# Synthesis of polyynes to model the *sp*-carbon allotrope carbyne

Wesley A. Chalifoux and Rik R. Tykwinski<sup>†\*</sup>

Carbyne is an allotrope of carbon composed of *sp*-hybridized carbon atoms. Although its formation in the laboratory is suggested, no well-defined sample is described. Interest in carbyne and its potential properties remains intense because of, at least in part, technological breakthroughs offered by other carbon allotropes, such as fullerenes, carbon nanotubes and graphene. Here, we describe the synthesis of a series of conjugated polyynes as models for carbyne. The longest of the series consists of 44 contiguous acetylenic carbons, and it maintains a framework clearly composed of alternating single and triple bonds. Spectroscopic analyses for these polyynes reveal a distinct trend towards a finite gap between the highest occupied molecular orbital and the lowest unoccupied molecular orbital for carbyne, which is estimated to be  $\sim$ 485 nm ( $\sim$ 2.56 eV). Even the longest members of this series of polyynes are not particularly sensitive to light, moisture or oxygen, and they can be handled and characterized under normal laboratory conditions.

By mass, carbon is the fourth most-abundant element in the universe, and it is essential to all known forms of life<sup>1</sup>. In terms of naturally occurring carbon, typically two forms (allotropes) are encountered, namely diamond ( $sp^3$ -hybridized carbon) and graphite ( $sp^2$ -hybridized carbon). In each case, it is the bulk properties of these allotropes that are of interest, properties that differ quite dramatically for each allotropic form. For example, diamond is the hardest known naturally occurring material and also a good electrical insulator. Graphite, however, is much softer and shows high conductivity<sup>2</sup>. Carbon allotropes discovered more recently, such as fullerenes, nanotubes and graphene (all composed of  $sp^2$ -hybridized carbon) each also show unique properties, often different from those of either diamond or graphite, and they offer the promise of technological breakthroughs<sup>3</sup>.

Next in the series of carbon-atom hybridization from  $sp^3$ - to  $sp^2$ is *sp*-hybridization and the least well-known of the carbon allotropes, carbyne (Fig. 1). Whereas diamond and graphite feature an atomic carbon skeleton with covalent bonding in three and two dimensions, respectively, carbyne is expected to feature a linear or approximately linear one-dimensional framework. Carbyne is proposed to exist in interstellar dust<sup>4</sup>, meteorites<sup>5</sup> and as a by-product of shock-fused graphite<sup>6</sup>. There are reports of the laboratory formation of carbyne by a number of routes, including gas-phase deposition techniques, electrochemical synthesis, dehydrohalogenation of polymers and others<sup>7,8</sup>. In all these cases, the resulting product or material is ill-defined, and the specific properties of carbyne thus remains unrealized.

In the absence of a well-defined or indisputable sample of carbyne, synthetic chemists addressed the question of its potential properties through the formation of homologous series of polyyne molecules (Fig. 1)<sup>9</sup>. More specifically, this involved the synthesis of oligomers of defined length and composed only of acetylenic segments that are end-capped with a sterically bulky group designed to stabilize the polyyne chain. For a particular set of oligomers, convergence of spectroscopic data versus length to a constant value (saturation) can be evaluated to shed light on the expected properties of the polymer carbyne. A notable example is that of the triethylsilyl end-capped series TES[n] (Fig. 1), which provided isolable

samples for n = 6 and 8 (ref. 10). The attempted synthesis of longer derivatives, however, gave **TES**[10], **TES**[12] and **TES**[16], which could be manipulated only in solution, but the longest molecule, **TES**[16], could not be purified<sup>10</sup>. Polyynes **Pt**[*n*] were isolated and characterized up to n = 14; **Pt**[14] is the longest isolable polyyne reported to date<sup>11</sup>.

Recognizing that large, sterically demanding end-capping groups should provide synthetic access to much longer polyynes, we used the tris(3,5-di-*t*-butylphenyl)methyl moiety (Tr\*, Fig. 1)<sup>12</sup> and report here the synthesis of polyynes with unprecedented length: the series **1a–1j** is composed of polyynes of up to n = 22 acetylenic bonds (that is, 44 contiguous *sp*-carbons). Structure–property analyses for polyynes **1a–1j** provide predictions as to the highest occupied molecular orbital (HOMO)–lowest unoccupied molecular orbital (LUMO) gap and structure of carbyne.



Figure 1 | Schematic structures of carbyne, polyyne and several polyyne species.

Department of Chemistry, University of Alberta, Edmonton, Alberta, T6G 2G2 Canada; <sup>†</sup>Present address: Institut für Organische Chemie, Friedrich-Alexander-Universität, Erlangen-Nürnberg, Henkestrasse 42, 91054 Erlangen, Germany. \*e-mail: rik.tykwinski@chemie.uni-erlangen.de

Tr*	H + Me <sub>3</sub> Si $- \frac{1}{b}$ Sii-	$\Pr_{3} \xrightarrow[MeOH/CH_{2}Cl_{2}]{Cu(OAc)_{2} \cdot H_{2}O}{(K_{2}CO_{3})}$	$\operatorname{Tr}^{*} - \left[ _{c} \operatorname{Sii-Pr}_{3} \right]_{c}$
Entry	Substrate 2	Substrate 3	4, % yield
1	<b>2b</b> ( <i>a</i> = 2)	<b>3c</b> ( <i>b</i> = 3)	<b>4e</b> (c = 5), 84
2	<b>2c</b> ( <i>a</i> = 3)	<b>3c</b> ( <i>b</i> = 3)	<b>4f</b> ( <i>c</i> = 6), 63
3	<b>2c</b> ( <i>a</i> = 3)	<b>3d</b> ( <i>b</i> = 4)	<b>4g</b> (c = 7), 73
4	<b>2d</b> ( <i>a</i> = 4)	<b>3d</b> ( <i>b</i> = 4)	<b>4h</b> (c = 8), 77
5	<b>2e</b> ( <i>a</i> = 5)	<b>3d</b> ( <i>b</i> = 4)	<b>4i</b> (c = 9), 42
6	<b>2f</b> ( <i>a</i> = 6)	<b>3d</b> ( <i>b</i> = 4)	<b>4j</b> (c = 10), 52
7	<b>2g</b> ( <i>a</i> = 7)	<b>3d</b> ( <i>b</i> = 4)	<b>4k</b> (c = 11), 35

#### **Results and discussion**

The formation of the synthetic precursors to polyynes 1a-1j is designed around a chain-extension sequence using the terminal polyynes 2b-2g in a reaction with differentially end-capped polyynes 3c and 3d, where *a* and *b* represent the number of acetylene units in precursors 2 and 3, respectively (Table 1). The building blocks 2b-2d and 3c and 3d were assembled readily based on reported methods (see Supplementary Information (pp S12-S18) and Jahnke and Tykwinski<sup>13</sup>), and the synthesis of 2e-2g is described below. This chain extension was accomplished efficiently using a modified Eglinton-Galbraith protocol to generate unsymmetrical  $Tr^{*}-[C\equiv C]_{c}$ -Sii-Pr<sub>3</sub> polyynes (4e-4k)<sup>14,15</sup>. Specifically, allowing a mixture of the appropriate terminal polyyne **2b-2g** (a = 2-7) and either **3c** (b = 3) or **3d** (b = 4) to react in the presence of excess Cu(OAc)<sub>2</sub>·H<sub>2</sub>O, K<sub>2</sub>CO<sub>3</sub> and 2,6-lutidine resulted in the formation of polyynes 4e-4k in good to moderate yields (Table 1, entries 1-4 and entries 5-7, respectively). Three notable aspects of this process are: (1) removal of the trimethylsilyl groups of 3c and 3d was effected in situ under the Eglinton-Galbraith reaction conditions, which eliminated the need for an independent desilylation step<sup>16</sup>, (2) use of 2,6-lutidine in place of the typical base, pyridine, was essential to the success of the reaction and (c) 4i, 4j and 4k represent the longest unsymmetrical polyynes isolated and characterized to date.

Removal of the terminal *i*-Pr<sub>3</sub>Si groups from polyynes 4c-4k to give terminal polyynes 2c-2k (Table 2) was first explored using tetrabutylammonium fluoride as a desilylation agent at low temperature. Under these conditions 4c was desilylated successfully, but significant decomposition was observed for longer derivatives. Ultimately, desilylation was achieved using CsF in a 5:1 mixture of tetrahydrofuran (THF)/H<sub>2</sub>O, which provided polyynes 2d-2k with little apparent decomposition based on ultraviolet–visible spectroscopic analyses (see Supplementary Fig. S1).

The stabilizing influence of the Tr\* end-capping group was already quite evident at this point, in that terminal polyynes up to seven triple bonds in length (**2g**) could be isolated neat and characterized by <sup>1</sup>H and <sup>13</sup>C NMR, infrared and ultraviolet–visible spectroscopy, as well as by mass spectrometry (although within minutes the terminal polyyne **2g** showed signs of decomposition as a solid). To our knowledge, the longest terminal polyyne previously isolable as a stable, neat solid was a tetrayne<sup>17</sup>. Given the instability observed for **2g** in the solid state, attempts were not made to isolate the longer terminal polyynes **2h–2k**. Thus, after formation by desilylation, the products **2g–2k**, after work up, were taken on directly to the formation of **1f–1j**. Polyynes **2h–2k** could, however, be obtained pure in solution and characterized by high-resolution matrix-assisted laser desorption/ ionization mass spectrometry, as well as by ultraviolet–visible and infrared spectroscopy.

Oxidative dimerization of precursors **2b-2k** was the final step towards the synthesis of polyynes **1a-1j** (Table 2). Standard Hay

Table 2	Synthesis	of polyynes	1a-1j (	n = <b>4-22)</b> .
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Tr*	$\frac{\text{Desilylation}}{\text{conditions}}$ Tr* $\frac{1}{1}$	$_{a}$ H $\frac{\text{Dimerization}}{\text{conditions}}$ Tr* $\frac{1}{1}$ Tr*
4	2	1
4	2, yield, %	1, yield, %
-	<b>2b</b> $(a=2)^{\dagger}$	<b>1a</b> (n = 4), 92 <sup>¶</sup>
<b>4c</b> (c = 3)*	<b>2c</b> ( <i>a</i> = 3), 91 <sup>‡</sup>	<b>1b</b> $(n = 6), 66^{\text{T}}$
<b>4d</b> $(c = 4)^*$	<b>2d</b> ( <i>a</i> = 4), 100 <sup>§</sup>	<b>1c</b> ( <i>n</i> = 8), 97 <sup>¶</sup>
<b>4e</b> (c = 5)	<b>2e</b> (a = 5), 99 <sup>§</sup>	<b>1d</b> ( <i>n</i> = 10), 26 <sup>#</sup> , 74 <sup>**</sup>
<b>4f</b> ( <i>c</i> = 6)	<b>2f</b> ( <i>a</i> = 6), 93 <sup>§</sup>	<b>1e</b> ( <i>n</i> = 12), 41 <sup>#</sup> , 86**
<b>4g</b> (c = 7)	<b>2g</b> $(a=7)^{\$\parallel}$	<b>1f</b> ( <i>n</i> = 14), 90**
<b>4h</b> (c = 8)	<b>2h</b> (a = 8) <sup>§∥</sup>	<b>1g</b> ( <i>n</i> = 16), 76**
<b>4i</b> (c=9)	<b>2i</b> ( <i>a</i> = 9) <sup>§∥</sup>	<b>1h</b> ( <i>n</i> = 18), 81**
<b>4j</b> (c = 10)	<b>2j</b> (a = 10) <sup>§∥</sup>	<b>1i</b> ( <i>n</i> = 20), 57**
<b>4k</b> (c = 11)	<b>2k</b> (a = 11) <sup>§</sup> ∥	<b>1j</b> ( <i>n</i> = 22), 18**

\*The synthesis of **4c** and **4d** is described in the Supplementary Information (pp S15–S17). <sup>†</sup>Diyne **2b** was produced by a different method, see Supplementary Information (pp S12–S13). <sup>†</sup>Desilylation using tetrabutylammonium fluoride, THF, room temperature. <sup>§</sup>Desilylation using CsF in tetrahydrofuran/H<sub>2</sub>O (5:1), room temperature. <sup>II</sup>Product carried on directly to next step. <sup>§</sup>Oxidative dimerization using CuCl, tetramethylethylenediamine, CH<sub>2</sub>Cl<sub>2</sub>, O<sub>2</sub>. <sup>#</sup>Oxidative dimerization using CuClAc)<sub>2</sub>:H<sub>2</sub>O (excess), pyridine, CH<sub>2</sub>Cl<sub>2</sub>. \*\*Oxidative dimerization using CuOAc)<sub>2</sub>:H<sub>2</sub>O (excess), 2,6-lutidine, CH<sub>2</sub>Cl<sub>2</sub>.

coupling conditions  $^{18}$  (CuCl, tetramethylethylenediamine,  $\rm CH_2Cl_2,$  $O_2$ ) readily converted **2b-2d** into tetrayne **1a**, hexayne **1b** and octayne 1c, respectively. These conditions were not, however, particularly successful when applied to the conversion of 2e and 2f into decayne 1d and dodecayne 1e, respectively. In these cases, the formation of side products rendered purification and isolation of the desired polyynes difficult. Switching to the typical Eglinton-Galbraith coupling conditions (Cu(OAc)<sub>2</sub>·H<sub>2</sub>O, pyridine, THF) gave decayne 1d and dodecayne 1e without the troublesome by-products encountered with the Hay protocol, but only in modest isolated yields of 26% and 41%, respectively. By using CH<sub>2</sub>Cl<sub>2</sub> as the reaction solvent and the less nucleophilic base 2,6-lutidine, however, the yields of 1d and 1e were improved significantly. Gratifyingly, this set of conditions was generally applicable, and terminal polyynes 2g-2k thus provided the remaining members of the series, 1f-1j. The isolated polyynes span a range of colours, starting with tetrayne 1a, which is colourless. Hexaand octaynes 1b and 1c are both yellow, and decayne 1d marks the transition from yellow to orange. Polyynes 1e-1h are all isolated as orange solids and the longest derivative, 1j, is orange-red in colour.

Ultraviolet-visible spectroscopy is a primary tool for polyyne characterization because these molecules typically display wellresolved spectra with distinctive vibrational fine structures<sup>19</sup>. Furthermore, the lowest energy wavelength of significant absorption,  $\lambda_{\max}$ , shows a steady shift to lower energy as a function of polyyne length, and this trend documents the lowering of the HOMO  $\rightarrow$  LUMO bandgap energy ( $E_{\rm g}$ ) as the conjugation length of the molecule increases. Numerous studies used  $\lambda_{\max}$  values for shorter polyynes with n = 2-14 as a predictive tool for carbyne based on the premise that the HOMO  $\rightarrow$  LUMO bandgap should eventually reach an asymptotic limit, which would be a reasonable estimate of  $\lambda_{\text{max}}$  (and thus  $E_{g}$ ) for carbyne. Typically, a plot of  $E_{g}$ versus 1/n was examined (where *n* is the number of acetylene units) and extrapolation to the y-intercept then provided an estimate of  $\lambda_{sat}$  for an infinite polyme chain (that is,  $\lambda_{sat} = \lambda_{max}$  at the saturation length of the series). For example, this analysis was reported for the series *i*-Pr<sub>3</sub>Si-[C=C]<sub>n</sub>-Si*i*-Pr<sub>3</sub> (n = 2-10) and predicted a limiting value of  $\lambda_{sat} = 570$  nm (ref. 20), nearly identical to the value predicted for aryl end-capped polyynes of the same lengths  $(\lambda_{\rm sat} = 570 \text{ nm})^{21}$ . Similarly, a plot of  $E_{\rm g}$  versus 1/n reported for the organometallic series Pt[n] up to n = 14 predicted a limiting value of  $\lambda_{sat} = 573$  nm (ref. 11), which is somewhat higher than that determined for an earlier series of Pt-terminated polyynes up to  $n = 12 \ (\lambda_{sat} \approx 492 - 527 \text{ nm})^{22}$ .

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**Figure 2** | Spectroscopic and crystallographic characterization of polyynes 1a–1j. a, Ultraviolet-visible spectra of 1a–1j polyynes as measured in hexanes; wavelength of  $\lambda_{max}$  for each derivative is shown in blue and vibrational separations of lowest energy absorptions for 1j are shown in magenta. a.u. = arbitrary units. **b**, Convergence of the absorption maxima in the series of 1a–1j polyynes according to equation (1), where  $\lambda_{\infty} = 486$  nm,  $\lambda_1 = 175$  nm and a = 0.116. **c**, <sup>13</sup>C NMR spectrum (in CDCl<sub>3</sub>) of 1j in the range 69–57 ppm, which shows convergence of the *sp*-carbon resonances to a median value of ~63.7 ppm (outlying signals at 68.66 and 57.38 ppm arise from the endmost acetylenic carbons). **d**, X-ray crystallographic structures for 4**f** and 1**b** shown as space-filling models (left) and ORTEP representations (right, 20% probability level, hydrogen atoms not shown); selected bond lengths (Å) for 4**f**: C(1)–C(2) 1.213(3), C(2)–C(3) 1.357(4), C(3)–C(4) 1.209(3), C(4)–C(5) 1.360(4), C(5)–C(6) 1.211(3), C(6)–C(7) 1.358(3), C(7)–C(8) 1.202(3), C(8)–C(9) 1.360(4), C(9)–C(10) 1.210(3), C(10)–C(11) 1.364(3), C(11)–C(12) 1.196(3); selected bond lengths (Å) for 1**b**: C(2)–C(3) 1.192(3), C(3)–C(4) 1.363(3), C(4)–C(5) 1.205(3), C(5)–C(6) 1.353(3), C(6)–C(7) 1.201(3), C(7)–C(7') 1.362(5).

The ultraviolet–visible spectra for **1a–1j** are plotted in Fig. 2a, and show the expected vibrational fine structure (for example, spacing of 1,534 and 1,695 cm<sup>-1</sup> for the three lowest energy absorptions of **1j**). The bathochromic shift in  $\lambda_{max}$  as a function of length spans from  $\lambda_{max} = 268$  nm for **1a** to  $\lambda_{max} = 458$  nm for **1j**. The extinction coefficients for  $\lambda_{max}$  increase almost uniformly as a function of length with values that approach one million for **1i** ( $\varepsilon = 866,000$  at  $\lambda_{max} = 451$  nm and  $\varepsilon = 936,000$  at  $\lambda = 420$  nm).

Following the usual procedure (see above), a plot of the solution optical bandgap  $E_{\rm g}$  versus 1/n gave a good fit to the data for series **1a–1j** and predicted a limiting value of  $\lambda_{\rm sat} \approx 564$  nm (2.20 eV; see Supplementary Fig. S2). There is, however, a potential problem with this estimation, as previously documented by Meier *et al.* for other rigid, conjugated oligo- and polymers<sup>23</sup>. A plot of  $E_{\rm g}$  versus 1/n assumes that  $E_{\rm g}$  continues to decrease uniformly as a function of length through to infinity. It is well-established,

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nential function (equation (1)).

$$\lambda_{(n)} = \lambda_{\infty} - (\lambda_{\infty} - \lambda_1) e^{-a(n-1)}$$
(1)

where *n* is again the number of acetylene units,  $\lambda_{(n)}$  is  $\lambda_{\max}$  for a polyyne of length *n* ( $\lambda_1$  is thus  $\lambda_{\max}$  of the monomer with *n* = 1, calculated according to equation (1)) and  $\lambda_{\infty}$  is the limiting value as  $n \to \infty$ . The parameter *a* indicates how fast saturation (convergence) is approached, and it can also be used to compare convergence characteristics between different oligomer series. Finally, a practical definition is required that relates the approach to  $\lambda_{\infty}$  and  $\lambda_{\text{sat}}$ , and the suggestion of Meier is adopted here<sup>23</sup>, taking into account the accuracy of the ultraviolet–visible spectrometer ( $\pm 1$  nm):  $\lambda_{\text{sat}}$  is defined by fulfilment of the relationship  $\lambda_{\infty} - \lambda_{(n)} \leq 1$  nm.

Applying the  $\lambda_{\text{max}}$  data for **1a–1j** into equation (1) generated the parameters  $\lambda_{\infty} = 486 \pm 5$  nm,  $\lambda_1 = 175 \pm 13$  nm and  $a = 0.116 \pm 100$ 0.008 (for details and error analysis, see Supplementary Table S2). Figure 2b shows the convergence of  $\lambda_{max}$  values for **1a-1j** based on equation (1) and predicts a  $\lambda_{sat}$  for carbyne of 485 nm (2.56 eV) by about n = 48. This  $\lambda_{sat}$  value is substantially higher in energy than that predicted by the linear extrapolation method of  $E_g$  versus 1/n ( $\lambda_{sat} \approx 564$  nm, 2.20 eV). Interestingly, the low value of a = 0.116 shows that saturation for polyynes was reached quite slowly in comparison to that for other conjugated oligomers, which probably reflects that effective conjugation for polyynes is not dependent on molecular conformation<sup>23</sup>. Given the extensive set of  $\lambda_{\text{max}}$  data for polyyne series **1a–1j**, we believe this to be the most accurate estimation to date for the finite HOMO-LUMO gap of carbyne. This analysis also provides excellent evidence that a polyyne structure of alternating single and triple bonds will be maintained for carbyne, that is, Peierls distortion will be upheld<sup>25,26</sup>.

All synthesized polyynes 1a-1j were sufficiently stable and soluble for <sup>13</sup>C NMR spectroscopic analysis, which provided additional insight into the prospective properties of carbyne (Table 3). <sup>13</sup>C NMR spectroscopy confirmed that the longest member of the series, 1j, was still a polyyne rather than carbyne; quite amazingly, 21 of the 22 resonances for unique sp-carbons were observed (two overlapping signals were found at 64.27 parts per million (ppm)). The two endmost alkyne carbons experienced the greatest influence from the end-capping group and appeared at  $\sim$ 69 and  $\sim$ 57 ppm. The remaining carbon signals resonated within a narrow window of 62.10-64.60 ppm and converged towards a median value of 63.7 ppm (Fig. 2c and Supplementary Fig. S3). Thus, these data suggest that the <sup>13</sup>C NMR spectrum of carbyne probably consists of a broadened signal centred at a chemical shift of  $\sim$ 63.7 ppm, a value quite consistent with that expected from a polyynic framework composed of alternating single and triple bonds. This proposed chemical shift is consistent with other predictions<sup>11,20–22</sup>, albeit more precise.

X-ray crystallographic analysis of **4f** and **1b** offered an opportunity to examine structural aspects and a means to visualize the steric shielding that results from the presence of the Tr\* groups (Fig. 2d). Molecule **4f** adopts a gentle bend in the solid state, and the bond angles for centrosymmetric **1b** show an S-shaped conformation; in both cases the C-C=C bond angles range from about 175 to  $179^{\circ}$ . Clearly evident in the structures is the dramatic difference

## Table 3 | <sup>13</sup>C NMR alkyne carbon resonances for polyynes 1a-1j.

Compound	Alkyne carbon chemical shifts in $CDCI_3$ (ppm)
1a	69.06, 63.00, 62.36, 57.19
1b	68.78, 62.96, 62.90, 62.75, 62.42, 57.29
1c	68.65, 63.55, 63.49, 62.97, 62.60, 62.50, 62.19, 57.33
1d	68.60, 63.96, 63.78, 63.49, 63.11, 62.69, 62.54, 62.28,
	62.06, 57.36
1e	68.57, 64.17, 63.92, 63.77, 63.49, 63.18, 62.86, 62.52, 62.50,
	62.17, 61.99, 57.36
1f	68.57, 64.28, 64.00, 63.93, 63.69, 63.45, 63.24, 63.01,
	62.74, 62.48, 62.44, 62.12, 61.96, 57.37
1g	68.56, 64.34, 64.05, 64.01, 63.81, 63.61, 63.45, 63.29, 63.12,
	62.93, 62.69, 62.48, 62.41, 62.10, 61.95, 57.37
1h	68.55, 64.36, 64.06, 64.04, 63.85, 63.68, 63.55, 63.42,
	63.31, 63.19, 63.05, 62.87, 62.65, 62.47, 62.39, 62.08, 61.94,
	57.37
1i	68.63, 64.50, 64.19, 64.19, 64.03, 63.88, 63.77, 63.67,
	63.58, 63.51, 63.43, 63.33, 63.20, 63.03, 62.81, 62.54, 62.54,
	62.20, 62.03, 57.39
1j	68.66, 64.60, 64.27, 64.27, 64.14, 64.00, 63.91, 63.83,
	63.77, 63.72, 63.66, 63.61, 63.54, 63.44, 63.33, 63.15, 62.93,
	62.64, 62.58, 62.29, 62.10, 57.38

in size between the Tr\* (diameter,  $d \approx 14$  Å) and *i*-Pr<sub>3</sub>Si ( $d \approx 7$  Å) groups. Thus, whereas the *i*-Pr<sub>3</sub>Si groups do little to shield the polyyne chain, the Tr\* groups extend well out over the *sp*-carbon framework, and offer significantly more steric shielding against the prospect of intermolecular interactions that reportedly lead to decomposition (schematically shown in red, Fig. 2d)<sup>9,27,28</sup>. Finally, analysis of bond lengths reveals that C–C and C=C bonds for both structures remain within the expected ranges<sup>26–28</sup>.

#### Conclusions

The experimental data reported here are thus consistent with the prediction of carbyne as a polyyne-like material with a structure composed of alternating single and triple bonds with a finite bandgap of  $\lambda_{\text{max}} = 485$  nm (2.56 eV). The successful synthesis of the longer derivates **1i** and **1j** as persistent molecules is significant, and suggests that longer polyynes, and even carbyne, are viable targets for stepwise synthesis.

#### Methods

**General procedure: modified Eglinton–Galbraith coupling.** To a solution of 2 (0.297 mmol) and 3 (1.49 mmol) in CH<sub>2</sub>Cl<sub>2</sub> (10 ml), MeOH (10 ml) and 2,6-lutidine (10 ml) was added K<sub>2</sub>CO<sub>3</sub> (1.49 mmol) and the mixture was stirred for 10 minutes at room temperature. To the mixture was added Cu(OAc)<sub>2</sub>-H<sub>2</sub>O (2.97 mmol) and the reaction was stirred at room temperature until deemed complete by thin-layer chromatography (TLC) (~17 hours). The reaction was quenched by the addition of saturated aqueous NH<sub>4</sub>Cl (25 ml) and the resulting mixture extracted with hexanes (100 ml). The organic phase was washed with saturated aqueous NH<sub>4</sub>Cl (25 ml), and brine (25 ml), dried over MgSO<sub>4</sub> and filtered. The resulting solution was filtered through a plug of silica, the solvent removed *in vacuo* and the crude product purified by column chromatography (silica, hexanes) to yield 4.

**General procedure: desilylation using CsF.** To a solution of 4 (1.288 mmol) in THF (100 ml) and H<sub>2</sub>O (20 ml) was added CsF (1.54 mmol). The reaction was stirred at room temperature until deemed complete by TLC analysis (~1 hour). The reaction was quenched by the addition of saturated aqueous NH<sub>4</sub>Cl (50 ml) and the resulting mixture extracted with hexanes (100 ml). The organic layer was washed with H<sub>2</sub>O (50 ml) and brine (50 ml), dried over MgSO<sub>4</sub> and filtered. The solvent was removed *in vacuo* and the crude product purified by passing through a plug of silica gel with CH<sub>2</sub>Cl<sub>2</sub>/hexanes (1:5) to yield 2 (*a* = 3–6).

**General procedure: formation of 1f–1j** ( $n \ge 14$ ). To a solution of 4 (0.10 mmol) in THF (25 ml) and H<sub>2</sub>O (5 ml) was added CsF (0.12 mmol) and the reaction monitored by TLC for completion (~20 minutes). The reaction was quenched by the addition of saturated aqueous NH<sub>4</sub>Cl (15 ml) and the resulting mixture extracted with hexanes (50 ml). The organic layer was washed with H<sub>2</sub>O (15 ml) and brine (15 ml), dried over MgSO<sub>4</sub> and filtered. The resulting solution was concentrated to

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 ${\sim}5$  ml, and the crude intermediate 2 was passed through a plug of silica gel with CH<sub>2</sub>Cl<sub>2</sub>/hexanes (1:5). The filtrate was concentrated to  ${\sim}25$  ml and diluted with CH<sub>2</sub>Cl<sub>2</sub> (25 ml). Cu(OAc)<sub>2</sub>:H<sub>2</sub>O (0.510 mmol) and 2,6-lutidine (1.0 ml) were added and the reaction mixture was stirred at room temperature for 24 hours. The reaction mixture was concentrated to  ${\sim}5$  ml, the crude product purified by passing through a plug of silica gel with CH<sub>2</sub>Cl<sub>2</sub>/hexanes (1:10) and the solvent removed *in vacuo* to yield **1**.

**Solid-state analyses.** Single crystals of **4f** suitable for X-ray crystallographic analysis were grown by slow evaporation of a CDCl<sub>3</sub>/hexanes (1:1) solution at room temperature.  $C_{70}H_{98}$ Si,  $M_r = 967.57$ , triclinic crystal system, space group  $P\bar{1}$  (No. 2), a = 10.2575(7) Å, b = 14.6634(11) Å, c = 21.9595(16) Å,  $\alpha = 76.520(1)^{\circ}$ ,  $\beta = 89.555(1)^{\circ}$ ,  $\gamma = 86.272(1)^{\circ}$ , V = 3,205.0(4) Å<sup>3</sup>, Z = 2,  $\rho_{calcd} = 1.003$  g cm<sup>-3</sup>,  $\mu = 0.073$  mm<sup>-1</sup>,  $\lambda = 0.71073$  Å, T = -100 °C,  $2\theta_{max} = 50.50^{\circ}$ , total data collected = 22,875,  $R_1 = 0.0635$  for 6,600 observed reflections with  $[F_o^2 \ge 2\sigma(F_o^2)]$ , w $R_2 = 0.1884$  for 586 variables and 11,578 unique reflections with  $[F_o^2 \ge -3\sigma(F_o^2)]$ , residual electron density = 0.407 and -0.258 e Å<sup>-3</sup>. CCDC no. 772622.

Single crystals of **1b** suitable for X-ray crystallographic analysis were grown by slow evaporation of a xylene solution at room temperature.  $C_{98}H_{126}$ ,  $M_r = 1,303.99$ , triclinic crystal system, space group  $P\overline{1}$  (No. 2), a = 10.5799(10) Å, b = 10.7732(10) Å, c = 19.6616(18) Å,  $\alpha = 84.7319(14)^\circ$ ,  $\beta = 75.3405(14)^\circ$ ,  $\gamma = 87.4321(14)^\circ$ , V = 2,158.4(3) Å<sup>3</sup>, Z = 1,  $\rho_{calcd} = 1.003$  g cm<sup>-3</sup>,  $\mu = 0.056$  mm<sup>-1</sup>,  $\lambda = 0.71073$  Å, T = -100 °C,  $2\theta_{max} = 50.76^\circ$ , total data collected = 15,603,  $R_1 = 0.0538$  for 4,479 observed reflections with  $[F_o^2 \ge 2\sigma(F_o^2)]$ ,  $wR_2 = 0.1535$  for 486 variables and 7,893 unique reflections with  $[F_o^2 \ge -3\sigma(F_o^2)]$ , residual electron density = 0.408 and -0.195 e Å<sup>-3</sup>. CCDC no. 772621.

The diameters of the endgroups for **4f** and **1b** were estimated from the radius as measured from the central carbon or silyl atom of the end-capping group to the outermost proton, based on the X-ray crystallographic structures, using Mercury CSD 2.2.

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#### Author contributions

W.A.C. designed the experiments, and performed the syntheses, characterization and property studies. W.A.C. and R.R.T co-wrote the paper. R.R.T conceived the project.

#### Additional information

The authors declare no competing financial interests. Supplementary information and chemical compound information accompany this paper at www.nature.com/ naturechemistry. Reprints and permission information is available online at http://npg.nature.com/reprintsandpermissions/. Correspondence and requests for materials should be addressed to R.R.T.