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Synthesis of CuO Wires from Layered Organic-Inorganic Hybrids

Young-Duk Huh* and Seok-Soon Kweon

Department of Chemistry, Dankook University, Seoul 140-714, Korea. *E-mail: ydhuh@dankook.ac.kr Received July 7, 2005

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Copper oxide (CuO) is a narrow band-gap semiconducting oxide that has been widely studied in electrochemical cells, in photothermal and photoconductive materials, and as a heterogeneous catalyst.¹⁻³ Various techniques have been used to fabricate CuO nanostructures.3-6 One-dimensional (1D) CuO can be obtained through thermal treatment by using the precursors Cu(OH)2 or Cu2(OH)2CO3 as a sacrificial template.7-9 Layered organic-inorganic hybrids are characterized by strong intralayer covalent or ionic bonding in an inorganic frame, and weak interlayer interactions such as van der Waals forces between organic molecules. 10-14 Most layered organic-inorganic hybrid materials fabricated to date were obtained by intercalation of organic molecules into the inorganic layer. In this work, we report the preparation of CuO from the layered organic-inorganic hybrid, Cu₂(OH)₃(C₁₂H₂₅SO₄). CuO wires are formed by a self-aggregation of nanoparticles. The morphologies of the aggregated CuO products are analyzed using scanning electron microscopy and transmission electron microscopy.

Experimental Section

The structures of $\text{Cu}_2(\text{OH})_3(\text{C}_{12}\text{H}_{25}\text{SO}_4)$ and CuO were analyzed by powder X-ray diffraction (XRD, Phillips PW 1710) using Cu K_α radiation. The morphology of the CuO was characterized by scanning electron microscopy (SEM, Phillips XL30 ESEM-FEG) and transmission electron microscopy (TEM, Jeol 2010). The thermal behavior of the precursor was studied by thermogravimetric analysis (TGA, Seiko Exstar 6000).

Results and Discussion

The copper hydroxide salts, $Cu_2(OH)_3X$ (X = exchange-

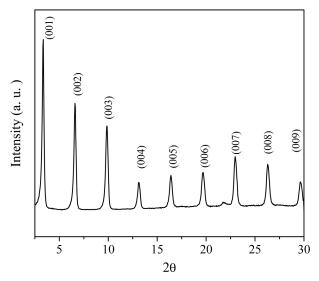


Figure 1. X-ray diffraction pattern and Miller indices of $Cu_2(OH)_3(C_{12}H_{25}SO_4)$.

able anion, NO_3^- , $RCOO^-$, etc.), show a botallackite-type structure, in which two distinct copper atoms lie in 4 + 2 (oxygen + X) and 4 + 1 + 1 (oxygen + oxygen + X) environments. The anion is located in the interlayer, while the molecular component coordinates the copper ion. The copper hydroxide salts form plate-like structures. The interlayer distance depends on the intercalated organic moiety. Figure 1 shows the X-ray diffraction pattern and Miller indices of $Cu_2(OH)_3(C_{12}H_{25}SO_4)$. The series of (001) peaks at regular intervals indicates a typical layered structure. The basal spacing is calculated to be 26.2 Å, which is equal to that reported by Rabu *et al.* 17,18 The interdigitated monolayer structure of the dodecylsulfate is oriented normal to the $[Cu_2(OH)_3]^+$ layer.

Figure 2 shows a typical TGA curve for the Cu₂(OH)₃-(C₁₂H₂₅SO₄) precursor in an atmospheric flow. The curve shows four pronounced weight loss steps. The first weight loss at 45 °C is due to the dehydration of water adsorbed within the precursor, where the actual empirical formula for the precursor is Cu₂(OH)₃(C₁₂H₂₅SO₄)·0.38H₂O. Between 180 °C and 250 °C, there is a major weight loss due to the decomposition of intercalated dodecylsulfate where the suggested product is the stable CuO·CuSO₄. ¹⁹ The calculated and measured weight losses are 46.9% and 48.6%, respectively. The third weight loss is observed in the temperature range of 250-410 °C. In this temperature range,

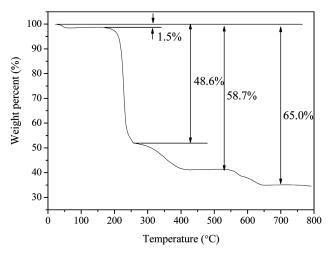


Figure 2. TGA curve of $Cu_2(OH)_3(C_{12}H_{25}SO_4)$.

CuO·CuSO₄ is partially decomposed to form the more stable $1.5 \text{CuO} \cdot 0.5 \text{CuSO}_4$ with calculated and measured weight losses of 59.3% and 58.7%, respectively. Above 650 °C, the $1.5 \text{CuO} \cdot 0.5 \text{CuSO}_4$ decomposes to form 2CuO with calculated and measured weight losses of 64.6% and 65.0%, respectively. The possible chemical reactions for the thermal decomposition of $\text{Cu}_2(\text{OH})_3(\text{C}_{12}\text{H}_{25}\text{SO}_4) \cdot 0.38 \text{H}_2\text{O}}$ are presented below:

$$Cu_{2}(OH)_{3}(C_{12}H_{25}SO_{4})0.38H_{2}O$$

$$\rightarrow Cu_{2}(OH)_{3}(C_{12}H_{25}SO_{4}) + 0.38H_{2}O$$
(1)

$$\begin{array}{ccc} Cu_2(OH)_3(C_{12}H_{25}SO_4) & \rightarrow & CuO\cdot CuSO_4 + C_{12}H_{25}OH \\ & + & H_2O \ (and \ other \ forms) \end{array} \tag{2}$$

$$CuO \cdot CuSO_4 \rightarrow 1.5CuO \cdot 0.5CuSO_4 + 0.5SO_3$$
 (3)

$$1.5CuO \cdot 0.5CuSO_4 \rightarrow 2CuO + 0.5SO_3 \tag{4}$$

Figure 3 shows the X-ray diffraction pattern and Miller indices of CuO obtained by the thermal decomposition of $\text{Cu}_2(\text{OH})_3(\text{C}_{12}\text{H}_{25}\text{SO}_4)\cdot 0.38\text{H}_2\text{O}$ at 850 °C for 1 h. The diffraction peaks correspond to pure CuO with a monoclinic

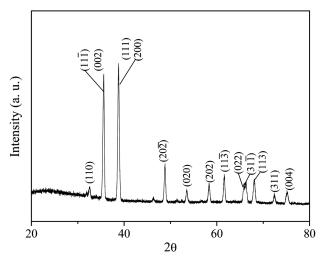


Figure 3. X-ray diffraction pattern and Miller indices of CuO.

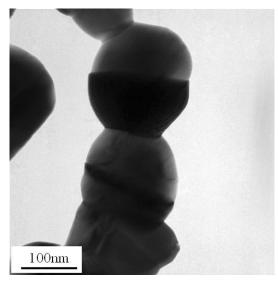
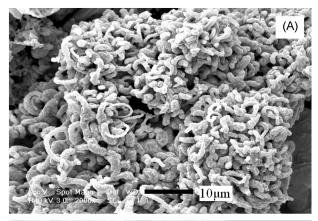


Figure 4. TEM image of CuO product obtained from the $Cu_2(OH)_3(C_{12}H_{25}SO_4)$ precursor by thermal treatment at 850 °C.

structure (a = 4.6837 Å, b = 3.4226 Å, c = 5.1288, $\beta = 99.54^{\circ}$, JCPDS 45-0937).²⁰ The XRD peaks exhibit no broadening since the particles are much larger than the nanometer range.

Figure 4 shows the TEM images of the CuO product obtained by thermal decomposition of the Cu₂(OH)₃-(C₁₂H₂₅SO₄) precursor at 850 °C. The image consists of a large number of particles with a fairly uniform diameter of 150 nm. Figure 5(A) shows the SEM morphology of the wire-like structure of the resulting CuO product. SEM examination shows that the CuO wires are $\sim 1 \mu m$ thick with lengths up to 20 μ m. Note that there are no flat or spherical structures. As can be seen from Figure 5(B), the CuO wire consists of particles with sizes on the order of a few hundred nanometers. The SEM image reveals that the micron-sized CuO wire is in fact an assembly of dozens of hundred nanometer-sized particles. Therefore, the CuO nanoparticles aggregate and assemble to form micron-sized CuO wires. If the aggregation and assembly of CuO particles were random, a spherical CuO product would result.

The $Cu_2(OH)_3(C_{12}H_{25}SO_4)$ precursor has a typical layered structure with a basal spacing of 26.2 Å. The CuO·CuSO₄ product is formed by thermal decomposition of the organic dodecyl sulfate in the Cu₂(OH)₃(C₁₂H₂₅SO₄) precursor at 200 °C. Since the interlayer distance is relative long, the layered precursor structure slowly collapses to form CuO·CuSO₄ at this temperature, with the CuO·CuSO₄ particles stacked layer by layer. At higher temperature, the CuO·CuSO₄ compounds finally decompose to form CuO particles. These CuO particles aggregate in a specific way to form micron-sized CuO wire. The formation mechanism of the micron-sized CuO wire is not clear. However, the formation of aggregated CuO wires depends on the morphology of the precursor. The decomposition of the layered Cu₂(OH)₃(C₁₂H₂₅SO₄) leads the CuO particles to integrate and to form wire-like CuO. The micron-sized CuO wires are



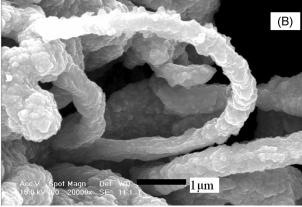


Figure 5. SEM images of CuO wire obtained from the $Cu_2(OH)_3(C_{12}H_{25}SO_4)$ precursor by thermal treatment at 850 °C; (A) × 2000 and (B) × 20000.

synthesized from two-dimensional (2D) $\text{Cu}_2(\text{OH})_3(\text{C}_{12}\text{H}_25\text{SO}_4)$ by simple thermal decomposition in the absence of a suitable template.

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References

- 1. Reitz, J. B.; Solomon, E. I. J. Am. Chem. Soc. 1998, 120, 11467.
- Wu, M. K.; Ashburn, R. J.; Torng, C. J.; Hor, P. H.; Meng, R. L.; Gao, L.; Huang, Z. J.; Wang, Y. Q.; Chu, C. W. *Phys. Rev. Lett.* 1987, 58, 908.
- 3. Anandan, S.; Wen, X.; Yang, S. Mater. Chem. Phys. 2005, 93, 35.
- Hsieh, C. T.; Chen, J. M.; Lin, H. H.; Shih, H. C. Appl. Phys. Lett. 2003, 82, 3316.
- Huang, L. S.; Yang, S. G.; Li, T.; Gu, B. X.; Du, Y. W.; Lu, Y. N.; Shi, S. Z. J. Cryst. Growth 2004, 260, 130.
- Zhu, C. L.; Chen, C. N.; Hao, L. Y.; Hu, Y.; Chen, Z. Y. Solid State Commun. 2004, 130, 681.
- 7. Du, G. H.; Tendeloo, G. V. Chem. Phys. Lett. 2004, 393, 64.
- 8. Wang, Z. L.; Kong, X. Y.; Wen, X.; Yang, S. J. Phys. Chem. 2003, B107, 8275.
- 9. Wen, X.; Zhang, W.; Yang, S. Langmuir 2003, 19, 5898.
- 10. Kurmoo, M.; Day, P.; Derory, A.; Estournes, C.; Poinsot, R.; Stead, M. J.; Kepert, C. J. J. Solid State Chem. 1999, 145, 452.
- 11. Ogata, S.; Miyazaki, I.; Tasaka, Y.; Tagaya, H.; Kadokawa, J. I.; Chiba, K. *J. Mater. Chem.* **1998**, *8*, 2813.
- 12. Khan, A. I.; O'Hare, D. J. Mater. Chem. 2002, 12, 3191.
- Yang, J. H.; Lee, S. Y.; Han, Y. S.; Park, K. C.; Choy, J. H. Bull. Korean Chem. Soc. 2003, 24, 499.
- Ryu, S. Y.; Yoon, M.; Choy, J. H.; Hwang, S. H.; Frube, A.; Asahi,
 T.; Masuhara, H. *Bull. Korean Chem. Soc.* 2003, 24, 446.
- 15. Fujita, W.; Awaga, K. Inorg. Chem. 1996, 35, 1915.
- Fujita, W.; Awaga, K.; Yokoyama, T. Inorg. Chem. 1996, 36, 196.
- Rabu, P.; Rouba, S.; Laget, V.; Hornick, C.; Drillon, M. J. Chem. Soc. Chem. Commun. 1996, 1107.
- Laget, V.; Hornick, C.; Drillon, M. J. Mater. Chem. 1999, 9, 169.
- 19. Dunn, J. G.; Muzenda, C. Thermochim. Acta 2001, 369, 117.
- 20. Asbrink, S.; Norbby, L. J. Acta Cryst. 1970, B26, 8.