# **Inorganic Chemistry**

# Structure and Bonding of Bi<sub>4</sub>Ir: A Difficult-to-Access Bismuth Iridide with a Unique Framework Structure

Anna Isaeva,<sup>†</sup> Michael Ruck,<sup>\*,†,‡</sup> Konrad Schäfer,<sup>§</sup> Ute Ch. Rodewald,<sup>§</sup> and Rainer Pöttgen<sup>\*,§</sup>

<sup>†</sup>Department of Chemistry and Food Chemistry, Technische Universität Dresden, 01062 Dresden, Germany <sup>‡</sup>Max Planck Institute for Chemical Physics of Solids, Nöthnitzer Straße 40, 01187 Dresden, Germany <sup>§</sup>Institut für Anorganische und Analytische Chemie, Universität Münster, Corrensstraße 36, 48149 Münster, Germany

Supporting Information

ABSTRACT: Crystals of Bi<sub>4</sub>Ir, a new intermetallic compound, were obtained by the reaction of an iridium-containing intermetallic precursor with liquid bismuth. X-ray diffraction on a single crystal revealed a rhombohedral structure  $[R\overline{3}m, a = 2656.7(2) \text{ pm}, \text{ and } c =$ 701.6(4) pm]. Bi<sub>4</sub>Ir is not isostructural to Bi<sub>4</sub>Rh but combines motifs of the metastable superconductor Bi14Rh3 with those found in the weak topological insulator Bi14Rh3I9. The two crystallographically independent iridium sites in Bi4Ir have square-prismatic and skewed-squareantiprismatic bismuth coordination with Bi-Ir distances of 283-287 pm. By sharing common edges, the two types of [IrBi<sub>8</sub>] units constitute a complex three-dimensional network of rings and helices. The bonding in the heterometallic framework is dominated by pairwise Bi-Ir interactions. In addition, three-center bonds are found in the bismuth triangles formed by adjacent [IrBi<sub>8</sub>] polyhedra. Density



functional theory based band-structure calculations suggest metallic properties.

# INTRODUCTION

The synthesis of bismuth-rich intermetallic compounds of highmelting transition metals is strongly hampered by the dramatic differences between the boiling point of bismuth (1883 K) and the much higher melting points of the transition metals, e.g., 2683 K for iridium or 2239 K for rhodium.1 Only small amounts of rhodium and iridium are soluble in liquid bismuth, and flux synthesis is slow. Early investigations of the Bi-Ir system by melt reactions and subsequent annealing yielded the binary phases Bi<sub>2</sub>Ir and Bi<sub>3</sub>Ir.<sup>2,3</sup> The structure types  $\beta$ -Bi<sub>2</sub>Rh and Bi3Ni were assigned on the basis of Debye-Scherrer diffractograms. Subsequent studies by Kjekshus<sup>4</sup> could not reproduce Bi<sub>2</sub>Ir either by annealing experiments or by chemicaltransport reactions. Phase-pure Bi<sub>2</sub>Ir nanoparticles were only recently synthesized via a microwave-assisted polyol process.<sup>5</sup> Bi<sub>2</sub>Ir is a weakly paramagnetic semimetal. Bi<sub>3</sub>Ir nanoparticles can be obtained by the same technique;<sup>6</sup> these nanoparticles reversibly intercalate oxygen up to the composition Bi<sub>3</sub>IrO<sub>2</sub>. The latter material appears to be a unique metallic oxide ion conductor and is also the first one that operates at room temperature.

Single crystals of  $Bi_3(Ir_{0.77}Cu_{0.17}Ni_{0.06})$  were recently obtained from intermetallic precursor compounds.<sup>7</sup> In metal flux synthesis, the broadly applied approach to intermetallics,  $^{8-10}$  bismuth has only scarcely been used for the growth of phosphides and germanides. Recent examples include  $Lu_3Ir_7P_5$ ,<sup>11</sup> ScRh<sub>6</sub>P<sub>4</sub>,<sup>12</sup> CeRh<sub>6</sub>Ge<sub>4</sub>,<sup>13</sup> and Ce<sub>2</sub>Rh<sub>3</sub>Ge<sub>5</sub>.<sup>14</sup> In the pursuit of expanding our explorative bismuth flux technique to

ternary antimonides, we have repeatedly obtained crystals of binary bismuth-iridium phases as byproducts. The crystal structure and chemical-bonding peculiarities of the hitherto unknown bismuth-richest phase, Bi<sub>4</sub>Ir, are reported herein.

# EXPERIMENTAL SECTION

Synthesis. Single crystals of Bi<sub>4</sub>Ir are accessible from the reaction of an iridium-containing intermetallic precursor with liquid bismuth. In a typical experiment, a sample of the nominal composition Sm<sub>3</sub>Ir<sub>3</sub>Sb<sub>7</sub> was synthesized by arc-melting<sup>15</sup> from the elements [samarium pieces (Smart Elements), iridium powder (Allgemeine Pforzheim), and antimony lumps (Johnson Matthey), all with purities >99.9%] under 700 mbar of argon (purified with titanium sponge at 900 K, with silica gel, and with molecular sieves). The crushed Sm<sub>3</sub>Ir<sub>3</sub>Sb<sub>7</sub> regulus was subsequently mixed with 40 mol equiv of bismuth shots (ABCR; >99.9%). The mixture was placed in a corundum crucible, which afterward was sealed in an evacuated silica tube. The ampule was rapidly heated up to 1370 K in a muffle furnace and kept at that temperature for 1 week. Then the temperature was decreased to 1070 K by radiative heat loss, and the sample was annealed at that temperature for another 1 week, followed by slow cooling to room temperature within 60 h. The bismuth excess was slowly dissolved in an equimolar solution of  $H_2O_2$  (Acros; 35%) and glacial acetic acid (VWR International). The resultant sample was washed with demineralized water. The reaction products were SmSb, elemental

Special Issue: To Honor the Memory of Prof. John D. Corbett

Received: September 11, 2014

iridium,  $Bi_3Ir$ , and  $Bi_4Ir$  as well as thin fibers of a samarium–iridium antimonide–bismuthide with an approximate composition SmIrSb<sub>5</sub>Bi<sub>2</sub> [energy-dispersive X-ray (EDX) data].<sup>16</sup> Various phases were easily separated from each other mechanically based on their crystal habits.

Certainly, the synthesis of  $Bi_4Ir$  was not targeted in the way presented. We repeatedly observed  $Bi_4Ir$  from bismuth flux experiments on related ternary antimonides. These data underline that  $Bi_4Ir$  is achievable and reproducible through this technique. This does, however, not mean that such a strategy can generally by applied to related binaries.

**EDX Analyses.** The composition of the Bi<sub>4</sub>Ir crystals was analyzed by EDX using a ZEISS EVO MA10 scanning electron microscope in variable pressure mode. Elemental iridium and bismuth were used as standards. No impurity elements heavier than sodium (detection limit of the instrument) were observed. The crystals exhibited the composition of 80 ± 1 atom % bismuth and 20 ± 1 atom % iridium, in excellent agreement with the proposed formula Bi<sub>4</sub>Ir.

Single-Crystal X-ray Diffraction. Small block-shaped single crystals of Bi4Ir were isolated from the flux by mechanical fragmentation. In order to check the crystal quality, Laue photographs of the crystals were collected on a Buerger precession camera (white molybdenum radiation, imaging plate technique, and Fujifilm BAS-1800). Intensity data were collected at room temperature using a Stoe IPDS-II imaging plate diffractometer (graphite-monochromatized Mo K $\alpha$  radiation;  $\lambda$  = 71.073 pm) in the oscillation mode. Numerical absorption correction ( $\mu = 131.2 \text{ mm}^{-1}$ ) was applied to the data set. Lattice metrics and systematic reflection conditions first suggested the monoclinic space groups C2/m, Cm, and C2. The centrosymmetric group was chosen for the first structure refinement. The starting atomic parameters were found by means of the SUPERFLIP program,<sup>17</sup> implemented in the JANA2006 package.<sup>18</sup> Critical inspection of the resultant structure model gave strong hints for higher, presumably rhombohedral, symmetry. Evaluation of the monoclinic structural model with the PLATON routine<sup>19</sup> readily led to the correct space group,  $R\overline{3}m$ . Examples of overlooked trigonal symmetry are well-known for diverse structures.<sup>20</sup> The setting obtained for Bi<sub>4</sub>Ir from PLATON was subsequently standardized with the *TIDY* routine,<sup>21,22</sup> and the structure refinement has smoothly converged to the residuals listed in Table 1. The final positional parameters and interatomic distances are listed in Tables 2 and 3. All sites are fully occupied. Further details on the structure determination are available from Fachinformationszentrum Karlsruhe, D-76344 Eggenstein-Leopoldshafen, Germany, by quoting CSD 428458.

**Quantum-Chemical Calculations.** Scalar-relativistic and fully relativistic density functional theory (DFT)-based calculations were performed using the full-potential linearized augmented plane wave (LAPW) method<sup>23</sup> within the local density approximation. Chemical bonding was characterized via topological analysis of the real-space electron localizability indicator (ELI-D,  $\gamma_D^{-24,25}$  that was performed in the *DGrid* program package<sup>26</sup> and visualized with the *Paraview* software.<sup>27</sup> Formal atomic charges were calculated via integration of the electron density ( $\rho$ ) in basins according to the quantum theory of atoms in molecules (QTAIM) developed by Bader.<sup>28</sup>

# RESULTS AND DISCUSSION

**Synthesis and Structure.** The fundamental obstacles that hamper direct crystal growth of bismuth–iridium phases have been ameliorated by activating iridium via a prereaction with antimony and samarium with the subsequent addition of bismuth to the system. The Bi<sub>4</sub>Ir crystals are one of the reaction products (see the Experimental Section) that grew out of the quaternary reaction mixture. The formation of byproducts cannot be avoided, and pure-phase samples cannot be synthesized by this method.

As by now, Bi<sub>4</sub>Ir represents the bismuth-richest phase in the Bi–Ir system. Bi<sub>4</sub>Ir is not isostructural to Bi<sub>4</sub>Rh<sup>29,30</sup> but crystallizes in a novel rhombohedral structure type with 18 formula units per unit cell [ $R\overline{3}m$ , a = 2656.7(2) pm, and c =

# Table 1. Crystal Data and Structure Refinement for Bi<sub>4</sub>Ir

empirical formula	Bi₄Ir
cryst syst	trigonal, rhombohedral
space group	R3m
Pearson symbol	hR135
fw, g mol <sup><math>-1</math></sup>	1028.1
unit cell dimensions, pm	a = 2656.7(2), c = 701.6(4)
cell volume, nm <sup>3</sup>	V = 4.2881
formula units per cell	Z = 27
calcd density, $g \text{ cm}^{-3}$	10.75
diffractometer type	IPDS-II
detector distance, mm	60
exposure time, min	3
$\omega$ range; increment, deg	0–180, 1.0°
integration param (A, B, EMS)	13.3, 3.1, 0.013
cryst size, $\mu m^3$	$20 \times 20 \times 60$
abs coeff, mm <sup>-1</sup>	131.2
transmn ratio (max/min)	3.54
F(000), e	11043
heta range for data collection, deg	3-30
range in <i>hkl</i>	±36, ±36, ±8
total no. of reflns	5339
indep reflns/R <sub>int</sub>	959/0.088
reflns with $I \ge 2\sigma(I)/R_{\sigma}$	477/0.120
no. of data/param	959/41
GOF on $F^2$	0.74
R1/wR2 for $I > 2\sigma(I)$	0.028/0.044
R1/wR2 (all data)	0.058/0.092
extinction coeff	$38(8) \times 10^{-5}$
largest diff peak/hole (e Å $^{-3}$ )	5.84/-5.17

701.6(4) pm]. Although the structure features a large unit cell, it is built by only two simple building units. The coordination spheres of two nonequivalent iridium atoms are constituted by eight bismuth atoms each. The Ir2-centered polyhedron is a slightly elongated cube (a square prism), while the Ir1-centered one is a skewed square antiprism (square faces are rotated by 31°). The Bi–Ir distances fall in the narrow range from 283 to 287 pm. They are somewhat longer than those in Bi<sub>3</sub>Ir (276–279 pm)<sup>7</sup> or than the sum of the covalent radii (278 pm).<sup>1</sup> The cubes and antiprisms occur in the ratio 1:2 and form a complex three-dimensional network  $\frac{3}{\infty}$ [IrBi<sub>8/2</sub>] by sharing common edges (Figures 1 and 2).

Cubic or square-antiprismatic coordination of electron-rich transition-metal atoms (M) by bismuth atoms is found in various bismuth-rich binary and ternary compounds. Solely skewed cubes (square faces rotated by 19°) are reported for the structure of  $\alpha$ -Bi<sub>4</sub>Rh.<sup>29,30</sup> Cubes and antiprisms in the ratio 2:1 constitute the isomorphous polyhedral networks of the metastable intermetallic compound Bi<sub>14</sub>Rh<sub>3</sub><sup>31,32</sup> as well as of the subhalides Bi<sub>12</sub>Rh<sub>3</sub>Cl<sub>2</sub><sup>32</sup> and Bi<sub>12</sub>Rh<sub>3</sub>Br<sub>2</sub>.<sup>33</sup> The intermetallic honeycomb layers in Bi<sub>14</sub>Rh<sub>3</sub>I<sub>9</sub>,<sup>34,35</sup> Bi<sub>13</sub>Pt<sub>3</sub>I<sub>7</sub>,<sup>36,37</sup> and Bi<sub>12</sub>Pt<sub>3</sub>I<sub>5</sub><sup>37</sup> are formed entirely by [MBi<sub>8/2</sub>] cubes. In all of the aforementioned cases, the local connectivity is the same. Each polyhedron is linked to four others, defining two trigonal prisms of bismuth atoms; i.e., there are two trigonal prisms per three polyhedra.

The polyhedral network of Bi<sub>4</sub>Ir combines structural motifs that resemble those in the topological insulator Bi<sub>14</sub>Rh<sub>3</sub>I<sub>9</sub> (sixmembered rings of polyhedra stacked along 0, 0, *z*) with those found in the superconductor Bi<sub>14</sub>Rh<sub>3</sub> (helices of alternating cubes and antiprisms around 1/3, 1/3, *z* and, with opposite

Table 2. Atomic Coordinates and Anisotropic Displacement Parameters (pm<sup>2</sup>) for Bi<sub>4</sub>Ir<sup>a</sup>

atom	site	x	у	z	$U_{11}$	U <sub>22</sub>	U <sub>33</sub>	$U_{12}$	$U_{13}$	$U_{23}$	$U_{ m eq}$
Bi1	36i	0.06061(4)	0.24008(4)	0.18365(13)	133(4)	131(4)	147(4)	49(4)	-6(4)	20(4)	145(3)
Bi2	36 <i>i</i>	0.13027(5)	0.38324(4)	0.03919(15)	229(5)	106(4)	195(5)	60(4)	-80(4)	-17(4)	187(4)
Bi3	18h	0.60481(3)	-x	0.0325(2)	129(4)	$U_{11}$	179(7)	53(5)	-14(3)	$-U_{13}$	151(4)
Bi4	18h	0.75189(3)	-x	0.08165(19)	177(5)	$U_{11}$	131(7)	97(6)	6(3)	$-U_{13}$	158(5)
Ir1	18g	0.16937(5)	0	<sup>1</sup> / <sub>2</sub>	123(4)	114(6)	151(7)	57(3)	-5(3)	-10(5)	130(4)
Ir2	9e	<sup>1</sup> / <sub>2</sub>	0	0	135(6)	71(8)	140(9)	35(4)	-2(3)	-5(7)	122(6)
<sup><i>a</i></sup> The anisotropic displacement factor exponent takes the form $-2\pi^2[(ha^*)^2U_{11} + + 2hka^*b^*U_{12}]$ . $U_{12} = U_{23} = 0$ .											

#### Table 3. Interatomic Distances (pm) in $Bi_4Ir^a$

Bi1–Ir1 (×1)	283.2	Bi3–Ir1 (×2)	284.9
Bi1–Ir2 (×1)	286.3	Bi3-Bi1 (×2)	321.5
Bi1-Bi1 (×1)	315.8	Bi3-Bi4 (×1)	331.8
Bi1-Bi3 (×1)	321.5	Bi3-Bi4 (×2)	352.7
Bi1-Bi2 (×1)	327.0	Bi4–Ir1 (×2)	284.6
Bi1-Bi2 (×1)	344.7	Bi4-Bi2 (×2)	318.3
Bi1-Bi2 (×1)	361.5	Bi4-Bi3 (×1)	331.8
Bi1-Bi4 (×1)	379.3	Bi4-Bi3 (×2)	352.7
Bi1-Bi1 (×1)	379.7	Bi4-Bi1 (×2)	379.3
Bi2–Ir1 (×1)	284.1	Ir1–Bi1 (×2)	283.2
Bi2–Ir2 (×1)	286.9	Ir1–Bi2 (×2)	284.1
Bi2-Bi4 (×1)	318.3	Ir1-Bi4 (×2)	284.6
Bi2-Bi2 (×1)	326.0	Ir1–Bi3 (×2)	284.9
Bi2-Bi1 (×1)	327.0	Ir2-Bi1 (×4)	286.3
Bi2-Bi1 (×1)	344.7	Ir2-Bi2 (×4)	286.9
Bi2-Bi1 (×1)	361.5		
Bi2-Bi2 (×1)	369.0		

"All distances within the first coordination spheres are listed. Standard deviations are all equal or less than 0.2 pm.



**Figure 1.** View of the Bi<sub>4</sub>Ir structure along the *c* axis. The edge-sharing  $[IrBi_{8/2}]$  cubes (Ir2, yellow) and square antiprisms (Ir1, blue) are emphasized.



Figure 2. Cutout of the  $Bi_4Ir$  structure with edge-sharing  $[IrBi_{8/2}]$  cubes (Ir2, yellow) and square antiprisms (Ir1, blue), emphasizing the mixed helix (central axis) and the attached six-membered rings of the antiprisms.

chirality, around  ${}^{2}/{}_{3}$ ,  ${}^{2}/{}_{3}$ , z). In contrast to Bi<sub>12</sub>Rh<sub>3</sub>X<sub>2</sub> (X = Bi, Cl, and Br),  ${}^{31-33}$  thorough analysis of the residual electron density inside the helices of the Bi<sub>4</sub>Ir structure does not reveal any additional "interstitial" atoms X on their central axes. Likewise, the hexagonal prisms defined by six [IrBi<sub>8</sub>] cubes are not centered by isolated atoms, as has been found in the abovementioned layered subiodides. Consistently, the inner diameters of the unfilled structural motifs in Bi<sub>4</sub>Ir are about 50–100 pm smaller. As in all of the other structures, the lone pairs of the bismuth atoms, which decorate the surface of the framework, are pointing into these voids (cf. Figure 3 and the discussion below).

Article



**Figure 3.** Fragment of the Bi<sub>4</sub>Ir structure with both types of iridium coordination polyhedra and trigonal prisms of bismuth atoms connecting them. Thicker lines signify the strongest bonds in the framework. ELI-D localization domains given at  $\gamma = 1.01$  [1, pairwise Bi–Ir bonds (0.8–0.9 electrons); 2, three-center bond slightly off the base of the trigonal prism formed by bismuth atoms (0.4 electrons)] and at  $\gamma = 1.08$  [3, Bi 6s lone pairs (1.6–2.4 electrons)].

**Electronic Structure and Chemical Bonding.** The slightly higher electronegativity of iridium in comparison to rhodium, as evidenced by the estimated values of the ionization potentials,<sup>38</sup> suggests that the iridium atom in Bi<sub>4</sub>Ir should bear a larger negative charge than the rhodium atom in Bi<sub>4</sub>Rh. The higher electronic withdrawal toward iridium, opposite to rhodium, is further confirmed by formal atomic charges (calculated according to the QTAIM): 1.0– for iridium and 0.2+ to 0.3+ for bismuth in Bi<sub>4</sub>Ir but 0.7– for rhodium and 0.3+ to 0.35+ for bismuth in Bi<sub>14</sub>Rh<sub>3</sub>I<sub>9</sub>.<sup>34</sup>

Apart from monosynaptic basins, which correspond to bismuth lone pairs and closed shells or the iridium penultimate and core shells, there exist two types of ELI-D maxima that designate bonding interactions: (a) eight pairwise Bi–Ir bonds per polyhedron; (b) three-center Bi–Bi bonds that reside in the upper and lower bases of every trigonal prism defined by edgesharing [IrBi<sub>8</sub>] polyhedra (Figure 3). The latter bonds manifest themselves also through the shorter Bi–Bi distances in the prisms (ca. 3.27 Å), as opposed to the other Bi–Bi contacts within the framework (3.4–3.6 Å). The bonding pattern strongly resembles that of  $Bi_{14}Rh_3I_{9}$ ,<sup>34</sup> in which the respective bonding basins comprise 1.1 and 0.8 electrons.<sup>28</sup> The corresponding electron count for  $Bi_4Ir$  yields about 0.8–0.9 and 0.4 electrons, respectively.

The conductivity pathway provided by the bonding network propagates along the entire framework and thus accounts for metallic properties predicted by the DFT-based calculations (Figures 4 and 5). Strongly mixed Ir d and Bi p states provide



**Figure 4.** Band structure of  $Bi_4Ir$  in the scalar-relativistic approximation. The Fermi level resides at 0 eV.



Figure 5. Total density of states and partial atomic contributions in  $\mathrm{Bi}_4\mathrm{Ir}.$ 

the dominant contribution to the density of states at the Fermi level (Figure 5) and thus account for the metallic conductivity. Bismuth atoms that occupy various sites show no notable differences either in their contributions to bands or in their homo- and heteroatomic distances. An interesting detail of the Bi<sub>4</sub>Ir band structure is a pseudogap that appears in the energy range between -1.0 and -0.2 eV (Figure 4) in the scalarrelativistic approximation. A gap opening under spin-orbit coupling, possibly indicating a transition in the topological state of matter,<sup>34,39</sup> has not been observed in the ensuing fully relativistic calculation. The spin-orbit coupling manifests itself in moderate energy splitting of the relevant bands.

# 

The bismuth-rich intermetallic compound  $Bi_4Ir$  is accessible in the form of block-shaped single crystals from bismuth flux reactions of different quaternary Sm-Ir-Sb-Bi starting compositions. The basic building units of the  $Bi_4Ir$  structure are square-prismatic and skewed-square-antiprismatic [IrBi<sub>8</sub>] units with Ir-Bi distances of 283–287 pm, which condense to a complex three-dimensional network that is dominated by strong Ir-Bi bonding.

#### ASSOCIATED CONTENT

#### Supporting Information

Crystallographic data in CIF format. This material is available free of charge via the Internet at http://pubs.acs.org.

#### AUTHOR INFORMATION

#### **Corresponding Authors**

\*E-mail: michael.ruck@tu-dresden.de.

\*E-mail: pottgen@uni-muenster.de.

#### **Author Contributions**

The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

#### Notes

The authors declare no competing financial interest.

# ACKNOWLEDGMENTS

This work was supported by the Deutsche Forschungsgemeinschaft.

# DEDICATION

In memory of John Corbett.

# REFERENCES

(1) Emsley, J. *The Elements;* Oxford University Press: Oxford, U.K., 1999.

(2) Zhuravlev, N. N.; Smirnova, E. M. Sov. Phys. Crystallogr. 1966, 10, 694–697.

(3) Massalski, T. B., Okamoto, H., Subramanian, P. R., Kacprzak, L., Eds. *Binary Alloy Phase Diagrams*, 2nd ed. (version 1.0.); ASM International: New York, 1996.

(4) Kjekshus, A. Acta Chem. Scand. 1971, 25, 411-422.

(5) Boldt, R.; Grigas, A.; Heise, M.; Herrmannsdörfer, T.; Isaeva, A.; Kaskel, S.; Köhler, D.; Ruck, M.; Skrotzki, R.; Wosnitza, J. Z. Anorg. Allg. Chem. 2012, 638, 2035–2043.

(6) Heise, M.; Ruck, M. Z. Anorg. Allg. Chem. 2012, 638, 1568.

(7) Heise, M.; Rasche, B.; Isaeva, A.; Baranov, A. I.; Ruck, M.; Schäfer, K.; Pöttgen, R.; Eufinger, J. P.; Janek, J. *Angew. Chem., Int. Ed.* **2014**, *53*, 7344–7348.

(8) Canfield, P. C.; Fisk, Z. Philos. Mag. B 1992, 65, 1117-1123.

(9) Kanatzidis, M. G.; Pöttgen, R.; Jeitschko, W. Angew. Chem., Int. Ed. 2005, 44, 6996–7023.

(10) Canfield, P. C. Philos. Mag. 2012, 92, 2398-2400.

(11) Pfannenschmidt, U.; Rođewald, U. Ch.; Pöttgen, R. Z. Anorg. Allg. Chem. 2010, 636, 314-319.

(12) Pfannenschmidt, U.; Rodewald, U. Ch.; Pöttgen, R. Monatsh. Chem. 2011, 142, 219–224.

(13) Voßwinkel, D.; Niehaus, O.; Rodewald, U. Ch.; Pöttgen, R. Z. Naturforsch. 2012, 67b, 1241–1247.

(14) Voßwinkel, D.; Pöttgen, R. Z. Naturforsch. 2013, 68b, 301-305.

(15) Pöttgen, R.; Gulden, Th.; Simon, A. GIT Labor-Fachz. 1999, 43,

133–137.

(16) Schäfer, K.; Pöttgen, R., unpublished results.

(17) Palatinus, L.; Chapuis, G. J. Appl. Crystallogr. 2007, 40, 786–790.

- (18) Petříček, V.; Dušek, M.; Palatinus, L. Z. Kristallogr. 2014, 229, 345.
- (19) Spek, A. L. Acta Crystallogr., Sect. D 2009, 65, 148-155.

(20) Cenzual, K.; Gelato, M. L.; Penzo, M.; Parthé, E. Z. Kristallogr. 1990, 193, 217-242.

(21) Parthé, E.; Gelato, L. M. Acta Crystallogr., Sect. A 1984, 40, 169– 183.

(22) Gelato, L. M.; Parthé, E. J. Appl. Crystallogr. 1987, 20, 139-143.

(23) The Elk FP-LAPW Code, 2009-2014, http://elk.sourceforge. net (last access date: 1/10/2014).

(24) Kohout, M. Faraday Discuss. 2007, 135, 43-54.

(25) Kohout, M. Int. J. Quantum Chem. 2004, 97, 651-658.

(26) Kohout, M. DGrid, version 4.7; Max Planck Institut: Radebeul, Germany, 2013.

(27) Sandia National Labs, Kitware Inc., and Los Alamos National Labs. *Paraview: Parallel visualization application*, 2008, http://paraview.org (last access date: 1/10/2014).

(28) Bader, R. F. W. Atoms in Molecules; Oxford University Press, Oxford, U.K., 1990.

(29) Glagoleva, V. P.; Zhdanov, G. S. Sov. Phys. JETP **1956**, 3, 155. (30) Grin, Yu.; Wedig, U.; von Schnering, H. G. Angew. Chem., Int. Ed. Engl. **1995**, 34, 1204–1206.

(31) Gu, Q. F.; Krauss, G.; Grin, Yu.; Steurer, W. J. Solid State Chem. 2007, 180, 940–948.

(32) Kaiser, M.; Rasche, B.; Ruck, M. Angew. Chem., Int. Ed. 2014, 53, 3254–3258.

(33) Ruck, M. Solid State Sci. 2001, 3, 369-375.

(34) Rasche, B.; Isaeva, A.; Ruck, M.; Borisenko, S.; Zabolotnyy, V.; Büchner, B.; Koepernik, K.; Ortix, C.; Richter, M.; van den Brink, J. *Nat. Mater.* **2013**, *12*, 422–425.

(35) Rasche, B.; Gerisch, A.; Kaiser, M.; Isaeva, A.; Van den Broek, W.; Koch, C.; Kaiser, U.; Ruck, M. *Chem. Mater.* **2013**, *25*, 2359–2364.

(36) Ruck, M. Z. Anorg. Allg. Chem. 1997, 623, 1535-1541.

(37) Kaiser, M.; Rasche, B.; Isaeva, A.; Kaiser, U.; Ruck, M. Chem.— Eur. J. 2014, in press.

(38) Wu, Z. J.; Kawazoe, Y. Chem. Phys. Lett. 2006, 423, 81-86.

(39) Rasche, B.; Ruck, M.; Richter, M.; Koepernik, K.; van den Brink, J., unpublished results.