

From branched self-assemblies to branched mesoporous silica nanoribbons†

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Branched left-handed twisted mesoporous silica nanoribbons were prepared *via* a template method.

Recently, branched, hyperbranched, Y- and star-shaped nanoparticles have attracted many researchers, because they can be potentially applied in the fields of electronics and photonics. However, only a few methods have been used to control the branched nanostructures. For example, branched ϵ -MnO₂ nanostructures were synthesized *via* a hydrothermal method.¹ Hyperbranched PbSe nanowire networks were synthesized *via* a vapor–liquid–solid mechanism.² Star-shaped Bi₂S₃ was formed by the splitting crystal growth mechanism.³ Star-shaped gold nanoparticles were synthesized using a surfactant direct method.⁴ Boron nanowire Y-junctions were synthesized in a self-assembled manner by fusing two individual boron nanowires grown inclined toward each other.⁵ Multibranched carbon nanotubes were synthesized using flow fluctuation.⁶ Y-junction carbon nanotubes were grown by a chemical vapor deposition method using nanochannel alumina as a template.⁷

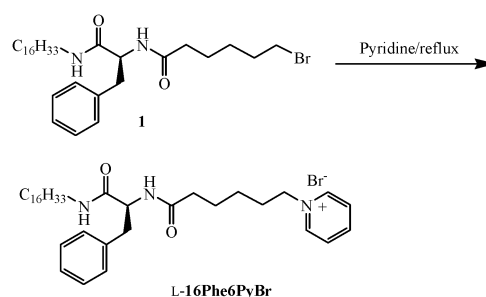
Low-molecular-weight amphiphiles show a variety of morphologies. Recent results indicated that sol–gel transcription is a powerful method to control inorganic, organic–inorganic hybrid, and organic nanostructures using the self-assemblies of these amphiphiles as templates.^{8–13} The morphologies of organic templates can control those of inorganic nanomaterials. Up to now, nanofibers, nanotubes, and mesoporous nanofibers have been prepared. For the preparation of branched nano-silicas through a sol–gel transcription process, branched organic self-assemblies should be selected as templates, at least, branched organic self-assemblies as intermediates can be formed during the sol–gel transcription process. Here we successfully obtained branched mesoporous silica nanostructures through a dynamic templating process. The branched mesoporous silicas have potential to be used as templates to prepare other nanomaterials.

Amphiphile L-16Phe6PyBr which is composed of L-phenylalanine was synthesized according to Scheme 1 (ESI †).¹² It can gel THF, benzene, nitrobenzene, 1,4-dioxane, and chlorobenzene. When 10 mg of L-16Phe6PyBr were dissolved

in 1.0 mL of pure water, a white suspension was obtained. An optical micrograph shows many short ribbons and fibers of μ m-size floating in water (Fig. 1a). It is interesting to find that huge branched organic self-assemblies were identified when 10 mg of L-16Phe6PyBr were dissolved in 1.0 mL of 1-propanol. According to our previous results,^{11,12} branched silica nanostructures are potentially obtained *via* the sol–gel transcription method.

Because it is hard to transfer the morphologies of the self-assemblies of L-16Phe6PyBr in pure 1-propanol, the sol–gel transcriptions were carried out in mixtures of water and 1-propanol. A typical synthetic procedure is as follows: compound L-16Phe6PyBr (10 mg, 0.016 mmol) was dissolved in a mixture of 0.2 mL of 1-propanol and 0.8 mL of NH₃ aq. (5.0 wt%), then 20 mg (0.096 mmol) of tetraethoxyl orthosilicate (TEOS) were dropped into the suspension under strong stirring at 0 °C. After it turned white at 90 s, stirring was stopped. The mixture was kept at 0 °C for 1 day and 80 °C for 4 days under static conditions. Finally, the template was removed by washing with a mixture of HCl aq. and methanol, and calcination was performed in a box furnace in air at 550 °C for 5 h, with a ramp rate of 1 °C min^{−1}.

To visually study the morphologies of the obtained silica, we used field emission scanning electron microscopy (FESEM).‡ Before taking FESEM images, 10 nm Pt–Pd was covered on the surface of samples. Although the obtained silica combined some nano-sheets, branched silica nanoribbons were identified (Fig. 2 and Supporting information, Fig. S1†). All of these nanoribbons are left-handed twisted with a uniform helical pitch of around 3.0 μ m. Generally, these nanoribbons are less than 10 μ m in length. Lamellar pore architecture was identified from transmission electron microscope (TEM) images (Fig. 3 and Supporting information, Fig. S2†). This nanostructure is different from the twisted ribbons constructed by many tubes in a single layer.¹¹ It is very interesting to find that the



Scheme 1 Synthesis of compound L-16Phe6PyBr.

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† Electronic supplementary information (ESI) available: Synthesis of compound L-16Phe6PyBr; FESEM and TEM images; SAXRD pattern. See DOI: 10.1039/b812016a

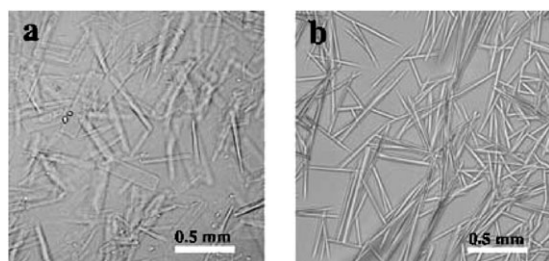


Fig. 1 Optical micrographs of L-16Phe6PyBr in pure water (a, 10 mg of L-16Phe6PyBr in 1.0 mL of water) and in 1-propanol (b, 10 mg of L-16Phe6PyBr in 1.0 mL of 1-propanol).

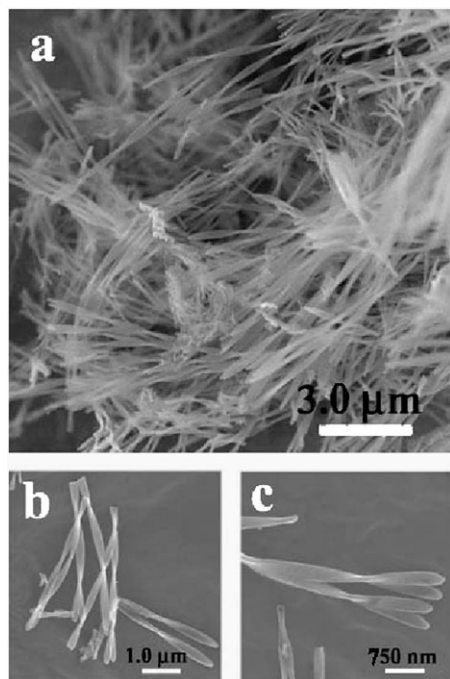


Fig. 2 FESEM (a, b, and c) images of branched mesoporous silica nanoribbons.

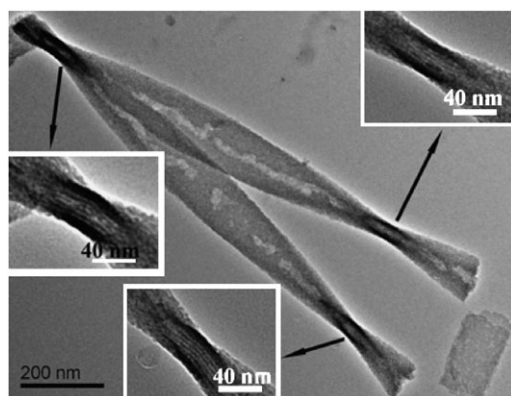


Fig. 3 TEM image of branched mesoporous silica nanoribbons after calcination.

thickness of the trunk and the branch of some ribbons is nearly the same (Fig. 2b and 3). The mesoporous nanoribbons exhibited a nitrogen Brunauer–Emmett–Teller (BET) surface area of 306 m² g^{−1} and the Barrett–Joyner–Halenda (BJH)

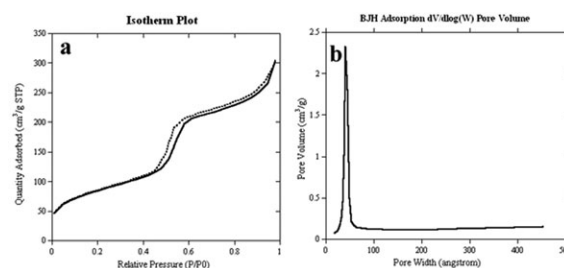


Fig. 4 N₂ adsorption–desorption measurements of left-handedly twisted mesoporous silica nanoribbons. Sorption isotherms (a) and BJH pore-size distribution plot (b) from the adsorption branch.

pore-size distribution plot determined from the adsorption branch shows one sharp peak at 4.0 nm (Fig. 4). This lamellar pore architecture shows high thermostability, even this silica was calcined at 550 °C for 5 h. The small-angle X-ray diffraction (SAXRD) pattern shows no sharp peaks (Supporting information, Fig. S3†). Several kinds of helical and coiled mesoporous silica nanostructures have been prepared recently.^{11,12} Only broad peaks were identified in SAXRD patterns. For the lamellar mesoporous silica nanoribbons shown here, possibly due to the twisted morphology and the ribbons formed only by several silica layers, no sharp peaks were identified in the SAXRD pattern.

The mechanism of the formation of the branched mesoporous ribbons was studied using TEM at different reaction times (Fig. 5).¹⁴ Before dropping TEOS, both ribbons and helical bundles were identified (Fig. 5a). After dropping TEOS, dendrimer-like organic self-assemblies were identified at 30 s

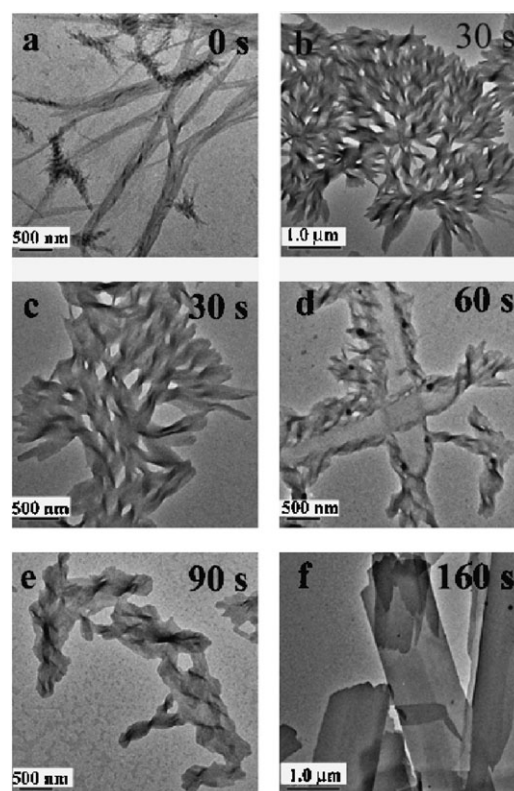


Fig. 5 TEM images of the reaction mixture at different reaction times (a, 0 s; b, 30 s; c, 30 s; d, 60 s; e, 90 s, and f, 160 s).

(Fig. 5b). Each twisted nanoribbon is separated into two twisted nanoribbons (Fig. 5c). Branched nanoribbons were still identified at 90 s, just after stopping stirring (Fig. 5e). These branched nanoribbons organized into non-coiled nanoribbons with increasing stirring time (Fig. 5d and e). The non-coiled nanoribbons started to form at 60 s with self-assembling of twisted nanoribbons (Fig. 5d). Then non-coiled nanoribbons and branched nanoribbons were obtained finally (Fig. 5e and Supporting information, Fig. S4†). Compared with Fig. 5a, c, and d, it was found that the helical pitch of the morphologies increased with increasing reaction time. The branched nanoribbon (Fig. 5b) was the intermediate between the helical bundle (Fig. 5a) and non-coiled nanoribbon (Fig. 5f). Because of the existence of 1-propanol in the reaction mixture, ethanol resulting from the hydrolysis of TEOS should not be the main factor to change the morphologies of the organic self-assemblies from the nanoribbons and helical bundles to the dendrimer-like structure. Thus, the adsorption of silicate anions and silica oligomers should be the main factors controlling this morphology change. After the polycondensation of the silica oligomers on the surface of the branched organic self-assemblies, branched silica nanoribbons were formed. Branched mesoporous silica nanoribbons were obtained by removing the templates. In all, the formation of branched mesoporous silica nanoribbons follows the dynamic template process.¹⁵ The morphologies of organic self-assemblies change during the sol-gel transcription process.

Circular dichroism (CD) and ultra-violet (UV) absorption spectra were taken to study the packing of L-16Phe6PyBr molecules (Fig. 6). For the UV absorption, two bands are identified from 240 to 280 nm. The CD sign at 229 nm originates from the carboxy groups. π - π interactions among aromatic rings play an important role in the formation of organic self-assemblies.¹⁶ Two induced CD signs are identified at 258 and 273 nm in the CD spectrum, which originate from the π - π interactions among the phenyl and pyridinium rings, respectively. It seems that the phenyl groups are packing in a right-handed manner. However, the pyridinium rings are packing in a left-handed manner. The interactions of the hydrophobic association of the hydrocarbon chains, hydrogen bonding, π - π interactions among the aromatic rings, and the adsorption of silicate anions and silica oligomers induce the

formation of non-coiled nanoribbons. Compared with the other gelators reported previously,^{11,12,14} the π - π interaction among the phenyl groups of L-16Phe6PyBr is the main factor forming the non-coiled nanoribbons.

In summary, branched mesoporous silica nanoribbons were fabricated through a dynamic templating process. These branched nanoribbons have a left-handed twist and show high thermostability. They are templated by the intermediates of the organic self-assemblies during the sol-gel transcription process. Based on the mechanism shown here, many other branched nanoparticles have the potential to be prepared, which could be used in electronics and photonics.

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Notes and references

† TEM images were obtained using a TecnaiG220. FESEM images were taken on a Hitachi S-5000. Specific surface area and pore-size distribution were determined by the BET and BJH methods using a N_2 adsorption isotherm measured by a Gemini V 2380 instrument. The SAXRD pattern was taken on an X'Pert-Pro MPD X-ray diffractometer. CD and UV spectra were taken on an AVIV 410 spectrometer.

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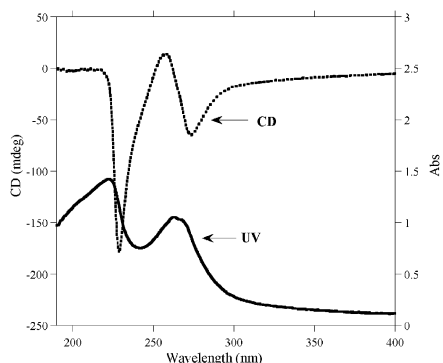


Fig. 6 CD and UV absorption spectra of L-16Phe6PyBr viscous aqueous solution (20 mg of L-16Phe6PyBr in 1.0 mL of water).