Accepted Manuscript

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PII: S0925-8388(18)30338-4

DOI: 10.1016/j.jallcom.2018.01.324

Reference: JALCOM 44788

To appear in: Journal of Alloys and Compounds

Received Date: 17 January 2017

Revised Date: 23 January 2018

Accepted Date: 25 January 2018

Please cite this article as: F. Huang, R. Meng, Y. Sui, F. Wei, J. Qi, Q. Meng, Y. He, One-step hydrothermal synthesis of a CoS₂@MoS₂ nanocomposite for high-performance supercapacitors, *Journal of Alloys and Compounds* (2018), doi: 10.1016/j.jallcom.2018.01.324.

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One-step hydrothermal synthesis of a CoS₂@MoS₂ nanocomposite for high-performance supercapacitors

Feng Huang ^a, Rongde Meng ^b, Yanwei Sui ^a, Fuxiang Wei ^a, Jiqiu Qi ^a, Qingkun

Meng^a, Yezeng He^{a,*}

^a School of Materials Science and Engineering, China University of Mining and Technology, Xuzhou 221116, People's Republic of China

^b Beijing Experimental School, Haidian, Beijing 100037, People's Republic of China

Abstract

A MoS₂-coated CoS₂ composite has been prepared successfully via a one-step hydrothermal method. This CoS₂@MoS₂ hybrid material has a distinct morphology with few-layer MoS₂ wrapping the CoS₂ nanoparticles. Serving as a supercapacitor electrode, the obtained CoS₂@MoS₂ hybrid material exhibits a remarkable specific capacitance of 1038 F/g, a high rate capability of 71.7%, and excellent cycling stability of 84.76% retention after 10000 cycles. The superior electrochemical properties of the CoS₂@MoS₂ nanocomposite are attributed to the synergistic effects between the layered MoS₂ and conductive CoS₂.

Key words: MoS₂; CoS₂; flower-like structure; supercapacitors

1. Introduction

Supercapacitors have attracted worldwide attention due to their excellent properties, such as rapid recharge ability, high power density, long cycle life, eco-friendly nature and low-cost preparation [1-3]. There are two major types of supercapacitors that are related to the different charge storage mechanisms. The first type is a double-layer capacitor, which accumulates charge at the electrode-electrolyte interface by establishing a Helmholtz double-layer [4-7], and the second type is a pseudo-capacitor based on a redox capacitive mechanism [8-11].

Graphene, a two-dimensional carbon allotrope, has attracted considerable attention due to its fascinating properties and broad applications in nanoelectronic devices since its discovery in 2004 [12]. However, graphene materials also have several disadvantages such as high costs, low yield and poor scalability [13,14]. MoS₂ *Corresponding author, E-mail: hyz0217@hotmail.com

is a transition metal sulfide with a layered structure. The ultrathin thickness and 2D morphology enable MoS₂ to achieve ultrahigh specific surface areas compared to its bulk counterpart [15,16]. Mo atoms can easily prompt redox faradaic reactions in the interlayer space due to oxidation states spanning a range from +2 to +6, and its theoretical specific capacitance could be as high as 1000 F/g [17,18]. Therefore, the layered structure and redox faradaic reactions play important roles in making the MoS₂ a promising electrode material used in the supercapacitors. To date, many studies regarding the use of MoS₂ in supercapacitors have been reported. For example, Ramadoss et al. prepared a mesoporous MoS₂ nanostructure by a hydrothermal method and obtained maximum capacitance values of 376 and 403 F/g at a scan rate of 1 mV/s in 1 M Na₂SO₄ and KCl electrolyte solutions, respectively [19]. Wang et al. synthesized hierarchical MoS₂ nanospheres for supercapacitors by a hydrothermal method and demonstrated a capacitance of 142 F/g at a current density of 0.59 A/g [20]. Zhou et al. reported flower-like MoS₂ nanospheres that were synthesized by a hydrothermal route and showed specific capacitances of 122 F/g at 1 A/g or 114 F/g at 2 mV/s [21]. Krishnamoorthy et al. synthesized few-layered MoS₂ nanosheets by using a mechanical milling method and demonstrated a MoS₂ based wire type solid state supercapacitors device with a specific capacitance of 119 μ F/cm⁻¹ [22]. Nevertheless, the preparation of a high-performance supercapacitors using MoS₂ as an electrode material is still a challenging task due to the low electrical conductivity and low number of accessible active sites in MoS₂.

To overcome these difficulties, MoS_2 should be combined with a conductive material with high electrochemical activity to enhance the electrical conductivity and increase content of accessible active sites [23,24]. As an important transition metal sulfide, CoS_2 has the best conductivity and thermal stability among the entire cobalt sulfide family, such as CoS, Co_3S_4 , Co_4S_3 , Co_2S_3 and Co_9S_8 [25,26]. CoS_2 has a superior capacitive behavior because its negative Gibbs free energy value causes a spontaneous thermodynamic reduction reaction that is useful for pseudo-capacitance supercapacitors [27,28]. Xing et al. have successfully prepared a type of octahedral-shaped CoS_2 with a specific capacitance of 236.5 F/g at 1 A/g in 2 M KOH

[29]. Ren et al. synthesized CoS_2 nanowires deposited on a graphite disc using a hydrothermal method with a specific capacitance of 828.2 F/g at 10 mV/s in 6 M KOH [30].

In this work, we synthesized a novel $CoS_2@MoS_2$ nanocomposite by a facile one-step hydrothermal technique. The $CoS_2@MoS_2$ nanocomposite electrode shows a high specific performance of 1038 F/g at 1 A/g in 2 M KOH solution. In addition, the $CoS_2@MoS_2$ nanocomposite electrode showed excellent cyclic stability and a high rate capability contrasted with pure MoS_2 and pure CoS_2 . The capacitance remained approximately 84.76% after 10000 cycles of charge-discharge at a current density of 5 A/g. The synergistic effect between MoS_2 and CoS_2 in the composite is expected to improve electrode performance when used as an electrode material for supercapacitors.

2. Experimental section

2.1 Materials

All the reagents used in this experiment were of analytical grade and were used without purification. Thiocarbamide (NH₂CSNH₂), sodium molybdatedihydrate (Na₂MoO₄·2H₂O), cobalt (II) nitrate hexahydrate (Co(NO₃)₂·6H₂O) and anhydrous ethanol (C₂H₆O) were purchased from Aladdin Industrial Corporation (ShangHai). The water was purified through a Lab Pure Water System (18.25 mΩ).

2.2 Synthesis of MoS₂, CoS₂ and CoS₂@MoS₂ nanocomposites

The $CoS_2@MoS_2$ nanocomposite was synthesized through a facile solvothermal method. Initially, 2.5 mmol Na₂MoO₄·2H₂O, 1.25 mmol Co(NO₃)₂·6H₂O (at a Mo:Co molar ratio of 2:1) and 6.25 mmol H₂NCSNH₂ were dissolved in a 30 ml mixed solution of DI water and anhydrous ethanol (1:1 in volume). The precursor solution was magnetically stirred by ultrasonication for 30 min. The mixture was then transferred into a Teflon-lined autoclave, sealed, and heated in an oven at 180 °C for 48 hours. After natural cooling to room temperature, the resulting black solid product was collected, then washed with DI water three times and ethanol three times before drying in vacuum at 70 °C (referred to as CM-2). The number "2" refers to the molar ratio of Na₂MoO₄·2H₂O and Co(NO₃)₂·6H₂O. CM-1 and CM-3 were synthesized by

the addition of the corresponding quantity of $Co(NO_3)_2 \cdot 6H_2O$ into the mixture with the same amount of $Na_2MoO_4 \cdot 2H_2O$ (2.5 mmol). As reference materials, pristine MoS_2 and CoS_2 were also prepared, following the same procedure in the absence of $Co(NO_3)_2 \cdot 6H_2O$ and $Na_2MoO_4 \cdot 2H_2O$, respectively.

2.3 Characterization

The $CoS_2@MoS_2$ (CM-2) samples were characterized for structural properties using X-ray diffraction (D8 Advance, Bruker, Germany) with Cu-K α radiation in the angle range from 5°-85° at a scan rate of 8 °/min. The surface area and nitrogen absorption-desorption isotherm were analyzed using an ASAP 2020 analyzer. XPS spectra were recorded on ESCALAB 250Xi using Al-K α radiation as the X-ray source. The morphology and structure of the products were characterized using field emission SEM (Hitachi SU-8000) equipped with high-resolution TEM (HRTEM, Tecnai G2F20).

2.4 Electrochemical measurements

Electrochemical measurements were performed on a CHI660D (Chenhua, Shanghai, China) electrochemical workstation in a 2 M KOH aqueous electrolyte solution using a three-electrode system that consists of a working electrode, a platinum plate counter electrode, and a saturated calomel electrode (SCE, reference electrode) at room temperature. The working electrode was prepared by mixing the as-prepared active material, acetylene black and polyvinylidene fluoride (PVDF) in N-methylpyrrolidinone (NMP) according to a ratio of 80:10:10. The resultant slurry was pasted on nickel foam $(1 \times 2 \text{ cm}^2)$ serving as a current collector, which was then dried at 80 °C for 12 h and pressed on 10 MPa. The mass of the activated material was obtained by a weight difference method using a sensitive microbalance. Finally, the loading masses of MoS₂, CoS₂ and CoS₂@MoS₂ on the Ni foam were 1.1, 0.78 and 0.9 mg/cm², respectively. Cyclic voltammetry measurements were carried out in a potential range from 0 to 0.65 V (vs. SCE) at different scan rates in a 2 M KOH electrolyte. Galvanostatic charge-discharge measurements were performed at different current densities (1, 2, 5, and 10 $A \cdot g^{-1}$) for the as-prepared electrodes. The specific capacitance can be calculated from the galvanostatic charge-discharge curve

according to Eq. (1)

$$C = \frac{I\Delta t}{m\Delta v} (1),$$

where C is the specific capacitance, I is the current, Δt is the discharge time, Δv is the potential window, and m is mass of the electroactive material. Electrochemical impedance spectroscopy (EIS) measurements were conducted by applying an AC voltage with 5 mV amplitude in the frequency range from 0.01 Hz to 100 KHz.

3. Results and discussion

The $CoS_2@MoS_2$ (CM-1, 2, 3) nanostructures were prepared using a low-temperature hydrothermal method. During the reaction, the active species (S²⁻ ions) are released from NH₂CSNH₂ according to Eq. (2). The S²⁻ ions react with dilute nitric acid to produce elemental sulfur (Eq. (3)). In addition, the elemental sulfur plays an important role in the chemical reactions (Eq. (4)). Lastly, the S²⁻ ions interacted with MoO₄²⁻ ions to produce MoS₂ (Eq. (5)).

$$H_2NCSNH_2 + H_2O \rightarrow 2NH_3 + 2H_2S + CO_2$$
(2)

$$6H_2S + 4HNO_3 \rightarrow 6S + 4NO + 5H_2O \tag{3}$$

$$H_2S + S + Co^{2+} \rightarrow CoS_2 + 2H^+$$
(4)

$$MoO_4^{2-} + H_2S \rightarrow MoS_2 + 3H_2 + SO_4^{2-}$$
 (5)

The crystal structure and phase purity of the as-prepared samples were characterized by XRD. Fig. 1a shows the XRD pattern of the as-prepared flowerlike $CoS_2@MoS_2$ nanostructured sample. The curve shows several sharp peaks at 14.3°, 29.2°, 32.6°, 39.5°, 49.7°, 52.08°, 58.43°, 60.1°, 68.4°, 72.7° and 75.9° which can be assigned to the (002), (004), (100), (103), (105), (018), (110), (008), (200), (203) and (116) planes of hexagonal MoS₂ (JCPDS 37-1942), respectively. The peaks at 32.3°, 36.23°, 39.8°, 46.3°, 60.2° and 69.8° correspond to the (200), (210), (211), (220), (230) and (410) planes of CoS_2 (JCPDS 65-3322), respectively [31,32]. Table 1 lists the textural properties (pore volume, BET surface area and pore size) of MoS_2 , $CoS_2@MoS_2$ and CoS_2 . Clearly, $CoS_2@MoS_2$ exhibits the largest pore volume, BET surface and pore size. Due to the presence of flowerlike MoS_2 , the BET surface of $CoS_2@MoS_2$ increases to 60.1 m²/g compared with a value of 51.2 for pure CoS_2 .

Moreover, the pores were mainly distributed in the range of 2-20 nm. Such a high-surface-area microstructure contains many active sites for electrochemical interactions with electrolyte ions, which is very useful for energy storage applications.

Materials	Pore volume (cm ³ /g)	BET surface area (m ² /g)	Pore size (nm)
MoS ₂	0.129	58.2	8.9
$CoS_2@MoS_2$	0.165	60.1	
(CM-2)			10.9
CoS_2	0.112	51.2	7.6

Table 1 Textural properties of MoS₂, CoS₂@MoS₂ and CoS₂.

XPS was carried out to obtain information regarding the structure and the electronic state of Mo, Co and S of the as-prepared nanocomposite. As shown in Fig. 2a, the survey scan of the $CoS_2@MoS_2$ nanostructured electrode exhibits characteristic peaks of the Mo 3d, Co 2p, S 2p, O 1s and C 1s levels at their corresponding binding energies. Fig. 2b shows that the high-resolution scans of the Mo 3d electrons exhibit two primary peaks at 228.9 and 232.2 eV that correspond to the Mo 3d_{3/2} and 3d_{5/2} in MoS₂, respectively, which further confirms the formation of MoS₂. The peak at 232.8 eV is attributed to the existence of a small amount of Mo⁶⁺ due to MoS₂ exposure to the air. The emergence of a peak at 226.0 eV corresponds to the S 2s orbitals of MoS₂. Fig. 2c shows two dominating peaks at 779.8 eV and 794.7 eV that correspond to Co 2p_{3/2} and 2p_{1/2}, respectively. The two peaks at 161.4 eV and 162.8 eV correspond to the S 2p_{3/2} and 2p_{1/2} levels of divalent sulfide ions (S²⁻). The peak at 163.4 eV indicates the presence of bridging S₂²⁻ or apical S²⁻ [33,34].

The morphologies of the $CoS_2@MoS_2$ nanocomposites (the samples are referred to as CM-1, CM-2 and CM-3) were examined by FESEM. As shown in Fig. 3, CM-2 presents a flowerlike morphology and is quite different from CM-1 and CM-3. In addition, inductively coupled plasma (ICP) analysis was conducted to determine the molar percentage between CoS_2 and MoS_2 in CM-1, CM-2 and CM-3, as shown in Table 2.

Samples Component	CM-1	CM-2	CM-3
MoS_2	42.9%	74.9%	87.5%
CoS_2	57.1%	25.1%	12.5%

Table 2. The molar content of MoS_2 and CoS_2 in CM-1, CM-2 and CM-3.

Fig. 4 shows the morphologies of MoS_2 , CoS_2 and $CoS_2@MoS_2$ (CM-2) nanocomposites. The images indicate that the $CoS_2@MoS_2$ nanocomposite has a similar morphology to MoS_2 . MoS_2 (Fig. 4a,b) and CM-2 (Fig. 4e,f) are both flowerlike, whereas CoS_2 is sphere-like. The size of the CM-2 nanocomposite is much smaller than that of either MoS_2 or CoS_2 . Hence, the CM-2 nanocomposite has a large surface area that provides plenty of active sites for an electrochemical reaction.

Fig. 5 shows the TEM images of the synthesized CM-2 composite. It can be seen that the MoS_2 sheets exhibit only a few layers located at the edge of CM-2 (Fig. 5a,b). The interconnected nanosheets of MoS_2 still maintain a flowerlike morphology, and several CoS_2 nanoparticles embedded among the flowerlike MoS_2 nanosheets are recognized in Fig. 5a (black area). The CoS_2 nanoparticles serve as substrates for the growth of MoS_2 nanoplates and are highly likely to increase the conductivity of the composites. The high-resolution TEM (HRTEM) images of CM-2 in Fig. 5 (c,d inset) demonstrate that the interlayer spacing of MoS_2 in the composite is 0.62 nm, and interplanar spacing of the CoS_2 sheets is 0.24 nm, which are both consistent with previous reports [35,36].

To explore the potential applications of the $CoS_2@MoS_2$ nanocomposite for supercapacitors, we have tested the electrochemical properties of the CM-1, CM-2 and CM-3 nanocomposite electrodes by cyclic voltammetry (CV) and galvanostatic charge/discharge (GCD) measurements, as shown in Fig. 6a,b. The CV curves of CM-1, CM-2 and CM-3 were obtained at a scan rate of 5 mV/s in the potential window of 0.0-0.65 V versus SCE in 2 M KOH. Fig. 6b illustrates the charge–discharge curves of CM-1, CM-2 and CM-3 in the potential range of 0–0.5 V at a current density of 1 A/g. The CM-2 nanocomposite shows the best charge storage properties. The specific capacitance can be calculated by Eq. (1), and the specific capacitance of the prepared CM-2 was calculated to be 1038 F/g, which is greater than that of both CM-1 (508 F/g) and CM-3 (468 F/g). Next, CM-2 was taken as an example for comparison with pure MoS₂ and pure CoS₂. As shown in Fig. 6c, the significant increase in the integrated area indicates that the composite has a higher capacity. For the curves of the CM-2 composite, the characteristic oxidation and reduction positions (oxidation peak around +0.52 V and reduction peak around +0.24 V) are related to faradaic redox reactions between the Co²⁺ and Co³⁺ states of cobalt in an alkaline KOH electrolyte [37,38]. The main reaction mechanism can be described as follows,

$$CoS_{2} + OH - \leftrightarrow CoS_{2}OH + H_{2}O + e^{-}$$

$$MoS_{2} + K^{+} + e^{-} \leftrightarrow MoS - SK^{+}$$

$$(6)$$

$$(7)$$

The supercapacitive characteristics can also be measured by a constant current charge-discharge cycling test. Fig. 6d shows the GCD curves for MoS₂, CoS₂ and CM-2 at a current density of 1 A/g. The specific capacitance values were calculated to be 116.8 and 464.4 F/g for the MoS₂ and CoS₂, respectively, while the specific capacitance of CM-2 composite attained 1038 F/g. Fig. 6e shows the specific capacitances of the MoS₂, CoS₂ and CM-2 electrodes as a function of the current density. The specific capacitance retention of CM-2 is higher than that of MoS₂ and CoS₂, which can be up to 71.7% when the current density changes from 1 A/g to 10 A/g. Typical impedance plots for this electrode are displayed in Fig. 7f. The equivalent series resistance (Rs) of the MoS₂, CoS₂@MoS₂ and CoS₂ electrodes are measured to be 0.56171, 0.34692 and 0.59254 Ω , respectively (Fig. 6f inset). The charge transfer resistance (Rct) of the MoS₂, CM-2 and CoS₂ electrodes are 0.8686, 0.1303 and 0.46718 Ω , respectively. These results indicate that the conductivity of CM-2 has increased significantly due to the existence of CoS₂ [39,40]. The smaller Rs and Rct values indicate a lower resistance and better capacitive behavior.

Cycle measurements were performed to further evaluate the electrochemical performance of the MoS_2 , CoS_2 and CM-2. The cycling stabilities of the MoS_2 , CoS_2

and the CM-2 electrodes were tested by repeating the GCD measurements between 0.0 and 0.5 V (vs. SCE) at a current density of 5 A/g for 10000 cycles. As shown in Fig. 7, the capacitance retention values of MoS_2 , CoS_2 and CM-2 in a 2 M KOH electrolyte are 57.02%, 80.71% and 84.76% after 10000 cycles, which indicates that CM-2 has the best cycling stability.

The excellent electrochemical properties of CM-2 are attributed to the synergistic effect of MoS_2 and CoS_2 : (1) Layered MoS_2 forms ion-buffering reservoirs to improve the ion diffusion rate within the prepared materials; (2) the nanoscale size of the CM-2 composite increases the specific surface area and greatly shortens the diffusion route through which both ions and electrons transfer during the charge-discharge process; (3) the presence of MoS_2 as a CoS_2 substrate in the composite system not only maintains a flower-like structure but also provides a path for the insertion and extraction of ions within the composite to increase the reaction rate; and (4) the conductive CoS_2 nanoparticles could efficiently lower the internal resistance and provide perfect continuous channels for electron transportation. Hence, the electrochemical performance of CM-2 is better than that of pure MoS_2 or CoS_2 due to the presence of two components with complementary properties.

4. Conclusions

In summary, the $CoS_2@MoS_2$ nanocomposite has been synthesized via a facile one-step hydrothermal method. In contrast to pure MoS_2 and pure CoS_2 , the specific capacitance and rate capability of the $CoS_2@MoS_2$ (CM-2) nanocomposite has been greatly enhanced. Furthermore, the $CoS_2@MoS_2$ (CM-2) nanocomposite displays an excellent long-time cycling stability, retaining 84.76% of the initial capacitance after 10000 cycles at a current density of 5 A/g. These excellent results indicate that the $CoS_2@MoS_2$ nanocomposite is a promising and suitable electrode material for high-performance supercapacitors.

Acknowledgments

The authors would like to acknowledge the support by the National Natural Science Foundation of China (Grant No. 51501221) and the China Postdoctoral Science Foundation (No. 2016M591954).

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Figure Captions

Fig. 1 (a) XRD pattern of the $CoS_2@MoS_2$ (CM-2) nanocomposite. N₂ sorption isotherms of (b) MoS_2 ; (c) $CoS_2@MoS_2$ (CM-2); and (d) CoS_2 . Inset: the pore size distribution curve.

Fig. 2 XPS spectra of the CoS₂@MoS₂ nanocomposite: (a) survey; (b) Mo 3d; (c) Co 2p; and (d) S 2p.

Fig. 3 SEM images of a,d) CM-1, b,e) CM-2, and c,f) CM-3.

Fig. 4 SEM images of (a,b) pure MoS_2 ; (c,d) pure CoS_2 ; and the (e,f) $CoS_2@MoS_2$ nanocomposite.

Fig. 5 TEM (a,b) and HRTEM(c,d) of the CoS₂@MoS₂ nanocomposite.

Fig. 6 (a) Cyclic voltammograms curves of CM-1, CM-2 and CM-3. (b) Galvanostatic charging/discharging curves of CM-1, CM-2 and CM-3. (c) Cyclic voltammograms of MoS_2 , CoS_2 and $CoS_2@MoS_2$. (d) Galvanostatic charging/discharging curves of MoS_2 , CoS_2 and $CoS_2@MoS_2$. (e) Rate capability curves of MoS_2 , CoS_2 and $CoS_2@MoS_2$. (e) Rate capability curves of MoS_2 , CoS_2 and $CoS_2@MoS_2$. (f) Nyquist plots of MoS_2 , CoS_2 and $CoS_2@MoS_2$, CoS_2 and $CoS_2@MoS_2$, CoS_2 and $CoS_2@MoS_2$, CoS_2 and $CoS_2@MoS_2$. (f)

Fig. 7 Cycling performance plots of MoS₂, CoS₂ and CoS₂@MoS₂.



Figure 2







Figure 4





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Highlights:

1. A MoS₂-coated CoS₂ composite has been prepared successfully via a one-step

hydrothermal method.

2. The $CoS_2@MoS_2$ composite exhibits specific capacitance of 1038 F/g and rate capability of 71.7%.

3. The capacitance retention of $CoS_2@MoS_2$ composite equals 84.76% after 10000 cycles.