

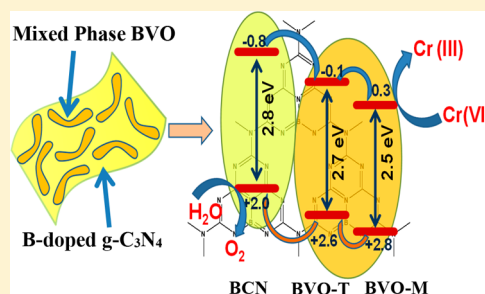
# Serendipitous Assembly of Mixed Phase BiVO<sub>4</sub> on B-Doped g-C<sub>3</sub>N<sub>4</sub>: An Appropriate p–n Heterojunction for Photocatalytic O<sub>2</sub> evolution and Cr(VI) reduction

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## Supporting Information

**ABSTRACT:** Type II p–n heterojunction B-doped g-C<sub>3</sub>N<sub>4</sub>/BiVO<sub>4</sub> moieties have been fabricated by depositing n-type BiVO<sub>4</sub> on the surface of p-type 1 wt % B-doped g-C<sub>3</sub>N<sub>4</sub> for the first time. The materials were characterized by PXRD, XPS, UV–vis DRS, IR, PL, and Raman analysis. The photocatalytic activities of as synthesized samples were studied toward reduction of Cr(VI) and water splitting reaction to generate oxygen. The results reveal that Type II p–n heterojunction considerably enhance the photocatalytic activity as compared to neat n-type BiVO<sub>4</sub> and p-type B-doped g-C<sub>3</sub>N<sub>4</sub>. 50% BiVO<sub>4</sub>/B-doped g-C<sub>3</sub>N<sub>4</sub> heterostructure exhibited the best photocatalytic activity, which is 7.9-fold higher than that of BiVO<sub>4</sub> and followed by a pseudo first order kinetics with apparent first order rate constant of 0.063 min<sup>−1</sup>. Again the heterojunction is able to produce 4.2 times higher oxygen evolution value as related to pristine BiVO<sub>4</sub>. The superior photocatalytic activity is attributed to higher visible light utilization and lower recombination of electron–hole pairs by creating a p–n junction. PXRD and HRTEM data suggest the formation of mixed phase monoclinic and tetragonal BiVO<sub>4</sub> thereby creating a heterojunction for the improvement of the photocatalytic performances. The formation of mixed-phase BiVO<sub>4</sub> is attributed to the high temperature calcination as well as surface energy of B-doped g-C<sub>3</sub>N<sub>4</sub>. The oxygen vacancy in the system is confirmed through XPS and Raman analysis. Moreover, the excellent photocurrent response by the designed photocatalyst at lower overpotential and decrease in carrier recombination as compared to bulk one, studied from LSV and electrochemical impedance spectroscopy, validate the unique photocatalytic activity of the catalyst. The formation of the p–n heterojunction is confirmed from a Mott–Schottky analysis. The work shed new light on the assembly of the p–n heterojunction, showing excellent photocatalytic properties in a simple way.



## INTRODUCTION

Since the photocatalytic process is regarded worldwide as a preeminent method to resolve the extreme problems of energy deficiency and environmental crisis, the design of a visible light active photocatalyst has been gaining more attention.<sup>1,2</sup> Semiconductor photocatalysis is a cutting-edge technology which can convert solar energy to chemical energy that could be used for removal of pollutant and various chemical transformation processes including water oxidation to generate O<sub>2</sub>.<sup>3,4</sup> TiO<sub>2</sub> has become a model photocatalyst due to its nontoxicity, higher stability, and tremendous oxidizing ability. However, due to its wide band gap characteristics, the catalyst is able to absorb only ultraviolet light, which covers only 4% of solar energy.<sup>5</sup> Moreover, the high density of the trap state, which leads to fast electron–hole recombination process, supplements another limitation of TiO<sub>2</sub>. Hence, catalyst design including visible light active catalyst is one of the most promising tasks in the current scenario of research.<sup>6,7</sup> In this context, BiVO<sub>4</sub><sup>8,9</sup> has drawn great attention as a visible light responsive catalyst due to its narrower band gap, good photostability, nontoxicity, and higher sunlight utilization ability. Nevertheless, the valence band of BiVO<sub>4</sub> is sufficiently positive and acts as a photoanode that facilitates the oxygen

evolution reaction. Generally, BiVO<sub>4</sub> exists in three different polymeric forms,<sup>10</sup> i.e., zircon-tetragonal, scheelite-tetragonal, and scheelite-monoclinic. Among these, the monoclinic phase<sup>11</sup> is the most interesting one due to its higher photocatalytic activity as compared to both tetragonal phases. The reason for this is due to the high band gap of tetragonal phase possessing an UV absorption band (2.9 eV) as compared to monoclinic system (2.4 eV) having both an UV and visible absorption band. The transition of the electron from O 2p to V 3d is responsible for generation of the UV band. The visible light band is assigned to the movement of electrons from the valence band<sup>12</sup> (available from hybrid orbital of Bi 6s and O 2p) to the conduction band (formed by V 3d orbital). Again separation efficacy of electron and hole pair is improved in case of monoclinic BiVO<sub>4</sub> due to the distortion of Bi–O bond. Further DFT calculation demonstrate that hole mobility<sup>13</sup> is an important parameter for the enhanced photocatalytic activity by monoclinic phase compared to tetragonal analogue. However, the bulk monoclinic BiVO<sub>4</sub> suffers from lower

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activity due to poor charge transfer<sup>14</sup> properties and weak surface adsorption characteristics.

Various strategies have been demonstrated so as to develop the photocatalytic property of monoclinic BiVO<sub>4</sub>, including heterojunction construction,<sup>15</sup> loading of cocatalyst,<sup>16</sup> and doping of impurity<sup>17</sup> to its lattice structure. Of these approaches, the construction of heterojunction with different crystalline phases<sup>18</sup> of BiVO<sub>4</sub> appears to be a new way. The heterojunction between phases generally have a higher interfacial charge transfer rate<sup>19</sup> and can be easily synthesized by adjusting the pH of the reaction medium<sup>20</sup> or controlling the heating temperature<sup>18</sup> during the crystallization process. Tan et al. has shown that the coexistence of a mixed phase structure has enhanced photocatalytic behavior toward RhB degradation.<sup>20</sup> Fan et al. has confirmed that the interface of monoclinic and tetragonal phase provides a suitable path for separation of photoinduced electrons<sup>19</sup> and also changes their direction of migration. Cheng et al. has synthesized BiVO<sub>4</sub> nanofibers by an electrospinning method<sup>18</sup> with controlled heat treatment. The improved photocatalytic activity toward RhB degradation was due to the formation of phase junction between the tetragonal and monoclinic phase. The existence of heterostructured material obviously leads to the enhanced charge separation and migration thereof, thereby improving the photocatalytic process.

Carbon materials such as graphitic carbon nitride<sup>21</sup> (g-C<sub>3</sub>N<sub>4</sub>) belongs to an important class of material in the domain of photocatalytic research owing to its unparalleled capability to utilize visible light and brilliant chemical and thermal strengths. The material consists of tris(s-triazine) repeating groups which are responsible for the good electrical conductivity due to high electron mobility.<sup>22</sup> The catalyst can easily be synthesized on a large scale just by heating simple organic molecules like urea and melamine. The polymer has a moderate band gap of 2.7 eV, which absorbs visible light of the solar spectrum.<sup>23,24</sup> However, bulk g-C<sub>3</sub>N<sub>4</sub> has a low specific surface area and also possesses fast electron–hole recombination due to the  $\pi$ – $\pi$  stacking among the aromatic layers. Recently, numerous approaches have been demonstrated to improve its photocatalytic property including the design of nanostructured array,<sup>25–27</sup> heterojunction construction,<sup>28,29</sup> and impurity doping<sup>23,30</sup> to its lattice structure. Doping of metal<sup>31</sup> or nonmetal<sup>32</sup> into the g-C<sub>3</sub>N<sub>4</sub> lattice may able to tune the valence and conduction band edge potential, enhancing higher photocatalytic activity. In this context, boron doping<sup>23,33</sup> is important because it not only alters the conductivity of g-C<sub>3</sub>N<sub>4</sub> from n- to p-type but also shifts its conduction band negative enough to facilitate various reduction reactions including chromium reduction and hydrogen evolution reaction. Again, heterojunction photocatalysts constructed on g-C<sub>3</sub>N<sub>4</sub> have been extensively designed to minimize the fast charge recombination of the polymer. In this perspective, Li et al. has constructed a monoclinic porous BiVO<sub>4</sub>/g-C<sub>3</sub>N<sub>4</sub> nanocomposite that shows excellent photocatalytic activity toward RhB degradation.<sup>34</sup> Samanta et al. has designed the BiVO<sub>4</sub>/g-C<sub>3</sub>N<sub>4</sub> heterostructure for the oxidation of aromatic alcohols with improved selectivity.<sup>35</sup> Coral-like direct Z-scheme BiVO<sub>4</sub>/g-C<sub>3</sub>N<sub>4</sub> has been synthesized by Sun et al. for photocatalytic oxidation of toluene.<sup>36</sup> Wang et al. has combined the electrospinning preparation of a BiVO<sub>4</sub>/g-C<sub>3</sub>N<sub>4</sub> composite film and explored its photoelectrochemical performances.<sup>37</sup> The g-C<sub>3</sub>N<sub>4</sub> nanosheet not only forms a heterojunction but also prevents the agglomeration of BiVO<sub>4</sub>.

Considering modified g-C<sub>3</sub>N<sub>4</sub>, Kong et al. has synthesized S-doped C<sub>3</sub>N<sub>4</sub>/BiVO<sub>4</sub> composite via impregnated coprecipitation method.<sup>38</sup> The nanocomposite shows 129 and 349% enhancement in water oxidation and charge carrier lifetime, respectively, as compared to pristine BiVO<sub>4</sub>. Hou Yang et al. have synthesized polymeric carbon nitride foam and explored its photocatalytic activity toward tetracycline removal in different natural water matrices including wastewater and natural water. Yan Wu et al. have demonstrated a Z-scheme-based 2D–2D heterostructure consisting of oxygen-doped carbon nitride and ultrathin Co–Al layered double hydroxide and tested its photocatalytic activity toward MO degradation. Again the group has synthesized petal-like CdS nanostructures coated with S-doped graphitic carbon nitride, which shows excellent activity toward RhB degradation and H<sub>2</sub> evolution. Yan Wu et al. have demonstrated hierarchical 2D–2D Zn<sub>3</sub>In<sub>2</sub>S<sub>6</sub>/fluorinated carbon nitride nanosheet for boosting photocatalytic MO degradation and H<sub>2</sub> evolution. Again the same group has constructed a Z-scheme-based p–n heterojunction by coupling LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> for Brilliant Blue degradation.<sup>39–43</sup>

This work demonstrates the construction of a novel and efficient heterojunction catalyst with mixed phase BiVO<sub>4</sub>(monoclinic-tetragonal) decorated on B-doped graphitic carbon nitride nanosheet for the first time. BVCN were synthesized by a facile impregnated coprecipitation method followed by calcination. As synthesized catalysts were tested for photocatalytic water oxidation and chromium reduction and show excellent catalytic activity compared to pristine BiVO<sub>4</sub>. The synthesis, characterization, and photocatalytic properties of the catalysts were discussed in detail.

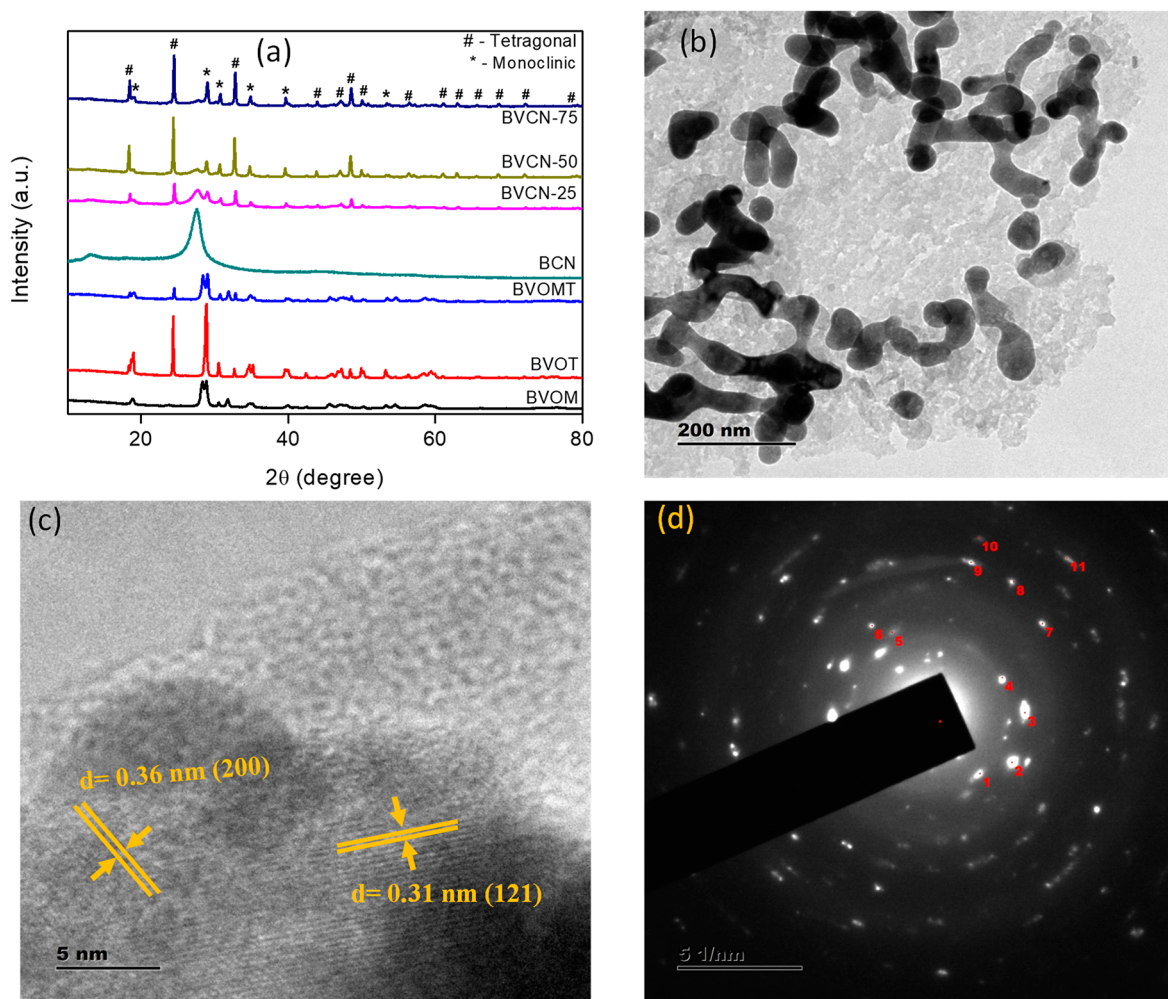
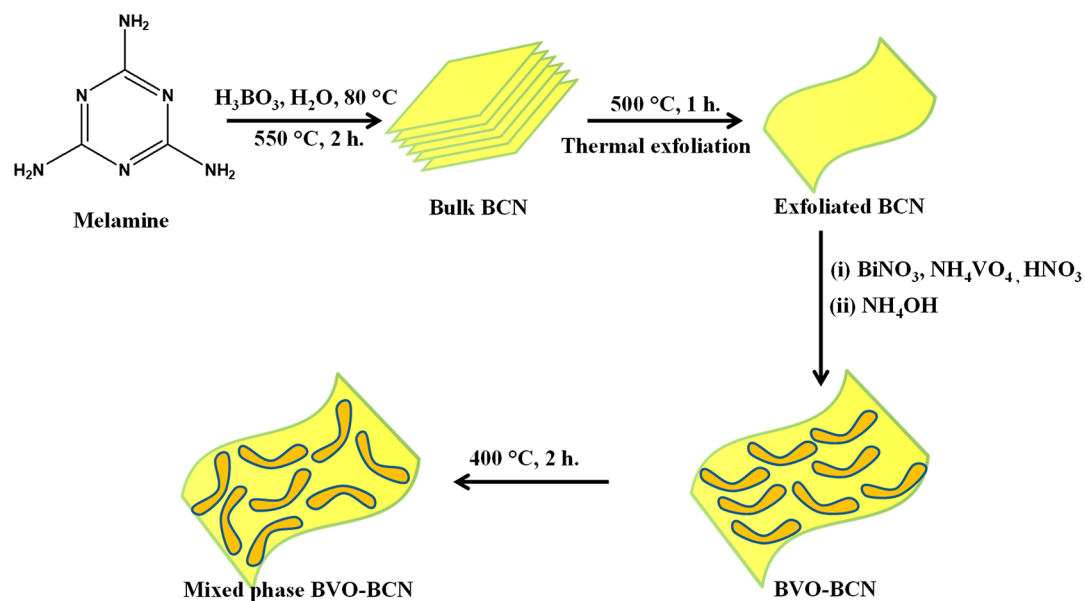
## ■ EXPERIMENTAL SECTION

**Materials.** Bismuth nitrate, ammonium vanadate, melamine, boric acid, silver nitrate, and potassium dichromate were purchased from Merck. All chemicals were used as received without any further purification. Double distilled water is used throughout all experiments.

**Synthesis of B-Doped g-C<sub>3</sub>N<sub>4</sub>(BCN).** B-doped g-C<sub>3</sub>N<sub>4</sub> (1 wt %) was synthesized via the reported procedure with slight modification.<sup>23</sup> A 10 g sample of melamine and 0.572 g of boric acid were taken in 300 mL of water and heated at 80 °C for 12 h whereupon a white crystalline compound is formed. The as obtained compound was calcined at 550 °C for 2 h and calcined at 450 °C for 1 h to give a light yellow color powder.

**Synthesis of Monoclinic BiVO<sub>4</sub>(BVOM).** First, 6 g of bismuth nitrate was put in 100 mL of water, and 0.8 mL of nitric acid was added into it. The whole solution was stirred for 30 min. In another beaker, 1.4 g of ammonium vanadate was added into 100 mL of water and the mixture stirred to give a yellow solution. Then the above solution was added dropwise to bismuth nitrate solution to give an orange color precipitate. After 30 min of continuous stirring, the pH of the solution was neutralized to 7 by addition of ammonium hydroxide solution. The light yellow color precipitate was filtered and dried to give the desired compound.<sup>44</sup> Tetragonal BiVO<sub>4</sub> was also synthesized by following literature procedures<sup>45</sup> described elsewhere (BVOT). Mixed phase BiVO<sub>4</sub> (BVOMT) was synthesized by calcining as synthesized BVO at 400 °C.

**Synthesis of BiVO<sub>4</sub>–B Doped g-C<sub>3</sub>N<sub>4</sub> (BVCN).** First, 0.73 g of bismuth nitrate and 1 g of B-doped g-C<sub>3</sub>N<sub>4</sub> were taken into 200 mL of water and sonicated for 30 min and stirred. To the above solution was added 0.5 mL of nitric acid, and stirring was continued for 1 h. In another beaker, 0.175 g of ammonium vanadate was dissolved in 20 mL of water, the yellow colored solution was added dropwise to the above solution, and stirring was continued for 30 min. Finally the pH of the solution was neutralized by adding ammonium hydroxide and the light yellow color precipitate was filtered, dried and calcined at

Scheme 1. Formation of  $\text{BiVO}_4$ -B-Doped  $\text{g-C}_3\text{N}_4$  p-n Heterojunction

**Figure 1.** (a) PXRD of as synthesized material showing mixed phase BVO in BVCN. (b) HRTEM images of BVCN where cashew nut shape BVO is dispersed in BCN nanosheet. (c) Lattice fringe width of BVCN matching with  $d$  values of both monoclinic and tetragonal phase of BVO. (d) SAED pattern of BVCN.



400 °C to get BVCN-50. BVCN-25 and BVCN-75 were synthesized by following the same procedure by varying the weight percent of  $\text{BiVO}_4$  precursor (Scheme 1).

**Photocatalytic Water Oxidation and Cr(VI) Reduction.** The properties of as synthesized photocatalysts were assessed through photocatalytic water oxidation and Cr(VI) reduction reaction upon visible light illumination. Photocatalytic water splitting reaction for oxygen generation was done in a 100 mL sealed quartz batch reactor fitted with a 150 W xenon lamp having a 420 nm cut off filter. Typically 20 mg of catalyst was spread in a 20 mL solution (5 mmol  $\text{AgNO}_3$ ).  $\text{AgNO}_3$  is used as electron acceptor<sup>46</sup> from the semiconductor photocatalyst and provided the hole for the water oxidation reaction. Proper dispersion of photocatalyst was achieved by stirring the solution with a magnetic stirrer, and the dissolved gases were removed by passing nitrogen gas a number of times over the reaction medium before light illumination. The experiment was performed at room temperature by circulating cold water, and developed gas was accumulated by downward displacement of water. Analysis of the collected gas was done via a gas chromatograph equipped with a 5 Å molecular sieve column with a thermal conductivity detector built-in and confirmed as oxygen. For photocatalytic Cr(VI) reduction, 50 mg of each catalyst was put in a 100 mL Pyrex glass containing 20 mL of Cr(VI) solution (20 mg  $\text{L}^{-1}$ ). Before light irradiation, the suspension was magnetically stirred for 30 min to confirm the establishment of adsorption/desorption equilibrium. Visible light was illuminated by using a 150 W xenon lamp fitted with a cutoff filter ( $\lambda > 420$  nm). By circulating cold water, the temperature of the reaction solution was maintained at room temperature. During illumination, 2 mL of solution was extracted through a syringe at given time interval and centrifuged to remove the catalyst. The conversion of Cr(VI) to Cr(III) was monitored colorimetrically by using diphenylcarbazide (DPC) method<sup>47</sup> where absorbance maximum is observed at 545 nm.

## ■ CHARACTERIZATION

X-ray diffraction patterns of as synthesized materials were recorded in the  $2\theta$  range 10–80° with a scan rate of  $2^\circ \text{ min}^{-1}$  on a Rigaku miniflex diffractometer equipped with  $\text{Cu K}\alpha$  radiation ( $\lambda = 1.45$  Å). High resolution transmission electron microscopy (JEOL-JEM 2010, Japan) obtained with an energy dispersive X-ray spectrometer (Oxford Instrument, INCA, U.K.) was used for the structure and morphology determination of the materials. X-ray photoelectron spectroscopy (XPS) was done in a XPS system in a Kratos Axis 165 instrument equipped with a Mg  $\text{K}\alpha$  source. DRUV spectra and photoluminescence (PL) spectra were carried out on JASCO UV–vis spectrophotometer and JASCO-FP-8300 spectrofluorometer, respectively. JASCO-FT-IR-4600 spectrometer was used to record the IR spectra of synthesized material taking KBr as reference and frequency range was set from 400 to 4000  $\text{cm}^{-1}$ . Electrochemical experiments were performed using a IVIUMnSTAT instrument where Pt and Ag/AgCl electrodes were taken as counter and reference electrodes, respectively. The electrophoretic deposition method has been taken into consideration for preparation of the working electrode. Typically 30 mg of catalyst and iodine were taken in 30 mL of acetone, and proper dispersion of catalyst was done by a sonicating the solution for 10 min. Two parallel fluorine doped tin oxide (FTO), separated with 15–20 mm distances, were immersed in the above solution, and 60 V bias was set for 3 min to deposit the catalyst on the FTO surface. The FTOs were calcined at 200 °C for 2 h to remove the impurities on the surface of the catalyst. The linear sweep voltammetry (LSV) was performed through applying the potential from 0.4 to 1.3 V (scan rate 25 mV/s) and illuminated with a 300 W xenon lamp. Electrochemical impedance spectroscopy (EIS) was performed at zero biased potential in the dark and with the range of frequency set from  $10^{-2}$  to  $10^5$ . All electrochemical experiments were carried out in 0.2 M  $\text{Na}_2\text{SO}_4$  solution.

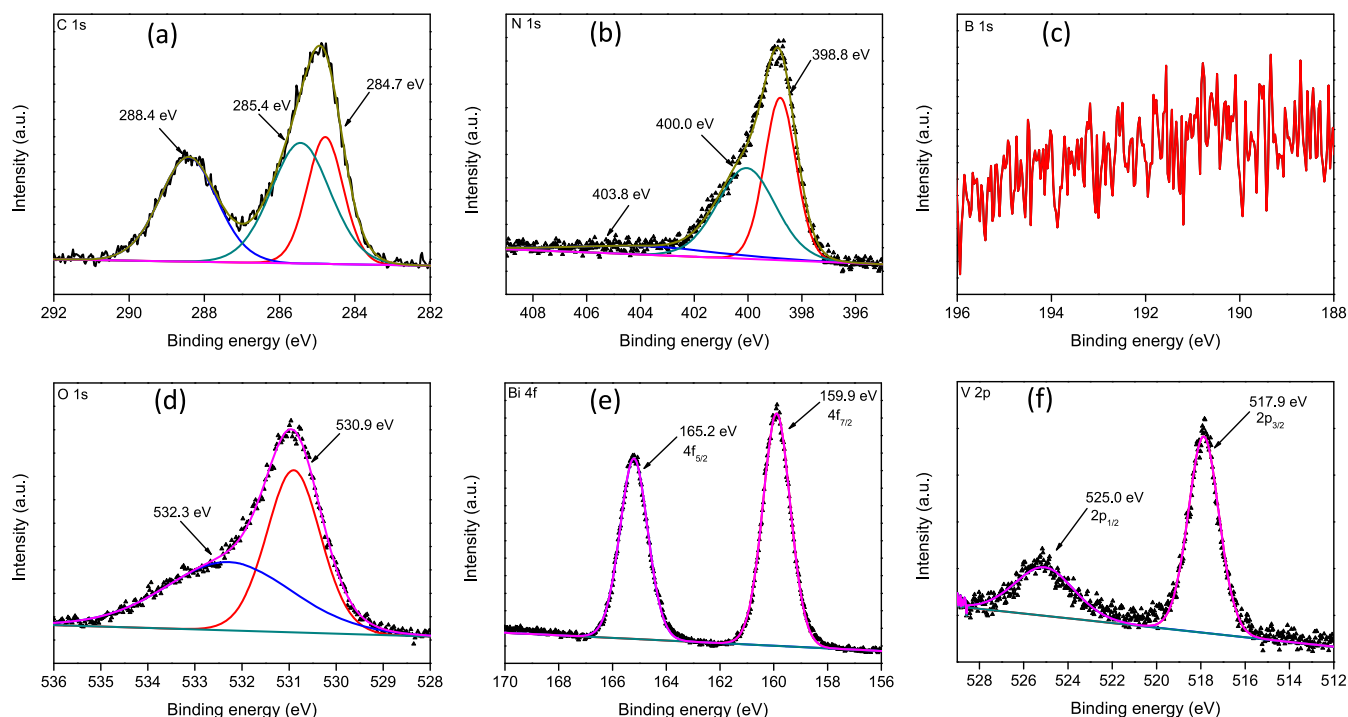
## ■ RESULTS AND DISCUSSION

**Crystal Structure.** The crystal structure, phase purity, and composition of as prepared catalyst were analyzed by powder

XRD.<sup>48</sup> As shown in Figure 1a two pronounced peaks at  $2\theta = 13.2^\circ$  and  $27.6^\circ$  were observed for BCN which are indexed to 100 and 002 planes of graphitic carbon nitride.<sup>49,23</sup> The 100 peak is attributed to stacked conjugated aromatic ring and 002 planes represent the in plane structural packing of triazine group. The well-known three polymorphs of  $\text{BiVO}_4$  are zircon structure with tetragonal system (z-t) and scheelite structure with monoclinic (s-m) and tetragonal (s-t) systems.<sup>10</sup> The pure BVO shows a series of sharp and narrow diffraction peaks which is the well-known monoclinic phase (JCPDS-00-014-0688, space group  $I2/a$ ) of bismuth vanadate with cell constant  $a = 5.195$ ,  $b = 11.701$ ,  $c = 5.092$ . BVO exhibits peaks at  $18.9^\circ$ ,  $28.4^\circ$ ,  $28.8^\circ$ ,  $30.5^\circ$ ,  $34.8^\circ$ ,  $39.9^\circ$ ,  $40.7^\circ$ ,  $42.4^\circ$ ,  $45.6^\circ$ ,  $47.6^\circ$ ,  $50.2^\circ$ ,  $53.2^\circ$ ,  $54.4^\circ$ ,  $58.6^\circ$ , and  $59.2^\circ$  which are assigned to (011), (−130), (−121), (040), (200), (211), (112), (051), (231), (042), (202), (−161), (013), (321), and (123), respectively. Interestingly in case of heterojunction such as BVCN-25, BVCN-50, and BVCN-75, a peak at  $24.3^\circ$  is observed which is attributed to zircon tetragonal phase<sup>50</sup> of BVO (JCPDS-00-014-0133, space group  $I41/amd$ ) with cell constant  $a = 7.299$ ,  $b = 7.299$ ,  $c = 6.457$ . The percentage of monoclinic phase<sup>51</sup> in BVCN heterostructure is calculated by using the following equation.

$$\eta_{\text{mono}} = \frac{I_{\text{mono}(121)}}{I_{\text{mono}(121)} + I_{\text{tetra}(200)}} \times 100$$

Here  $\eta_{\text{mono}}$ ,  $I_{\text{mono}(121)}$ , and  $I_{\text{mono}(200)}$  represent the percentage of monoclinic phase in mixed phase BVO, the relative intensity of (121) peak for monoclinic phase, and the (200) peak for the tetrahedral phase, in that order. The calculated  $\eta_{\text{mono}}$  were found to be 66, 30.9, 17.4, and 29.2% for BVOMT, BVCN-25, BVCN-50, and BVCN-75 respectively. The experiments were repeated for three times and it was found that the decrease in percentage of monoclinic phase in BVCN composite is attributed to the surface energy of BCN. Here surface energy plays a pivotal role in creating the crystal structure of certain phase. The interfacial interaction between functionalized graphene<sup>52</sup> sheets and metal oxide nanoparticles is believed to control the nucleation energy for different phases of  $\text{TiO}_2$ . So we assume that the free amino group ( $-\text{NH}_2$ ) present on the edge of  $\text{g-C}_3\text{N}_4$  will interact strongly to the  $\text{Bi}^{3+}$  and controlled the growth of lattice by forming mixed phase BVO. Again due to the high temperature calcination of the material BVCN, few of the B could be inserted into BVO lattice during the nucleation process, which is responsible for the formation of mixed phase BVO. The assumption correlates to the formation of noisy peak in XPS spectrum of B 1s from 188 to 196 eV which relates to the B–N coordination *vide infra*. Again crystallite size of as synthesized samples were calculated using Scherrer's equation<sup>53</sup> and found to be 15.9, 50.3, 29.8, 4.6, 27.5, 36.4, and 37.3 nm for BVOM, BVOT, BVOMT, BCN, BVCN-25, BVCN-50, and BVCN-75, in that order. FT-IR were obtained in order to explore the molecular structure<sup>54</sup> of the as synthesized materials. Analogous to BCN, the additional heterostructures have prominent vibration peak of triazine unit around  $800 \text{ cm}^{-1}$  and stretching mode of aromatic heterocycles<sup>23</sup> around  $1200$ – $1600 \text{ cm}^{-1}$  (Figure S1a). The peak around 990, 825, and  $733 \text{ cm}^{-1}$  links to the symmetric and asymmetric stretching vibrations of V–O bond. The band around  $530 \text{ cm}^{-1}$  is attributed to the stretching vibration of Bi–O bond. The peaks from  $3000$  to  $3600 \text{ cm}^{-1}$  links to the stretching mode of vibration of  $-\text{NH}_2$  and  $-\text{OH}$  groups

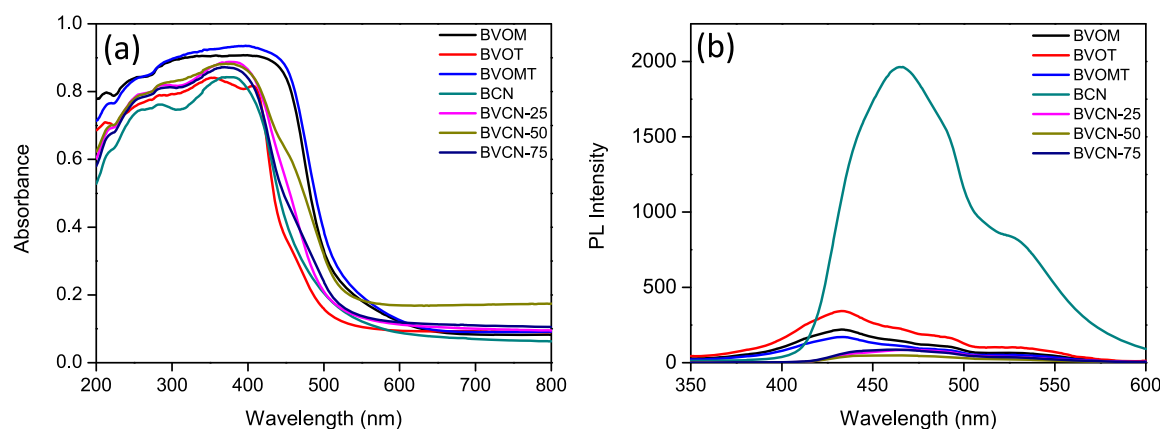


**Figure 2.** XPS spectra of BVCN (a) C 1s, (b) N 1s, (c) B 1s, (d) O 1s, (e) Bi 4f, (f) V 2p.

having hydrogen bonding. Again Raman spectroscopy<sup>55,56</sup> has been conducted in order to elucidate the structure and bonding of the constituents of heterostructure through their characteristic vibration and shown in Figure S1b. The monoclinic phase of BVOM<sup>57,58</sup> exhibits five noticeable peaks at 829.6, 734.9, 342.0, 223.3, and 130.3  $\text{cm}^{-1}$  which are attributed to the vibrational features of  $\text{VO}_4$  tetrahedron.<sup>59</sup> The intense peak at 829.6  $\text{cm}^{-1}$  and weak peak at 734.9  $\text{cm}^{-1}$  are assigned to the shorter symmetric and long asymmetric V–O stretching mode of vibration with  $A_g$  and  $B_g$  symmetry in that order. The peak present at 341.3  $\text{cm}^{-1}$  resembles to asymmetric bending vibration of the  $\text{VO}_4$  tetrahedron. The external mode including rotation/vibration of BVO is assigned to the peaks at 223.3 and 130.3  $\text{cm}^{-1}$ . Interestingly the peak at 829.6  $\text{cm}^{-1}$  has been shifted to lower frequency<sup>60</sup> in case of composite and is ascribed at 828.0, 825.0, and 828.7  $\text{cm}^{-1}$  for BVCN-25, BVCN-50, and BVCN-75 respectively. Such results correspond to the elongation of V–O bond leading to structural distortion which occurs due to the modification of electronic band structure of BVOM in the heterostructure. The shifting of symmetric stretching mode of vibration to a lower frequency is more in case of BVCN-50 and intensity also decreases to a larger extent indicating that high degree of crystallinity and oxygen vacancy in the system. This result can be corroborated with PXRD and XPS analysis and links to the higher photocatalytic activity by BVCN-50 as compared to other synthesized catalysts.

**Morphology Study.** High resolution transmission electron microscopy (HRTEM) has been done so as to appreciate the morphological<sup>61</sup> as well as crystalline<sup>62</sup> nature of the photocatalyst. The Figure 1b reveals the formation of cashew nut shaped BVOs which are well dispersed on BCN nanosheet. The dark and light gray area in TEM images represents BVO and BCN respectively. The BCN not only acts as support but also restricts the agglomeration of BVO that relates to the extraordinary photocatalytic activity of the materials. Again the

lattice fringes with  $d$  spacing value 0.31 and 0.36 nm are attributed to the (121) and (200) lattice planes of both monoclinic and tetragonal phase of BVO shown in Figure 1c. The polycrystalline nature of the material is yet again confirmed from selected area electron diffraction (SAED) pattern of BVCN which consists of numbers of diffraction rings (Figure 1d). The concentric rings are assigned as 1, 2, 3, 4, 5, 6, 7, 8, and 9 which represent (200), (121), (211), (002), (051), (213), (420), (123), and (224) lattice planes of BVCN as revealed from Figure 1d. The (200), (211), (213), (420), and (224) planes represent the tetragonal phase and (121), (051), and (123) link to the monoclinic phase of BVO whereas the (002) crystal plane stands for BCN. The TEM observation confirmed the formation of a tight heterojunction between the mixed phase BVO and BCN in BVCN and the existence of both monoclinic and tetragonal phase of BVO, which is in accordance with PXRD data. FESEM were performed in order to explore the morphology of BVO, BCN, and BVCN-50. From the Figure S2 it is clear that spherical shape BVOM (Figure S2a) has been formed where as in case of BCN (Figure S2c), crumpled nanosheet with worm-hole mesoporosity is present. When spherical BVO is deposited on to the BCN nanosheet, wrapping of nanosheet takes place to form BVCN-50 (Figure S2e) p–n heterojunction. The EDX spectra show the successful formation of BVO, BCN and BVCN-50 (Figure S2, parts b, d, and f). Elemental mapping shows the uniform distribution of constituent atoms of BVCN-50 (Figure S3). Further, In order to relate the surface area with the photocatalytic activity of the photocatalysts, nitrogen adsorption–desorption isotherm has been performed for BVO-MT, BCN and BVCN-50. From the Figure S4, it is revealed that all photocatalysts show type-IV isotherm according to BDDT classification, which relates to the presence of mesopores with 2–50 nm. The H3 type hysteresis loop suggested the formation of worm-hole mesoporosity, which links to the excellent photocatalytic activity by the photocatalyst. The pore



**Figure 3.** (a) UV-vis spectra and (b) photoluminescence spectra of BVOM, BVOT, BVOMT, BCN, BVCN-25, BVCN-50, and BVCN-75.

volume of BVO-MT, BCN, and BCN-50 was found to be 0.0316, 0.0635, and 0.0655 cm<sup>3</sup>/g. The specific surface area of BVO-MT, BCN, and BVCN-50 were 7.62, 10.55, and 19.76 m<sup>2</sup>/g. The excellent photocatalytic activity of BVCN-50 for O<sub>2</sub> evolution and Cr(VI) reduction may be due to its highest surface area.

**X-ray Photoelectron Spectroscopy.** To explore the surface elemental composition and chemical status of as synthesized photocatalyst, X-ray photoelectron spectroscopy<sup>23,63</sup> measurement has been carried out. The XPS survey spectra of photocatalyst consists of Bi, V, O, C, N, and B elements, the primary constituents of the photocatalyst. The C 1s, Bi 4f, V 2p, O 1s, N 1s, and B 1s core-level could be easily observed in the XPS survey spectra (Figure S5f). Additionally the chemical statuses of the elements were analyzed by their analogous high resolution XPS spectra and fitted by means of CASA. The peak positions of all the XPS spectra were calibrated with C 1s with 284.7 eV. High resolution C 1s core (Figure 2a) of BVCN consist of three distinct peaks at 284.7, 285.4, and 288.4 eV that are attributed to carbide, unintended carbon species<sup>64</sup> from the XPS instrument itself, and sp<sup>2</sup>-hybridized carbon attached to nitrogen of the aromatic ring (N-C=N) of g-C<sub>3</sub>N<sub>4</sub> framework respectively. Similarly in case of N 1s (Figure 2b) the peak present at 398.8 eV is ascribed to nitride or cyanide (N<sup>3-</sup> or -C≡N).<sup>38,65</sup> The peak at 400.0 eV corresponds to the sp<sup>2</sup> hybridized nitrogen attached to a carbon atom (C=N-C) and the peak at 403.8 eV associate with the tertiary nitrogen relates to N-(C)<sub>3</sub> group existing in the aromatic ring of carbon nitride. The core level of B 1s (Figure 2c) was found to be noisy due to its low atomic percent doping in carbon nitride.<sup>23</sup> To confirm the successful boron doping, XPS spectra of neat BCN has been done and represented in Figure S5a–e. High resolution O 1s XPS spectrum (Figure 2d) of BVCN has 2 peaks positioned at 530.9, 532.3 eV corresponds to the V–O bond and oxygen vacancy<sup>60,66</sup> in BVO lattice respectively. The O 1s peak at 532.3 eV is related to oxygen vacancy in crystal lattice due to high temperature calcination of the material. The core level of Bi 4f<sup>65,67</sup> (Figure 2e) spectrum shows two peaks having binding energy at 159.9 and 165.2 eV which corresponds to Bi 4f<sub>7/2</sub> and Bi 4f<sub>5/2</sub> in that order. Similarly V 2p core level (Figure 2f) XPS spectrum is fitted into two characteristics peaks, positioned at 517.9 and 525.0 eV link to V 2p<sub>3/2</sub> and V 2p<sub>1/2</sub> in that order.

**Optical Properties.** The photophysical<sup>68</sup> characteristics of all photocatalysts were studied using UV-DRS and photo-

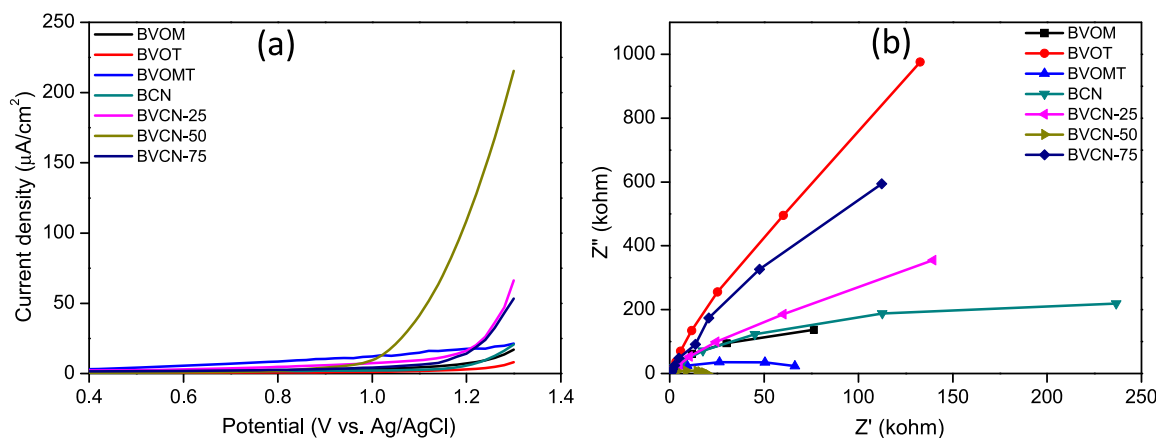
luminescence spectroscopy. From the Figure 3a, it is clear that all photocatalysts were able toward absorption of visible light of solar spectrum. It is interesting to note that BVCN-50 has more absorption tail as compared to other photocatalysts. BCN, BVOM, BVOT, BVOMT, BVCN-25, BVCN-50, and BVCN-75 absorb visible light with onset potential of 430, 530, 458, 534, 544, 600, and 551 nm. Upon light illumination, electrons of BVO are excited from the valence band containing O 2p into the conduction band of V 3d orbitals.<sup>69</sup> The enhancement in the absorption tail of BVCN-50 suggests it to be the foremost photocatalyst compared to all. Furthermore, the band gap of the as synthesized materials was calculated using Tauc's plot (Figure S6a).<sup>23</sup>

$$\alpha h\nu = A(h\nu - E_g)^n$$

Here  $\alpha$  stands for the absorption coefficient,  $\nu$  is the frequency of light,  $A$  is proportionality constant and  $E_g$  links to the band gap energy. The value of  $n$  in Tauc's equation relates to the kind of transition in semiconductor photocatalyst ( $n = 1/2$  for direct transition and  $n = 2$  for indirect transition). By extrapolating the linear portion of the  $(\alpha h\nu)^2$  curve vs  $h\nu$ , the value of  $E_g$  for different photocatalysts were determined. The band gap for BCN, BVO, BVCN-50 were 2.8, 2.5, and 2.6 eV in that order. The optical evidence suggests that BVCN-50 has improved solar light utilization capability as compared to other photocatalysts.

**Photoluminescence Spectroscopy.** In order to survey the separation efficiency of photogenerated charge carrier<sup>70</sup> and the tendency of electron hole pairs recombination on the surface as well as at the interface of photocatalysts, photoluminescence spectroscopy was done for all photocatalysts at an excitation of 330 nm and depicted in Figure 3b. It is clear from the figure that a substantial decrease in PL intensity takes place while going from BCN to BVO to BVCN series. Higher PL intensity stands for the higher extent of radiative process related to charge pair recombination. The emission intensity of BCN is broad and centered at 465 nm and is attributed to that band to band transition.<sup>24</sup> Similarly, the cases of BVOM, BVOT, and BVOMT show emission intensity at 432 and 530 nm, which was corroborated by earlier reports (Figure S6b). This double band emission<sup>71</sup> is related to the recombination of electron in the V 3d band with the hole in the O 2p band and Bi<sup>3+</sup> to V center charge transfer, respectively. The same figure also described that the emission peaks are at 469, 464, and 460 nm for BVCN-25, BVCN-50, and BVCN-75, in that order. A significant decrease in PL intensity in the case of BVCN-50





**Figure 4.** (a) Linear sweep voltammetry and (b) electrochemical impedance spectra of all photocatalysts.

indicates the rate of electron–hole recombination is lowest as compared to others and hence related to the highest photocatalytic activity by BVCN-50.

**Photoelectrochemical Properties.** Photoelectrochemical analyses of as designed photocatalysts were examined by a standard three electrode system by using 0.2 M  $\text{Na}_2\text{SO}_4$  solution as electrolyte and depicted in Figure 4a. As expected from a n-type semiconductor,<sup>12</sup> BVO electrode acts as photoanode which generates anodic photocurrent by using the photoinduced holes for water oxidation at the semiconductor-electrolyte interface. All photocatalysts show almost negligible current (nA range) in the dark (Figure S7a). Upon light illumination, BVOM and BCN show 8.7 and 20.4  $\mu\text{A}$  current in anodic direction. The higher photocurrent generation by BCN as compared to BVOM is attributed to the thin nanosheet nature of BCN where electrons are channelized more easily. Since BCN is identified to improve electron transportation in a composite due to the large quantities of delocalized electrons from the conjugated  $\text{sp}^2$  hybridized carbon network.<sup>21</sup> Similarly photocurrent generation by BVOT and BVOMT was found to be 8.7 and 20.8  $\mu\text{A}$ . BVCN-25 and BVCN-75 generate 65.7 and 53.4  $\mu\text{A}$  current in the anodic direction upon light irradiation. More interestingly, BVCN-50 shows 215  $\mu\text{A}$  anodic current at a potential of 1 V, and the overpotential<sup>72</sup> decreases up to 0.24 V as compared to other photocatalysts. The abrupt decrease in the overpotential is attributed to the staggered energy band and formation of high quality heterostructures between BVOMT and BCN. The catalyst BVCN-50 shows highest oxygen evolution activities with a strong negative shift in the onset potential, which links to the lower surface recombination.<sup>73</sup> The higher photocurrent relates to the better charge separation efficiency of BVCN-50. The catalyst has more ability for charge separation and generates 30- and 10.5-fold photocurrent as compared to neat BVO and BCN respectively. The highest photocurrent and negative shifting in overpotential relates to the superior photocatalytic activity by BVCN-50. Moreover, the photoelectrochemical reversibility of the electrode was assessed from transient photocurrent response analysis and depicted in Figure S7b. A sharp increment in the photocurrent response was observed upon light illumination, which relapses back to the original state under dark. Among the various photocatalysts examined in this study, BVCN-50 exhibits the highest photocurrent that shows 2.5 times higher than BCN and 3.5-

fold higher than BVOM. These photoelectrodes retain excellent reversibility and photostability of the materials.

The interfacial properties between the electrode (BCN, BVOM, BVOT, BVOMT, BVCN-25, BVCN-50, and BVCN-75) and electrolyte were examined by electrochemical impedance spectroscopy<sup>74</sup> measurement. The Nyquist plot generally consists of semicircle<sup>75</sup> (i.e., arch in the present study), where  $x$  and  $y$  axes represents the real part ( $Z'$ ) and negative of imaginary part ( $Z''$ ). The semicircle generally represents the charge-transfer process, while the diameter of the semicircle links to the charge transfer resistance. It is well established that the large value of impedance signifies the poor conductivity of electron across the electrode and electrolyte interface.<sup>23</sup> From the Figure 4b, it is clear that the charge transfer resistance for BVOM, BVOT, BVOMT, BCN, BVCN-25, BVCN-50, and BVCN-75 were 76.4, 132.7, 66.2, 113.0, 139.5, 19.1, and 112.5  $\text{k}\Omega$  respectively. The smallest diameter in case of BVCN-50 is attributed to the lesser charge transfer resistance due to the tremendous electrical conductivity by the material. Again the staggered energy band and formation of high quality heterostructures in case of BVCN-50 are responsible for the excellent electrical conducting nature, which makes it the best photocatalyst among all synthesized materials. Mott–Schottky measurement was implemented in 0.2 M  $\text{Na}_2\text{SO}_4$  (pH = 6.1) so as to know the band edge potential and type of the semiconductor<sup>76,23</sup> photocatalyst and is depicted in Figure S8. Parts a and b of Figure S8 reveal that straight lines having positive slope is extrapolated to the potential axis in case of both monoclinic and tetragonal phase of BVO, representing the materials to be intrinsic n-type<sup>12</sup> semiconductors. Similarly, a negative slope is drawn to the potential axis for BCN (Figure S8c), which indicate the intrinsic p-type character of the material. The CB position of n-type and VB position of p-type semiconductors were evaluated using following equation.

$$E_{fb(\text{vs.NHE})} = E_{fb(\text{pH}=0, \text{vs. Ag/AgCl})} + E_{\text{AgCl}} + 0.059 \text{ pH}$$

The CB potential for both monoclinic and tetragonal BVO was +0.3 and −0.1 eV, and for BCN, the VB potential was 2.0 eV. The corresponding VB potential for monoclinic and tetragonal BVO and CB position for BCN were calculated from UV–vis results by means of the equation  $E_{VB} = E_g + E_{CB}$ . Excitingly, in the case of BVCN-50 (Figure S8d), an inverted U shaped curve is obtained, and both positive and negative slope can be plotted which are located at different potential regimes. This is

a symbolic case for the coexistence of together p- and n-type properties in the photocatalyst<sup>23</sup> where a suitable p–n photochemical heterojunction is established at the interface of BVO and BCN. The Fermi level of both p- and n-type domains occurs at the interface, creating a built-in electric field. The built-in electric field is responsible for the migration of photogenerated electrons in opposite directions and improves the separation of electron hole pairs and ultimately triggers the photocatalytic activity.

**Photocatalytic Activity.** In addition to the application of water in photooxidation, an excellent photocatalyst should simultaneously have a photoreduction ability. The health hazard to both humans and animals is due to the presence of heavy metal ions like Cr(VI) in aquatic bodies causing pollution problems. Here, we have chosen Cr(VI) as a model pollutant for photocatalytic reduction as it is not only carcinogenic and mutagenic to humans but also highly toxic<sup>77</sup> due its mobility in aqueous medium. For the reduction experiment, 50 mg of catalyst was placed in 20 mL (20 ppm, pH = 2, adjusted by adding 0.1 M HCl) of Cr(VI) solution and sonicated for 10 min. The solutions were stirred in the dark for 30 min to attain an equilibrium adsorption state prior to visible light illumination. Blank experiment without photocatalyst reveals the negligible photolysis of Cr(VI) solution under light irradiation. The rate constant for Cr(VI) reduction reaction by all synthesized catalysts are depicted in the Table 1. From

**Table 1. First Order Fitting Results of Cr(VI) Reduction over All Photocatalysts**

catalyst	R <sup>2</sup>	k <sub>obs</sub> (min <sup>−1</sup> )	t <sub>1/2</sub> (min)	% of degradation
BVOM	0.998	0.0115	60.26	17
BVOT	0.994	0.0081	85.56	15
BVOMT	0.998	0.0237	29.24	33
BCN	0.999	0.0433	16.01	75
BVCN-25	0.999	0.0469	14.77	78
BVCN-50	0.999	0.0533	13.00	87
BVCN-75	0.998	0.0311	22.28	66

Figure 5a, it is confirmed that after light illumination for 30 min, there was a significant decrease in Cr(VI) concentration for BVOM (17%), BVOT (15%), BVOMT (33%), BCN (72%), BVCN-25 (77%), BVCN-50 (85%), and BVCN-75 (66%). The decrease in photocatalytic activity in case of BVCN-75 is attributed to the presence of large quantity of BVO on the surface of BVCN. A higher percentage of BVO in BVCN-75 retards the light penetration ability of the composite and also restricts the interfacial charge transfer. To explore the kinetics of Cr(VI) reduction, experimental data were fitted with a first order model by using the following formula.

$$\ln(C/C_0) = -kt$$

Here C<sub>0</sub> and C are the initial and final concentration after time t, and k is the rate constant for the reduction reaction. The rate constant follows the order BVOT < BVOM < BVOMT < BVCN-75 < BCN < BVCN-25 < BVCN-50 (Figure 5b). The rate constant for BVCN-50 is 7.9-fold higher than that of BVO. Figure 5c represents the spectral changes of Cr(VI) solution after different time intervals over BVCN-50. The pH of the solution has a significant role for the absorption and reduction of Cr(VI) to Cr(III). In addition to this, the Cr(VI) reduction reaction has been performed in river water where the experimental conditions were kept the same as in the case of

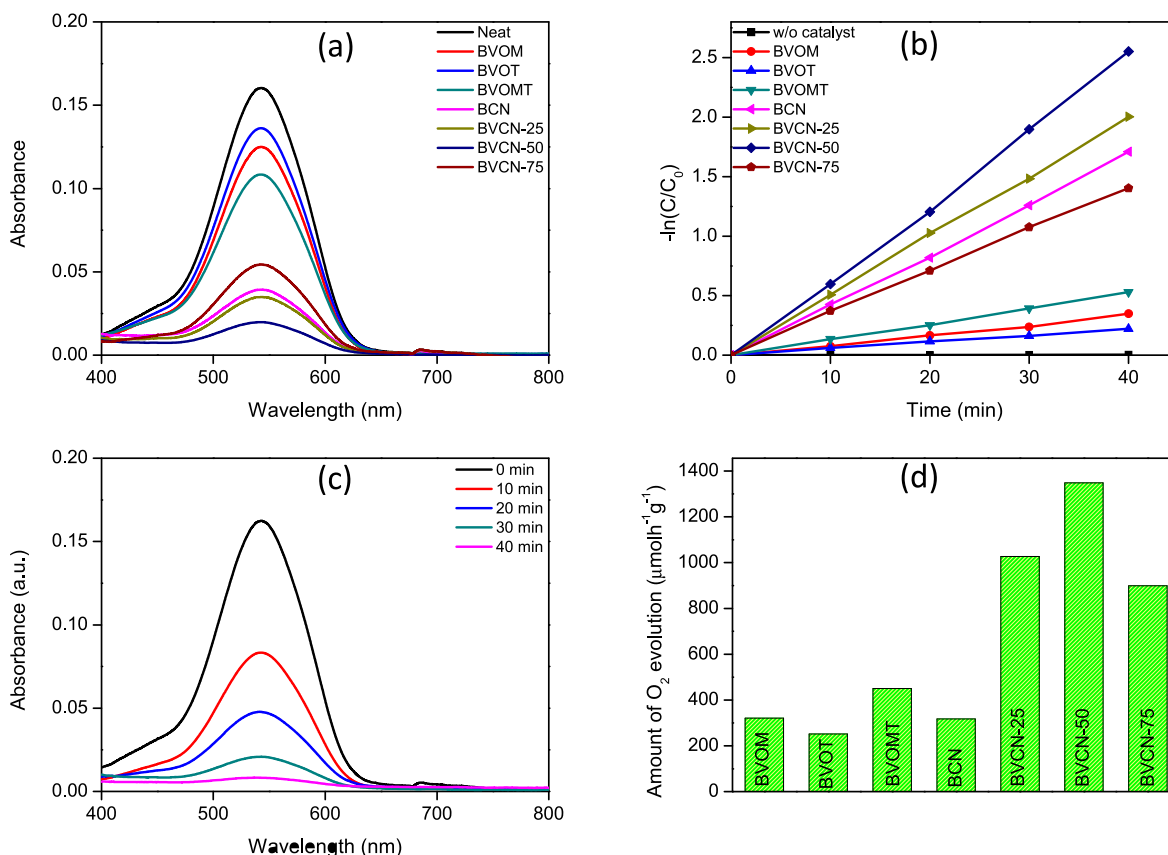
distilled water. From Figure S9a,b, it is clear that, after light illumination, the Cr(VI) concentration decreases significantly for BVOM (14%), BVOT (11%), BVOMT (25%), BCN (73%), BVCN-25 (75%), BVCN-50 (80%), and BVCN-75 (60%). The decrease in the percentage of Cr(VI) reduction in river water as compared to distilled water may be attributed to the presence of different ions, which interfere during the adsorption and photoreduction process. The percentage of Cr(VI) reduction in wastewater by the foremost catalyst BVCN-50 was found to be 62%, and the decrease in photocatalytic activity as compared to in distilled and river water medium is attributed to the presence of a large numbers of interfering ions.

The photocatalytic reduction process was found to be decreased upon increasing the pH of the solution.<sup>78</sup> The surface of the photocatalyst was positively charged at lower pH due to protonation and attracts negatively charged HCrO<sub>4</sub><sup>−</sup> ions, subsequently facilitating the adsorption process. The same reaction has been performed at different pHs and is depicted in Figure S10a,b, and the highest reduction rate was found at pH = 2. This is attributed to the adsorption of greater numbers of protons on the surface of the photocatalyst, increasing its proton exchange capacity. Upon increase in pH, the HCrO<sub>4</sub><sup>−</sup> species is converted to more negatively charged species like CrO<sub>4</sub><sup>2−</sup> and Cr<sub>2</sub>O<sub>7</sub><sup>2−</sup>. Therefore, the reduction rate decreases significantly due to the formation of negatively charged surface, which causes repulsion between the catalyst surface and HCrO<sub>4</sub><sup>−</sup> ion and decreases the adsorption rate. So as to know the photostability of the catalyst, repeated experiments were performed with an interval of 40 min and results were shown in Figure S10c.

The photocatalytic water splitting reaction has been achieved in a 100 mL sealed quartz batch reactor and analyzed by gas chromatography under visible light illumination. Both BVOM and BCN were able to generate similar oxygen evolution rates, 321 and 318 μmol h<sup>−1</sup>g<sup>−1</sup>, respectively, whereas in the case of BVOT and BVOMT, the oxygen production rates were 256 and 452.8 μmol h<sup>−1</sup>g<sup>−1</sup> (Figure 5d). BVOM has a band gap of 2.4 eV, which is able to absorb visible light, but the photocatalytic water oxidation has been inhibited due to its poor hole mobility. The oxygen evolution rate of BCN is akin to BVOM even though the band gap is 2.8 eV for BCN which links to the better charge mobility.<sup>33,38</sup> The composite heterojunction photocatalyst have improved charge separation and migration which results in enhancement in the photocatalytic activity. The oxygen evolution rate of BVCN-25, BVCN-50, and BVCN-75 were 1027.2, 1348.2, and 898.8 μmol h<sup>−1</sup>g<sup>−1</sup>. The oxygen evolution activity by BVCN-50 is 4.2 times higher than that of BVO whereas photocatalytic activity decreases in case of BVCN-75. Repeated experiments were performed in order to know the photostability of the catalyst with an interval of 3 h and depicted in Figure S10d. After reusability test of the photocatalyst, the crystal structure and morphology of BVCN-50 remain intact as revealed from PXRD (Figure S11) and FESEM image (Figure S12). Decrease in the photocatalytic activity by BVCN-75 is attributed to agglomeration of BVO due to its higher content that causes blocking of light. Therefore, BVCN-50 is regarded as a saturated composite catalyst between BVO and BCN. For comparative study, various BVO-based p–n heterojunction photocatalysts and their activities were tabulated (Table S1).

**Mechanism.** The improved photocatalytic activity of the heterostructure can be described by assuming the formation of

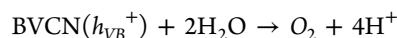
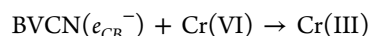




**Figure 5.** (a) UV-vis spectra of Cr(VI) reduction over all synthesized photocatalysts. (b) First order kinetics plot. (c) spectral changes of Cr(VI) solution after different time interval over BVCN-50. (d) Photocatalytic oxygen evolution by all synthesized photocatalysts.

p–n heterojunction. In order to study the type of semiconductor, Mott–Schottky analysis was performed, and from Figure S8, it is confirmed that the slope of the linear  $1/C^2$  potential curve of BVO and BCN were positive and negative. The results demonstrate that BCN has an intrinsic p-type<sup>33</sup> characteristic whereas BVO an intrinsic n-type.<sup>12</sup> The Fermi level ( $E_F$ ) of BCN (p-type) remains close to the VB, and for BVO (n-type), the Fermi level ( $E_F$ ) is close to CB.<sup>79</sup> When BVO is assembled on to the BCN surface (Scheme S1), a type-II p–n heterojunction is formed. In order to establish equilibrium in the  $E_F$  value of both semiconductors<sup>80</sup> forming p–n heterojunction, the energy levels of BVO move downward and energy levels of BCN shift upward. Ultimately, the conduction band edge potential of BVO is lower as compared to BCN. Upon visible light irradiation, both BVO and BCN in BVCN heterostructure could simultaneously be excited to form electron–hole pairs. Alongside this, the internal electric field is also responsible for the improvement in the immigration of photoinduced charge carriers. Due to the presence of both monoclinic and tetragonal phase, the photoinduced electrons from the CB of tetragonal phase can easily channelized in to the CB of monoclinic phase facilitating reduction reaction. Consequently, the BVCN p–n heterojunction not only promotes the separation of the photoinduced electron–hole pairs but also reduces the recombination of charge carriers, thus enhancing its photocatalytic activity. The excited electron can easily migrate from the CB of BCN to the CB of BVO where it reduces Cr(VI) to Cr(III). Simultaneously holes get transferred from VB of BVO

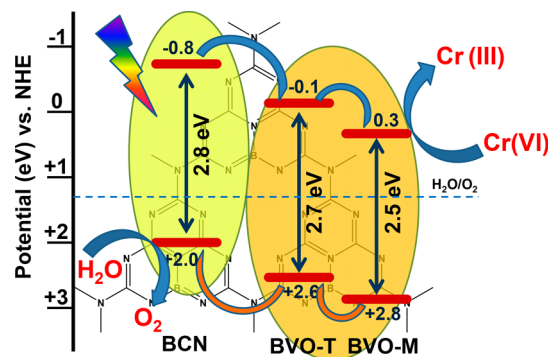
to the VB of BCN and oxidize  $H_2O$  to  $O_2$  (Scheme 2). The molecular reaction mechanism is summarized below.



## CONCLUSION

Graphitic-like boron-doped  $g\text{-C}_3\text{N}_4$  nanosheet was fabricated and  $\text{BiVO}_4$  nanostructure deposited on to it to construct p–n heterojunction photocatalyst for effective simultaneous reduction of Cr(VI) and oxidation of  $H_2O$  to generate  $O_2$ .

### Scheme 2. Plausible Mechanism for Photocatalytic Cr(VI) Reduction and Oxygen Evolution on BVCN-50



Monoclinic and tetragonal mixed phase  $\text{BiVO}_4$  is obtained due to thermal calcination of the catalyst at 400 °C as well as surface energy of BCN and confirmed from PXRD and HRTEM data. The best photocatalytic activity of 50%  $\text{BiVO}_4/\text{B-doped g-C}_3\text{N}_4$  for the reduction of  $\text{Cr(VI)}$  is 3.9-fold higher than that of  $\text{BiVO}_4$ . Again the photocatalyst is able to generate 4.2 times higher oxygen evolution rate as compared to  $\text{BiVO}_4$ . The excellent enhancement in the photocatalytic activity of  $\text{BiVO}_4/\text{B-doped g-C}_3\text{N}_4$  is ascribed to its strong absorption of visible light and low recombination rate of electron–hole pairs due to the formation of type-II p–n heterojunction between  $\text{BiVO}_4$  and B-doped  $\text{g-C}_3\text{N}_4$ . The built in electrostatic field at the junction is also responsible for the generation of excellent photocurrent by the heterojunction as compared to neat material. The results were corroborated by UV–vis, PL, photocurrent response, and electrochemical impedance spectroscopy. Moreover oxygen vacancy in the photocatalyst is also accountable for the excellent photocatalytic activity by the heterostructure and is validated from XPS and Raman analysis. The photocatalyst also exhibits excellent photostability and recyclability. The present work validates a facile way for the construction of p–n heterojunctions with improved visible light driven photocatalytic properties.

## ■ ASSOCIATED CONTENT

### Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.inorgchem.9b02309.

FTIR, Raman, FESEM, and EDS, elemental mapping of BVCN-50, surface area of BVOMT, BCN, and BVCN-50, XPS spectra of BCN, Tauc's plot and PL spectra of all photocatalysts, current density in dark and transient photocurrent density, Mott–Schottky plot, UV–vis spectra of  $\text{Cr(VI)}$  reduction and first order kinetics plot in river water, effect of pH on  $\text{Cr(VI)}$  reduction, reusability curve of  $\text{Cr(VI)}$  reduction and photocatalytic  $\text{O}_2$  evolution, PXRD plot and FESEM plot after reusability test, literature survey based on various p–n heterojunctions, formation mechanism of p–n heterojunctions, and probable charge separation process in BVCN heterostructure (PDF)

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### Notes

The authors declare no competing financial interest.

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