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Two-Dimensional Perovskite Chiral Ferromagnets

Bing Sun,[†] Xiao-Fei Liu,[†] Xiang-Yang Li,[†] Yamin Zhang,[†] Xiangfeng Shao,^{*,†} Dezheng Yang,^{*,§} and Hao-Li Zhang^{*,†,‡}

[†]State Key Laboratory of Applied Organic Chemistry (SKLAOC), Key Laboratory of Special Function Materials and Structure Design (MOE), College of Chemistry and Chemical Engineering, Lanzhou University, Lanzhou 730000, P. R. China

[§]Key Laboratory for Magnetism and Magnetic Materials of Ministry of Education, School of Physical Science and Technology, Lanzhou University, Lanzhou 730000, P. R. China

[‡]Tianjin Key Laboratory of Molecular Optoelectronic Sciences, Collaborative Innovation Center of Chemical Science and Engineering, Tianjin University, Tianjin 300072, P. R. China

ABSTRACT: Magnetic molecular materials with a chiral configuration are attractive candidates for sensing, information storage and spintronics. Herein, we report the first example of two-dimensional (2D) hybrid perovskite chiral ferromagnets, (*R*-MPEA)₂CuCl₄ and (*S*-MPEA)₂CuCl₄. These two compounds exhibit strong oppositely signed circular dichroism (CD) signals and clear ferromagnetic behaviors. Magnetic measurements revealed high saturation magnetization up to 12.5 emu g⁻¹. The coexistence of strong chirality and ferromagnetism enabled successful study on their magneto-chiral dichroism (MChD) spectra. These findings demonstrate a new materials platform for future magneto-optical and spintronic applications, providing insights to the structure-property correlation of chiral ferromagnetic perovskite.

Great efforts have been devoted to the material systems whose time-reversal and space-inversion symmetries are simultaneously broken, i.e. in a medium with magnetism and chirality.¹ Magnetism and chirality are directly connected through an antisymmetric exchange of Dzyaloshinskii-Moriya interaction (DMI). In a chiral ferromagnet, chirality breaks the space inversion and mirror symmetry,² which contributes to new types of interesting cross-effects, such as chiral magnetic effect (CME),³ chiral-induced spin selectivity⁴ (CISS) and magneto-chiral dichroism (MChD).⁵ Materials with both magnetism and chirality properties can be central to fundamental advances in multiferroics, current-induced spin-orbit torques and topological magnetic structures.⁶

Chiral ferromagnets may play a crucial role in spintronic devices, as a spin-polarized current flowing through chiral magnetic structures will cause a variety of excitations or manipulations of the magnetization.⁷ The interplay between crystallographic chirality and magnetism also enables physical insight into fundamental understanding of anomalous Hall effect (AHE), magnetizationinduced second-harmonic generation (MSHG) and MChD, and can give rise to new quantum particles, such as skyrmions.⁸ Chiral ferromagnetic systems possess DMI are of fundamental interest in the field of molecular magnetism,⁹ and is becoming more important for new applications in spintronics.¹⁰ However, the examples of chiral ferromagnets are still very limited.¹¹⁻¹⁵

Recently, organic-inorganic hybrid perovskites have been explored as a platform for developing new multifunctional materials.¹⁶⁻²⁰ Metal-halide octahedra form layers separated by the organic molecules, these materials are commonly referred to as two-dimensional (2D) hybrid perovskites.²¹ 2D hybrid perovskites have been exploited in a wide range of applications, owing to their multiplequantum-well structures, relatively high stability and impressive compositional tunability. Very few ferromagnetic perovskites have been prepared so far, which showed some interesting properties, such as spin-reorientation²² and thermochromic ferromagnets.²³ The unique structural tunability of hybrid perovskites offers an unprecedented opportunity to directly incorporate chiral organic molecules to enable perovskite's chirality.24-28 It has been demonstrated that spin-polarized photoluminescence occurred in chiral perovskites without an external magnetic field.²⁹ Spin transport in perovskite can be effectively manipulated upon the chirality of the (R/S/rac-MBA)₂PbI₄ via the CISS mechanism.⁴ Moreover, large spin-orbit coupling (SOC),³⁰ controllable Rashba split-ting,^{31,32} large Stark effect^{33,34} and optical spin selection³⁵ have been discovered in hybrid perovskites, highlighting their potential in spin-based applications. Theoretical investigations have predicted more attractive applications of chiral perovskite in quantum informatics and spintronics.36,37

Despite the extensive investigations, perovskite chiral ferromagnet remains blank to date. Herein, for the first time, we synthesized a pair of enantiomeric $_{2}D$ hybrid perovskites, (*R*-MPEA)_2CuCl₄ and (*S*-MPEA)_2CuCl₄. Circular dichroism (CD) and magnetic measurements confirm that these two perovskites possess both chirality and ferromagnetism. MChD investigation on chiral perovskites was carried out for the first time, which revealed potential of these materials for future optomagnetic and spintronic applications.



Figure 1. Optical microscopic images of $(R-MPEA)_2CuCl_4$ (a) and $(S-PEA)_2CuCl_4$ (b). Single crystal structures viewed along the c-axis $(R-MPEA)_2CuCl_4$ (c) and $(S-MPEA)_2CuCl_4$ (d) with color codes Cu (Indigo), Cl (Red), C (gray), N (blue). The CuCl₆-units are displayed as polyhedra, and H atoms are omitted for clarity. Experimental and calculated powder Xray diffraction (PXRD) patterns of (e) $(R-MPEA)_2CuCl_4$, (f) $(S-MPEA)_2CuCl_4$.

Direct combination of the chiral cations and CuCl₂•2H₂O mixing in ultrapure water at 80 °C followed by slow cooling of the solutions to room temperature yields yellow flake crystals, as seen in Figure 1a and 1b. Detailed synthesis procedures are listed in the experimental section. The crystal structure and refinement data are provided in Figure 1c, 1d and Table S1. Powder X-ray diffraction (PXRD) patterns of (*R*-MPEA)₂CuCl₄ and (*S*-MPEA)₂CuCl₄ all exhibit sharp peaks and show almost identical features as simulated spectra based on their single crystal structures, confirming the phase purity of the as-synthesized crystals (Figure 1e and 1f). Thermogravimetric analysis (TGA) measurements indicate that (*R*-MPEA)₂CuCl₄ and (*S*-MPEA)₂CuCl₄ possesses high thermal stability, with identical decomposition temperature

above 450 K, while $(rac-MPEA)_2CuCl_4$ is thermally stable up to 440 K (Figure S1).

Figure 1c and 1d illustrate the single-crystal structures of the (*R*-MPEA)₂CuCl₄ and (*S*-MPEA)₂CuCl₄. Both crystals have 2D layered structures, in which successive corner-sharing inorganic [CuCl₆]⁴⁻ octahedral layer is intercalated by two layers of chiral organic cations. In such a way, the multiple-quantum-wells structure is naturally formed with the chiral molecules embedded in.³⁸ Compounds (*R*-MPEA)₂CuCl₄ and (*S*-MPEA)₂CuCl₄ are antisymmetrically isostructural and crystallize in the chiral space group of *C*₂ assigned to the monoclinic crystal system. The lattice parameters of (*R*-MPEA)₂CuCl₄ and (*S*-MPEA)₂CuCl₄ are *a* = 35.470 Å, *b* = 5.4017 Å and *c* = 5.3432 Å; *a* = 35.450 Å, *b* = 5.4065 Å and *c* = 5.3466 Å, respectively.

It is known that the Cu²⁺ ions in the perovskite structure are strongly Jahn-Teller (JT) active, giving rise to changes in the Cu-Cl bond lengths and hence to structural distortions of the CuCl₆ octahedra.³⁹ The distortion levels (Δd) of the individual [CuCl₆]⁴⁻ octahedron is calculated using Equation (1):

$$\Delta d = \frac{1}{6} \sum \left[\frac{d_n - d}{d} \right]^2 \tag{1}$$

where *d* is the mean Cu-Cl bond length (Table S₂ and Table S₃) and *d*_n are the six individual Cu-Cl bond length.⁴⁰ The distortions of CuCl₆ octahedral in the perovskite structure are listed in Table S₄. The values of Δd is calculated to be 3.32×10^{-4} and 3.07×10^{-4} for the (*R*-MPEA)₂CuCl₄ and (*S*-MPEA)₂CuCl₄, respectively, indicating that there is no substantial difference in structural distortion.



Figure 2. Absorbance and CD spectra of the chiral perovskites. UV-vis-NIR absorption spectra (a) and CD spectra (b) of (*R*-MPEA)₂CuCl₄, (*S*-MPEA)₂CuCl₄ and (*rac*-MPEA)₂CuCl₄.

UV-vis-NIR absorption spectra of $(R-MPEA)_2CuCl_4$, $(S-MPEA)_2CuCl_4$ and racemic compound are shown in Figure 2a, which are nearly identical. The absorption peak at ~379 nm near the absorption edge corresponds to 2D perovskite excitonic feature, which is characteristic for 2D perovskites. The 2D hybrid perovskite structures are natural quantum wells with semiconducting slabs alternating with dielectric slabs.⁴¹ Therefore, this absorption feature is derived from quantum and dielectric confinement effects of the carriers, previously shown in other 2D semiconductors such as the $(CH_3CH_2NH_3)_4Pb_3Br_{10-x}Cl_x$ series and $(PA)_2(MA)_{n-1}Pb_nI_{3n+1}$.^{42.43} The high energy absorption band observed at ~274 nm is associated with $\pi \rightarrow \pi *$ transition of the organic cations.⁴⁴ The absorption edges of the chi-

59 60 ral perovskites locate around 567 nm. A broad band appears in the near-infrared spectrum ranging from 600 to 1200 nm, indicating the appearance of broad gap states.⁴⁵ This weak, broad band centered at approximately 800 nm in the absorption spectra results from Laporte forbidden Cu²⁺ *d-d* transitions.^{46,47} Photoluminescence spectra of (*R*-MPEA)₂CuCl₄, (*S*-MPEA)₂CuCl₄ and (*rac*-MPEA)₂CuCl₄ are shown in Figure S2. A broad emission band peaking at 555 and 617 nm was observed for these three compounds, indicating that nearly identical photoluminescence properties.

Figure 2b reveals strong CD signals appear at the same wavelengths (at 324, 372 and 436 nm) for the (R-MPEA)₂CuCl₄ and (S-MPEA)₂CuCl₄ films but with opposite signs, while (rac-MPEA)₂CuCl₄ shows a featureless flat signal. The CD spectra indicate strong chirality of the two new 2D perovskites. As shown in Figure S₃, opposite CD absorption features are observed in *R*-MPEA·HCl and S-MPEA•HCl. The racemic blend of rac-MPEA•HCl exhibits no absorption difference between left- and righthanded polarized light. The CD signals for these chiral organic ammonium salts mainly located before 300 nm. However, when they are incorporated into the chiral perovskite, the resulting perovskites exhibit strongly opposite CD values in the visible region (Figure 2b). These observations indicate that the chirality of organic cations directly transferred to the perovskite framework, thereby endowing the corresponding 2D perovskites with intrinsic chirality.⁴⁸ Notably, the CD peaks were located before the absorption edge at 567 nm, suggesting a lifting of the spin degeneracy within the band edge electronic states induced by the chiral molecules.4



Figure 3. Magnetic properties of the (*R*-, *S*-, *rac*-MPEA)₂CuCl₄. (a) Temperature dependence of the magnetizations of (*R*-MPEA)₂CuCl₄, (*S*-MPEA)₂CuCl₄ and (*rac*-MPEA)₂CuCl₄ in applied fields of 100 Oe. (b, c, d) The hysteresis loops of (*R*-MPEA)₂CuCl₄, (*S*-MPEA)₂CuCl₄ and (*rac*-MPEA)₂CuCl₄ at 2 K, 5 K, 10 K and 15 K.

Temperature-dependent magnetic susceptibility in a 100 Oe magnetic field for the $(R-MPEA)_2CuCl_4$, (*S*-MPEA)_2CuCl₄ and $(rac-MPEA)_2CuCl_4$ show clearly the

transition from ferromagnet to paramagnet (Figure 3a). A long-range ferromagnetic ordering below 6 K was observed for (R-MPEA)₂CuCl₄, (S-MPEA)₂CuCl₄ and (rac-MPEA)₂CuCl₄. The Curie temperatures of these 2D perovskites are similar to those observed on (BED)₂CuCl₆²³ and CuCl₄(C₆H₅CH₂CH₂NH₃)₂.⁴⁹ The hysteresis loops confirm the presence of spontaneous magnetization (Figure 3 b, c and d). The magnetization nearly saturates at a value of 12.5 emu g^{-1} in a field of 2500 Oe at 2 K in all compounds. Furthermore, the ferromagnetic hysteresis loops show a quick saturation, therefore supporting the presence of significant interactions below the magnetic ordering transition.⁵⁰ When the temperature is higher than the Curie temperature, the hysteresis loops gradually disappears. It is noted that the saturated magnetization is the highest value in the ever reported in perovskite materials.

To date, only a few perovskite compounds have been reported to possess a ferromagnetic coupling. It is interesting to study the Jahn-Teller activity of Cu-based complexes in the hybrid perovskites. From the configuration of Cu $d_{x^2-y^2}$ orbitals in the Cu-Cl octahedral layer (Figure S4), each octahedron is prolonged along the Jahn-Teller z-axis that lies in the CuCl plane. In an organic-inorganic perovskite system, the magnetic spin originates from the unoccupied Cu $d_{x^2-y^2}$ orbital.⁵¹ For these hybrid perovskites, the $d_{x^2-y^2}$ type symmetry orbitals of the individual octahedra in the a-b basal plane are orthogonal to each other. Jahn-Teller-active ions cause cooperative antiferrodistortive arrangements of the neighboring orthogonal octahedra, which produces ferromagnetic interactions. In fact, the system shows characteristics of a 2D Heisenberg ferromagnet.⁵⁰ The 2D copper halide perovskites presented herein are ideal model systems for the study of Jahn-Teller activity in Heisenberg systems, which may enable deeper understanding to the structure-magnetic relationships.52



Figure 4. MChD investigation and the energy level diagram of *d*-orbitals for Cu^{2+} . MChD recorded on $(R-MPEA)_2CuCl_4$ and $(S-MPEA)_2CuCl_4$ at 2 K.

MChD is a magneto-optical effect in which the absorption coefficient of a chiral compound for an unpolarized light beam differs depending on whether an externally applied magnetic field is parallel or antiparallel to the propagation direction of the light beam.⁵³ MChD is described as a manifestation of magneto-chiral anisotropy, which can be formalized by expanding the frequency dependence (ω) of the absorption coefficient (ϵ) with magnetic field H and the light propagation vector k (Equation (2)).

$$(\omega \kappa \mathbf{B}) = \varepsilon_0(\omega) \pm \alpha^{d/l}(\omega) \kappa \pm \beta(\omega) \mathbf{B} \pm \gamma^{d/l}(\omega) \kappa \cdot \mathbf{B}$$
(2)

Here, ε_0 is associated to the normal absorption of light at zero field, $\alpha^{d/l}$ to the natural circular dichroism, β to the magnetic circular dichroism, and $\gamma^{d/l}$ to the MChD, with dand l that refer to right- and left-handed medium, and +/the right- and left-handed circularly polarized waves, respectively.⁵⁴ MChD signal is associated with spin excitations and can be used to probe molecular magnetism systems for store and read the information.^{55,56} Investigation on the microscopic origins of MChD from organic/inorganic hybrid materials is so far scarce and limited to only a few examples.^{55,53,54,57,59}

Owing to the coexistence of chirality and ferromagnetism, the two enantiomers of $(R-MPEA)_2CuCl_4$ and $(S-MPEA)_2CuCl_4$ are suitable for MChD study. The wavelength dependence of the optical transmission in the region of 200-900 nm with an applied magnetic field (B = 1 T) at 2 K was tested (Figure 4). Measurements performed in the same conditions for $(R-MPEA)_2CuCl_4$ and $(S-MPEA)_2CuCl_4$ provide several strong MChD signals with opposite signs in their optical response, resulting in perfect mirror images. The maximum absolute ΔA value of these perovskite chiral ferromagnets is around 1.9×10^{-3} cm⁻¹, which is similar to that observed on Prussian blue analogue.⁵

Sharp MChD peaks are discernible at $\lambda = 262$ and 490 nm, as well as λ = 545 and 764 nm, corresponding to intra-atomic *d-d* transitions in the Cu²⁺ ions.^{60,61} The inset shows the energy level diagram of *d*-orbitals for Cu²⁺. According to ligand field theory, the ground state electronic distribution of Cu^{2+} is $\operatorname{t}_{2g}{}^6e_g{}^3$ which yields 2E_g term, where the excited electronic state is $t_{2g} {}^5e_g {}^4$ which corresponds to ${}^{2}T_{2g}$ term.⁶² Due to Jahn-Teller effect, ${}^{2}E_{g}$ ground state is split into ${}^{2}A_{1g}(d_{x^{2}-y^{2}})$ and ${}^{2}A_{2g}(d_{z^{2}})$ whereas ${}^{2}T_{2g}$ splits into ${}^{2}B_{1g}(d_{xy})$, ${}^{2}B_{2g}(d_{xz})$ and ${}^{2}B_{3g}(d_{yz})$ states. In our case, the observed electronic transitions at 262, 490 and 764 nm for the two enantiomers can be assigned to the $^{2} A_{1g}(d_{x^{2}-y^{2}}) \rightarrow ^{2} B_{3g}(d_{yz}), \quad ^{2} A_{1g}(d_{x^{2}-y^{2}}) \rightarrow ^{2} B_{2g}(d_{xz}),$ ${}^{2}A_{1g}(d_{x^{2}-y^{2}}) \rightarrow {}^{2}A_{2g}(d_{z^{2}})$, respectively. Figure 4 shows that the different transitions produce MChD signal is different in strength and lineshape. The MChD signal around 545 nm gives the strongest intensity, which is corresponding to the ${}^{2}A_{1q}(d_{x^{2}-y^{2}}) \rightarrow {}^{2}B_{1q}(d_{xy})$ transition, 63,64 while the signals associated with the other transitions are less intense

As discussed above, the orbitals on neighboring Cu²⁺ ions are orthogonal to each other, and the spins experience ferromagnetic superexchange interaction between Cu^{2+} (S = 1/2)⁶⁵ (Figure S4). In the UV-Vis region, the MChD is generally understood as an interference effect between electric dipole (*E*1) and magnetic dipole transitions (*M*1) resulting from the SOC.^{66,67} Based on these mechanisms, the MChD signal of the *d*-*d* transition for Cu^{2+} is expressed by the summation of the *E*1-*M*1 interference term of optical transition from the Cu^{2+} ions.⁶¹ Additionally, the dissymmetry factor for MChD can be defined as:

$$g_{MChD} = \frac{A(B^{\uparrow\uparrow}\kappa) - A(B^{\downarrow\uparrow}\kappa)}{\frac{1}{2}(A(B^{\uparrow\uparrow}\kappa) + A(B^{\downarrow\uparrow}\kappa))} = 2\frac{\Delta A_{MChD}}{A}$$
(3)

where $A = (A_- + A_+)/2$, A_- and A_+ are the absorbances for LCP and RCP in the absence of a magnetic field, respectively.⁵⁹ An essential characteristic of MChD is that g_{MChD} should be of opposite sign for two enantiomers.⁵⁹ We have calculated g_{MChD} between 250 nm and 500 nm according to Equation (3). Figure S5 plots the experimental results for g_{MChD} against the wavelength for the two enantiomers, and the absolute values of the g_{MChD} are summarized in Table S5. We observed absolute values of g_{MChD} = 0.51 T^{-1} for the (*R*-MPEA)₂CuCl₄ and g_{MChD} = 0.59 T⁻¹ for (*S*-MPEA)₂CuCl₄ at 405 nm, confirming significant magnetochiral anisotropy.⁶⁸ It is noted that these g_{MChD} coefficients are much higher than that reported for $Eu((\pm)tfc)_3$ complex $(5 \times 10^{-3} \text{ T}^{-1})^{59}$ and chiral Ni nanomagnets⁶⁸ $(7.3 \times 10^{-3} \text{ T}^{-1})^{59}$ 10⁻⁴ T⁻¹). This intense absolute configuration-dependent signal is direct experimental proof of the existence of the MChD effect in these two perovskite materials.5,54

In summary, we report the first pair of 2D hybrid perovskites that exhibit strong chirality and ferromagnetism simultaneously. Magnetic measurements reveal that these enantiomers have a saturation magnetization value up to 12.5 emu g⁻¹, which is the highest in the ever reported perovskite materials. Moreover, this work presents the first successful MChD measurement in perovskite materials, which indicates that large values of magneto-chiral anisotropy can be obtained in perovskite chiral ferromagnets. Our results about MChD-featured chiral ferromagnets suggest that these materials hold potentials for the development of magneto-optical devices, magneto-resistive sensors and spintronic devices. This work provides a new strategy for creating novel chiral ferromagnets by exploiting perovskite structure, which may lay a foundation for the further development of magnetic materials for spintronic applications.

EXPERIMENTAL SECTION

Materials. *R*-MPEA = *R*- β -methylphenethylamine (97%, ee 98%), *S*-MPEA = *S*- β -methylphenethylamine (97%, ee 98%) was purchased from Alfa Aesar. *rac*-MPEA = *rac*- β -methylphenethylamine (98%) was purchased from J&K Scientific co., LTD. CuCl₂•2H₂O (99.99%) was purchased from ALADDIN. Hydrochloric acid (36%-38%) was purchased from Xilong chemical reagent co., LTD. All reagents and solvent were used without further purification.

Synthesis of *R*-MPEA·HCl and *S*-MPEA·HCl single crystals. *R*-MPEA·HCl was synthesized by reacting *R*-MPEA and

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59 60 hydrochloric acid with the molar ratio of 1:1. The hydrochloric acid was added dropwise into the *R*-MPEA in a singlemouth flask with a round bottom in an ice bath. It takes an hour to complete. Then, the resulting solution was evaporated at 80 °C in a rotary evaporator to remove the solvent. As the solvent is removed, a large number of solids appeared. The crystal was recrystallized in anhydrous ethanol two times, then dried in a vacuum oven at 60 °C for 12 h. For the synthesis of *S*-MPEA•HCl, the steps were conducted the same procedure as above described.

Synthesis of (R-MPEA)₂CuCl₄ and (S-MPEA)₂CuCl₄ single crystals. The synthesis of these enantiomeric crystals through self-assembly crystallization from an aqueous solution of chiral ammonium salt (*R*-MPEA•HCl and *S*-MPEA•HCl) and CuCl2•2H2O salts, by slowly cooling crystallization. Take (R-MPEA)₂CuCl₄ as an example, *R*-MPEA•HCl and CuCl₂•2H₂O with the molar ratio of 2:1 were dissolved in an aqueous solution (2 mmol/mL) and stirred at 80 °C for 20 min. Then, the clear solution was kept in a quiet environment without destabilization for slowly cooling from high temperature to low temperature. After about two hours, many yellow plate-like crystals as large as several millimeters were obtained.

Synthesis of *rac***-MPEA·HCl and** (*rac***-MPEA**)₂**CuCl**₄ **compounds.** For the synthesis of *rac*-MPEA **·**HCl, the steps were conducted the same procedure as the synthesis of *R*-MPEA **·**HCl and *S*-MPEA **·**HCl. By applying the racemic *rac*-MPEA **·**HCl in the synthesis, we also obtained a racemic compound (*rac*-MPEA)₂CuCl₄. Although the crystal quality was not good enough for single-crystal structural analysis, we confirmed that (*rac*-MPEA)₂CuCl₄ crystallizes in a different structure from those of (*R*-MPEA)₂CuCl₄ and (*S*-MPEA)₂CuCl₄ by comparing PXRD patterns (Figure S6).

Thermogravimetric analyses (TGA). TGA was carried out on a Linseis STA PT1600. A heating rate of 10 °C min⁻¹ under flowing N_2 was used from room temperature to high temperature to investigate the thermal stabilities.

Powder X-ray diffraction (PXRD). The X-ray diffraction data were obtained at X'Pert PRO made by Panalytical Company with a wavelength of 1.5406 Å.

Single crystal X-ray diffraction (SCXRD). Single crystals of $(R-MPEA)_2CuCl_4$ and $(S-MPEA)_2CuCl_4$ with good quality were carried out on a Bruker D8 VENTURE Kappa Duo PHOTON II CPAD diffractometer with graphitemonochromatized Cu K α radiation ($\lambda = 1.54184$ Å), operating at 50 kV and 40 mA under N2 flow. Crystal data were integrated and corrected for absorption (numerical) using the STOE X-AREA programs. Crystal structures were solved by direct methods and refined by full-matrix least-squares on F2 using the OLEX2 program package.

Circular dichroism spectra. The CD spectra were collected using a CD spectrometer (J-1500, JASCO) at room temperature. KBr pellets were employed as the matrix for performing solid-state CD measurements.

Magnetic measurement. The magnetic properties of the powdered samples were measured using a superconducting

quantum interference device magnetometer with a sensitivity better than 10⁻⁸ emu (Quantum Design MPMS-XL, USA).

Magneto-chiral dichroism. Manual grinding powder form (R-MPEA)₂CuCl₄ and (S-MPEA)₂CuCl₄ mm-sized crystals were used for magneto-chiral dichroism investigation. Magneto-chiral dichroism spectra measurements were made on Applied Photophysics Chirascan Magnetic circular dichroism spectrometer in the presence of a magnetic field with both parallel and antiparallel fields. Unpolarized light was directed onto the samples using optical fibers. The two beams of circularly polarized light with left and right rotation are monochromated and focused. After that, the sample has the same energy, the same light intensity, the same circular degree of polarization, the same spot size and the same position. The wavelength dependence of the optical transmission in the region of 200-900 nm with an applied magnetic field (B = 1 T) at 2 K was measured. The sample is placed in a cryostat and the cooling rate is 5 °C/min. The averaged magnetochiral dichroism spectra were obtained from the difference between the parallel and antiparallel transmission spectra.

ASSOCIATED CONTENT

Supporting Information.

This material is available free of charge via the Internet at <u>http://pubs.acs.org</u>.

TGA; Photoluminescence spectra; CD spectra; Configuration of Cu $d_{x^2-y^2}$ orbital; Crystal data and structure refinement of (*R*-MPEA)₂CuCl₄ and (*S*-MPEA)₂CuCl₄; Distortions of CuCl₆ octahedral; Summary of the $|g_{MChD}|$ (PDF); X-ray crystallographic file for (*R*-MPEA)₂CuCl₄ and (*S*-MPEA)₂CuCl₄ (CIF)

AUTHOR INFORMATION

Corresponding Author

*shaoxf@lzu.edu.cn *yangdzh@lzu.edu.cn *haoli.zhang@lzu.edu.cn Notes

The authors declare no competing financial interests.

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