

# Raman spectroscopic study of the ternary system $\text{Sb}_2\text{S}_3$ - $\text{As}_2\text{S}_3$ - $\text{Tl}_2\text{S}$

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## Abstract

Raman spectroscopy has been used to study the local structure in glassy compositions of the ternary system  $\text{As}_2\text{S}_3$ - $\text{Sb}_2\text{S}_3$ - $\text{Tl}_2\text{S}$ . The evolution of the positions and shapes of the bands observed is attributed to modifications in the environment of arsenic and antimony corresponding to structural changes as the composition is varied. © 1997 Elsevier Science S.A.

## 1. Introduction

A number of recent studies have been devoted to glasses formed from arsenic, antimony and thallium sulfides [1–8]. These have demonstrated that, as a function of the stoichiometry, the properties [1,2] and structure [3–8] of the glass can be modified.

Our particular interest in this ternary system springs from its broad glassy domain [1], which is shown in Fig. 1. Such glasses have high thermal stability, are transparent over a wide range in infrared [2], and present an absorption

edge which can be tuned between the near infrared to ca. 2.5 eV [3] by varying the composition. The use of a range of complementary techniques is necessary if a structural model of an amorphous solid is to be proposed. We present here results obtained on  $\text{As}_2\text{S}_3$ - $\text{Sb}_2\text{S}_3$ - $\text{Tl}_2\text{S}$  glasses using Raman spectroscopy, and propose a correlation between our conclusions and those of previous works [5–8].

## 2. Experimental section

### 2.1. Glass preparation

Binary antimony, arsenic and thallium sulfides were first prepared by direct reaction from the elements in sealed, evacuated ( $10^{-5}$  torr) silica tubes. Mixtures of the binary sulfides were then heated to 1173 K in sealed, evacuated silica tubes and stirred for 1 h. The tubes were quenched in a water–ice–salt bath. The glasses were then stabilised by heating at 50 °C below the glass transition temperature for 48 h, in order to eliminate the strain produced by quenching. The compositions of the samples studied are given in Table 1.

### 2.2. Characterisation

The glassy nature of the samples can be shown by X-ray diffraction. Diffractograms are characterised by several diffuse rings and present no Bragg diffraction lines. Scanning electron microscopy was used to verify the homogeneity of the samples.

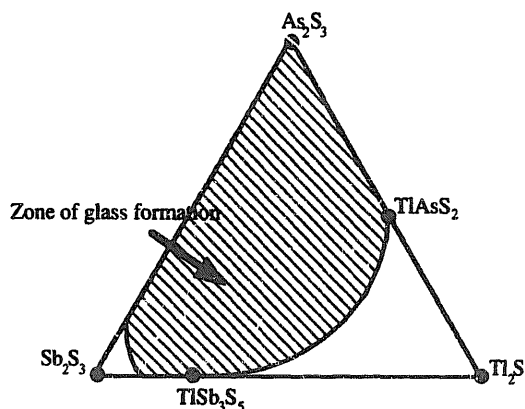


Fig. 1. The region of glass formation.

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Table 1  
Composition of amorphous samples.

Vitreous compound	Sb <sub>2</sub> S <sub>3</sub> %	As <sub>2</sub> S <sub>3</sub> %	Tl <sub>2</sub> S %
1	80	20	0
2	60	40	0
3	50	50	0
4	0	100	0
5	0	95	5
6	0	90	10
7	0	80	20
8	0	70	30
9	0	60	40
10	0	50	50
11	83.3	0	16.7
12	80	0	20
13	75	0	25
20	50	30	20
21	35	45	20
22	20	60	20
30	40	20	40
31	20	40	40
32	10	50	40

### 2.3. Raman spectroscopy

Raman spectra were recorded at room temperature in the range 200–4000 cm<sup>-1</sup> using a Bomem FT spectrometer DA8. Spectra of samples 1, 2, 3 and 4 were obtained using a Dilor triple monochromator spectrometer. These instruments are equipped with YAG (1064 nm) and argon (488 nm) lasers respectively.

## 3. Results and discussion

### 3.1. As<sub>2</sub>S<sub>3</sub>-Sb<sub>2</sub>S<sub>3</sub>

Raman spectra of glasses in the As<sub>2</sub>S<sub>3</sub>-Sb<sub>2</sub>S<sub>3</sub> system are shown in Figure 2. That of glassy As<sub>2</sub>S<sub>3</sub> (Figs. 2–4) presents a single broad and intense line, denoted A. When

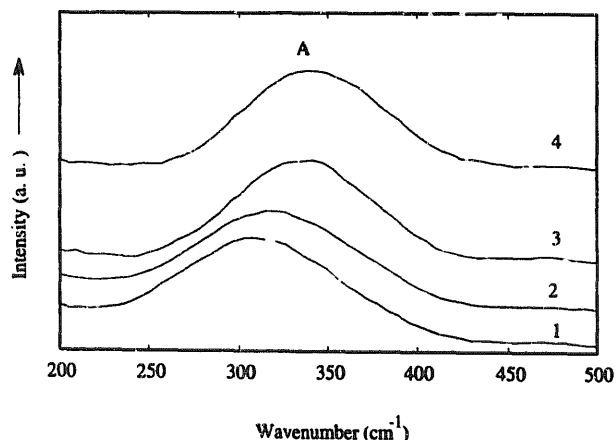


Fig. 2. Raman spectra of the vitreous compounds of the Sb<sub>2</sub>S<sub>3</sub>-As<sub>2</sub>S<sub>3</sub> binary system.

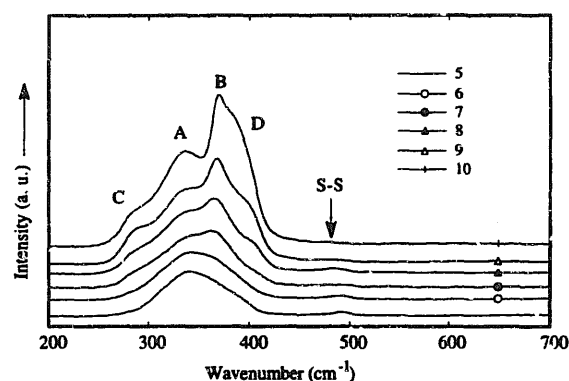


Fig. 3. Raman spectra of the vitreous compounds of the As<sub>2</sub>S<sub>3</sub>-Tl<sub>2</sub>S binary system.

the concentration of Sb<sub>2</sub>S<sub>3</sub> increases, the line shape of A is seen to broaden further, and its maximum is displaced to lower wavenumbers. As a first approximation, these spectra can be interpreted in terms of a molecular model [9], in which the Raman lines are assigned according to the constituent base units. In the case of glassy arsenic and antimony sulfides, the base unit generally used is pyramidal XY<sub>3</sub> (X=As or Sb; Y=S), having C<sub>3v</sub> symmetry. Using this model, band A can be assigned to the symmetric stretching modes of the AsS<sub>3</sub> pyramid, which is in good agreement with previous studies [5–8]. Lucovsky et al. [10] have interpreted the four modes of pyramidal AsS<sub>3</sub> by reference to the AsCl<sub>3</sub> molecule.

The broadening of band A and its shift to lower wavenumbers as the concentration of Sb<sub>2</sub>S<sub>3</sub> increases can be explained by existence of the symmetric stretching mode of the pyramidal SbS<sub>3</sub> units. Several different positions are given in the literature for the latter. Using Raman spectroscopy on compounds similar to those studied here, Kato et al. [11] reported this mode at 297 cm<sup>-1</sup>. Barnier et al. [12] assigned a shoulder at 330 cm<sup>-1</sup> in the infrared spectrum of amorphous Sb<sub>2</sub>S<sub>3</sub> to symmetrical vibrations of the SbS<sub>3</sub> pyramid, by reference to the

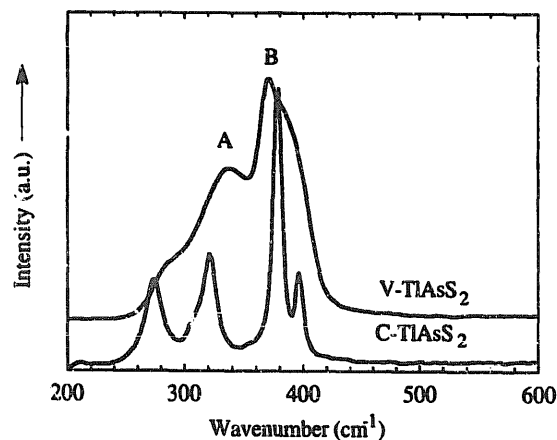


Fig. 4. Comparison of Raman spectra of the crystalline and vitreous compounds TlAsS<sub>2</sub>.

$\text{SbCl}_3$  molecule. Koudelka et al. [13] assign this mode to a line at  $302\text{ cm}^{-1}$  in the Raman spectra of Ge-Sb-S glasses.

Analysis of these spectra indicate that there are two type of mode assigned to pyramidal ( $C_{3v}$ )  $\text{AsS}_3$  and  $\text{SbS}_3$ . Previous studies [8] of this binary system have shown the substitution of arsenic by antimony.

### 3.2. $\text{As}_2\text{S}_3\text{-Ti}_2\text{S}$

The Raman spectra of representative samples  $\text{Ti}_{2x}\text{As}_{2-2x}\text{S}_{3-2x}$  for  $0 < x < 0.5$  are shown in Fig. 3. On increase of the  $\text{Ti}_2\text{S}$  content, a line (B) appears at  $367\text{ cm}^{-1}$ . At the same time, intensity of the line A decrease. In addition, two shoulders at  $270$  and  $390\text{--}400\text{ cm}^{-1}$  appear, denoted C and D.

A weak line at ca.  $490\text{ cm}^{-1}$  is seen in the spectra of samples rich in  $\text{AsS}_3$ , and is assigned to the symmetrical stretching vibration of As-S-S-As structural units [13]. Similar effects have been observed in the spectra of other chalcogenide glasses [14,15], and are reported to be related to the synthesis conditions used.

Fig. 4 shows the Raman spectrum of polycrystalline and glassy  $\text{TiAsS}_2$ . The number of lines, and their maxima, are identical, suggesting that the local structure is analogous. This result agrees with those obtained using X-ray absorption spectroscopy, which demonstrated that arsenic maintained the same environment in the two phases. The spectrum of polycrystalline  $\text{TiAsS}_2$  has lines at  $275$ ,  $325$ ,  $375$  and  $395\text{ cm}^{-1}$  but, in the absence of a single crystal, their precise assignment is difficult.

Crystalline  $\text{As}_2\text{S}_3$  [16] is characterised by two pyramidal arsenic sites ( $C_{3v}$  symmetry). Its Raman spectrum [17] has lines at  $311$ ,  $352$  and  $382\text{ cm}^{-1}$ , the first two of which are assigned to stretching vibrations of the  $\text{AsS}_3$  pyramid, and the last to a coupling mode between two  $\text{AsS}_3$  units. The absence of band A in the spectra of  $\text{TiAsS}_2$  phases might be interpreted as indicating the absence of  $\text{AsS}_3$  units in these materials.

It is known from X-ray diffraction that arsenic atoms occupy two different sites of  $C_s$  symmetry in  $\text{TiAsS}_2$  [18]. In the crystal, the reduction in symmetry of pyramidal units  $\text{AsS}_3$  from  $C_{3v}$  to  $C_s$  leads to degeneracy of the asymmetric modes E, which can be related to the appearance of additional lines in the spectrum of  $\text{TiAsS}_2$ . Table 2 shows the relationship between these symmetries.

The intense line B, observed at  $370\text{ cm}^{-1}$ , can be assigned to the vibration  $\nu'_2$  of type  $A'$  in symmetry  $C_s$  using the molecular model of  $\text{AsS}_3$ .

The line at  $325\text{ cm}^{-1}$ , also reported by Liu Jun et al. [19] in the Raman spectra of Ag-As-S glasses, is assigned to the presence of short chains  $(\text{AsS}_{3/2})_n$ , and results from the  $\nu'_1$  vibration of type  $A'$  of  $\text{AsS}_3$  units in  $C_s$  symmetry.

Shoulders C and D have been related to the infrared active mode [8]. The shoulder C is assigned to a coupling mode between two  $\text{AsS}_3$  units. The weak band D appears in a region intermediate between bending and stretching

Table 2

Relationship between the  $C_{3v}$  and  $C_s$  symmetries.

$C_{3v}(XY_3)$	$\nu_1(A_1)$ $\nu_s(XY)$	$\nu_2(A_1)$ $\delta_s(YXY)$	$\nu_3(E)$ $\nu_d(XY)$	$\nu_4(E)$ $\delta_s(YXY)$
	$\nu_1(A')$	$\nu_2(A')$	$\nu_3(A'')$	$\nu_4(A'')$
	$\nu_s(XZ)$	$\delta_s(YXY)$	$\nu_s(XY)$	$\nu_{as}(XY)$
$C_s(ZXY_2)$				
	$\nu_1(A')$	$\nu_2(A')$	$\nu_3(A'')$	$\nu_4(A'')$
	$\nu_s(XZ)$	$\delta_s(YXY)$	$\nu_s(XY)$	$\nu_{as}(XY)$

modes of the  $\text{AsS}_3$  pyramid and is provisionally assigned to the  $\nu'_4$  vibration ( $A''$ ) in  $\text{AsS}_3$  units in  $C_s$  symmetry.

In the following, lines A and B will be used as a probe for the presence of  $\text{AsS}_3$  motifs of symmetry either  $C_{3v}$  or  $C_s$  respectively. Their simultaneous presence in the spectra of other compounds shows the coexistence of entities with these symmetries. The inversion of intensity of these bands at high concentrations of  $\text{Ti}_2\text{S}$  is explained by the progressive transformation from an entity of symmetry  $C_{3v}$  to one of symmetry  $C_s$ .

In glasses of the  $\text{As}_2\text{S}_3\text{-Ti}_2\text{S}$  system, the initial glassy matrix is dominated by a pyramidal  $\text{AsS}_3$  arrangement of symmetry  $C_{3v}$ . The introduction of Ti-S bonds reduces the length of adjacent As-S covalent bonds, and is accompanied by a lowering of the local symmetry. EXAFS data at the arsenic K-edge [5] and XANES measurement at the sulphur K-edge [6], obtained on compounds of identical composition as these, confirm this hypothesis.

### 3.3. $\text{Sb}_2\text{S}_3\text{-Ti}_2\text{S}$

The glassy region in this binary system is limited to compositions rich in  $\text{Sb}_2\text{S}_3$  [11]. Corresponding Raman spectra are shown in Fig. 5, where a broad line, denoted E, is seen, centred at  $317\text{ cm}^{-1}$ , and bearing a shoulder (F) at

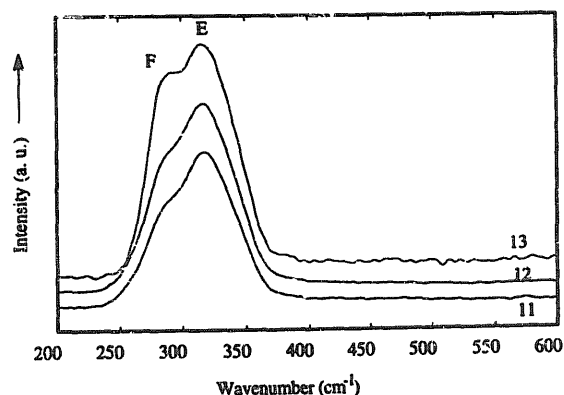


Fig. 5. Raman spectra of the vitreous compounds of the  $\text{Sb}_2\text{S}_3\text{-Ti}_2\text{S}$  binary system.

285  $\text{cm}^{-1}$ . The similarity of all spectra suggests that the local structure in all compositions is comparable.

The spectra of polycrystalline and glassy  $\text{TlSb}_3\text{S}_5$  are compared in Fig. 6, where the assignment of lines at 270, 293, 308 and 330  $\text{cm}^{-1}$  remains tentative.  $\text{TlSb}_3\text{S}_5$  is characterised by three different sites for antimony, two  $\text{SbS}_3$  units and one  $\text{SbS}_4$  unit [20]. Comparison of the spectra of Fig. 6 shows that band E seen in the spectrum of the glass is split into two components at 308 and 330  $\text{cm}^{-1}$  in that of the crystalline compound. These two lines are observed also in the Raman spectrum of  $\text{Ag}_3\text{SbS}_3$  and have been assigned [21] to the symmetric modes of the  $\text{SbS}_3$  pyramid. The shoulder F and the broadening of line E towards lower wavenumbers can be explained by the presence of two other modes in the crystalline compound. The intensity of E suggests that the glassy matrix of these compounds is dominated by a structural arrangement of the  $\text{SbS}_3$  type. EXAFS studies at the  $L_{\text{III}}$  edge of antimony [7] has indicated that the coordination shell corresponds largely to an  $\text{SbS}_3$  pyramid, in good agreement with the results presented here.

### 3.4. $\text{As}_2\text{S}_3\text{-Sb}_2\text{S}_3\text{-Tl}_2\text{S}$

Two series of samples, having fixed  $\text{Tl}_2\text{S}$  contents of 20% (series I) and 40% (series II) were examined. Their Raman spectra are shown in Fig. 7 Fig. 8. The position of the intense line B is characteristic of the  $\nu'_2$  mode ( $\nu'_2$ ) of the  $\text{AsS}_3$  entities; having  $C_s$  symmetry. Its intensity decreases as the  $\text{Sb}_2\text{S}_3$  content increases, suggesting that the number of units of  $C_s$  symmetry is concomitantly reduced.

A second line, denoted G, is observed between 350 and 317  $\text{cm}^{-1}$ . Generally associated with  $\text{AsS}_3$  and  $\text{SbS}_3$  units of  $C_{3v}$  symmetry are observed in this region (345 and 317  $\text{cm}^{-1}$  respectively). Line G probably results from an overlap of lines given by both species. The shift of band G from 345  $\text{cm}^{-1}$  for composition 7 to 317  $\text{cm}^{-1}$  for composition 12 is explained by an inversion in the number

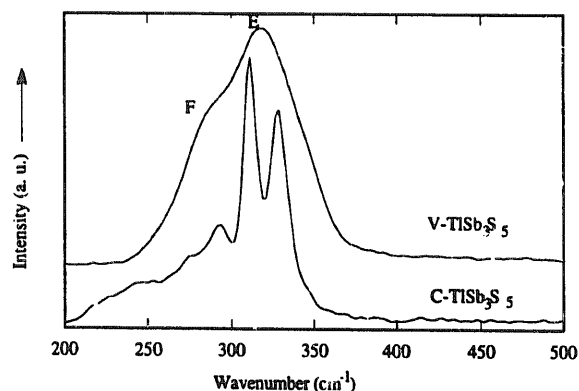


Fig. 6. Comparison of Raman spectra of the crystalline and vitreous compounds  $\text{TlSb}_3\text{S}_5$ .

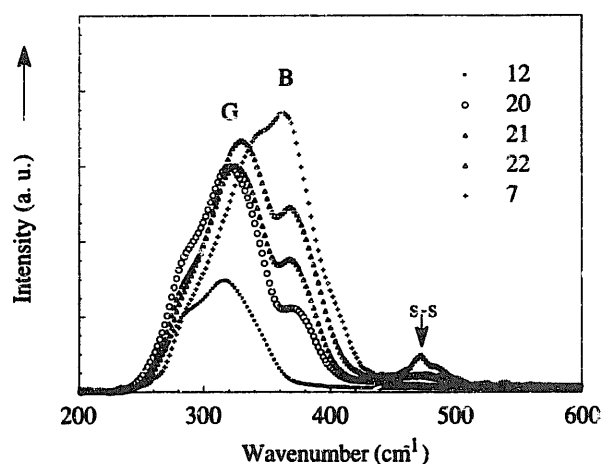


Fig. 7. Raman spectra of the vitreous compounds of the  $\text{Sb}_2\text{S}_3\text{-As}_2\text{S}_3\text{-Tl}_2\text{S}$  ternary system for a constant concentration of 20%.

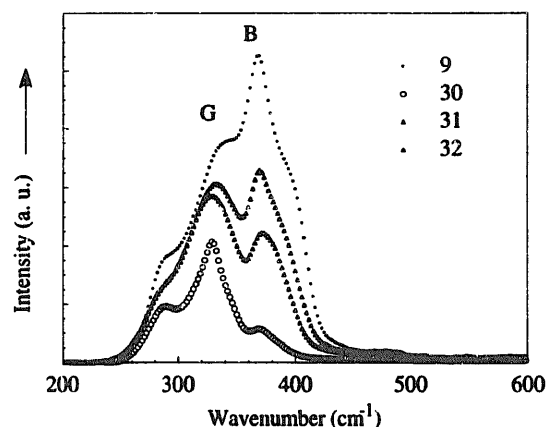


Fig. 8. Raman spectra of the vitreous compounds of the  $\text{Sb}_2\text{S}_3\text{-As}_2\text{S}_3\text{-Tl}_2\text{S}$  ternary system for a constant concentration of 40%.

of  $\text{SbS}_x$  ( $x=3, 4, 5$ ) and  $\text{AsS}_3$  units. Fig. 9 shows that the position of G moves non-linearly to lower frequencies as the  $\text{SbS}_3$  content increases, and that two regions can be distinguished. For compositions with less than 30%  $\text{Sb}_2\text{S}_3$ ,

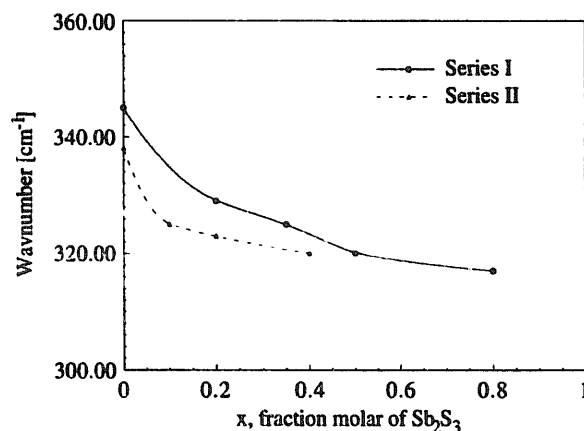


Fig. 9. Wavenumbers of line G with  $\text{Sb}_2\text{S}_3$  concentration in the  $\text{Sb}_2\text{S}_3\text{-As}_2\text{S}_3\text{-Tl}_2\text{S}$  ternary system for constant concentration lines of 20% and 40%  $\text{Tl}_2\text{S}$ .

the shift in the position of G is strongly correlated to the  $\text{Sb}_2\text{S}_3$  content; water correlation is found above this composition, where a less shifted frequency is observed as the  $\text{Sb}_2\text{S}_3$  content increases.

#### 4. Conclusion

Raman spectra of glasses in the binary system  $\text{As}_2\text{S}_3$ - $\text{Sb}_2\text{S}_3$  have been interpreted on the basis of a model making use of pyramidal  $\text{AsS}_3$  and  $\text{SbS}_3$  units, of  $C_{3v}$  symmetry. In  $\text{As}_2\text{S}_3$ - $\text{Ti}_2\text{S}$ , the appearance of additional lines (B, C, D) is attributed to the presence of  $\text{AsS}_3$  motifs of  $C_s$  symmetry, and the inversion of the intensity of lines A and B to transformation of the symmetry of these units. The dominant nature of band G in Raman spectra of glasses in the  $\text{Sb}_2\text{S}_3$ - $\text{Ti}_2\text{S}$  binary system indicates that the glassy matrix is characterised by the presence of pyramidal  $\text{SbS}_3$  units.

The results obtained on the three binary systems have enabled interpretation of the spectra given by the ternary system, and the evolution of the spectra as a function of  $\text{Sb}_2\text{S}_3$  concentration at two different contents (20 and 40%) of  $\text{Ti}_2\text{S}$  have been described. The reduction in intensity of line B is interpreted as indicating a reduction in the number of  $\text{AsS}_3$  units of symmetry  $C_s$ , and the shift of line G by a modification in the coordination at antimony.

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#### References

- [1] A. Bouaza, A. Ibanez, J. Olivier-Fourcade, E. Philippot, M. Maurin, *Mat. Res. Bull.* 22 (1987) 973–984.
- [2] J. Olivier-Fourcade, A. Bouaza, J.C. Jumas, M. Maurin, *J. Non-Cryst. Solids* 111 (1989) 277–284.
- [3] A. Bouaza, T. Olivier-Fourcade, J.C. Jumas, M. Maurin, H. Dexpert, *J. Chim. Phys.* 86 (1989) 1579–1585.
- [4] A. Gheorghiu, I. Lampre, S. Dupont, C. Sénémaud, M.A. El Idrissi Raghni, P.E. Lippens, J. Olivier-Fourcade, *J. Alloys Comp.* 228 (1995) 143–147.
- [5] J.M. Durand, J. Olivier-Fourcade, J.C. Jumas, M. Womes, P. Parent submitted to *J. Mater. Sci.*
- [6] J.M. Durand, P.E. Lippens, J. Olivier-Fourcade, J.C. Jumas, M. Womes, *J. Non-Cryst. Solids*, (1996) in press.
- [7] J.M. Durand, P.E. Lippens, J. Olivier-Fourcade, J.C. Jumas, M. Womes, *J. Non-Cryst. Solids* 194 (1996) 109.
- [8] M.A. El Idrissi Raghni, J.M. Durand, B. Bonnet, L. Hafid, J. Olivier-Fourcade, J.C. Jumas, *J. Alloys Comp.* 239 (1996) 8.
- [9] G. Lukovsky, J.P. deNeufville, F.L. Galeener, *Phys. Rev. B* 9 (1974) 1591.
- [10] G. Lucovsky, R.M. Martin, *J. Non-Cryst. Solids* 8–10 (1972) 185.
- [11] M. Kato, S. Onari, T. Arai, *Jpn. J. Appl. Phys.* 22 (1983) 1382.
- [12] S. Barnier, M. Guittard, C. Jullien, A. Chilouet, *Mat. Res. Bull.* 28 (1993) 399.
- [13] L. Koudelka, M. Fruner, M. Pisarcik, *J. Non-Cryst. Solids* 41 (1980) 171.
- [14] L. Koudelka, M. Pisarcik, *J. Non-Cryst. Solids* 64 (1984) 87.
- [15] I. Zitkovsky, P. Boolchand, *J. Non-Cryst. Solids* 114 (1989) 70.
- [16] D.J.E. Muller, W. Nowacki, *Z. Kristallogr.* 136 (1972) 48.
- [17] R. Zallen, M.L. Slade, A.T. Ward, *Phys. Rev. B* 36 (1987) 980.
- [18] M.E. Fleet, *Z. Kristallogr.* 138 (1973) 147.
- [19] L. Jun, J.J. Videau, B. Tanguy, J. Partier, J.M. Reau, P. Hagenmuller, *Mat. Res. Bull.* 23 (1988) 1315.
- [20] M.E. Fleet, *Z. Kristallogr.* 138 (1973) 147.
- [21] H.H. Byer, L.C. Bobb, I. Lefkowitz, E.S. Denver, *J. Ferroelec.* 5 (1973) 207.