  $\text{V}_2\text{O}_5$  phases (cf. Fig. 4). The fact that the thermal decomposition of  $\text{AlVO}_4$  has not been detected in the DTA curves of various solids (Figs. 1–3) indicated that such

a process occurred very slowly. Increasing the calcination temperature of the mixed oxide sample to 1000°C effected a phase transformation of  $\delta$ -alumina to  $\alpha$ -alumina and decreased the crystallinity of the  $V_2O_5$  phase. These results, clearly indicate a mutual effect between the  $Al_2O_3$  and  $V_2O_5$  phases. The complete phase transformation of  $\delta$ -alumina to  $\alpha$ -alumina has been induced by  $V_2O_5$  which lost some of its crystallinity.

The X-ray diffraction patterns, not presented here, of the solid of composition 0.2  $V_2O_5$ :1  $Al_2O_3$  and heated in air at 1000°C revealed the presence of a mixture of  $\delta$ - and  $\alpha$ -aluminas together with well-crystalline  $V_2O_5$ . These results show that increasing the amount of  $V_2O_5$  favoured the phase transformation process. It is well known that  $\alpha$ -alumina can be produced by heating pure  $Al_2O_3$  in air at temperatures above 1200°C [12–14]. The transformation of  $Al_2O_3$  into  $\alpha$ -alumina at 1000°C by treating with  $V_2O_5$  pointed to the role of this oxide in enhancing or catalyzing such a phase transformation which depended on the molar ratio of the reacting oxides. It has been shown by MacKenzie and Hossini [15] that the thermal transformation of  $\gamma$ - $Al_2O_3$  to  $\alpha$ - $Al_2O_3$  was effectively enhanced by an external d.c. electric field.  $\alpha$ - $Al_2O_3$  was formed in the region of the positive electrode. This behaviour was thought to be due to the electric withdrawal of protons from the anode region and their migration to the cathode, where they stabilized the defect spinel structure of  $\gamma$ - $Al_2O_3$ . The fact that the phase transformation of  $\delta$ -alumina to  $\alpha$ -alumina, occurring at 1000°C, has not been detected in the DTA curve of various solids, indicated that such a process took place with a relatively low rate.

The results of the thermal behaviour and X-ray diffraction studies of different solids showed that  $NH_4VO_3$  supported on  $Al_2O_3$  underwent thermal decomposition by heating in air giving  $V_2O_5$  via the formation of an unstable intermediate compound. The formed oxides interacted in the solid state at 650°C giving  $AlVO_4$ . Below such a temperature some kind of a solid solution, not easily detected by a simple X-ray diffraction analysis, might exist between  $Al_2O_3$  and  $V_2O_5$ . The ionic radii of  $Al^{3+}$  and  $V^{5+}$  ions are 0.50 and 0.59 Å, respectively [11]. Some of the  $Al^{3+}$  ions could be dissolved in the  $V_2O_5$  lattice. Such a process might proceed according to: location in interstitial positions and/or in cationic vacancies; by substituting some of the pentavalent vanadium ions of the  $V_2O_5$  lattice. These processes can be simplified, adopting Kröger's notions [16], as follows



$Al\boxed{\Delta}$  represents an aluminium ion located in an interstitial position or in a cationic vacancy, and  $Al(V^{5+})$  a trivalent aluminium ion located in the position of host  $V^{5+}$  cations present in the  $V_2O_5$  lattice. Reactions (4) and (5)

are accompanied by the transformation of some  $V^{5+}$  ions into  $V^{3+}$  and  $V^{4+}$  ions with the departure of a corresponding amount of  $O_2$  from the  $V_2O_5$  lattice. In other words, reactions (4) and (5) cause a weight loss of  $V_2O_5$ . In contrast, reaction (6) effects a gain in weight of  $V_2O_5$  due to fixation of some of atmospheric oxygen into the solid with the subsequent transformation of trivalent vanadium ions into  $V^{5+}$  ions. Such a reaction requires the presence of  $V^{3+}$  ions in the  $V_2O_5$  lattice [17]. This reaction cannot account for the experimental results obtained, simply because a weight loss has been detected in DTG and DTA curves of different solids (cf. the exothermic peak at  $430^\circ\text{C}$ , Figs. 1–3). It can be concluded that a portion of  $Al_2O_3$  has been dissolved in the  $V_2O_5$  lattice at temperatures around  $430^\circ\text{C}$  via location in interstitial positions and/or in cationic vacancies leading to the transformation of some  $V^{5+}$  ions to  $V^{3+}$  and/or  $V^{4+}$  ions. The solid solution thus formed underwent a chemical transformation by heating at  $650^\circ\text{C}$  giving  $AlVO_4$ .

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