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## Hierarchical Bi<sub>2</sub>MoO<sub>6</sub> nanosheet-built frameworks with excellent photocatalytic properties†

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Herein, we demonstrate for the first time the fabrication of one-dimensional (1D)  $\rm Bi_2MoO_6$  inter-crossed nanosheet-built frameworks by using  $\rm MoO_3$  nanobelts as the growth templates and molybdate source. Especially, this novel  $\rm Bi_2MoO_6$  framework structure exhibits remarkably enhanced photocatalytic activity toward the degradation of organic dyes under visible-light irradiation, far exceeding that of conventional  $\rm Bi_2MoO_6$  nanoplates and nanoparticles. The photoelectrochemical study suggests that the hierarchical framework structure could facilitate the photoinduced charge separation and transfer from the inter-crossed  $\rm Bi_2MoO_6$  nanosheets, which may make a significant contribution to the enhanced photocatalytic activity.

Semiconductor photocatalysts have attracted considerable attention over the past decades due to their potential applications in solving current energy and environmental issues by utilizing the abundant solar energy. Up to now, various semiconductors, especially TiO21-3 and ZnO,4-7 have been extensively studied and proven to be powerful catalysts for diverse photo-chemical reactions, such as splitting of water into hydrogen, degradation of organic waste, and artificial photosynthesis. However, it should be noted that limited by a large band gap (3.2 eV), these photocatalysts can only absorb the ultraviolet light ( $\lambda \leq 400 \text{ nm}$ ), which greatly restricts their efficiencies for the utilization of solar energy. To address this issue, much effort has been devoted to the exploration and fabrication of novel semiconductor materials with appropriate band-gaps for improving the photocatalytic performances. Among them, Bi<sub>2</sub>MoO<sub>6</sub> has recently become a popular target owing to its typical  $E_{\rm g}$  (2.6 eV) and excellent utilization of visible-light absorption. More specifically, as an important aurivillius oxide possessing the perovskite-like layered

Furthermore, these demonstrations clearly reveal that the

controlled fabrication of hierarchical inter-crossed frameworks

could serve as an alternative strategy for further improving the

photocatalytic performance.

structure, Bi<sub>2</sub>MoO<sub>6</sub> plate-like structures could be more easily

obtained, which could effectively enhance the photocatalytic properties as a result of the increased active sites.<sup>8</sup> More recently,

the Bi<sub>2</sub>MoO<sub>6</sub> nanosheet-built hierarchical structures, including

nanofibers, 9,10 hollow spheres, 11,12 and flower-like structures, 13

have been extensively reported, and their photocatalytic behaviours

could be further improved. However, note that the previous work

Fig. 1A shows the typical transmission electron microscope (TEM) image of single-crystalline  $MoO_3$  nanobelts synthesized through a modified hydrothermal method, <sup>14</sup> which would serve as the growth templates for the subsequent fabrication of 1D  $Bi_2MoO_6$  nanosheet-built frameworks. It can be clearly seen from both TEM and SEM images (shown in Fig. S1 and S5A, ESI†) that the as-prepared  $MoO_3$  nanobelts possess rectangular cross-sections with an average width of  $\sim$  120 nm and a length of about tens of micrometers. More specifically, it is confirmed that the  $MoO_3$  nanobelts grow along the [001] direction and two sets of crystal lattice fringes correspond to the {100} and {001} atomic spacings. Interestingly, when these  $MoO_3$  nanobelts reacted with  $Bi(NO_3)_3$  in aqueous solutions at 120 °C, the hydrolysis

mostly focused on building hierarchical structures with densely packed Bi<sub>2</sub>MoO<sub>6</sub> nanosheets. Up to now, the rational design, fabrication, and photocatalytic study of hierarchical architectures consisting of inter-crossed Bi<sub>2</sub>MoO<sub>6</sub> nanosheets have been rarely reported.

Herein, we demonstrate for the first time the controlled fabrication of 1D Bi<sub>2</sub>MoO<sub>6</sub> nanosheet-built framework (HBNF) structures with uniform nanochannels by simple hydration treatment of MoO<sub>3</sub> nanobelts, in which, MoO<sub>3</sub> nanobelts serve as both growth templates and molybdate source. Furthermore, it has been found that Bi<sub>2</sub>MoO<sub>6</sub> nanosheet-built frameworks exhibit much higher photocatalytic performance than conventional Bi<sub>2</sub>MoO<sub>6</sub> nanoplates and nanoparticles for the degradation of organic contaminants under visible light irradiation.

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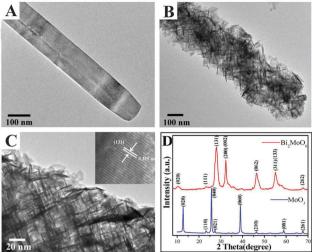


Fig. 1 The TEM images of (A) MoO<sub>3</sub> nanobelts, (B) and (C) Bi<sub>2</sub>MoO<sub>6</sub> nanosheet-built frameworks (HBNF) (inset: the HRTEM image of HBNF), (D) XRD patterns of the precursor and the product.

reaction process of MoO<sub>3</sub> into MoO<sub>6</sub><sup>2-</sup> ions occurred immediately, which would react with Bi3+ ions to form Bi2MoO6 hierarchical architectures (Fig. S3, ESI†). The TEM image of Bi<sub>2</sub>MoO<sub>6</sub> nanoproducts is shown in Fig. 1B and Fig. S2 (ESI†), clearly indicating that a novel framework structure with an average diameter of 200 nm has been obtained by this simple reflux process. More specifically, the enlarged TEM image (Fig. 1C) clearly reveals that this framework structure is constructed by numerous inter-crossed ultra-thin nanosheets with an average thickness of 6 nm (inset of Fig. S2B, ESI†). The corresponding high resolution TEM (HRTEM) image of the nanosheet building block (inset of Fig. 1C) demonstrates that the planar spacing is measured to be 0.315 nm, corresponding to the (131) crystal planes of orthorhombic Bi<sub>2</sub>MoO<sub>6</sub>, indicating that these nanosheets grow vertically (131) facet and aggregate into a framework structure.11

Furthermore, the X-ray diffraction (XRD) patterns (Fig. 1D) clearly reveal that the diffraction peaks of MoO3 nanobelts can be indexed to the orthorhombic structure of α-MoO<sub>3</sub> (JCPDS No. 05-0508). However, after the heat treatment, all the diffraction peaks of nanosheet-built frameworks could be indexed to the orthorhombic structure of  $\gamma$ -Bi<sub>2</sub>MoO<sub>6</sub> (JCPDS. No. 21-0102), and no any diffraction peak of MoO3 has been observed. Moreover, X-ray photoelectron spectroscopy (XPS) has also been performed to investigate the surface composition as well as the valence state of Bi<sub>2</sub>MoO<sub>6</sub> nanosheet-built frameworks. As shown in Fig. S4 (ESI†), all the peaks correspond to Bi, Mo, O and C elements (in which the C element was used for calibration) can be clearly detected and their valence states are consistent with the corresponding standardized binding energies. The above demonstrations clearly reveal that the MoO<sub>3</sub> nanobelts could be successfully and completely transformed into 1D Bi<sub>2</sub>MoO<sub>6</sub> nanosheet-built frameworks through this simple template growth process.

Furthermore, the morphological evolution of intermediate products at different reaction times has been studied in detail

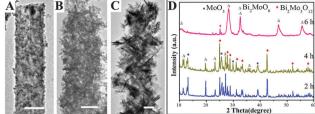
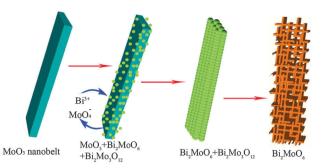


Fig. 2 TEM images of samples reacted for (A) 2 h, (B) 4 h, (C) 6 h and (D) the corresponding XRD pattern (the scale bar in the TEM images represent 100 nm)

to address the possible growth process of Bi<sub>2</sub>MoO<sub>6</sub> frameworks (Fig. 2, Fig. S5, ESI†). The TEM image shown in Fig. 2A clearly reveals that when the MoO3 nanobelt precursors reacted with Bi(NO<sub>3</sub>)<sub>3</sub> aqueous solution for 2 h, the surface of MoO<sub>3</sub> nanobelts have been completely covered by small nanoparticles, and the colour of reaction solution transformed gradually from white to yellow. However, further increasing the reaction time up to 4 h, the regular rectangular nanobelts have been completely transformed into a 1D disordered nanostructure consisting of numerous small particles. At 6 h, these nanoparticles have grown into nanosheets and a novel tree-like nanostructure has been formed (shown in Fig. 2C). Finally, Bi<sub>2</sub>MoO<sub>6</sub> nanosheet-built frameworks with a uniform and perfect structure evolved as a result of the Ostwald ripening process (Fig. 1B). Furthermore, the evolution of the composition and crystalline structure of these intermediate products during the growth process have been clarified (Fig. 2D). It is obvious that except for MoO3, the diffraction peaks of both Bi<sub>2</sub>MoO<sub>6</sub> and Bi<sub>2</sub>Mo<sub>3</sub>O<sub>12</sub> could be observed obviously at 2 h, indicating that the formed hetero-particles on MoO3 nanobelts could be attributed to the composite of Bi<sub>2</sub>MoO<sub>6</sub> and Bi<sub>2</sub>Mo<sub>3</sub>O<sub>12</sub>. When the hydrothermal reaction was carried out for 4 h, the diffraction peaks of MoO<sub>3</sub> were gradually reduced, and the peaks of Bi<sub>2</sub>MoO<sub>6</sub> and Bi<sub>2</sub>Mo<sub>3</sub>O<sub>12</sub> were significantly enhanced. At 6 h, the diffraction peaks of MoO3 have completely disappeared, and the main peaks could be indexed to Bi<sub>2</sub>MoO<sub>6</sub>. Moreover, some small diffraction peaks corresponding to Bi<sub>2</sub>Mo<sub>3</sub>O<sub>12</sub> could also be detected. Finally, the novel nanosheet-built framework structure consisting of pure orthorhombic Bi<sub>2</sub>MoO<sub>6</sub> crystals has been obtained at 8 h.

Based on these results, we proposed a possible growth process for the transformation of MoO<sub>3</sub> nanobelts into HBNF under the present synthetic conditions, which is illustrated in Scheme 1. It is well established that the MoO<sub>3</sub> crystals in water can be gradually hydrated to form H<sub>2</sub>MoO<sub>4</sub> (eqn (1)). Thereby, the formed H<sub>2</sub>MoO<sub>4</sub> molecules on the surface of MoO<sub>3</sub> nanobelts could directly react with Bi3+ anions to form Bi2MoO6 (eqn (2)) and Bi<sub>2</sub>Mo<sub>3</sub>O<sub>12</sub> (eqn (3)) nanoparticles via an ionexchange process, respectively. Upon increasing the reaction time, the MoO<sub>3</sub> nanobelts have been completely hydrated and transformed into Bi<sub>2</sub>MoO<sub>6</sub> and Bi<sub>2</sub>Mo<sub>3</sub>O<sub>12</sub> nanoparticles with an incompact 1D structure. Under the heating conditions, Bi2MO3O12 gradually converted into Bi<sub>2</sub>MoO<sub>6</sub> nanosheets (eqn (4)). <sup>15,16</sup> Moreover, during the transformation process, the anisotropic growth of

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Scheme 1 Schematic representation of the synthesis of hierarchical  $Bi_2MoO_6$  nanosheet-built frameworks through the anion-exchange reaction of the  $MoO_3$  nanobelt with  $Bi^{3+}$  anions under controlled conditions.

 $\rm Bi_2MoO_6$  nanosheets make them interconnected with each other. Finally, when  $\rm Bi_2Mo_3O_{12}$  has completely transformed into  $\rm Bi_2MoO_6$ , a kinetically stable  $\rm Bi_2MoO_6$  nanosheet-built framework has been fabricated. However, the exact growth mechanism of  $\rm Bi_2MoO_6$  frameworks cannot be completely understood and a more detailed study is still underway.

$$MoO_3 + H_2O \rightarrow H_2MoO_4$$
 (1)

$$H_2MoO_4 + 2Bi(NO_3)_3 \cdot 5H_2O \rightarrow Bi_2MoO_6 + 6HNO_3 + 8H_2O$$
 (2)

$$3H_2MoO_4 + 2Bi(NO_3)_3 \cdot 5H_2O \rightarrow Bi_2Mo_3O_{12} + 6HNO_3 + 10H_2O$$
(3)

$$Bi_2Mo_3O_{12} \rightarrow Bi_2MoO_6 + 2MoO_3$$
 (4)

To further study the surface area and nanochannels in the as-fabricated HBNF, the Brunauer–Emmett–Teller (BET) nitrogen adsorption isotherm experiments have been performed and the results are shown in Fig. 3A. Moreover, the related experiments on the BMNS have also been measured for comparison (Fig. 3B). It can be clearly observed that the nitrogen adsorption–desorption isotherms on HBNF and BMNS could be both identified as type IV, revealing the existence of abundant mesoporous structures in these two samples. <sup>17</sup> More specifically, the as-prepared HBNF possess a larger adsorption volume and a specific surface area (44.64 m $^2$  g $^{-1}$ ) than that of the BMNS sample (8.40 m $^2$  g $^{-1}$ ). Furthermore, the distribution of nanochannels and pores in these two samples have also been studied and shown in the inset of Fig. 3. It can be clearly seen

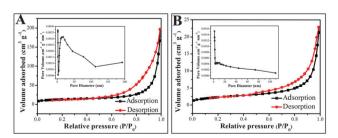


Fig. 3 Nitrogen adsorption—desorption isotherms measured at 76 K from the obtained HBNF (A) and BMNS (B). Insets: the pore size distribution of the corresponding crystals.

from Fig. 3A that the main pore size distribution of HBNF is generally in the range of 15–20 nm, which may be generated as a result of the inter-crossed nanosheets. However, in the case of BMNS, no primary pore size distribution has been observed due to their pure sheet-like structure (the inset image of Fig. 3B). These demonstrations clearly reveal that the rational assembly of  $\rm Bi_2MoO_6$  nanosheets into inter-crossed frameworks could not only greatly increase the specific surface area, but also form the nanopores as well as nanochannels.

Their photocatalytic behaviours were explored for the degradation of Rhodamine B (RhB) dyes under visible-light irradiation (Fig. S6, ESI†). To the best of our knowledge, this novel Bi<sub>2</sub>MoO<sub>6</sub> nanosheet-built framework structure was used, for the first time, as the catalyst for these photocatalytic reactions. For a comparison, the photocatalytic performances of Bi<sub>2</sub>MoO<sub>6</sub> nanosheets, nanoparticles, and Degussa-P25 have also been studied and compared in Fig. 4A, Fig. S7, and S8A (ESI†). It can be clearly seen that except for Degussa-P25, 18 all the Bi<sub>2</sub>MoO<sub>6</sub> photocatalysts exhibit excellent photocatalytic activities for the RhB degradation. Among them, the Bi<sub>2</sub>MoO<sub>6</sub> nanosheet-built framework structure exhibits the highest photocatalytic activity, which can completely degrade the RhB dye in only 30 min. Furthermore, the degradation efficiency of the RhB dye over Bi<sub>2</sub>MoO<sub>6</sub> nanosheets is about 80%, while only 35% of the RhB dye can be degraded over the Bi<sub>2</sub>MoO<sub>6</sub> nanoparticles. Moreover, the photocatalytic behaviours for the degradation of colorless phenol under visible-light irradiation have also been explored. It can be clearly seen from Fig. 4B and Fig. S8B (ESI†) that Bi<sub>2</sub>MoO<sub>6</sub> nanosheet-built frameworks still exhibit the highest photocatalytic activity for phenol degradation, which is consistent with the experimental results of RhB degradation. Thereby, it can be concluded that the enhanced performance of Bi<sub>2</sub>MoO<sub>6</sub> frameworks could be ascribed to their specific hierarchical nanosheet-built structure, large surface area and nanochannels, which could provide more active sites and thus facilitate diffusion of reactants and products during the photocatalytic reaction. Fig. S9 (ESI†) shows the transient photocurrent responses of the samples under intermittent 300 W Xe lamp irradiation. It can be seen that the Bi<sub>2</sub>MoO<sub>6</sub> nanosheet-built framework exhibits a much higher photocurrent intensity than Bi<sub>2</sub>MoO<sub>6</sub> nanosheets and nanoparticles, indicating the more efficient photoinduced charge separation and transfer in the hierarchical Bi<sub>2</sub>MoO<sub>6</sub> nanosheet-built framework, which is in good agreement with the above mentioned photocatalytic activities.

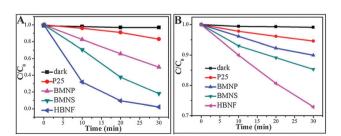
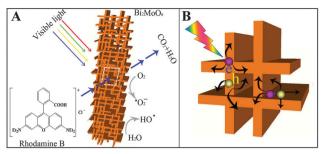


Fig. 4 The degradation curves of RhB (A) and phenol (B) over different photocatalysts under visible-light irradiation.

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Scheme 2 Schematic illustration (A and B) of photodegradation of RhB by HBNF.

Based on the above results, it can be confirmed that the photocatalytic and photoelectric performances of  $\mathrm{Bi}_2\mathrm{MoO}_6$  semiconductors can be further optimized by the construction of inter-crossed nanosheet-built framework structures.

Based on the above results, we identified some possible reasons to clarify the higher photocatalytic activity of HBNF structure than both Bi<sub>2</sub>MoO<sub>6</sub> nanosheets and nanoparticles. As shown in Scheme 2A, the frameworks could effectively enhance the visible light absorption as a result of their unique inter-crossed structures and a large specific surface area compared with nanosheets and nanoparticles. Moreover, the nanochannels in the range of 15-20 nm could facilitate the rapid transfer of organic molecules from both the exterior and interior of the framework, which can greatly increase the active sites. 19,20 More importantly, as shown in Scheme 2B, the nanosheet-built framework could facilitate the hole and electron rapid transfer from different orientations under visible light irradiation as a result of its unique inter-crossed nanosheets, which could greatly promote the effective separation of photoexcited electron-hole pairs and decrease the probability of electronhole recombination confirmed by the photoelectric results.

In summary, we have developed a facile and rapid process for fabricating 1D Bi<sub>2</sub>MoO<sub>6</sub> nanosheet-built framework structures by simply reacting MoO<sub>3</sub> nanobelts with Bi(NO<sub>3</sub>)<sub>3</sub> aqueous solution using a reflux process. Moreover, these Bi<sub>2</sub>MoO<sub>6</sub>

nanosheet-built frameworks exhibit a much higher visible-light-driven photocatalytic activity and photoelectric property than  $\rm Bi_2MoO_6$  nanosheets and nanoparticles. More importantly, the rational fabrication of the hierarchical nanosheet-built framework may be an effective technique for the development of highly efficient visible-light sensitive photocatalysts. The extension of this simple self-template strategy for the transformation of  $\rm MoO_3$  nanobelts to the synthesis of other hierarchical framework structures is underway in our research group.

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