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Ti-doping effect on the MI transition of hollandite vanadate, K₂V₈O₁₆

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Abstract

Ti-doping effect on the metal-insulator transition of hollandite vanadate, $K_2V_8O_{16}$ has been investigated. The transition is accompanied by the charge ordering and the structural change from tetragonal to monoclinic structures with the increase of volume in the low-temperature monoclinic insulator phase. A slight amount of Ti-substitution for V atoms raises the transition temperature up to 270 K at x = 0.5 in $K_2V_{8-x}Ti_xO_{16}$ from 170 K at x = 0. This significant rise of transition temperature can be understood as a negative chemical pressure effect caused by the Ti-substitution. On the other hand, the disorder effect from the Ti-substitution makes the transition broad and finally the transition smears out around x = 1.0. © 2006 Elsevier B.V. All rights reserved.

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Vanadium oxides exhibit a rich variety of electromagnetic properties. The most characteristic property among them is metal-insulator (MI) transition as a function of temperature, as typically observed in binary vanadium oxides [1]. The MI transitions in mixed valent vanadium oxides such as V_4O_7 are accompanied by the charge separation and charge order, and spin singlet $V^{4+}-V^{4+}$ pairs are formed in the low-temperature insulator phases, resulting in the large reduction of magnetic susceptibility at the MI transitions [2]. In our current study on vanadium oxides, an MI transition at 170 K was recently found in a hollandite vanadium oxide K₂V₈O₁₆ [3]. "Hollandite" is one of manganese minerals. Hollandite-type compounds have the general chemical formula $A_x M_8 O_{16}$ ($x \leq 2$). The hollandite structure consists of the tubular M8O16network and A-cations at the tunnel sites of the M₈O₁₆framework, as shown in the inset of Fig. 1. The M_8O_{16} framework is constructed from the double-chains running along the *c*-axis in the tetragonal structure and each doublechain is formed by sharing the edges of MO₆ octahedra. The MI transition of K₂V₈O₁₆ is accompanied by two step jumps about three orders in the resistivity along the c-axis and the reduction of magnetic susceptibility down to a small value

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(see Fig. 2), suggesting the formation of $V^{4+}-V^{4+}$ singlet pairs and $V^{3+}-V^{3+}$ pairs in the low-temperature insulator phase [3]. The transition is of first order accompanied by the structural change from a tetragonal to a monoclinic structure, and the low-temperature phase has a superlattice of $\sqrt{2a} \times \sqrt{2a} \times 2c$, where a and c denote the primitive cell parameters for the simple tetragonal hollandite structure, suggesting a charge ordering of V^{4+} and V^{3+} [3]. From these results we proposed a possible charge order model, where two kinds of double-chain formed by V^{3+} and V^{4+} chains and by single V^{4+} chains, respectively, order in a manner that gives a superlattice of $\sqrt{2} \times \sqrt{2}$ in the *a*-*b* plane, and the V⁴⁺-V⁴⁺ and $V^{3+}-V^{3+}$ pairs are formed in each chain along the *c*-axis, resulting in a duplication of the c-axis [3]. In order to elucidate the origin of the transition, we have investigated the substitution effect of Ti for V sites. We found the significant enhancement of transition temperature by the substitution. This means the transition is not of a simple charge order transition in which a significant damage on the transition is expected by the introduction of disorder. In this short paper, we report the structural and magnetic properties of $K_2V_{8-x}Ti_xO_{16}$ and discuss the Ti-substitution effect on the transition.

Powdered samples were prepared by a solid-state reaction of KVO_3 , V_2O_3 , V_2O_5 and TiO_2 under high

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Fig. 1. Composition (x) dependence of lattice parameters in $K_2V_{8-x}Ti_xO_{16}$. The inset shows a schematic structure of $K_2V_8O_{16}$ viewed from the *c*-axis of the tetragonal structure.



Fig. 2. Temperature dependence of magnetic susceptibility for various x in $K_2V_{8-x}Ti_xO_{16}$.

pressure of 4 GPa at 1473 K for 1 h. Starting reagents were ground thoroughly and the mixture was wraped by platinum foil and then put into a pressure cell. KVO_3 was obtained by a reaction of K_2CO_3 and V_2O_5 . V_2O_3 was prepared by the reduction of V_2O_5 under flowing H₂ gas at 1173 K. A cubic type apparatus was used for high pressure synthesis. The sample identification and the determination of lattice parameters were made by powder X-ray diffraction. Magnetic susceptibility of powdered sample was measured by using a SQUID magnetometer.

All samples of $K_2V_{8-x}Ti_xO_{16}$ ($0 \le x \le 8$) are obtained as a single phase with a simple tetragonal hollandite structure at room temperature. This means Ti forms an entire solid solution of $K_2V_{8-x}Ti_xO_{16}$. Fig. 1 shows the lattice parameters as a function of x in $K_2V_{8-x}Ti_xO_{16}$ at room temperature. The lattice parameters of both a and c increase with x, namely the volume increases with x. Fig. 2 shows temperature dependence of magnetic susceptibility (χ) for K₂V_{8-x}Ti_xO₁₆. The most remarkable point is a significant rise of transition temperature by a slight substitution of Ti. As mentioned above, the MI transition of $K_2V_8O_{16}$ is characterized by the reduction of magnetic susceptibility. The onset temperature of the reduction of magnetic susceptibility goes up to 270 K at x = 0.5 from 170 K at x = 0. Simultaneously, the transition becomes broad and smears out around x = 1.0. On the other hand, the magnetic susceptibility of low-temperature phase increases with x and then above x = 1, the magnetic susceptibility continuously changes as a function of temperature. There is no anomaly suggesting any transition.

In general, phase transitions accompanied by some kinds of order suffer a significant damage by the introduction of disorder and the transition temperature could be lowered. $K_2V_8O_{16}$ is a mixed valent compound with $V^{3+}/V^{4+} = \frac{1}{3}$. Since a stable valence state of Ti is Ti^{4+} in the coexistence with V^{3+}/V^{4+} , the substitution of Ti for V must give a significant disorder in the charge ordering at the transition. Therefore, the observed enhancement of transition temperature is not understood in terms of such general disorder effect. The transition of K₂V₈O₁₆ is accompanied by a volume increase about 0.14% in the low-temperature phase. On the other hand, the substitution of Ti for V also increases the volume. This means the Ti-substitution gives a negative pressure to the system as a chemical pressure, and as a result the transition temperature might be raised by a slight substitution of Ti. However, the further Tisubstitution leads to a significant disorder, and finally the transition smears out. The (chemical) pressure effect on the transition is dominant at the initial stage of the Tisubstitution, while the disorder effect becomes dominant with the progress of the Ti-substitution.

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