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# Magnetization and characteristic fields in $Nd_{0.8}Ba_{0.2}FeAsO_{0.6}F_{0.4}$ with binary doping

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ABSTRACT

the Ic vs H curves.

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## 1. Introduction

Among the family of oxypnictide based superconductors, the  $RFeAsO_{1-y}$  (FeAs-1111, R is the rare earth element) exhibits a relatively high superconducting critical temperature, reaching 55 K [1] in the case of electron doping such as SmFeAsO<sub>1-x</sub> $F_x$ . The parent compounds of both 1111 and 122 type, normally show the magnetic order phase of spin density wave (SDW), but no superconductivity [2–4]. They may evolve into superconductors as a consequence of the normative symmetry breaking of 1111 type SDW due to the element doping, such as SmFe<sub>0.9</sub>Co<sub>0.1</sub>AsO [5], Ca<sub>1-x</sub>Na<sub>x</sub>Fe<sub>2</sub>As<sub>2</sub> [6], and so on. Noting the iron site doping generally suppresses the critical transition temperature, and enhances the disorder of magnetism as well.

Apart from one element or one site substitution, people also suggest that the superconductivity is tailorable by the second element doping in other site simultaneously [7–9], such as  $Ce_{1-x}Gd_xFeAsO_{0.84}F_{0.16}$  [7], and  $SmFe_{1-x}Ru_xAsO_{0.85}F_{0.1}$  [8]. It is reported that the second site of doping could enhance transition temperature as well as the upper critical field [10].

We recently look into the effect of binary doping on the performances of the oxypnictide superconductors,  $Nd_{1-x}Ba_xFeAsO_{1-2x}F_{2x}$  at a series of doping contents (x = 0.02, 0.05, 0.1, 0.15, and

\* Corresponding author. E-mail address: cbcai@staff.shu.edu.cn (C.B. Cai). 0.2) [11]. Temperature dependences of electric resistivity and magnetic susceptibility reveal that the superconductivity for the studied system emerges at x = 0.1, and enhances together with  $H_{c2}(0)$  as the doping content x increases further. In case of x = 0.2, the superconducting critical temperature reaches as high as 50 K, which is the first demonstration of superconductivity with a high fluorine doping induced by both electron and hole doping in this family.

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In the present work, the iron oxypnictide superconductor of the namely  $Nd_{0.8}Ba_{0.2}FeAsO_{0.6}F_{0.4}$  is prepared

with respect to the observation of distinct magnetization characteristics arising from both electron and

hole doping. A magnetothermal phase diagram is given for the present iron oxypnictide system, based

on the irreversibility fields  $(H_{irr})$  and the upper critical fields  $(H_{c2})$  obtained from magnetotransport

measurements, as well as the lower critical fields  $(H_{c1})$  evaluated by magnetization loops at various

temperatures. High  $H_{c2}(0)$  are revealed at low temperature range, which is in consistent with the

observations of comparatively high critical current  $(I_c)$  and the temperature dependent peak effect in

In the present paper, we focus on the sample of x = 0.2, which exhibits the highest  $T_c$  in Nd<sub>1-x</sub>Ba<sub>x</sub>FeAsO<sub>1-2x</sub>F<sub>2x</sub> series. A lot of efforts have been made to understand its distinct magnetization behaviors and magnetothermal phase diagrams, together with various characteristic magnetic fields. Note that such a special system with both electron-doping and hole-doping may allow more breaking of original symmetry in composite and structure, giving rise to more complicated superconducting and magnetic behaviors.

# 2. Experimental

Polycrystalline bulk of nominal  $Nd_{0.8}Ba_{0.2}FeAsO_{0.6}F_{0.4}$  was synthesized by a conventional solid state reaction. Firstly, the intermediate phases of Fe<sub>2</sub>As, NdAs and FeO were prepared by the reactions of the mixed commercial powders of Nd, Fe, As and Fe<sub>2</sub>O<sub>3</sub>, sealed in a vacuum quartz tube, annealed at 600 °C for 6 h and then 850 °C for 12 h. After that, the pressed pellets were sealed in a vacuum quartz tube for final annealing at 1160 °C. More details for sample preparation can be found in the Ref. [12].





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Fig. 1. (Color online) X-ray diffraction pattern for the  $Nd_{0.8}Ba_{0.2}FeAsO_{0.6}F_{0.4}$  compound with its lattice parameters.



**Fig. 2.** (Color online) Temperature dependence of the resistivity for the present sample of  $Nd_{0.8}Ba_{0.2}FeAsO_{0.6}F_{0.4}$ , the arrow indicates the critical transition temperature. The inset shows the temperature dependence of the resistance in various applied magnetic fields up to 9 T.

The superconducting properties were measured by using a physical property measurement system (Quantum Design PPMS) in magnetic fields up to 9 T. For the magnetization measurements, rectangular specimens were prepared nearly in the same dimension of about  $6 \times 2 \times 1 \text{ mm}^3$ , and the applied magnetic field was normal to the long axis direction of the specimen.

#### 3. Result and discussion

#### 3.1. Magnetotransport performances

Fig. 1 shows the room temperature X-ray diffraction (XRD) patterns for the present sample of Nd<sub>0.8</sub>Ba<sub>0.2</sub>FeAsO<sub>0.6</sub>F<sub>0.4</sub> alone with their Rietveld refinement. It is observed that all main peaks are well indexed based on their basis of space group P4/nmm, although little extra peaks at around 30° and 49° of low intensity (marked with \*). Based on the index, the lattice parameters for such a compound, is calculated to be a = 3.962(7) and c = 8.536(0).

Fig. 2 shows the superconducting critical temperature, identified from the *R* vs. *T* curves. It is revealed that the present sample exhibits the superconductivity 50 K at x = 0.2. Moreover, the superconducting transition width  $\Delta T$ , determined by the temperatures corresponding to 90% and 10% of the normal state resistivity, decreases with increasing doping content, being as low as 1.86 K at



**Fig. 3.** (Color online) The magnetization curves M(H) of the present sample of Nd<sub>0.8</sub>Ba<sub>0.2</sub>FeAsO<sub>0.6</sub>F<sub>0.4</sub> at various temperatures from 4 to 80 K.

x = 0.2. The resistivity exhibits the linear behavior above  $T_c$ , and the residual resistance ratio (*RRR* = *R*(300 K)/*R*( $T_c$ ) reaches 5.5, demonstrating the good quality of the present sample.

The inset of Fig. 2 shows the temperature dependences of electric resistance in various applied magnetic fields. The superconducting transition temperature is determined by using a criterion [13] at which the resistivity becomes negligible (about  $10^{-4}$  Ohm). It is found that the transition temperature decreases with applied magnetic field, roughly at a slope rate of 0.85 K/T, i.e.  $dT_c/dH$ –0.85 K/T, in contrast to 1 K/T of  $dT_c/dH$  for SmFe<sub>0.85</sub>Co<sub>0.15</sub>AsO [13]. Obviously, the latter is larger than the former, implying the higher  $H_{c2}$  and stronger pinning effect probably existing in the present sample.

As well, this value is much lower than that of YBCO  $(dT_c/dH-4 \text{ K}/T)$  and MgB<sub>2</sub>  $(dT_c/dH-1.5 \text{ K}/T)$  samples [14,15], suggesting a high value of upper critical field  $H_{c2}$  in these compounds [16]. One may suppose that the oxypnictides appear as a new class of high field superconductors with  $H_{c2}$  surpassing the corresponding values of Nb<sub>3</sub>Sn and MgB<sub>2</sub>. In reality, the upper critical magnetic fields of the iron oxypnictides superconductors may surpass the 100 T, a benchmark of the high  $T_c$  cuprate superconductors. The values of  $H_{c2}$  and  $H_{irr}$  for the present sample will be evaluated and discussed in the section below.

# 3.2. Critical field and critical current

Fig. 3 shows the magnetic hysteresis curves measured at temperatures of from 4 to 80 K in the field of -0.5 T < H < 0.5 T. The experimental M(H) curves can be understood as the superposition of a superconducting contribution and a paramagnetic background emerges above  $T_c$ . The arrow indicates the characteristic magnetic fields,  $H_{c1}$ , which are about 1200 Oe at 4 K. Note that the peak of magnetization is broadened with the decreasing of the temperature.

It is nontrivial to notice that the M vs. H is not symmetric, with the peak deviation from the y axis. This scenario is frequently observed in iron oxypnictides superconductors, not only in single crystal [17] but also in polycrystalline samples [13,18,19]. Another feature is the peak deflection from the origin point [13,20] after deleting the paramagnetic background determined by M vs. H at the temperatures above  $T_c$ .

Fig. 4 shows the temperature dependence of  $H_{c1}$ . One can find the value surpasses 0.1 T at 4 K taken above. The values of  $H_{c1}$  decreasing with the increasing of temperature can be fitted well by using the expression of  $H_{irr}(T) = H_{irr}(0)^*(1 - T/T_c)^{\beta}$ .

Based on the magnetotransport measurements (shown in the inset of Fig. 2), the other two characteristic magnetic fields,



**Fig. 4.** (Color online) The temperature dependence of  $H_{c1}$  evaluated from M(H). The inset shows the phase diagram characterized by the temperature dependences of  $H_{c2}$ ,  $H_{irr}$  and  $H_{c1}$ .



**Fig. 5.** (Color online) The extended temperature dependences of  $H_{c1}$ ,  $H_{irr}$  and  $H_{c2}$  at a wide range of temperatures, based on the values fitted by the formula of  $H_{irr}(T) = H_{irr}(0)^*(1 - T/T_c)^\beta$  and the G-L formula of  $H_{c2}(T) = H_{c2}(0) \frac{1 - (T/T_c)^2}{1 - (T/T_c)^2}$ .

irreversibility fields ( $H_{irr}$ ) and upper critical fields ( $H_{c2}$ ) are determined by the criterion of 10% and 90% of normal state resistance, respectively. Noting that similar evaluation is widely applied to YBaCuO, MgB<sub>2</sub>, the Fluorin doped LaFeAsO and Strontium doped PrFeAsO polycrystalline samples [2,21,22]. The inset shows the relations between the reduced temperature and characteristic magnetic fields including  $H_{c1}$ ,  $H_{irr}$  and  $H_{c2}$ . It is revealed that the  $H_{irr}(T)$ curve is close to the  $H_{c2}(T)$  curve, implying that the strong flux pinning effect hinders the dissipating resistance due to flux jumping.

To further investigate the various characteristic magnetic fields and their relation, temperature dependences are taken into account through the whole temperature range below the critical transition temperature. The results are shown in Fig. 5, where the  $H_{irr}(0)$  and  $H_{c2}(0)$  are evaluated by the formula below, respectively.

$$H_{c2}(T) = H_{c2}(0) \frac{1 - (T/T_c)^2}{1 + (T/T_c)^2}$$
(1)

$$H_{c2}(0) = -0.693T_c \left(\frac{\mathrm{d}H_{c2}}{\mathrm{d}T}\right)_{T=T_c}$$
<sup>(2)</sup>

$$H_{\rm irr}(T) = H_{\rm irr}(0)^* (1 - T/T_c)^{\beta}.$$
 (3)

The first one follows the Ginzburg–Landau theory, with which the value of upper critical field  $H_{c2}(0)$  is evaluated to be



**Fig. 6.** (Color online) The field dependence of the critical current density at 20 K estimated from magnetization curves. Peak effect is marked with arrow.



**Fig. 7.** (Color online) The critical current density vs. the applied magnetic field up to 5000 Oe at several given temperatures. It is revealed that the peak position shifts towards the low magnetic field with increasing the temperature, until it disappears near the  $T_c$ .

162.5 T. This value is very close to the 162.8 T deduced by the Werthamer–Helfand–Hohenberg theory, the second formula. This suggests that the present sample exhibits good quality, regardless of lower than corresponding values for iron oxypnictides 42 622 family [23] and single step synthesized NdFeAsO<sub>0.8</sub>F<sub>0.2</sub> [24]. However, the slope of  $dH_{c2}(T)/dT|_{T_c}$  for the present sample is much larger than that of hole doped  $Pr_{1-x}Sr_xFeAsO$  [2], and the electron doped LaFeASO<sub>1-x</sub>F<sub>x</sub> [21,22].

Also one can get  $H_{irr}(0)$  of 146 T from the third formula with  $\beta = 1.42$  which corresponds to the value ( $\beta$ ) of YBaCuO [25,26], and this value is higher than that of Sr(Fe<sub>1-x</sub>Co<sub>x</sub>)<sub>2</sub>As<sub>2-y</sub> with  $H_{irr}(0)$  of 70 T, but the value of  $\beta$  is consistent, while remaining lower than that ( $\beta \sim 2$ ) predicted by the melting theory [27].

To further understand the pinning characteristic of present sample, the critical current density in various magnetic fields is studied as well. Fig. 6 shows the magnetic field dependence of magnetization critical current ( $J_c$ ) estimated from magnetization curves and the conventional critical state model, i.e. Bean model. The center deviated peaks occur for all the magnetization curves. As marked by an arrow, a typical peak appears near 3.2 T in the case of the applied temperature 20 K.

Fig. 7 shows the magnetic  $J_c$  derived from the magnetic hysteresis vs. the applied magnetic field up to 5000 Oe at several given temperatures. Noting that the maximum applied field is lower than that of at 20 K in Fig. 6, to look into the low field

effect purposely. It is obvious that a distinct peak is present, and its position shifts towards the low magnetic field with increasing the temperatures, until it disappears near the  $T_c$ . This should be reasonable as the irreversibility fields decreases, and thus the pinging effect occurs at a relatively low field with increasing temperatures.

As for the origination of peak effects, one may attribute it to the flux pinning, either from the coexisting weak magnetic heterogeneous phases which are probably most effective at the intermediate fields or from the high density defects or nonsuperconducting phases being enhanced at matching induced magnetic fields. For the other oxypnictide based superconductors, there are few reports with regard to the peak effect. At present it is unclear whether or not this is due to the peak effect as observed in most bulks of high  $T_c$  cuprates [28].

It is noted that the present  $J_c$  at 5 K is more than  $3 * 10^4$  A/cm<sup>2</sup>, which is similar to the  $J_c$  of  $2.5 * 10^4$  A/cm<sup>2</sup> at the same temperature for the polycrystalline sample of 122 type Sr<sub>0.6</sub>K<sub>0.4</sub>Fe<sub>2</sub>As<sub>2</sub> with Ag addition reported by Lei Wang et al. [9]. This value of  $J_c$  is comparable with the single crystal single crystal Ba(Fe<sub>1-x</sub>Co<sub>x</sub>)<sub>2</sub>As<sub>2</sub>, its  $J_c$  (1.8 K) reaches  $1.8 * 10^4$  A/cm<sup>2</sup> [29].

The direct result of the nonsymmetrical *M* vs. *H* curve is that the main peaks of  $J_c$  move to the right from the center. In reality, the second peak effect in the magnetization loop is frequently observed for the iron oxypnictide superconductors. For instance, single crystal Ba<sub>0.6</sub>K<sub>0.4</sub>Fe<sub>2</sub>As<sub>2</sub> [18] in a magnetic field of 1000 Oe, more details should be given in polycrystalline SmFeAsO<sub>0.85</sub>F<sub>0.15</sub>, and Sr<sub>1-x</sub>Nd<sub>x</sub>FeAsF [15,16,30]. The underlying antiferromagnetic phases probably make the main contribution to asymmetric magnetization, leading to the additional pinning effect at immediate field range. Further studies are required to understand the related vortex pinning mechanism [31].

It is interesting to note that the main peak moves to the left with increasing temperature, until it disappears at temperatures close to  $T_c$  such as 40 K. This implies that the effective pinning centers are subject to applied magnetic fields and are visualized at relatively low temperatures.

#### 4. Conclusions

A special iron oxypnictide based system, namely Nd<sub>0.8</sub>Ba<sub>0.2</sub>FeAs O<sub>0.6</sub>F<sub>0.4</sub> with both electron and hole doping is investigated with respect to the distinct magnetization behaviors and magnetothermal phases. While a high  $H_{c2}(0)$  of about 163 T is estimated by both GL theory and WHH formula, a large slope of  $dH_{c2}(T)/dT|_{T_c}$  is obtained as well, implying that the present system with both electron doping and hole doping gives rise to more breaking of original symmetry, and then more modification for magnetic and superconducting order parameters. The comparatively high critical current ( $J_c$ ) and the temperature dependent peak effect suggest the complicated flux pinning for the present sample.

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